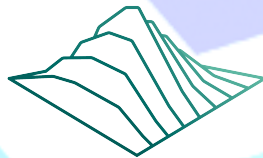


NOVEL MATERIALS DISCOVERY



MAX-PLANCK-GESELLSCHAFT

Learning Descriptors for Materials Properties with Symbolic Regression and Compressed Sensing



Luca M. Ghiringhelli

FRITZ-HABER-INSTITUT

MAX-PLANCK-GESELLSCHAFT

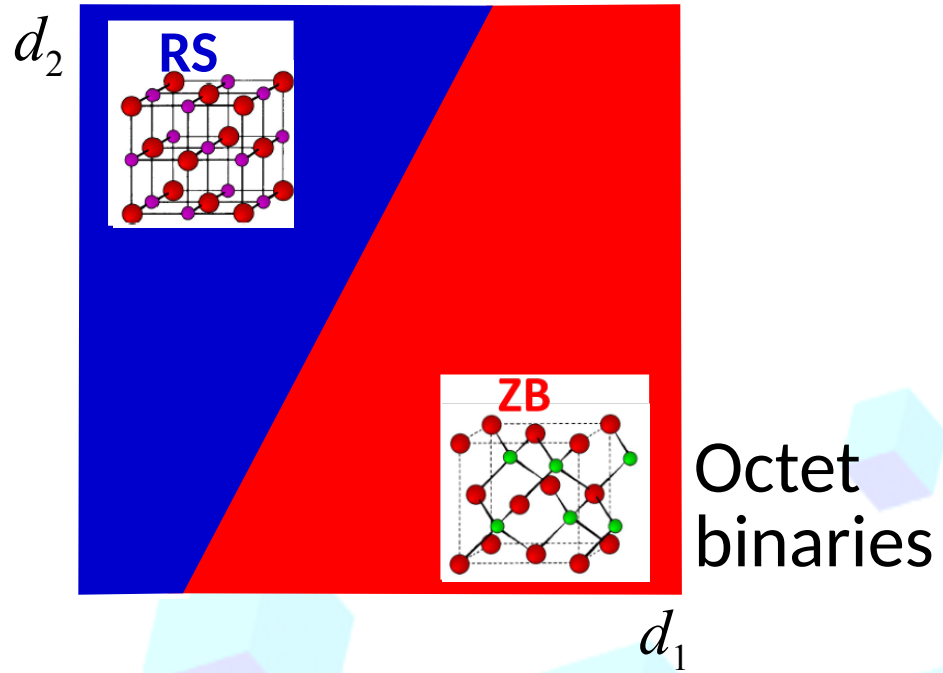
Big Data Summer

A summer school of the BiGmax Network
Platja d'Aro, Spain, September 9 - 13, 2019

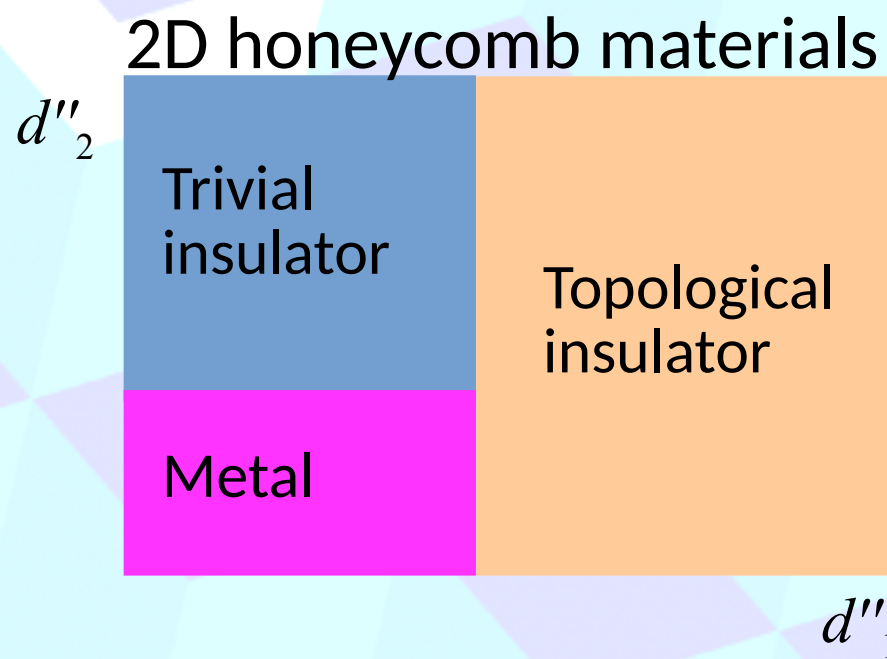
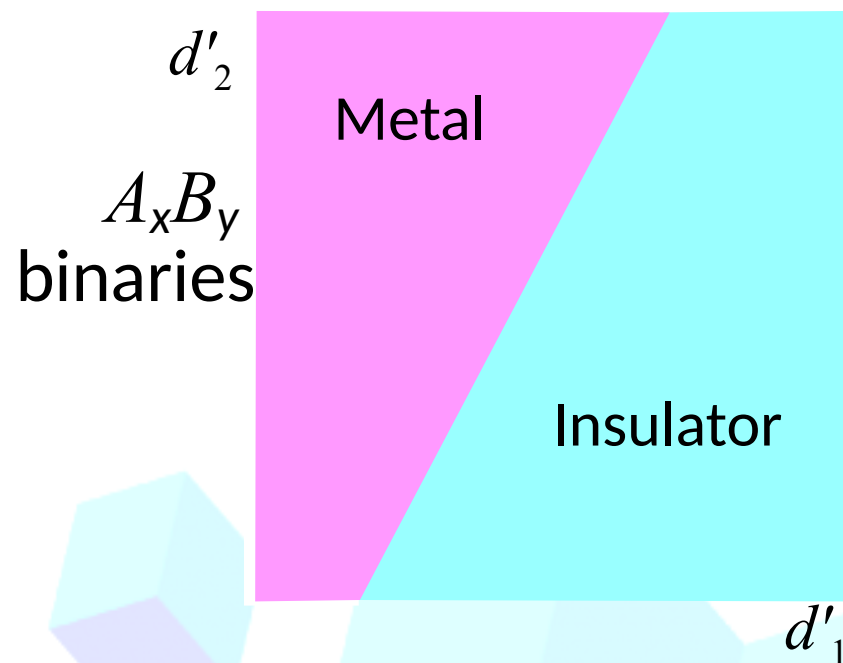
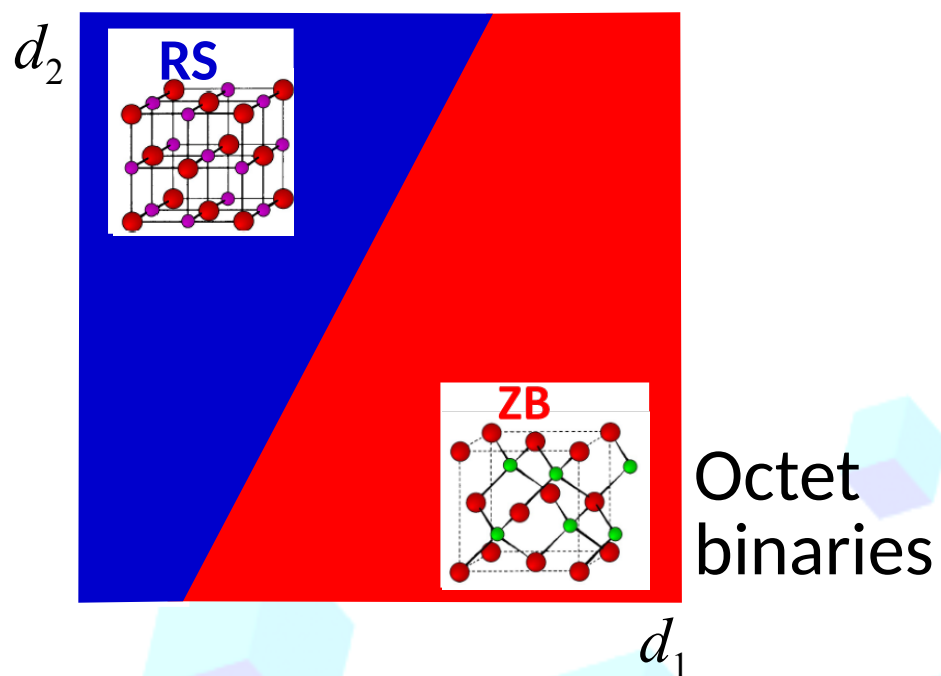
Charts/maps of materials



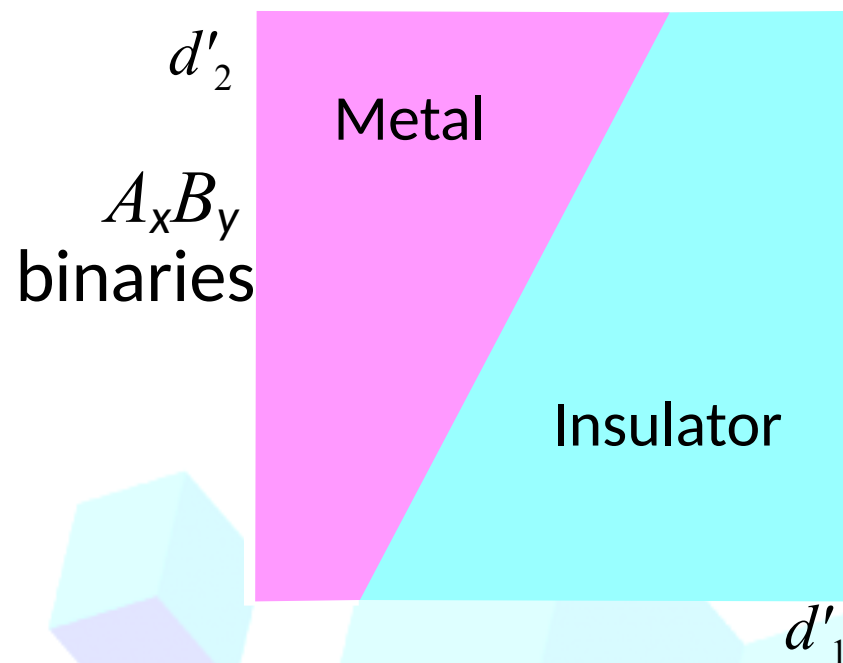
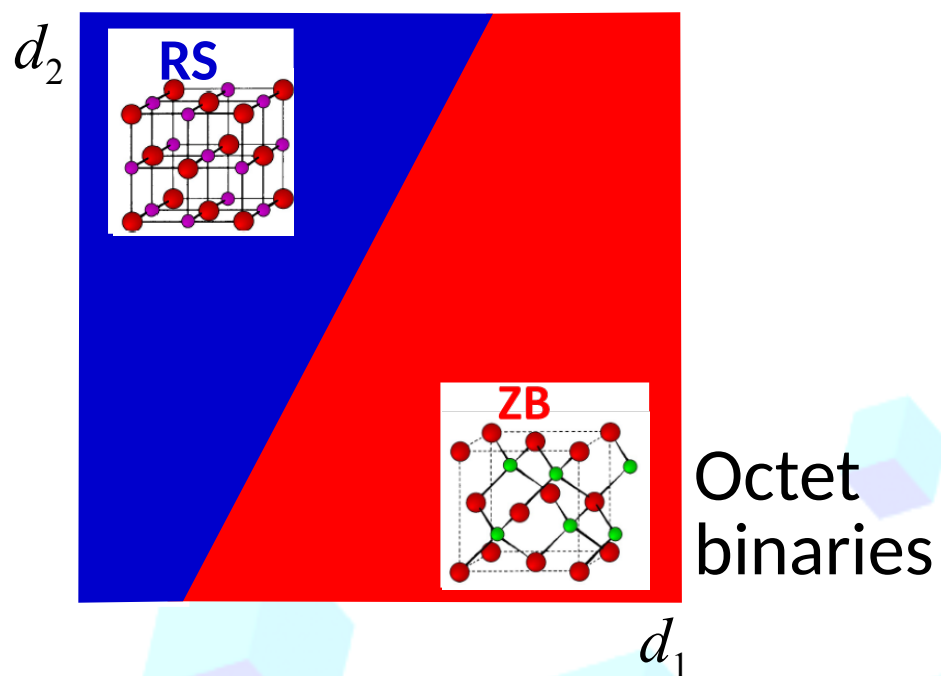
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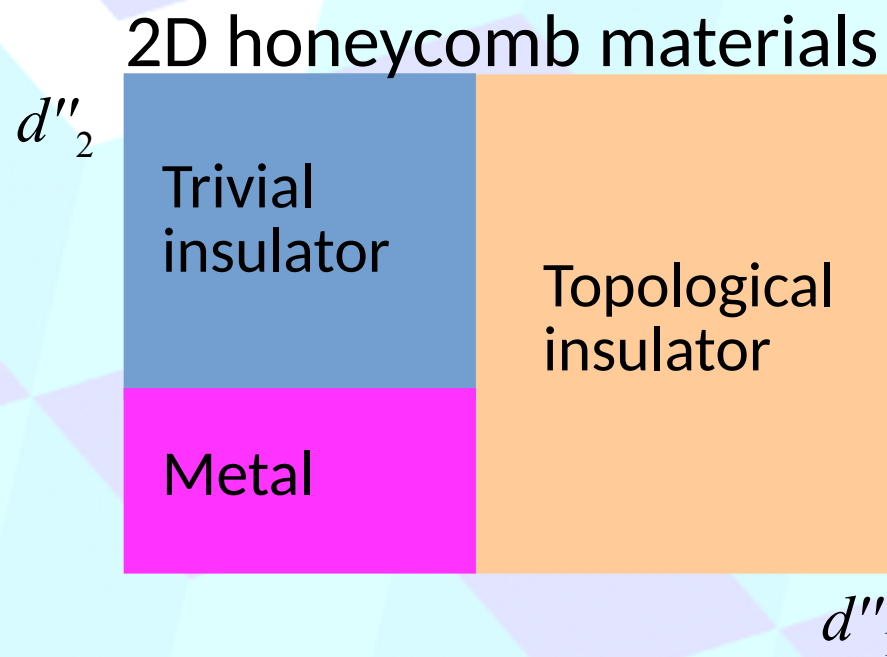
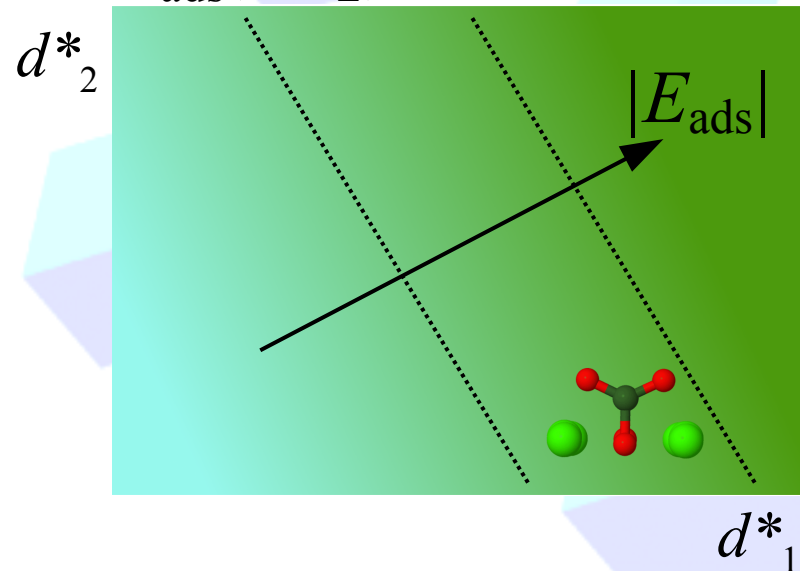
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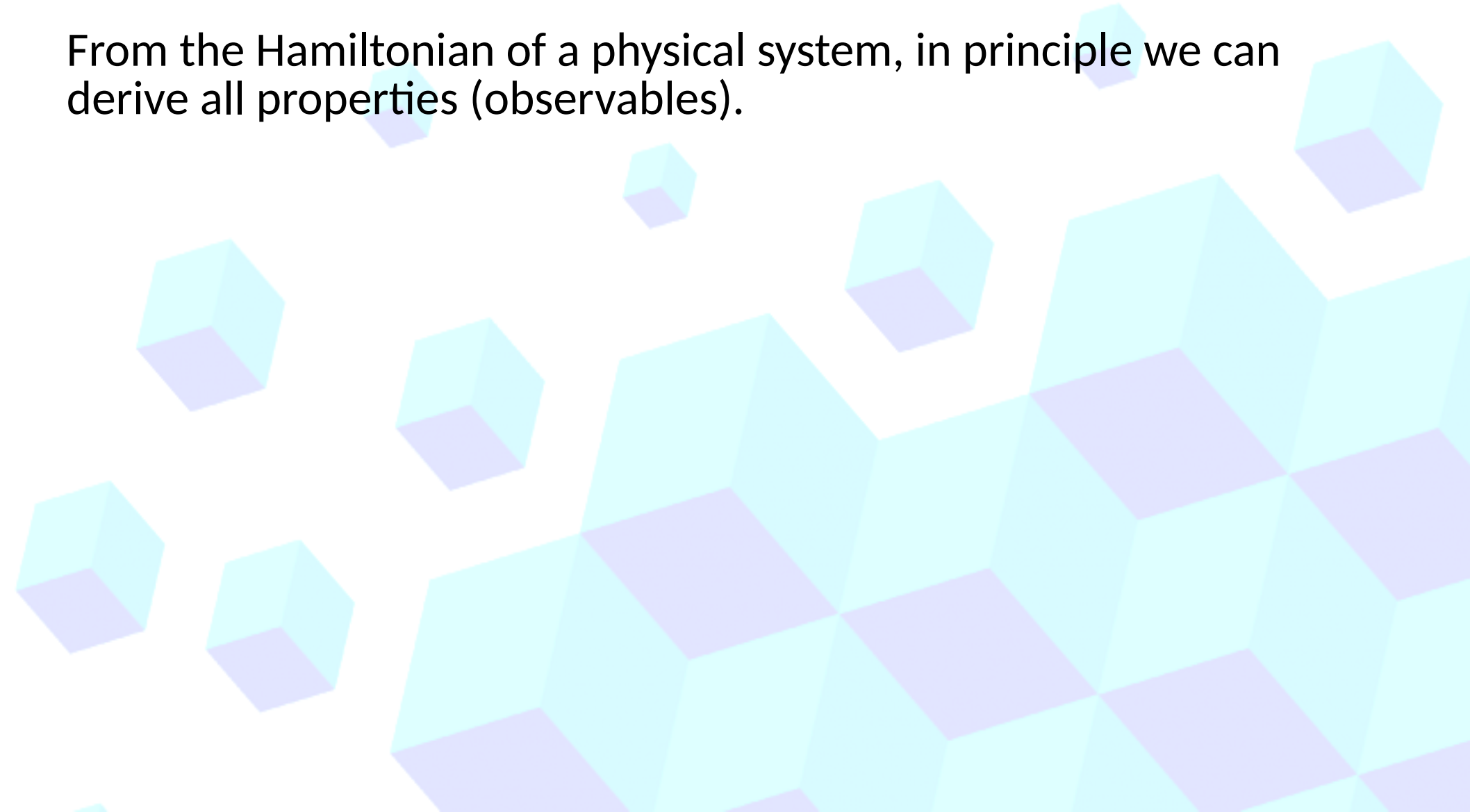
$E_{\text{ads}}(\text{CO}_2)$ on oxides



Building maps of materials properties

A quantum many-body problem

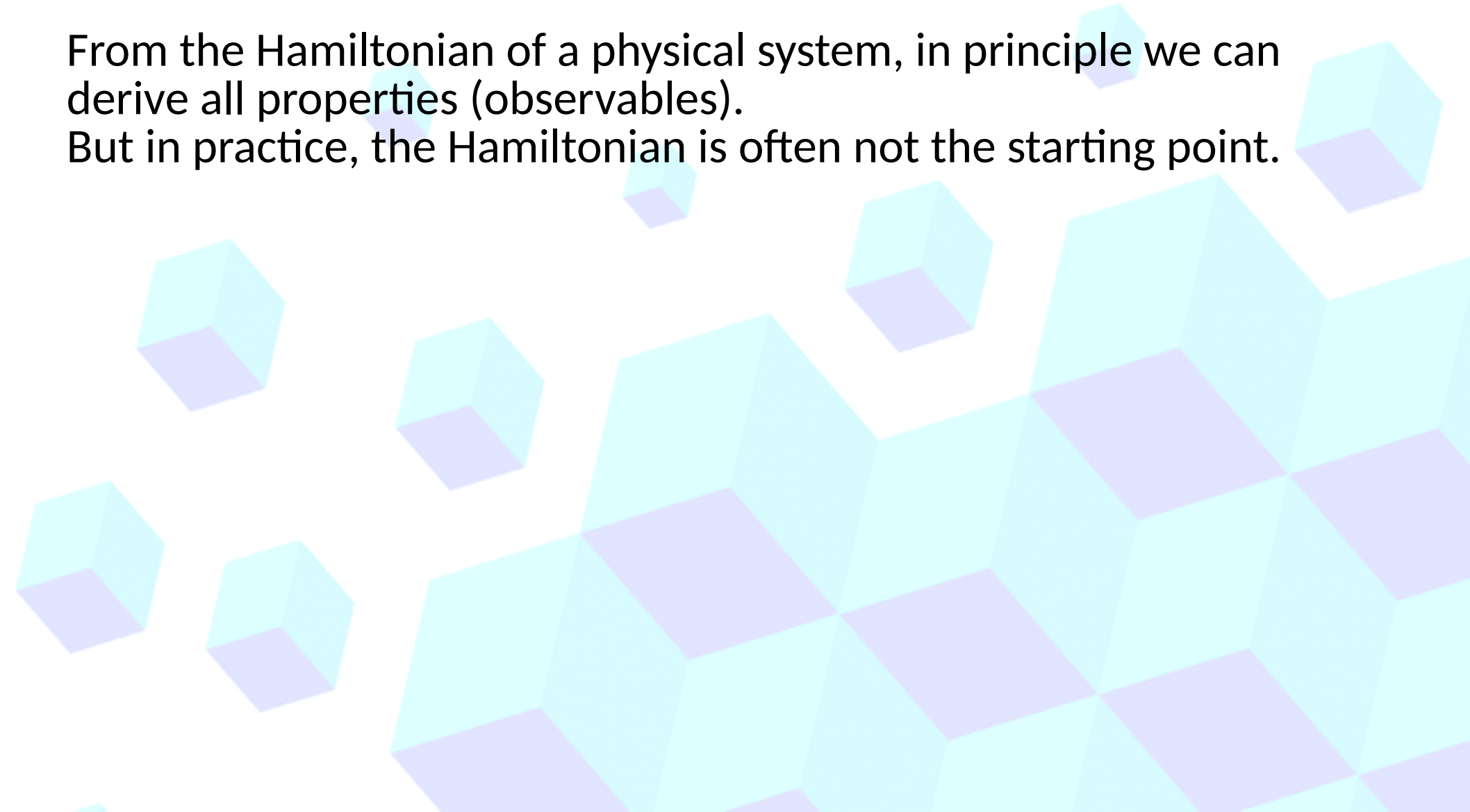
From the Hamiltonian of a physical system, in principle we can derive all properties (observables).



Building maps of materials properties

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But in practice, the Hamiltonian is often not the starting point.

For instance, given a class of chemical compositions (e.g., via prototype formula, such as ABX_3):

- what is the most stable crystal structure of each material in the class?
- which materials are metals / topological insulators / superconductors ?
- which material has the highest melting point?
- which materials has a surface optimal for catalysing some chemical reaction?

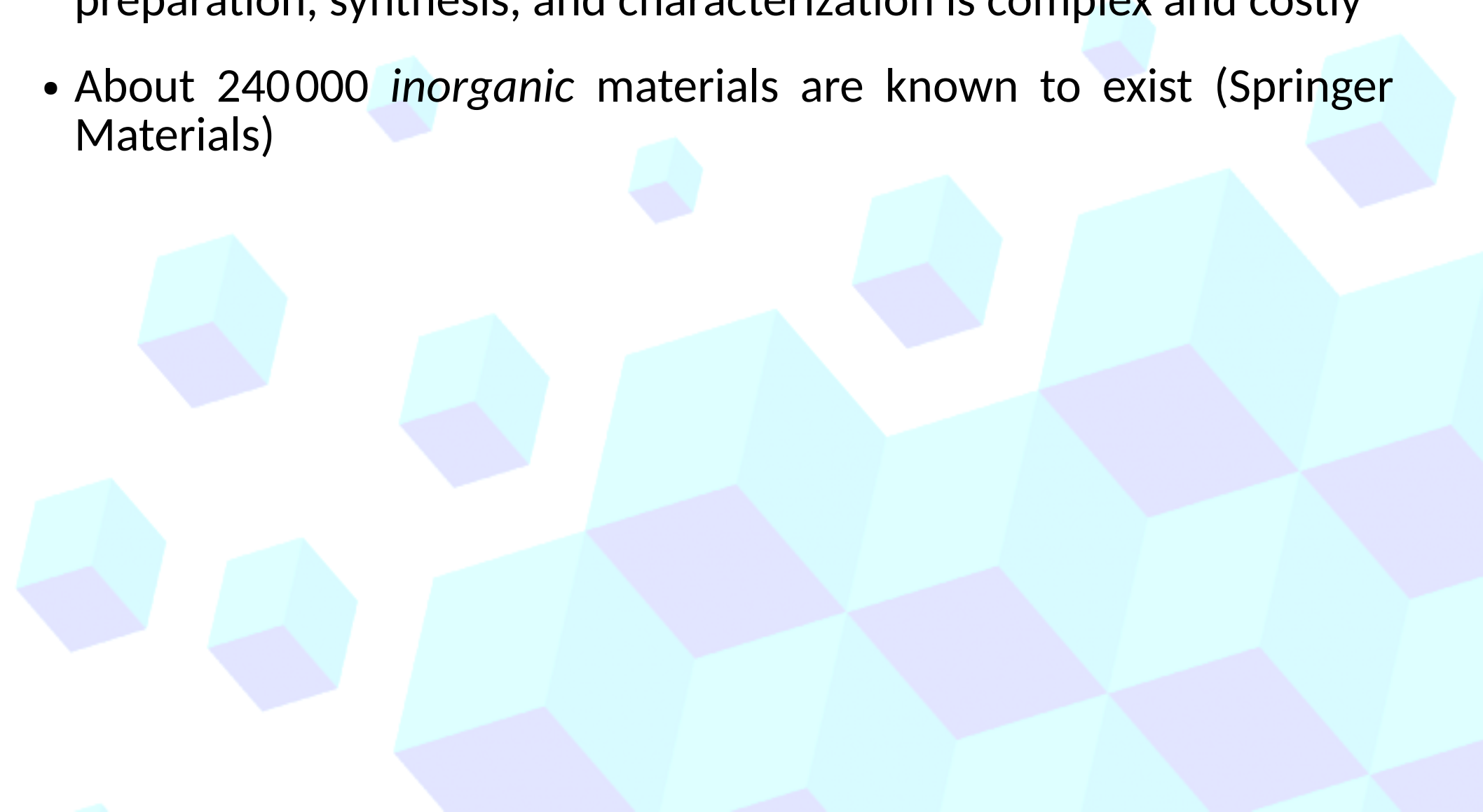
The Big Picture

- Design of new materials: preparation, synthesis, and characterization is complex and costly

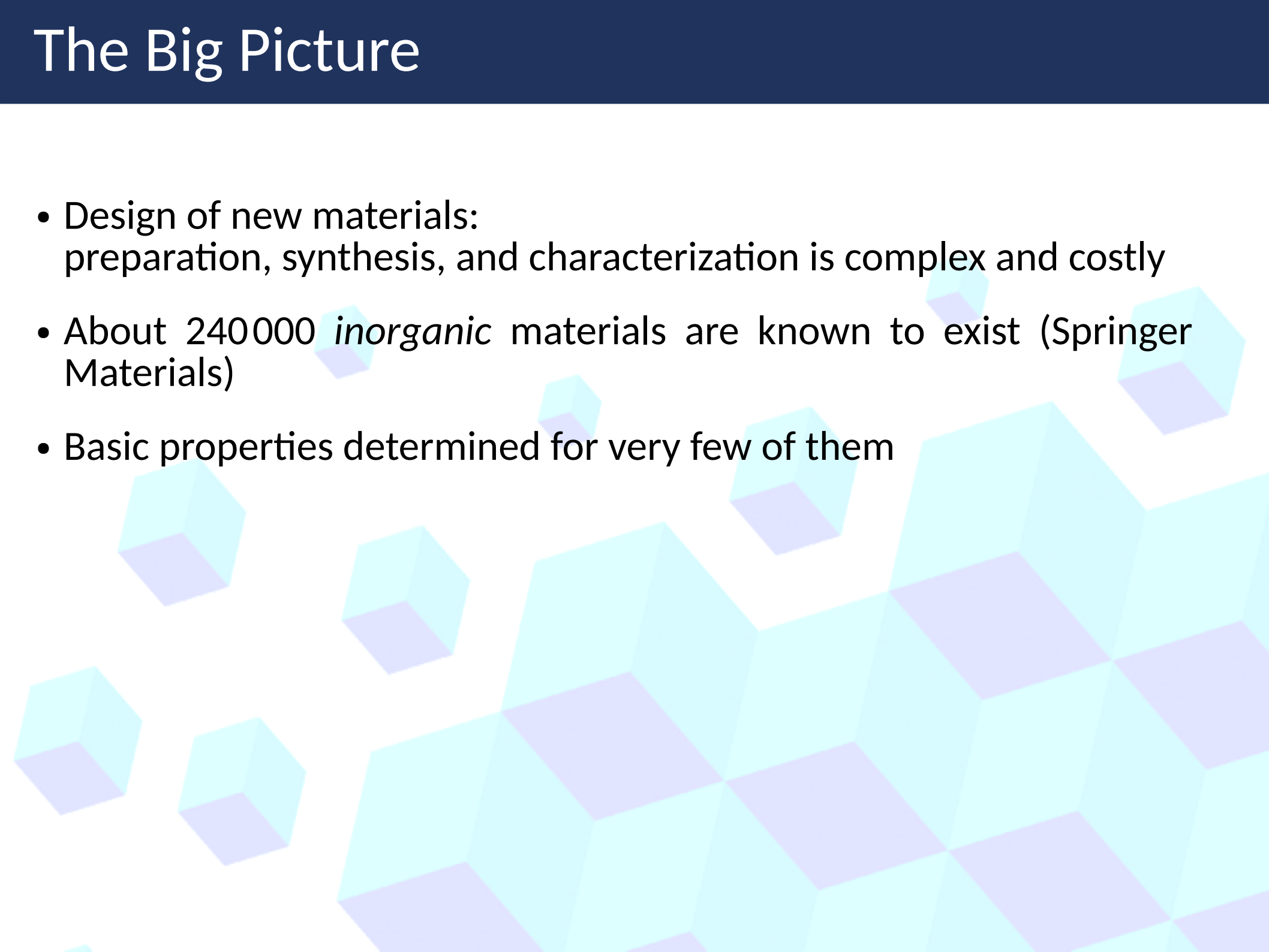


The Big Picture

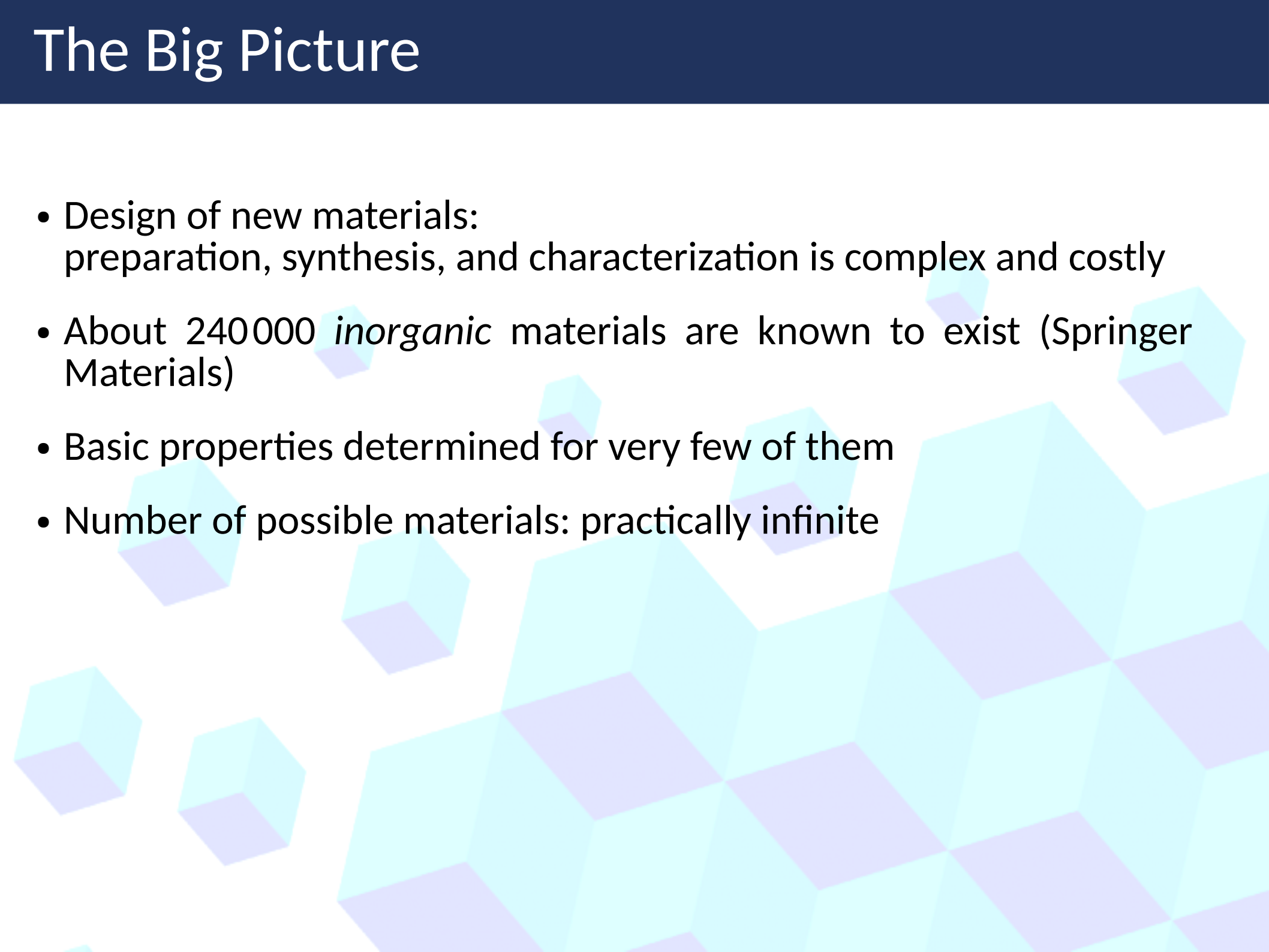
- Design of new materials: preparation, synthesis, and characterization is complex and costly
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The Big Picture

- Design of new materials: preparation, synthesis, and characterization is complex and costly
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 - Basic properties determined for very few of them
- 
- The background features a collection of 3D cubes in two colors: cyan and purple. Some cubes are scattered in the upper left and middle sections, while a large, dense cluster of overlapping cubes is located in the lower right corner, creating a geometric pattern.

The Big Picture

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- 
- The background of the slide features a decorative pattern of 3D cubes. The cubes are rendered in two colors: a light cyan and a light purple. They are scattered across the slide, with some appearing as small, isolated cubes and others forming larger, overlapping clusters. The overall effect is a modern, geometric aesthetic.

The Big Picture

- Design of new materials: preparation, synthesis, and characterization is complex and costly
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- ⇒ New materials with superior properties exist but not yet known
- Data analytics tools will help to identify trends and anomalies in data and guide discovery of new materials

We have a dream

From the **periodic table of the elements**
to **charts of materials**



We have a dream

From the **periodic table of the elements**
to **charts of materials**

Reihen	Gruppe I. — R ² O	Gruppe II. — RO	Gruppe III. — R ² O ³	Gruppe IV. RH ⁴ RO ²	Gruppe V. RH ³ R ² O ⁵	Gruppe VI. RH ² RO ³	Gruppe VII. RH R ² O ⁷	Gruppe VIII. — RO ⁴
1	H=1							
2	Li=7	Be=9.4	B=11	C=12	N=14	O=16	F=19	
3	Na=23	Mg=24	Al=27.3	Si=28	P=31	S=32	Cl=35.5	
4	K=39	Ca=40	—=44	Ti=48	V=51	Cr=52	Mn=55	Fe=56, Co=59, Ni=59, Cu=63.
5	(Cu=63)	Zn=65	—=68	—=72	As=75	Se=78	Br=80	
6	Rb=85	Sr=87	?Yt=88	Zr=90	Nb=94	Mo=96	—=100	Ru=104, Rh=104, Pd=106, Ag=108.
7	(Ag=108)	Cd=112	In=113	Sn=118	Sb=122	Te=125	J=127	
8	Cs=133	Ba=137	?Di=138	?Ce=140	—	—	—	— — — —
9	(—)	—	—	—	—	—	—	
10	—	—	?Er=178	?La=180	Ta=182	W=184	—	Os=195, Ir=197, Pt=198, Au=199.
11	(Au=199)	Hg=200	Tl=204	Pb=207	Bi=208	—	—	
12	—	—	—	Th=231	—	U=240	—	— — — —

Mendeleev's 1871 periodic table

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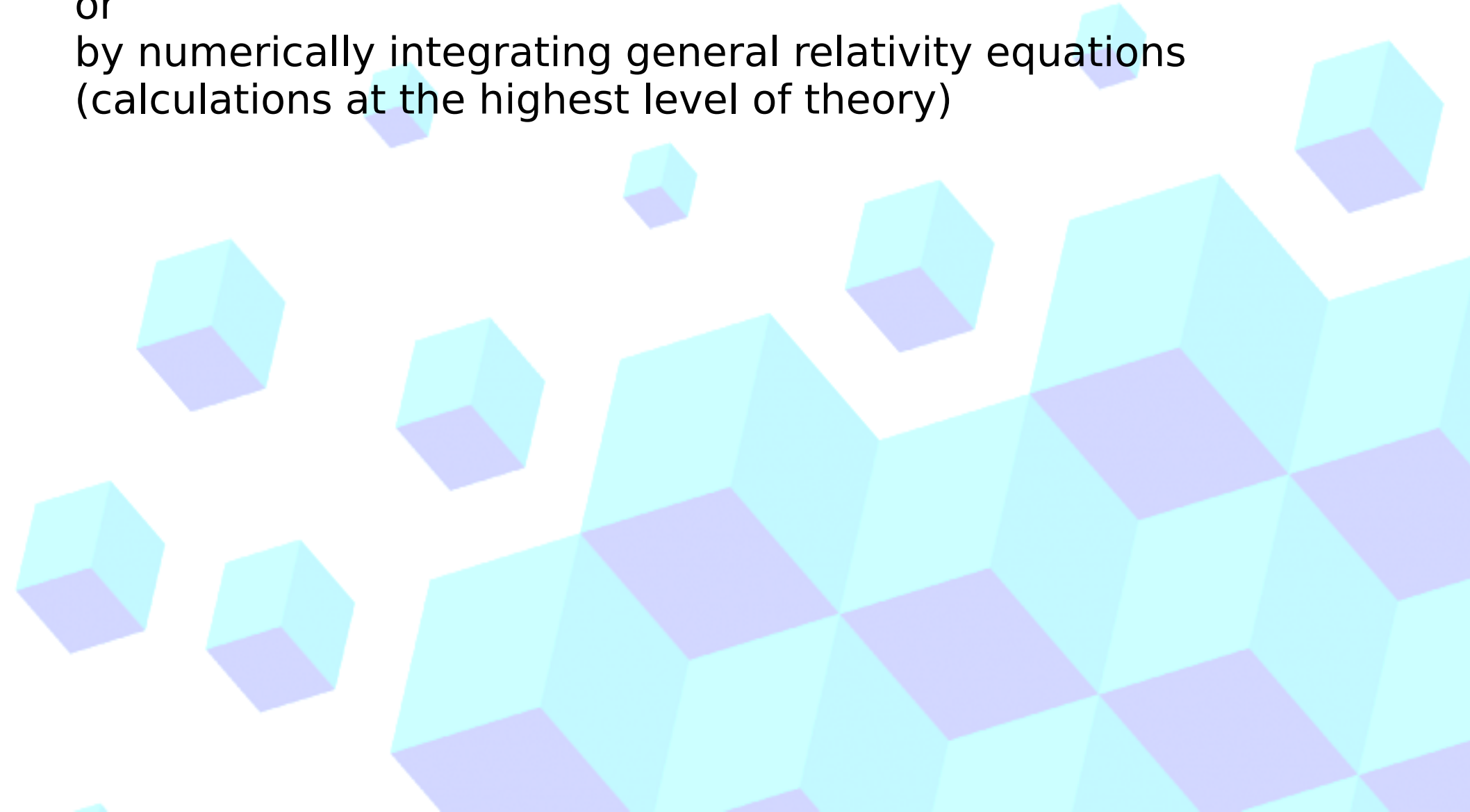
Mendeleev's 1871 periodic table

Learning → Discovery

Suppose

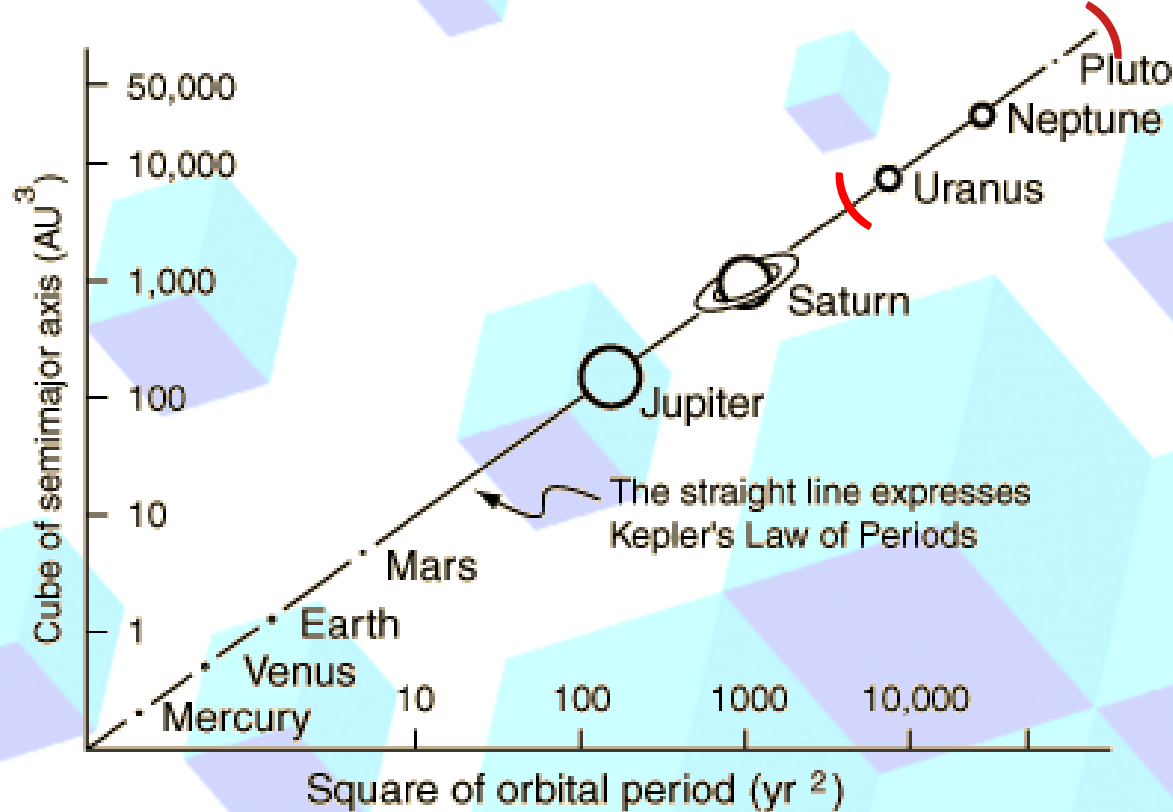
to know the trajectories of all planets in the solar system,
from accurate observations (experiment)

or
by numerically integrating general relativity equations
(calculations at the highest level of theory)



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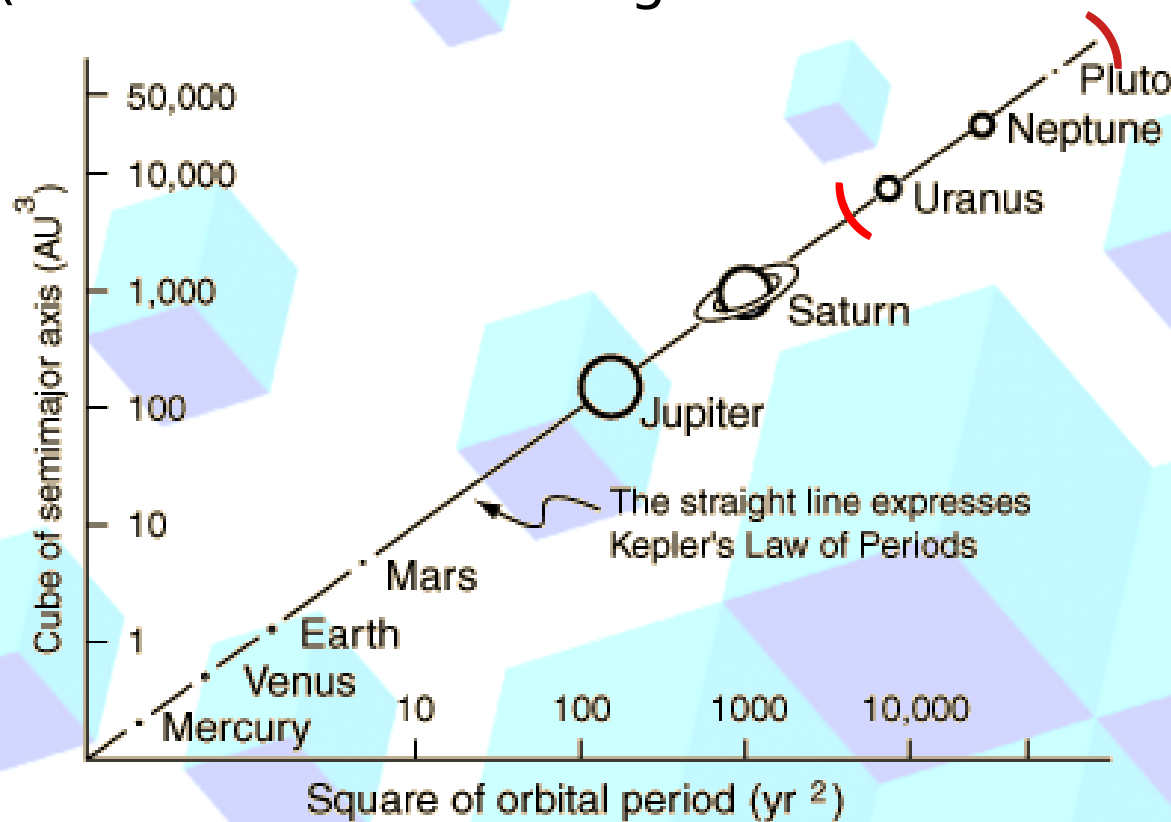
$$(\text{Orbital period})^2 = C (\text{orbit's major axis})^3$$

Learning → Discovery

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$$(\text{Orbital period})^2 = C (\text{orbit's major axis})^3$$

Data
(collected by
Tycho Brahe)

Statistical learning
(performed by
Johannes Kepler)

Physical law
(assessed by
Isaac Newton)

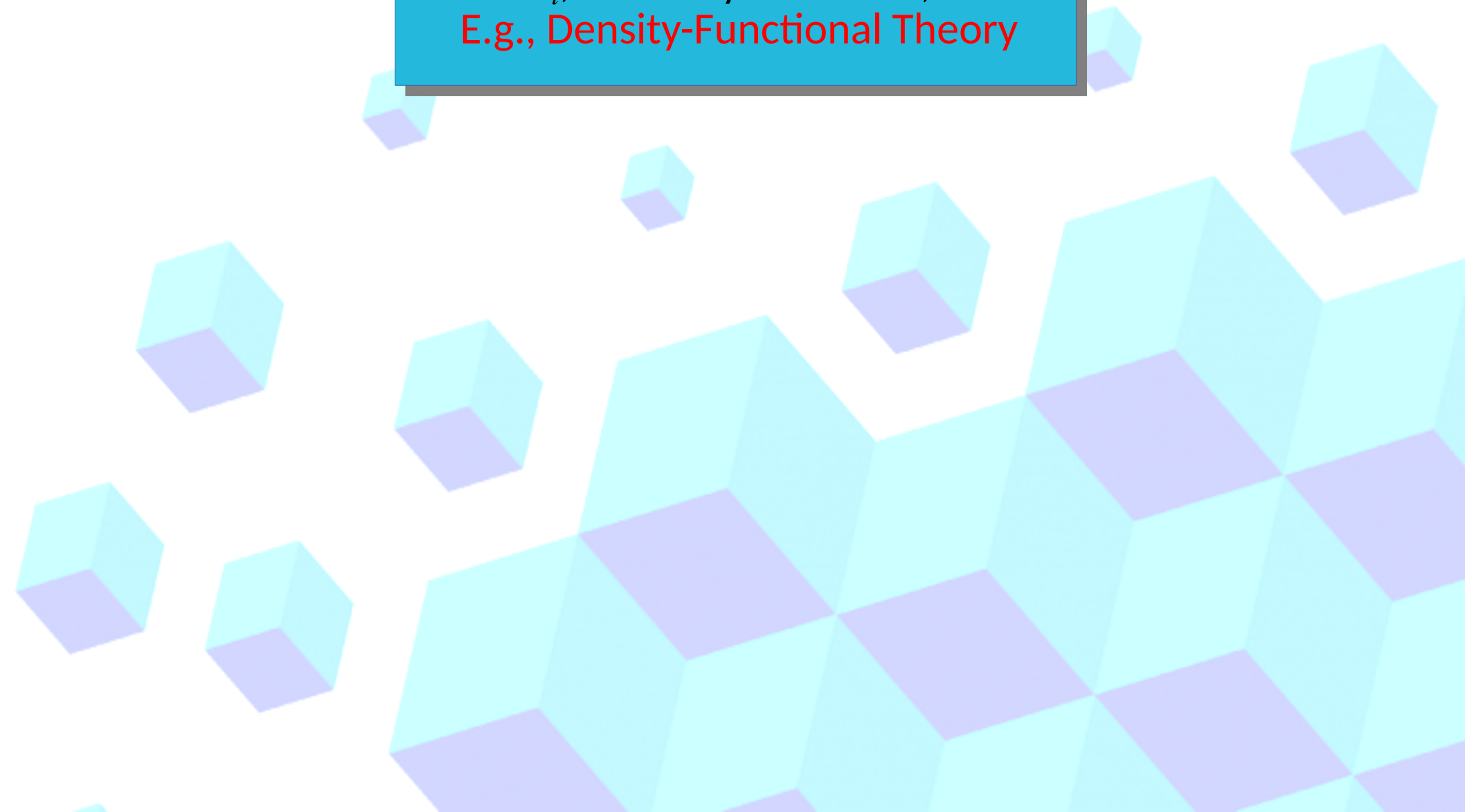
Supervised (big-)data analysis: a flow chart

Training set

Calculate properties and functions

P_i , for many *materials*, i

E.g., Density-Functional Theory



Supervised (big-)data analysis: a flow chart

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Find the *appropriate* descriptor d_i ,
build a table:

i	d_i	P_i
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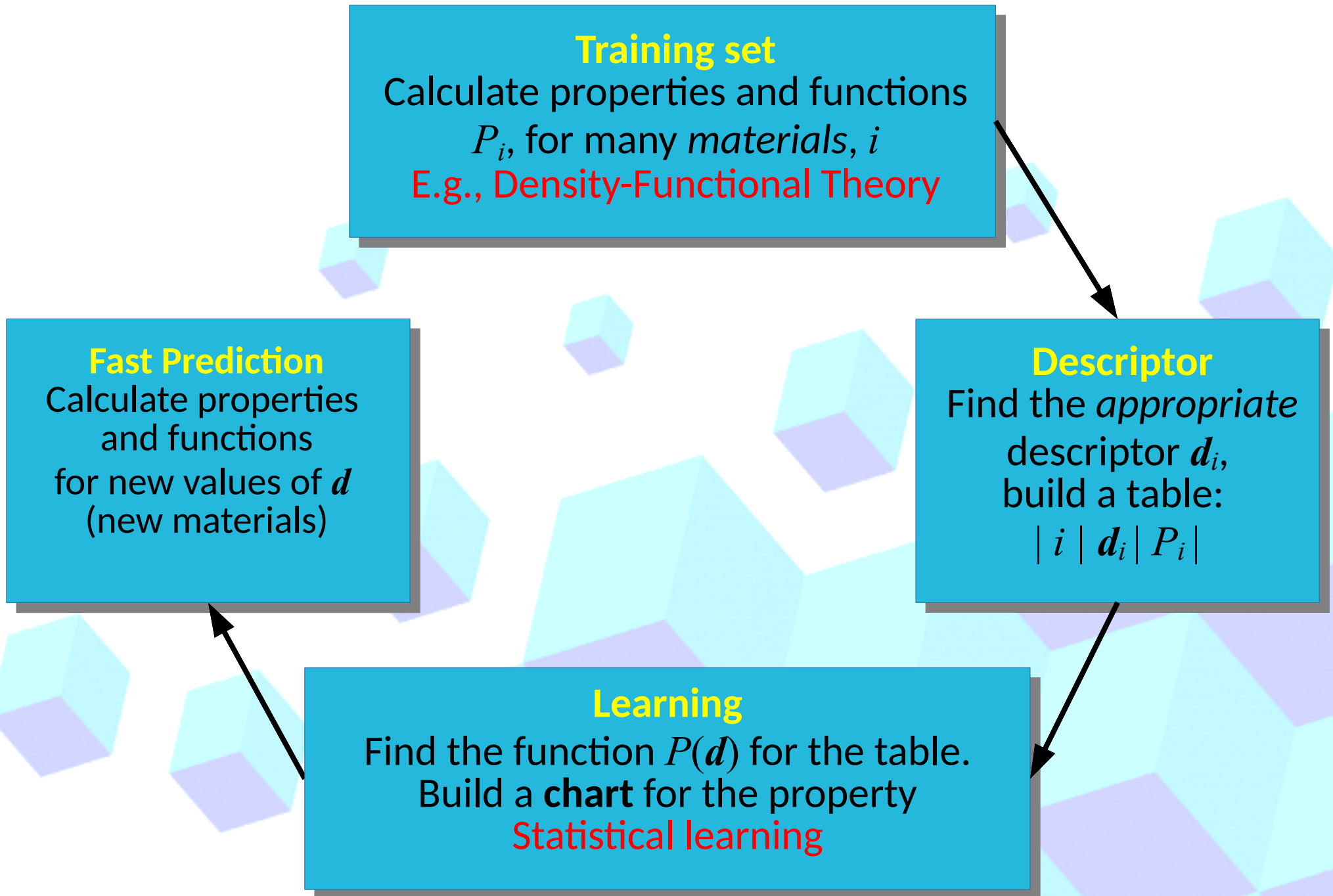
Learning

Find the function $P(d)$ for the table.

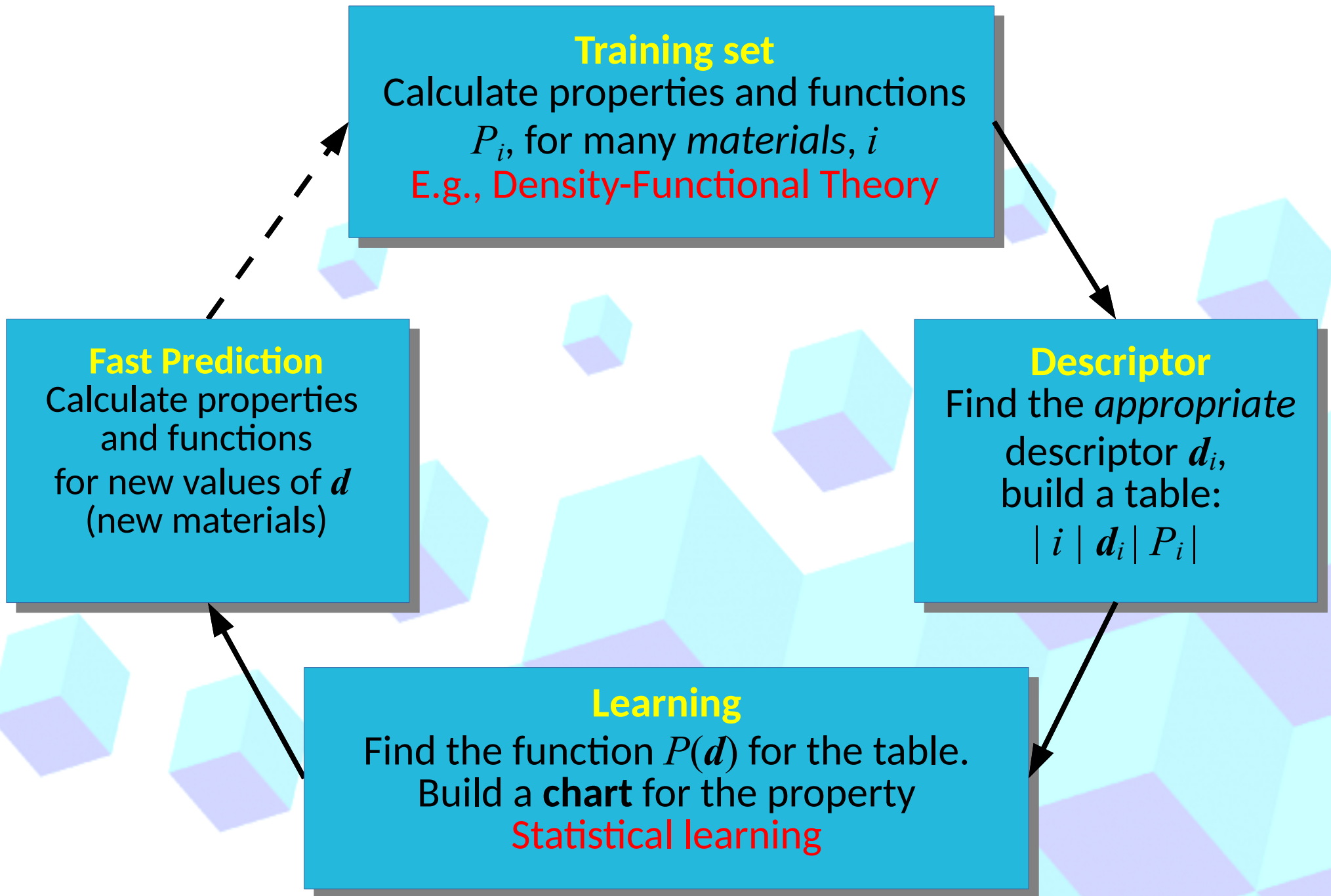
Build a **chart** for the property

Statistical learning

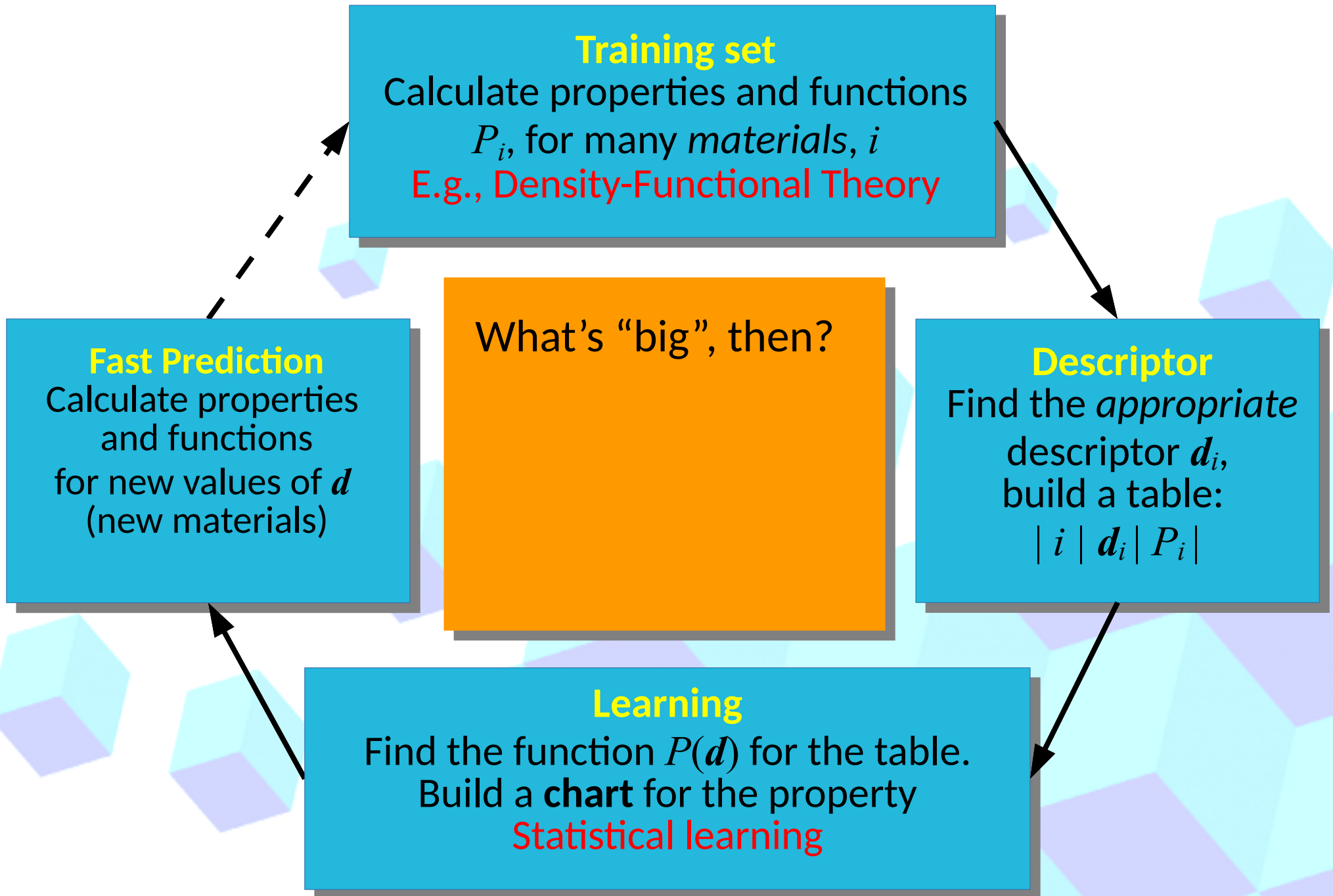
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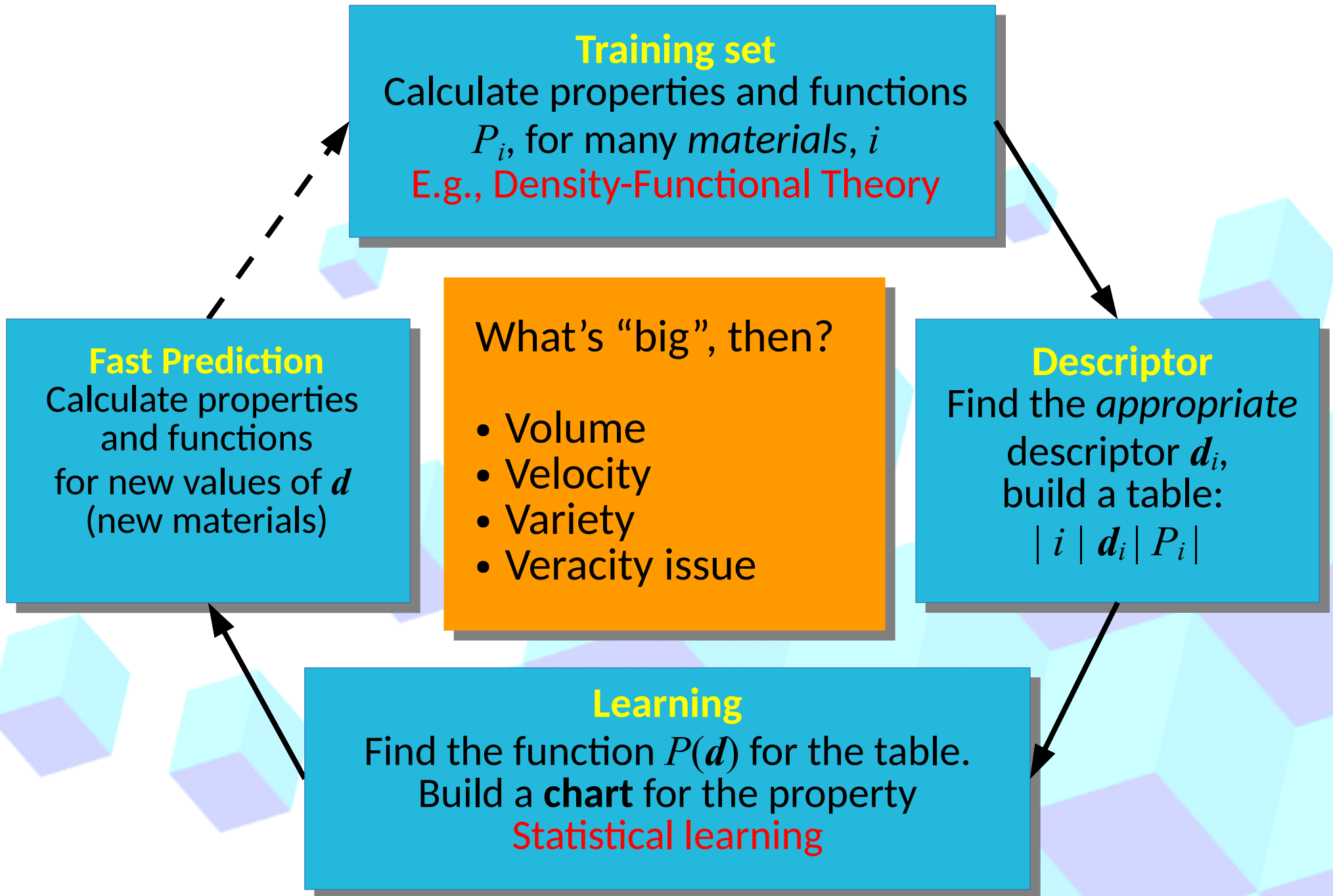
Supervised (big-)data analysis: a flow chart



Supervised big-data analysis: a flow chart



Supervised big-data analysis: a flow chart



Descriptor? Don't we know it from the start?

Training set

Calculate properties and functions
 P_i , for many *materials*, i
E.g., Density-Functional Theory

$\{R_I, Z_I\} \rightarrow$ Hamiltonian

$\{R_I\} \rightarrow$ Geometry

- translational, rotational, permutational invariant
- coarse graining $\{R_I\}$?

$\{Z_I\} \rightarrow$ Chemistry

Descriptor

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i	d_i	P_i
-----	-------	-------

Learning

Find the function $P(d)$ for the table.
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Statistical learning

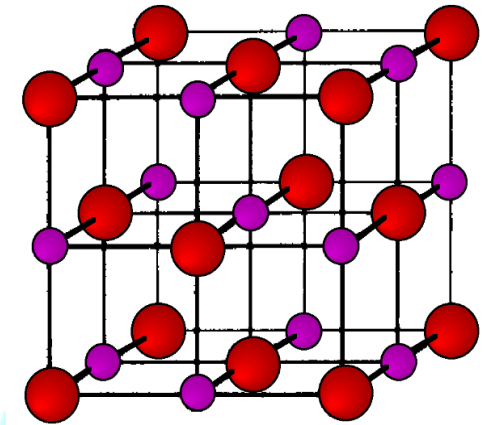
An example: predicting crystal structures from the composition

82 octet AB binary compounds

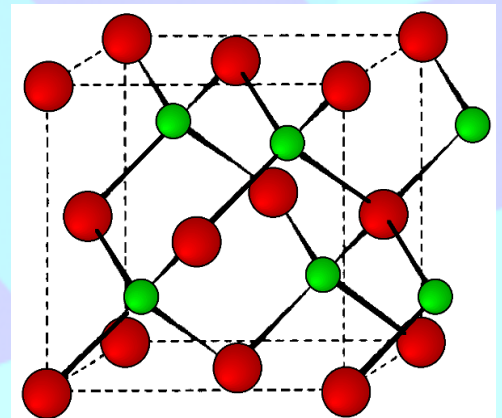
hydrogen 1 H																			helium 2 He 4.0026
lithium 3 Li 6.941	beryllium 4 Be 9.0122									boron 5 B 10.811	carbon 6 C 12.011	nitrogen 7 N 14.007	oxygen 8 O 15.999	fluorine 9 F 18.998	neon 10 Ne 20.180				
sodium 11 Na 22.990	magnesium 12 Mg 24.305									aluminum 13 Al 26.982	silicon 14 Si 28.086	phosphorus 15 P 30.974	sulfur 16 S 32.065	chlorine 17 Cl 35.453	argon 18 Ar 39.948				
potassium 19 K 39.098	calcium 20 Ca 40.078	scandium 21 Sc 44.956	titanium 22 Ti 47.867	vanadium 23 V 50.942	chromium 24 Cr 51.996	manganese 25 Mn 54.938	iron 26 Fe 55.845	cobalt 27 Co 58.933	nickel 28 Ni 58.693	gallium 31 Ga 69.723	germanium 32 Ge 72.61	arsenic 33 As 74.922	selenium 34 Se 78.96	bromine 35 Br 79.904	krypton 36 Kr 83.80				
rubidium 37 Rb 85.468	strontium 38 Sr 87.62	yttrium 39 Y 88.906	zirconium 40 Zr 91.224	niobium 41 Nb 92.906	molybdenum 42 Mo 95.94	technetium 43 Tc [98]	ruthenium 44 Ru 101.07	rhodium 45 Rh 102.91	palladium 46 Pd 106.42	indium 49 In 114.82	tin 50 Sn 118.71	antimony 51 Sb 121.76	tellurium 52 Te 127.60	iodine 53 I 126.90	xenon 54 Xe 131.29				
caesium 55 Cs 132.91	barium 56 Ba 137.33	lanthanum 57 La 138.91	cerium 58 Ce 140.12	praseodymium 59 Pr 140.91	neodymium 60 Nd 144.24	promethium 61 Pm [145]	samarium 62 Sm 150.36	europium 63 Eu 151.96	gadolinium 64 Gd 157.25	gold 79 Au 196.97	mercury 80 Hg 200.59	thallium 81 Tl 204.38	lead 82 Pb 207.2	bismuth 83 Bi 208.98	polonium 84 Po [209]	astatine 85 At [210]	radon 86 Rn [222]		
francium 87 Fr [223]	radium 88 Ra [226]	actinium 89 Ac [227]	thorium 90 Th 232.04	protactinium 91 Pa 231.04	uranium 92 U 238.03	neptunium 93 Np [237]	plutonium 94 Pu [244]	americium 95 Am [243]	curium 96 Cm [247]	unilithium 110 Uun [271]	unnilium 111 Uuu [272]	ununium 112 Uub [277]	ununquadium 114 Uuq [289]						

lanthanide series

** Actinide series



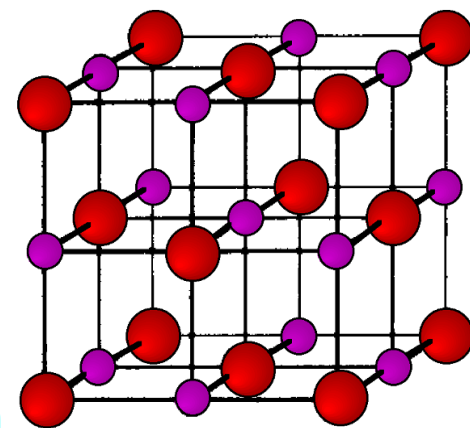
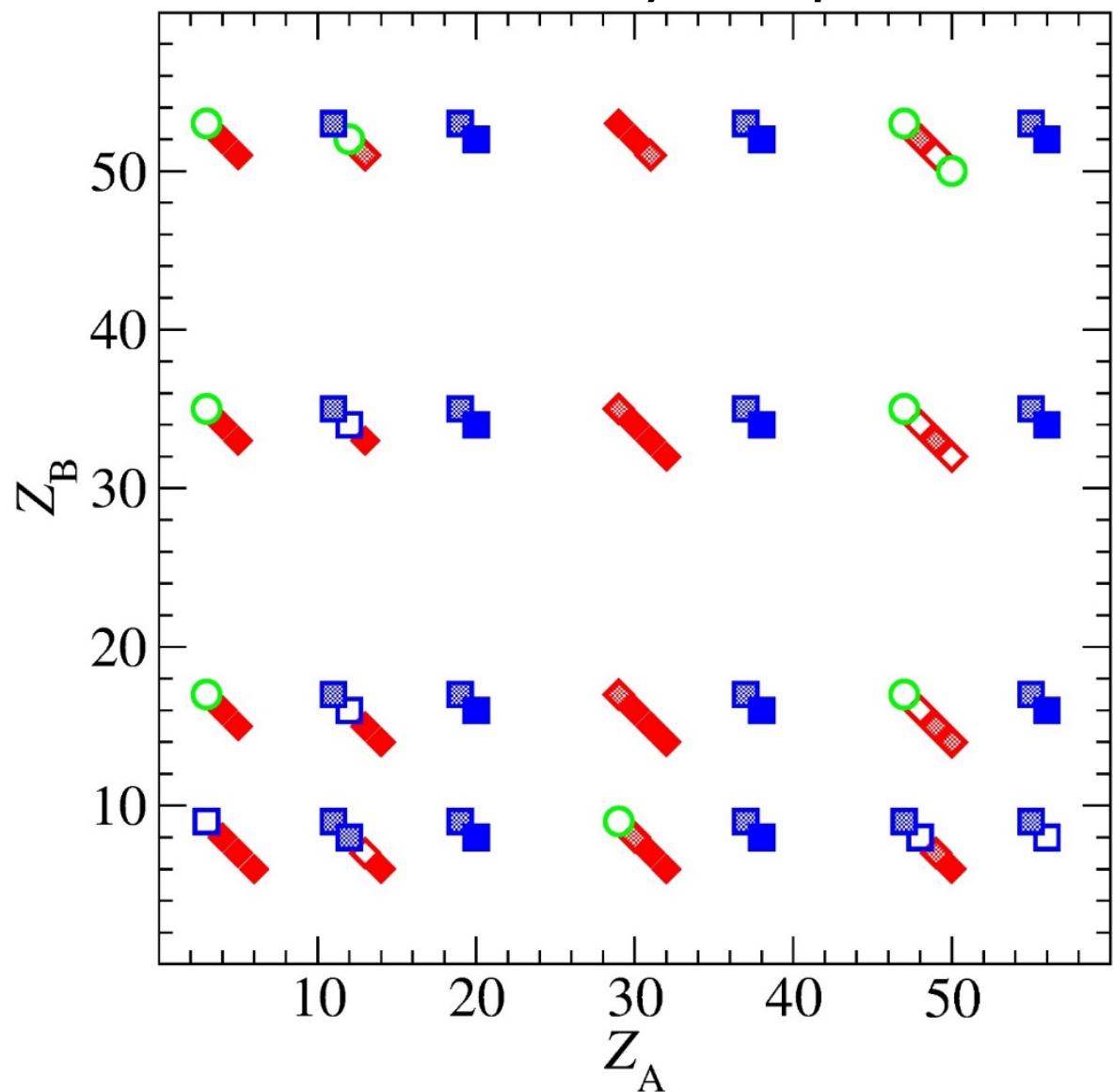
Rock salt



Zinc blende

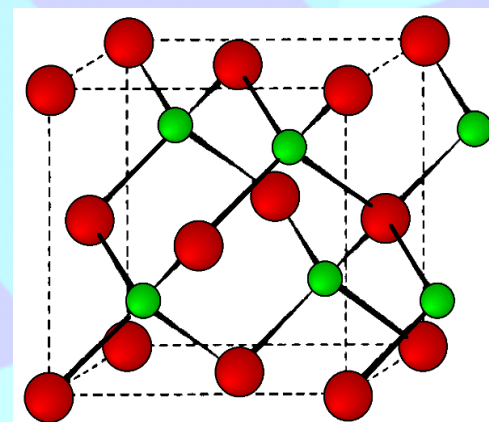
An example: predicting crystal structures from the composition

82 octet AB binary compounds



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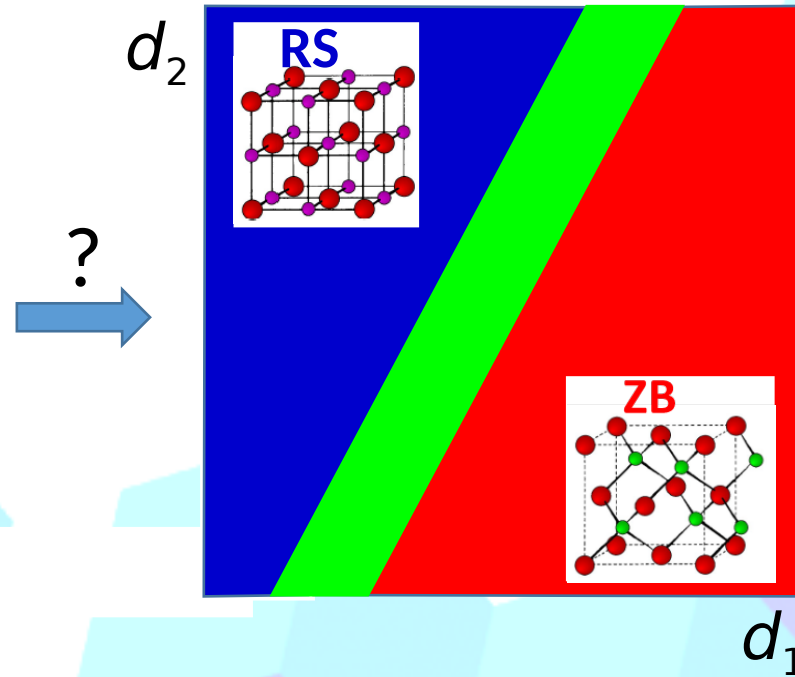
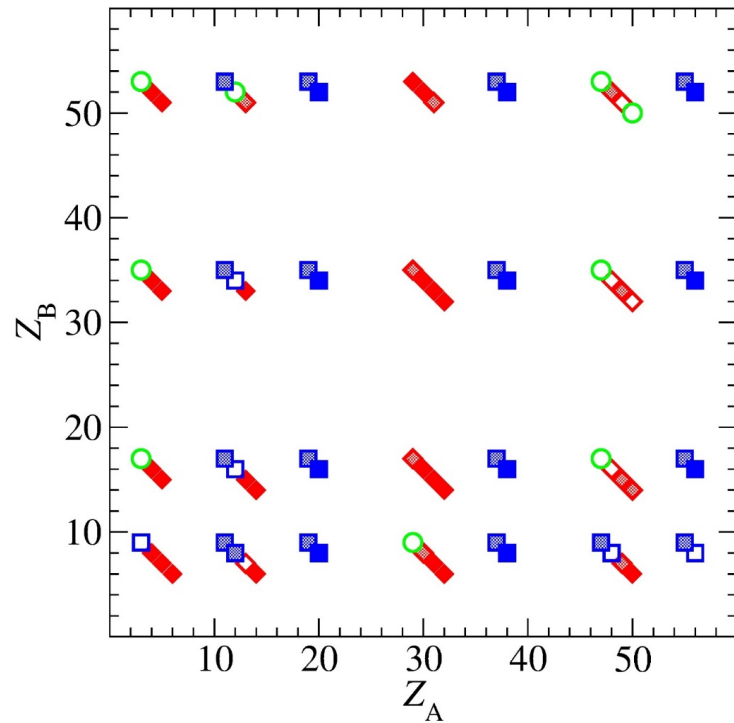
- Rock salt
- Rock salt/Zinc blende
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Zinc blende

An example: predicting crystal structures from the composition

82 octet AB binary compounds

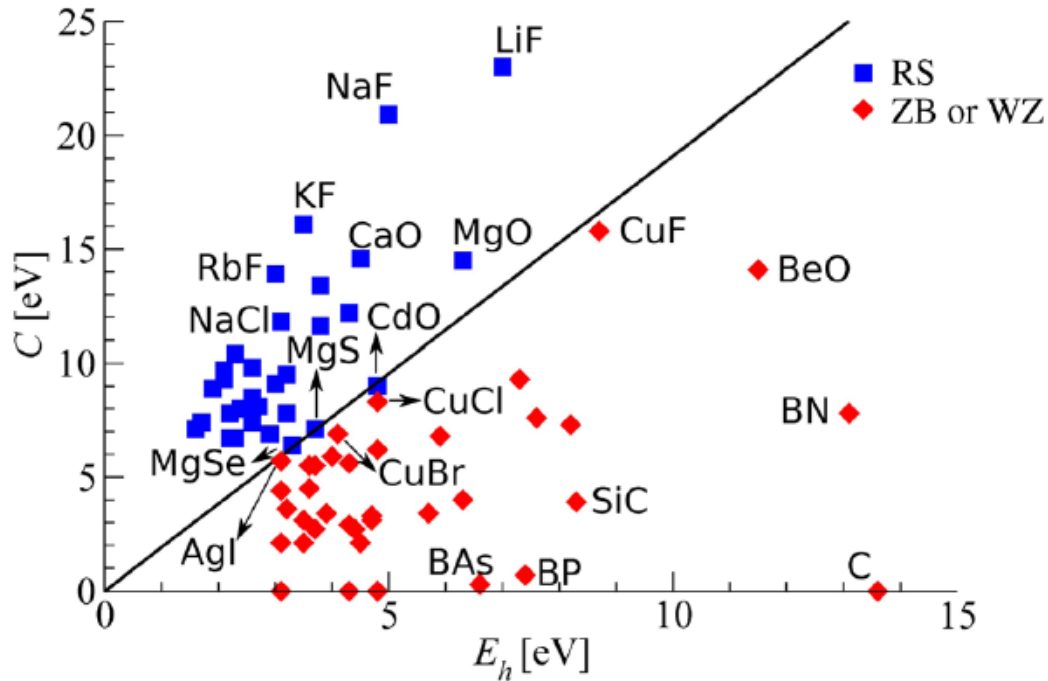


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The descriptor proposed by Phillips and van Vechten in 1969-70 depends on:

- lattice parameter
- electrical conductivity

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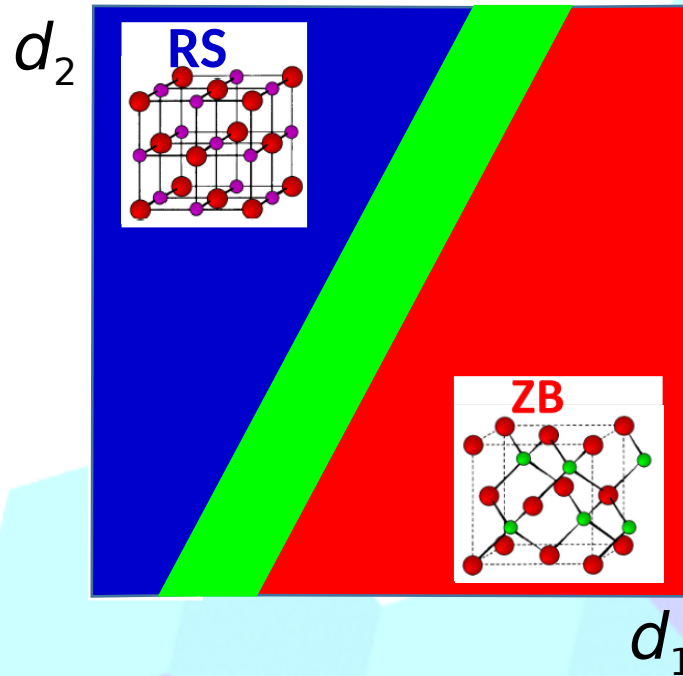
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An example: predicting crystal structures from the composition

82 octet AB binary compounds

Ansatz: atomic features

- HOMO
- LUMO
- Ionization Potential
- Electron Affinity
- Radius of valence s orbital
- Radius of valence p orbital
- Radius of valence d orbital
- ... ?



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$E(\text{Rock salt}) - E(\text{Zinc blende})$

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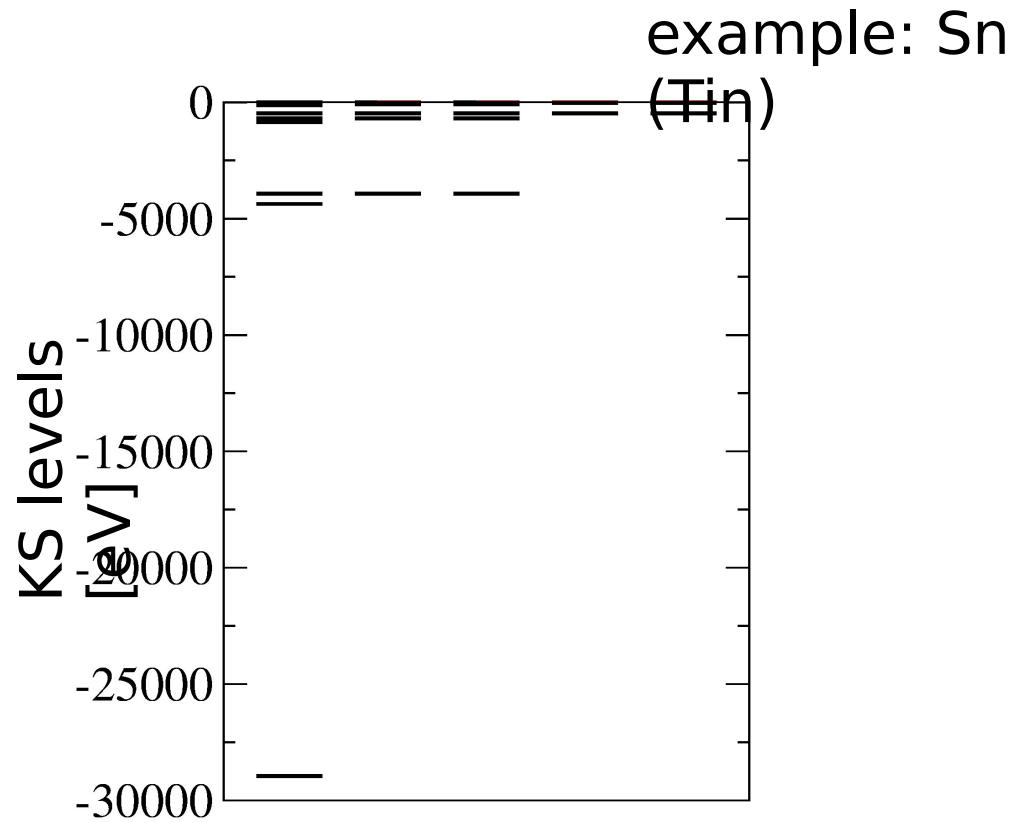
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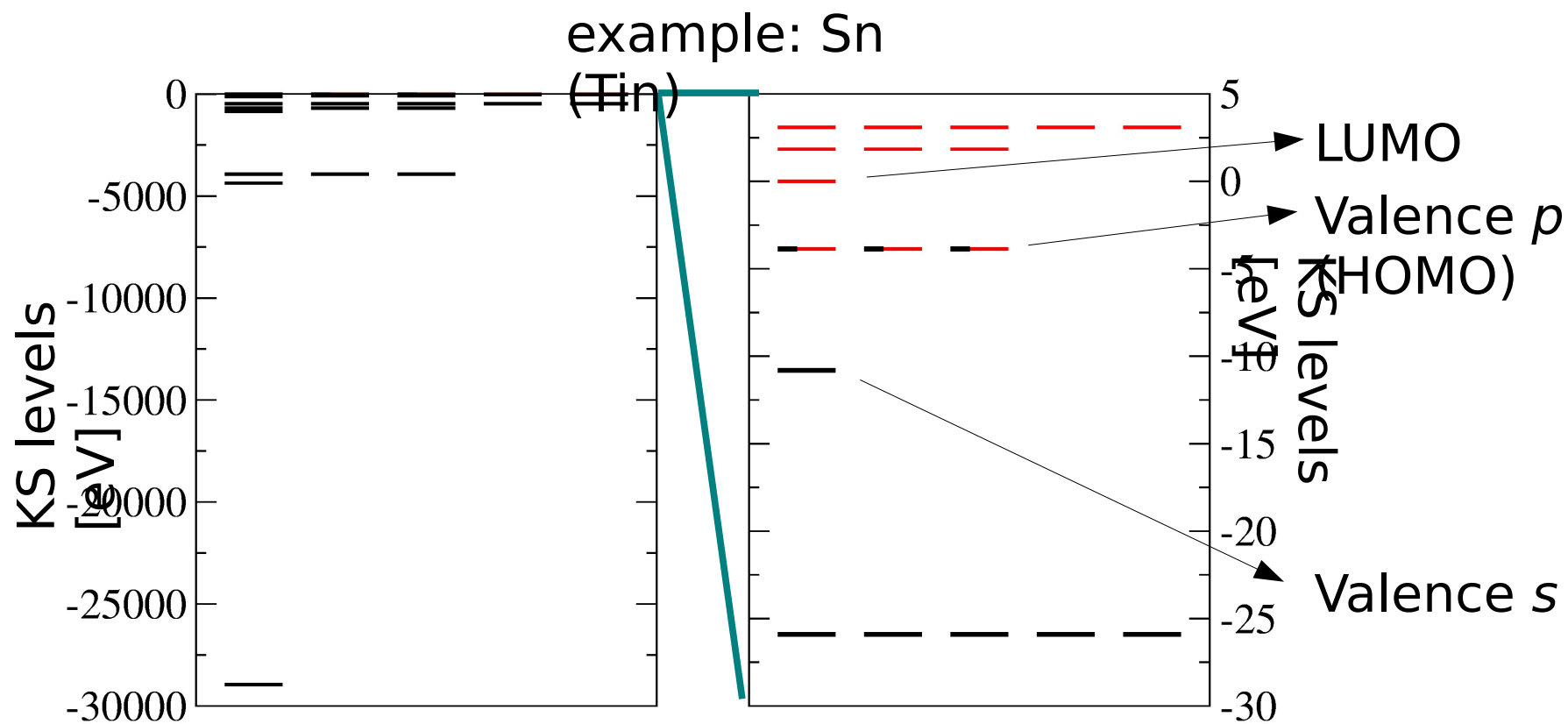
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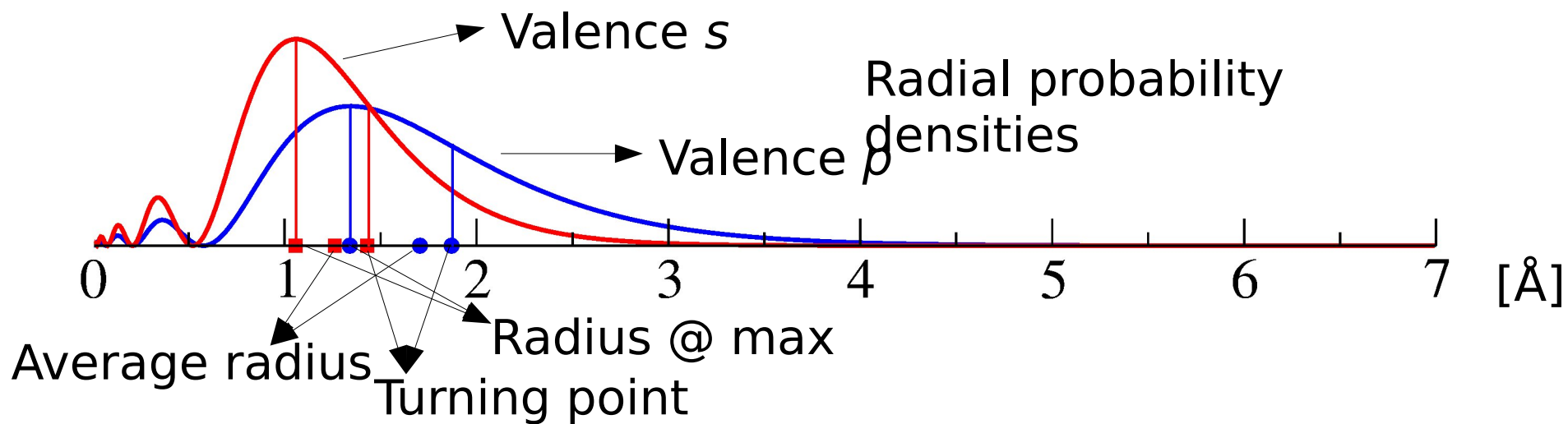
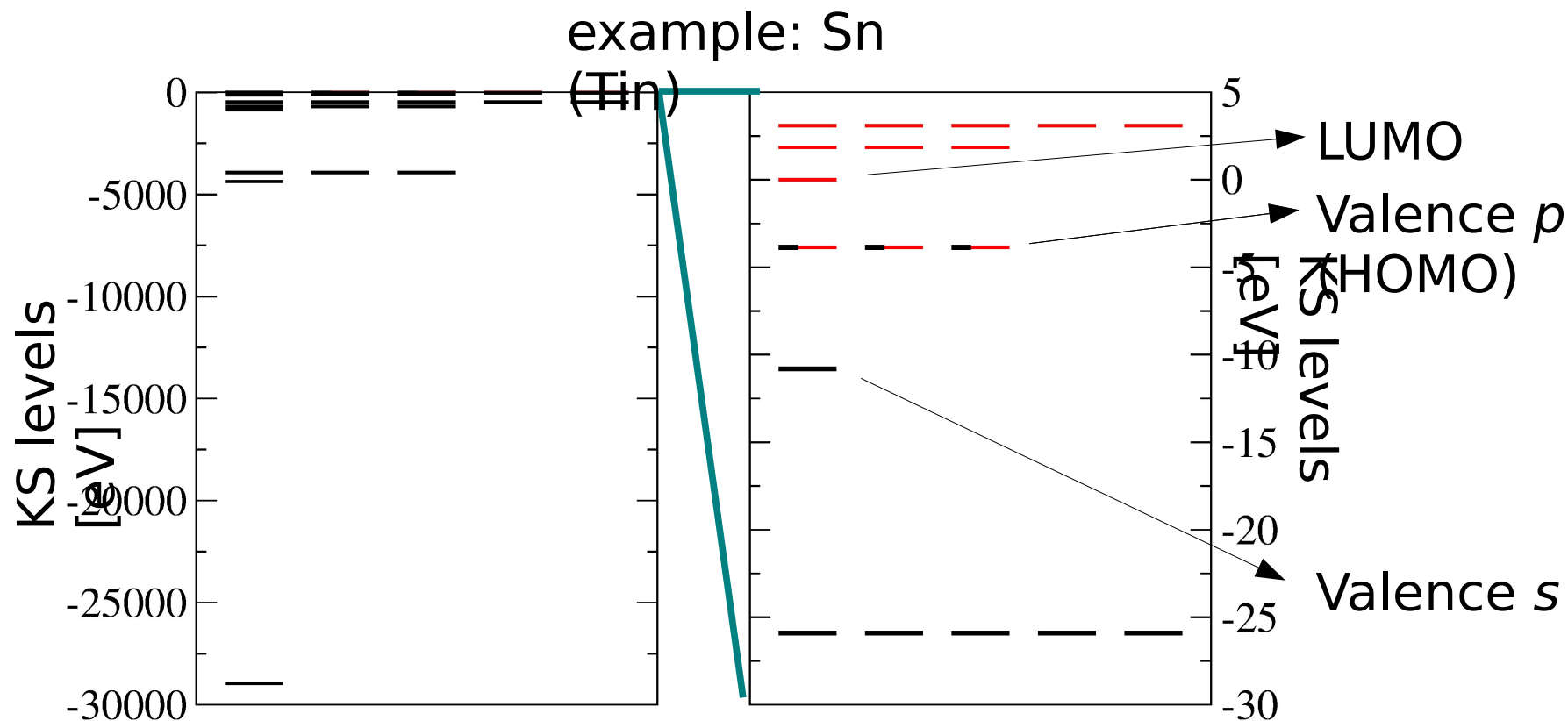
Primary (atomic) features



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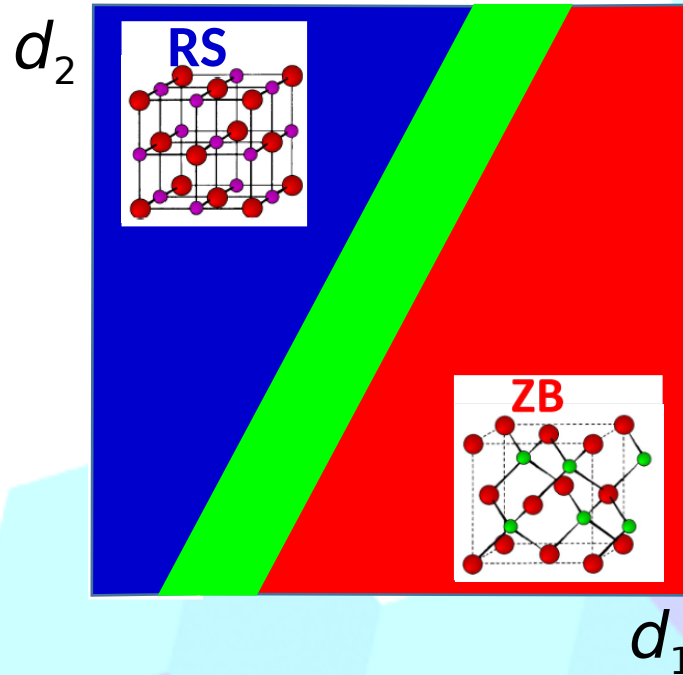


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- Radius of valence d orbital
- ... ?



- Rock salt
- Rock salt/Zinc blende
- ◆ Zinc blende

$E(\text{Rock salt}) - E(\text{Zinc blende})$

- J. A. van Vechten, *Phys. Rev.* 182, 891 (1969).
J. C. Phillips, *Rev. Mod. Phys.* 42, 317 (1970).
J. John and A.N. Bloch, *Phys. Rev. Lett.* 33, 1095 (1974)
J. R. Chelikowsky and J. C. Phillips, *Phys. Rev. B* 33, 2453 (1978)
A. Zunger, *Phys. Rev. B* 22, 5839 (1980).
D. G. Pettifor, *Solid State Commun.* 51, 31 (1984).
Y. Saad, D. Gao, T. Ngo, S. Bobbitt, J. R. Chelikowsky, and W. Andreoni, *Phys. Rev. B* 85, 104104 (2012).

Compressed sensing

Aim: finding descriptors and learning predictive models

Ansatz:

$$\mathbf{P} = c_1 \mathbf{d}_1 + c_2 \mathbf{d}_2 + \dots + c_n \mathbf{d}_n$$

Where

\mathbf{P} is the property of interest

$\mathbf{d}_1, \dots, \mathbf{d}_n$ are candidate features, i.e., nonlinear functions of primary features (EA, IP, ...)

c_1, \dots, c_n are unknown coefficients, with the extra constraint that these (nonzero) coefficients should be as few as possible.

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With a foreword on
dimensionality reduction

Linear dimensionality reduction: Principal components

Pearson, K. "On Lines and Planes of Closest Fit to Systems of Points in Space".
Philosophical Magazine 2, 559 (1901)

Linear dimensionality reduction: Principal components

Pearson, K. "On Lines and Planes of Closest Fit to Systems of Points in Space".
Philosophical Magazine 2, 559 (1901)

Orthonormal transformation of coordinates, converting a set of (possibly) linearly correlated coordinates into a new set of linearly uncorrelated (called principal or normal) components, such that the first component has the largest variance and each subsequent has the largest variance constrained to being orthogonal to all the preceding components

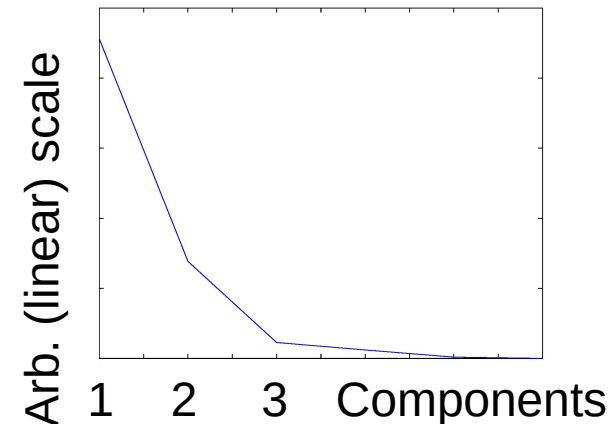
Linear dimensionality reduction: Principal components

Ansatz: atomic features

- Valence number Z_v
- Energy of valence s orbital E_s
- Energy of valence p orbital E_p
- Radius of valence s orbital r_s
- Radius of valence p orbital r_p

$r_s, r_p, E_s/\sqrt{Z_v}, E_p/\sqrt{Z_v}$,
for A and B atoms

linearly uncorrelated (called principal or normal) components, such that the first component has the largest variance and each subsequent has the largest variance constrained to being orthogonal to all the preceding components



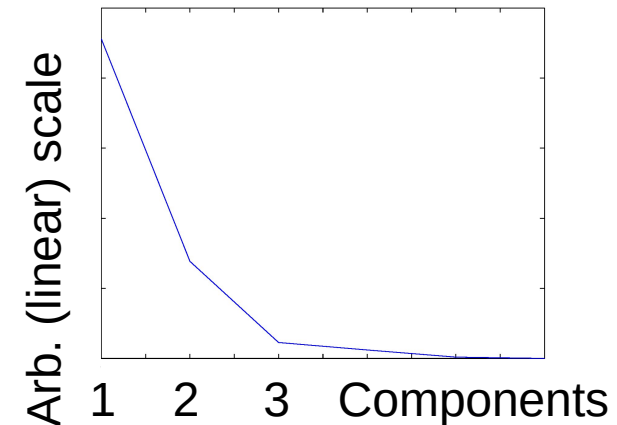
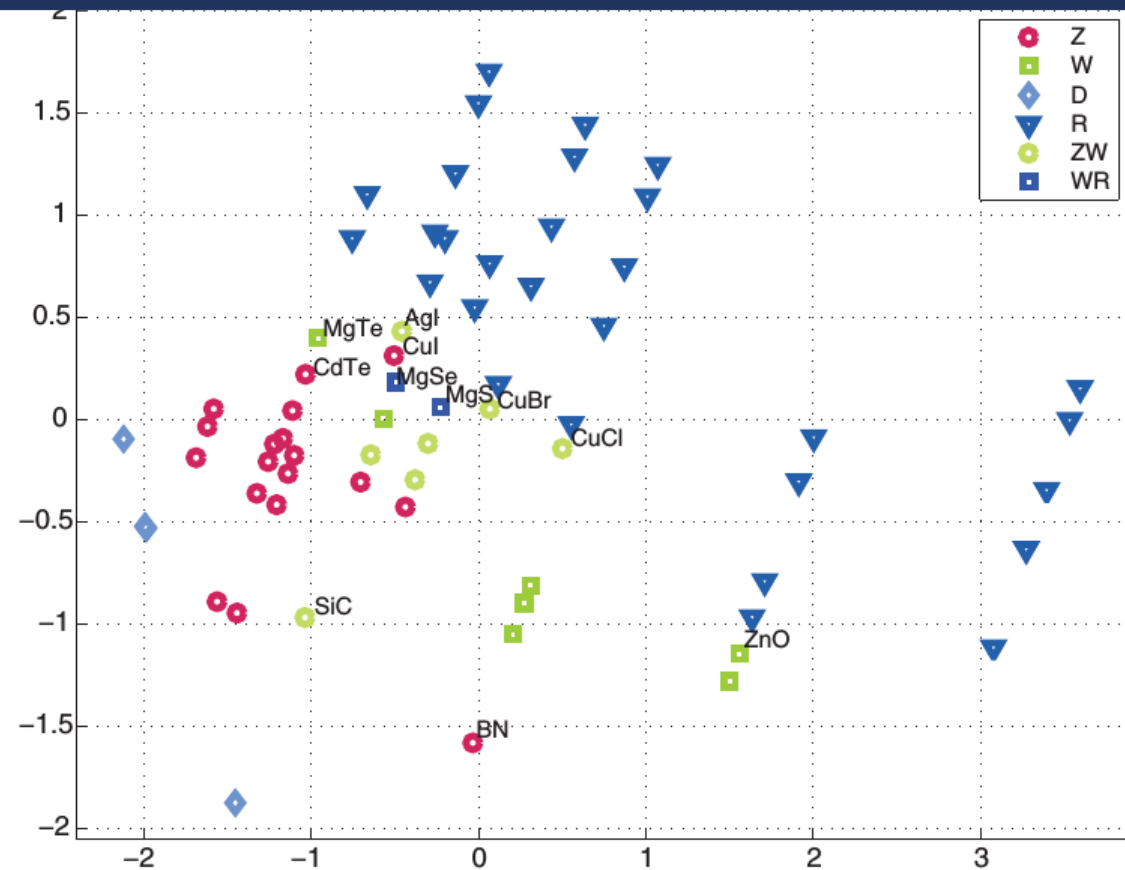
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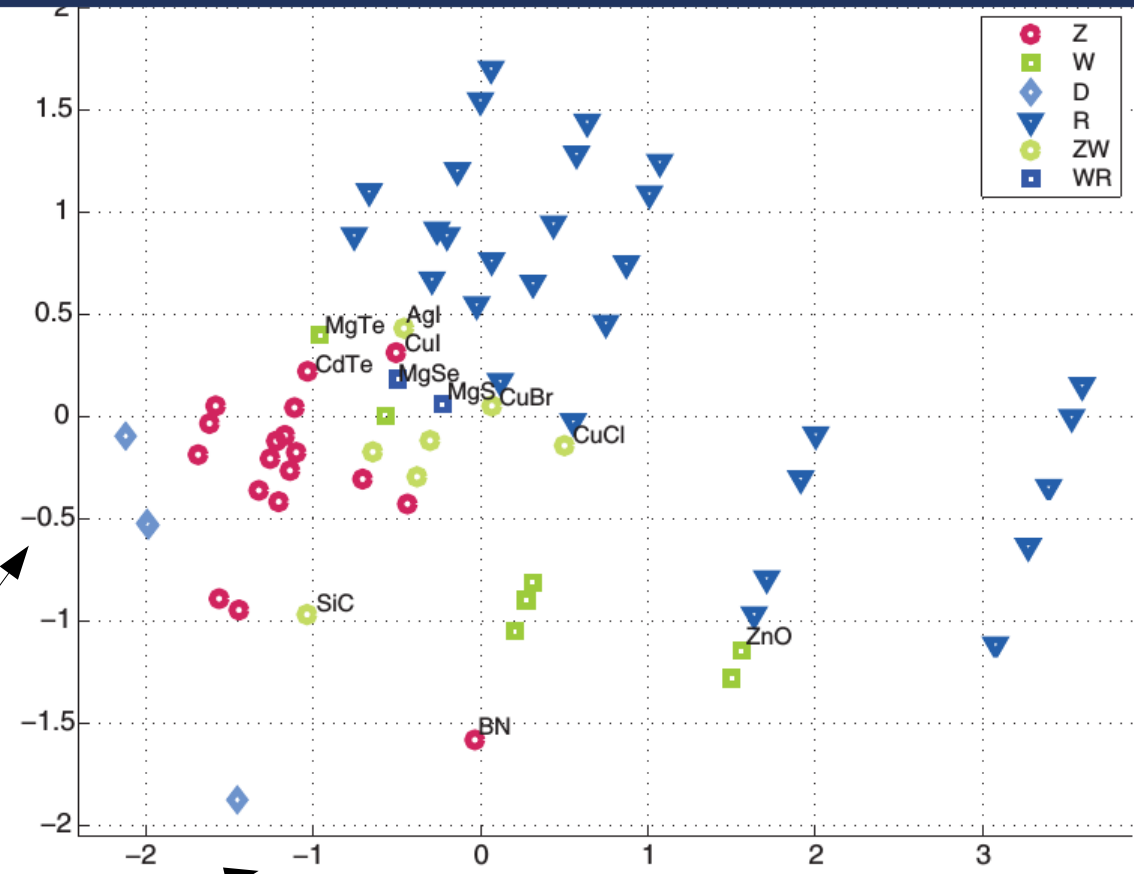
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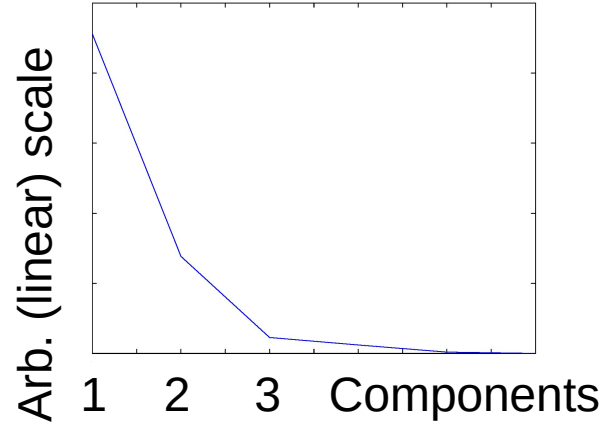
Z_v
 E_s
 E_p
 r_s
 r_p

$r_s, r_p, E_s/\sqrt{Z_v}, E_p/\sqrt{Z_v}$
for A and B atoms



What's on the axes?

Linear combination of (possibly all) the initial dimensions

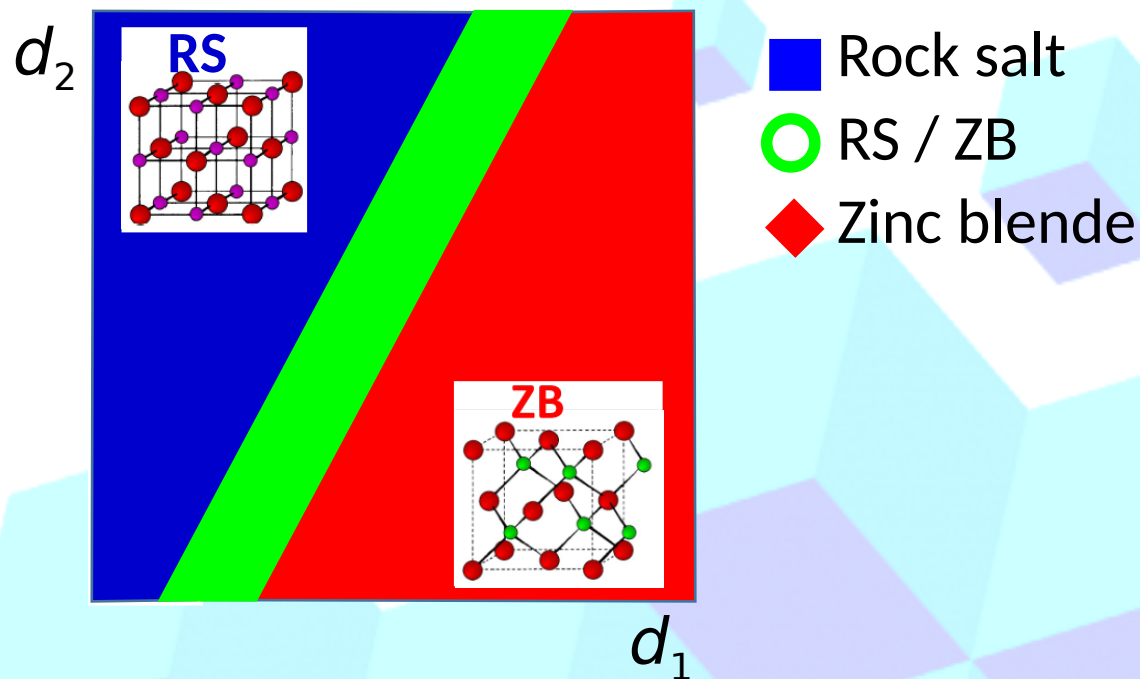


Compressed sensing: the quest for descriptors and predictive models

82 octet AB binary compounds

Ansatz: atomic features

- HOMO
- LUMO
- Ionization Potential
- Electron Affinity
- Radius of valence s orbital
- Radius of valence p orbital
- Radius of valence d orbital
- Thousands to billions of non-linear functions of the above



$$P = c_1 d_1 + c_2 d_2 + \dots c_n d_n$$

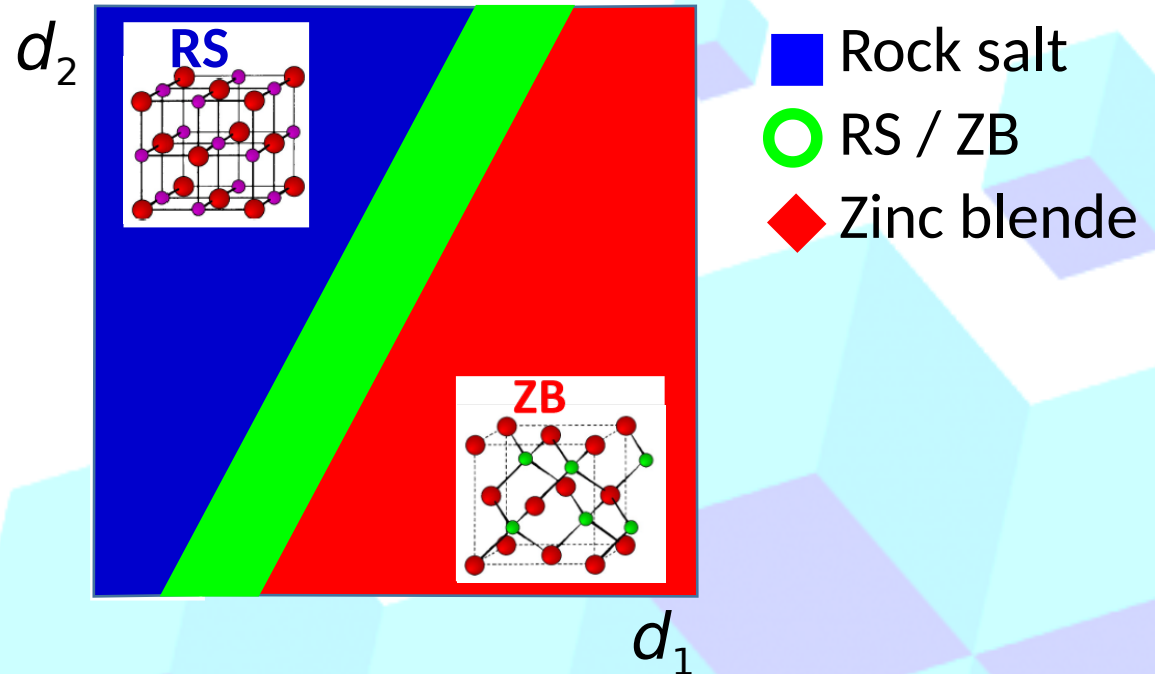
$E(\text{Rock salt}) - E(\text{Zinc blende})$

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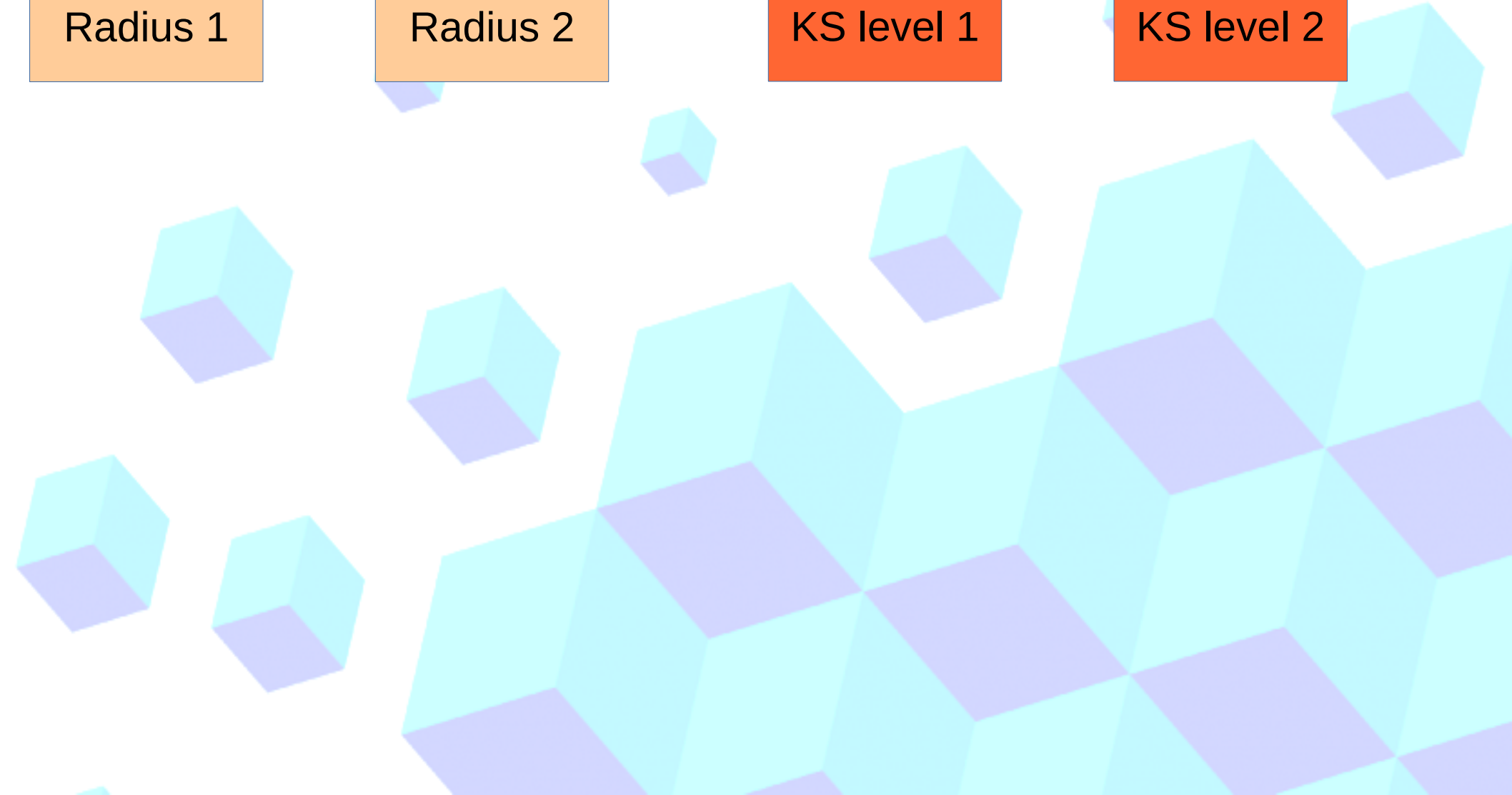
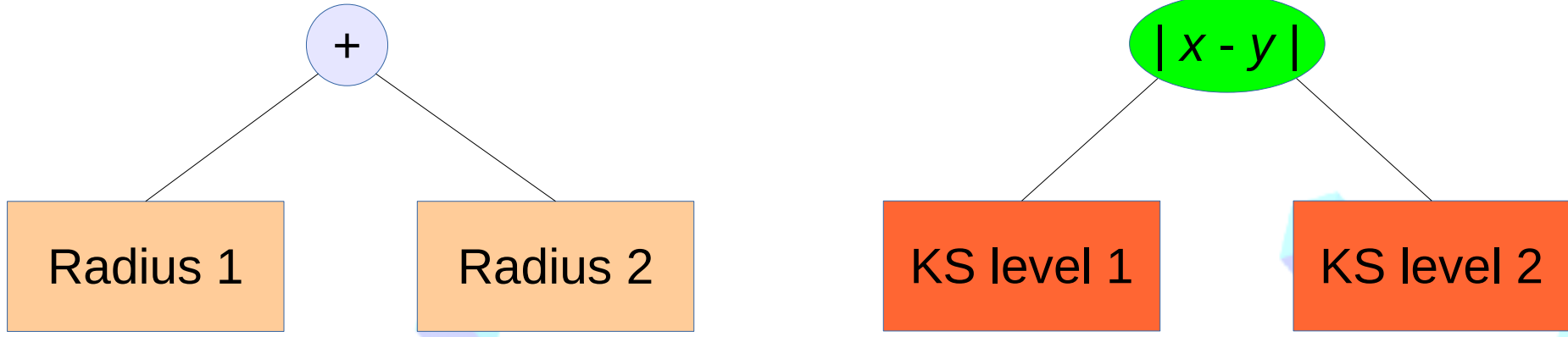


Symbolic Regression

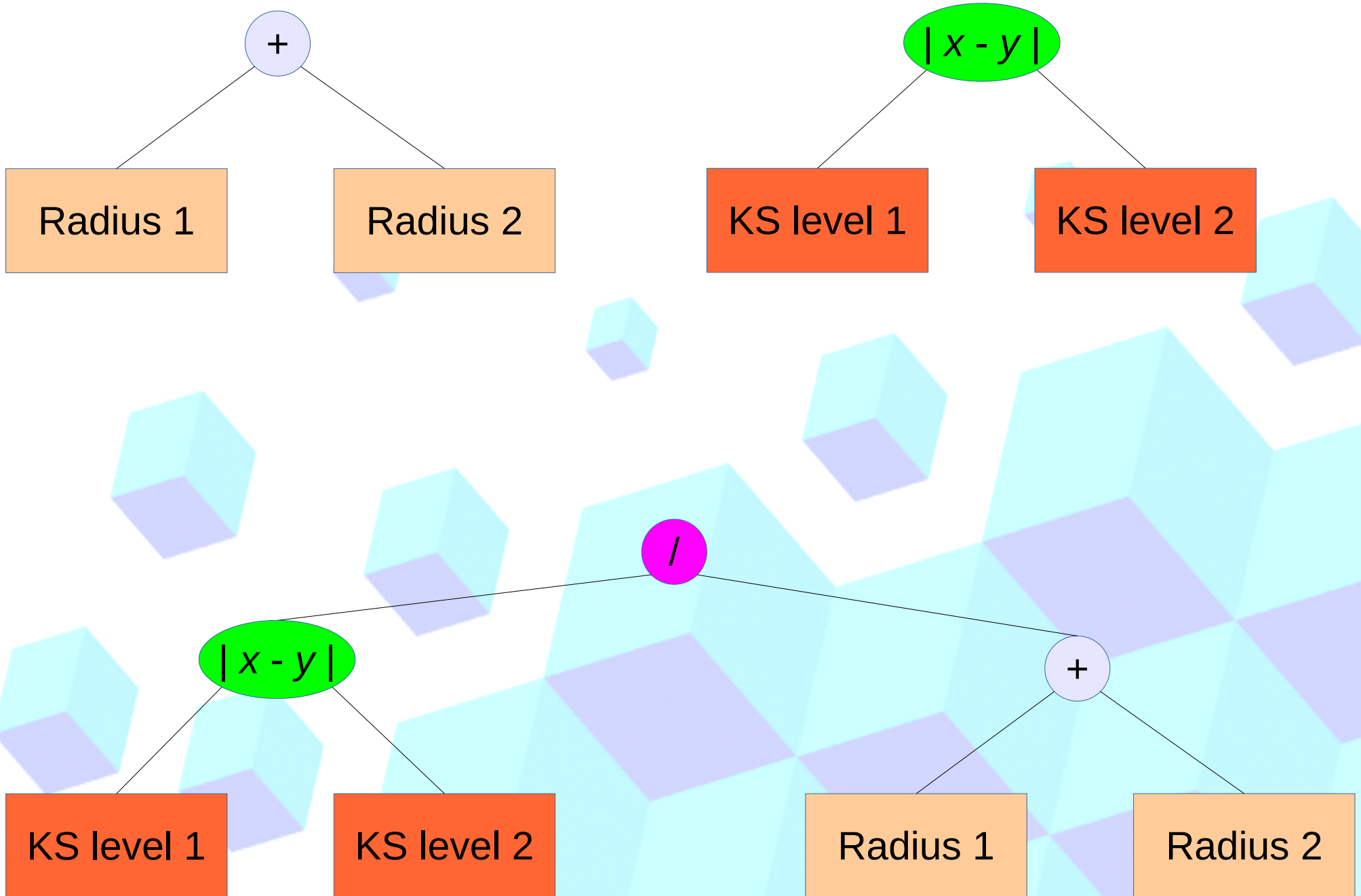
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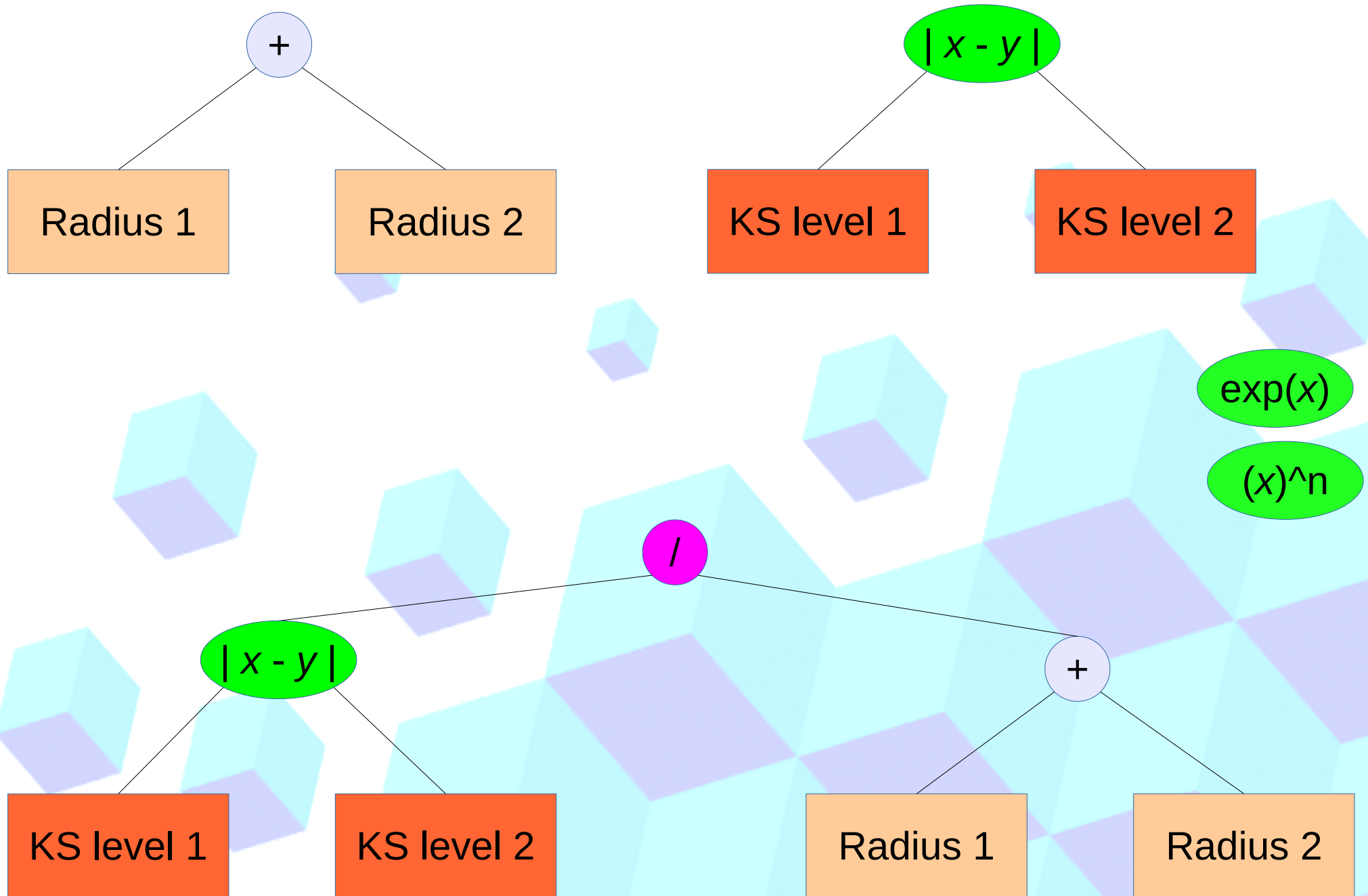
Systematic construction of the feature space



Systematic construction of the feature space



Systematic construction of the feature space



Systematic construction of the feature space: EUREQA

EUREQA: genetic programming software.
Global optimization (genetic algorithm).
Schmidt M., Lipson H., Science, Vol. 324, No. 5923, (2009)

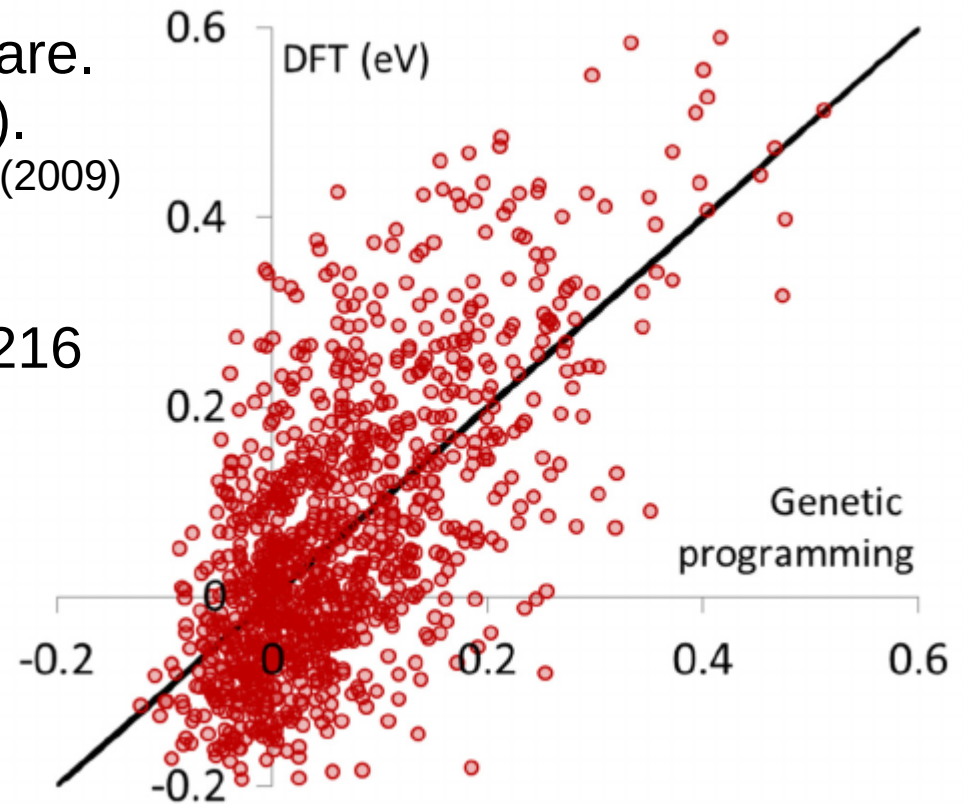
T. Müller et al. PRB **89** 115202 (2014):
Data: ~1000 amorphous structures of 216
Si atoms (saturated)

Property: hole trap depth

$$\frac{\min(1.66355, a) \max(5.37551, c) - f - bd}{g} - h \max(3.42929, e),$$

Descriptor (candidates: 242)

- a The largest distance between a H atom and its nearest Si neighbor
- b The shortest distance between a Si atom and its sixth-nearest Si neighbor
- c The maximum bond valence sum on a Si atom
- d The smallest value for the fifth-smallest relative bond length around a Si atom
- e The fourth-shortest distance between a Si atom and its eighth-nearest neighbor
- f The second-shortest distance between a Si atom and its fifth-nearest neighbor
- g The third-shortest distance between a Si atom and its sixth-nearest neighbor
- h The H-Si nearest-neighbor distance for the hydrogen atom with the fourth-smallest difference between the distances to the two Si atoms nearest to a H atom



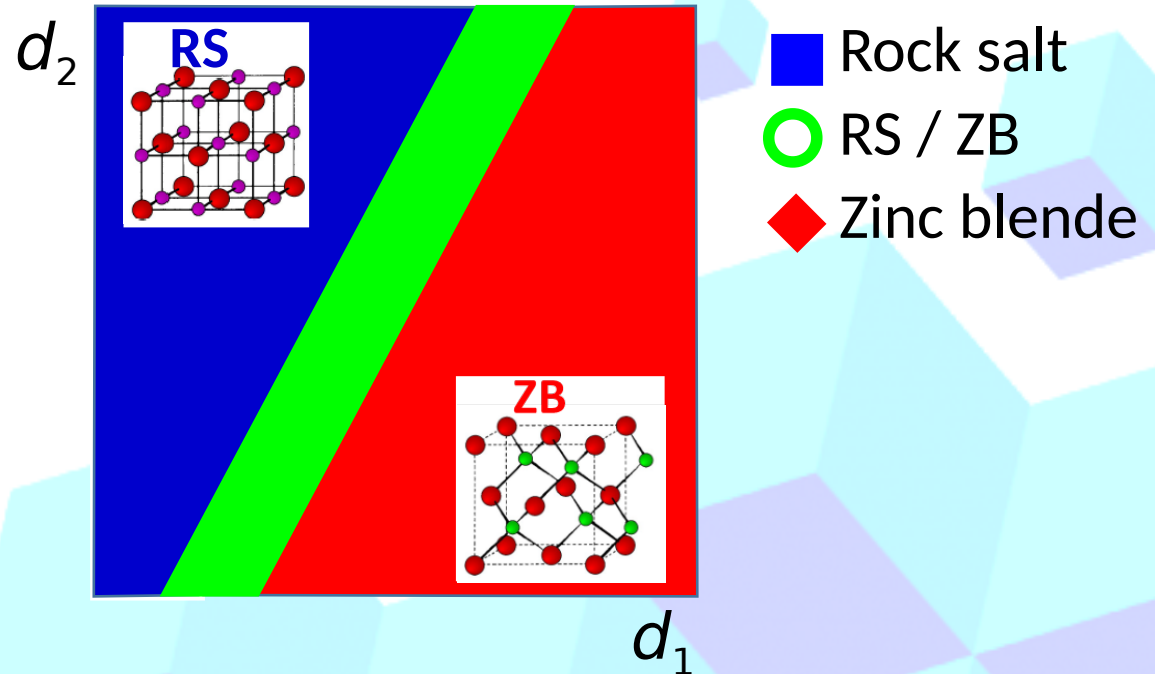
Building block	
Constant value	Exponential
Input variable	Natural logarithm
Addition	Power
Subtraction	Square root
Multiplication	Logistic function
Division	Minimum
Negation	Maximum
	Absolute value

Compressed sensing: the quest for descriptors and predictive models

82 octet AB binary compounds

Ansatz: atomic features

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- Radius of valence d orbital
- Thousands of non-linear functions of the above



$$P = c_1 d_1 + c_2 d_2 + \dots + c_n d_n$$

$$\operatorname{argmin}_{c \in \mathbb{R}^M} \|P - Dc\|_2^2 + \lambda \|c\|_0$$

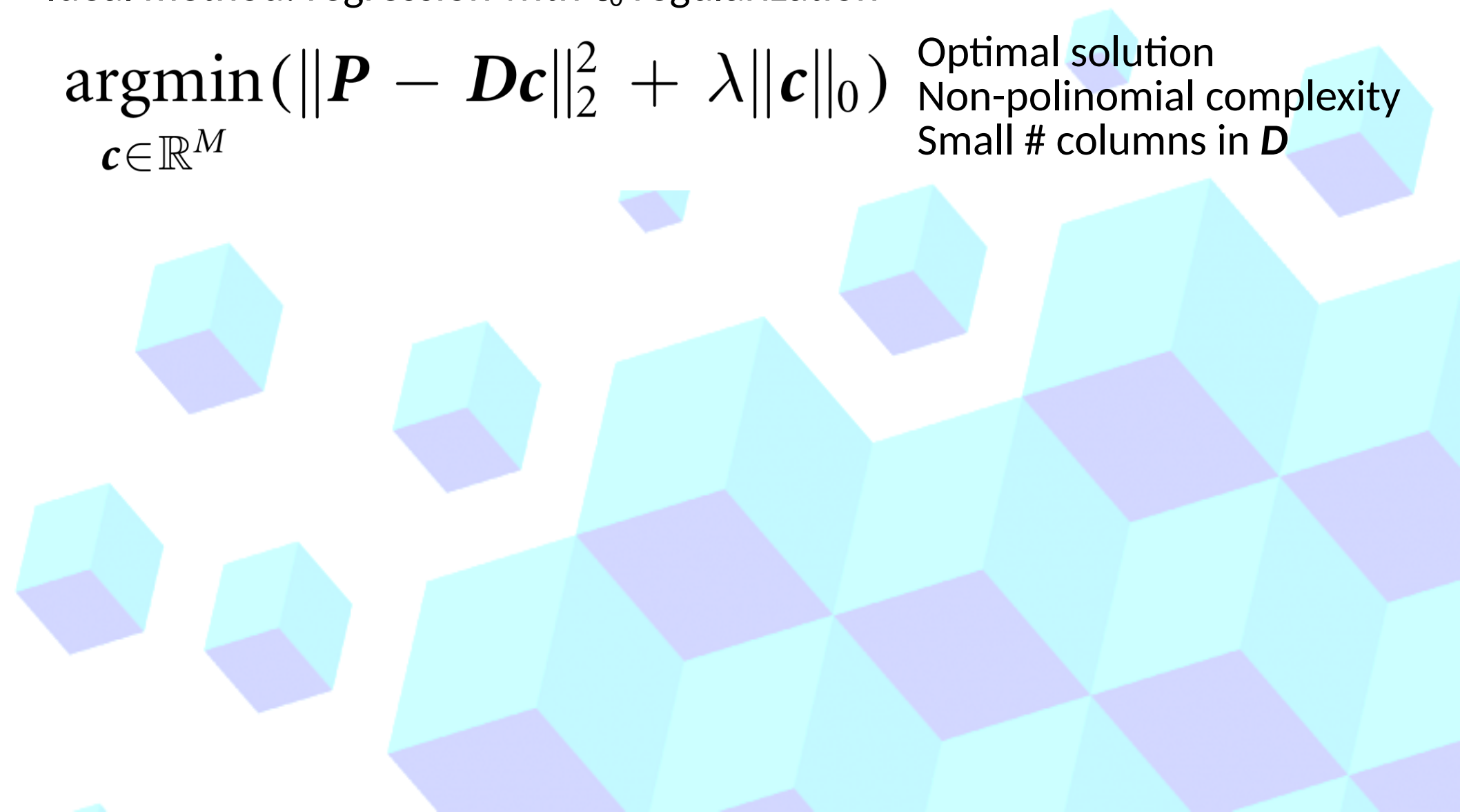
$E(\text{Rock salt}) - E(\text{Zinc blende})$

Compressed sensing: the quest for descriptors and predictive models

Ideal method: regression with ℓ_0 regularization

$$\operatorname{argmin}_{\mathbf{c} \in \mathbb{R}^M} (\|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_0)$$

Optimal solution
Non-polynomial complexity
Small # columns in \mathbf{D}



Compressed sensing: the quest for descriptors and predictive models

Ideal method: regression with ℓ_0 regularization

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Optimal solution
Non-polynomial complexity
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$\ \mathbf{c}\ _0$	# of nonzero elements of \mathbf{c}
$\ \mathbf{c}\ _2$	Euclidean. Square root of sum of squares of the elements of \mathbf{c}

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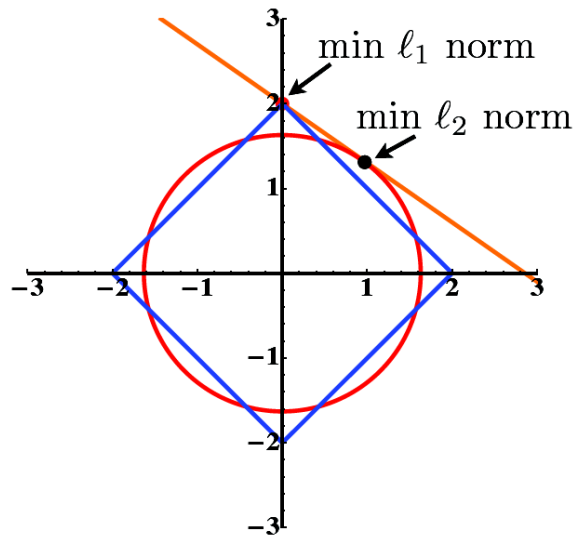
For matrices \mathbf{D} with uncorrelated columns: LASSO

$$\operatorname{argmin}_{\mathbf{c} \in \mathbb{R}^M} \|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_1$$

(Possibly) optimal solution
Convex optimization
Moderate # columns in \mathbf{D}

$\|\mathbf{c}\|_1$ “Manhattan”. Sum of absolute values of the elements of \mathbf{c}

Compressed sensing: the quest for descriptors and predictive models



$$\operatorname{argmin}_{\mathbf{c} \in \mathbb{R}^M} \|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda \|\mathbf{c}\|_1$$

(Possibly) optimal solution
Convex optimization
Moderate # columns in \mathbf{D}

$\|\mathbf{c}\|_1$ “Manhattan”. Sum of absolute values of the elements of \mathbf{c}

Lattice Anharmonicity and Thermal Conductivity from Compressive Sensing of First-Principles Calculations

Fei Zhou (周非)

Physical and Life Sciences Directorate, Lawrence Livermore National Laboratory, Livermore, California 94550, USA

Weston Nielson, Yi Xia, and Vidvuds Ozoliņš

Department of Materials Science and Engineering, University of California, Los Angeles, California 90095-1595, USA

(Received 22 April 2014; published 27 October 2014)



Compressed modes for variational problems in mathematics and physics

Vidvuds Ozoliņš^{a,*}, Rongjie Lai^{b,1}, Russel Caflisch^{c,1}, and Stanley Osher^{c,1,2}

Departments of ^aMaterials Science and Engineering, and ^cMathematics, University of California, Los Angeles, CA 90095-1555; and ^bDepartment of Mathematics, University of California, Irvine, CA 92697-3875

Contributed by Stanley Osher, October 8, 2013 (sent for review September 3, 2013)

PHYSICAL REVIEW B 87, 035125 (2013)

Compressive sensing as a paradigm for building physics models

Lance J. Nelson and Gus L. W. Hart

Department of Physics and Astronomy, Brigham Young University, Provo, Utah 84602, USA

Fei Zhou (周非) and Vidvuds Ozoliņš*

Department of Materials Science and Engineering, University of California, Los Angeles, California 90095, USA

(Received 26 June 2012; revised manuscript received 26 September 2012; published 18 January 2013)

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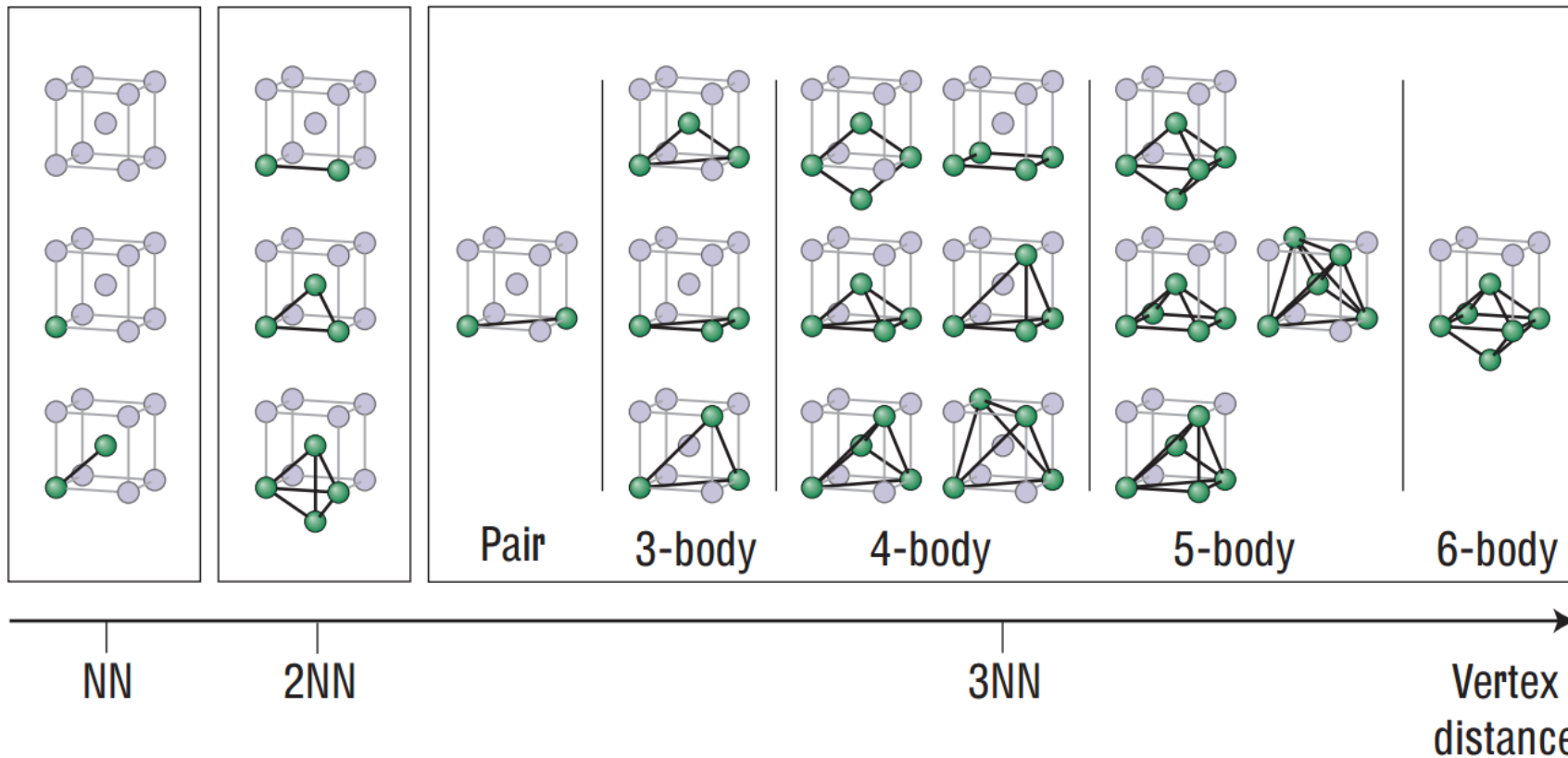
Department of Physics and Astronomy, Brigham Young University, Provo, Utah 84602, USA

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$$E(\sigma) = E_0 + \sum_f \bar{\Pi}_f(\sigma) J_f$$

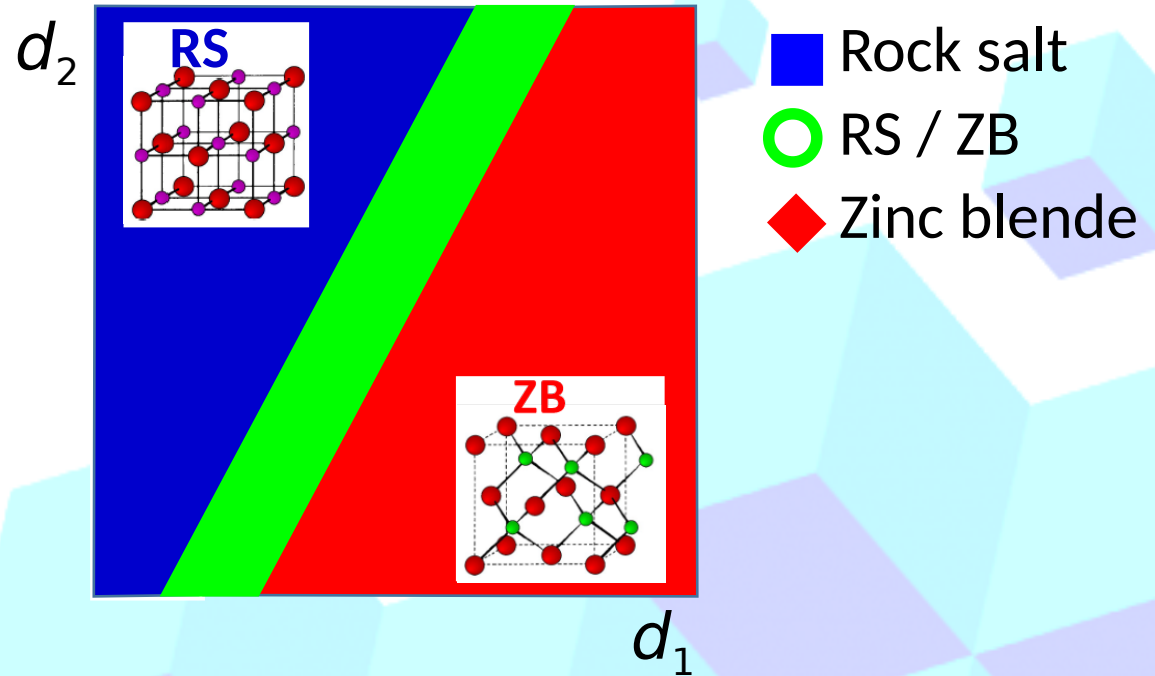


Compressed sensing: the quest for descriptors and predictive models

82 octet AB binary compounds

Ansatz: atomic features

- HOMO
- LUMO
- Ionization Potential
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- Billions of non-linear functions of the above



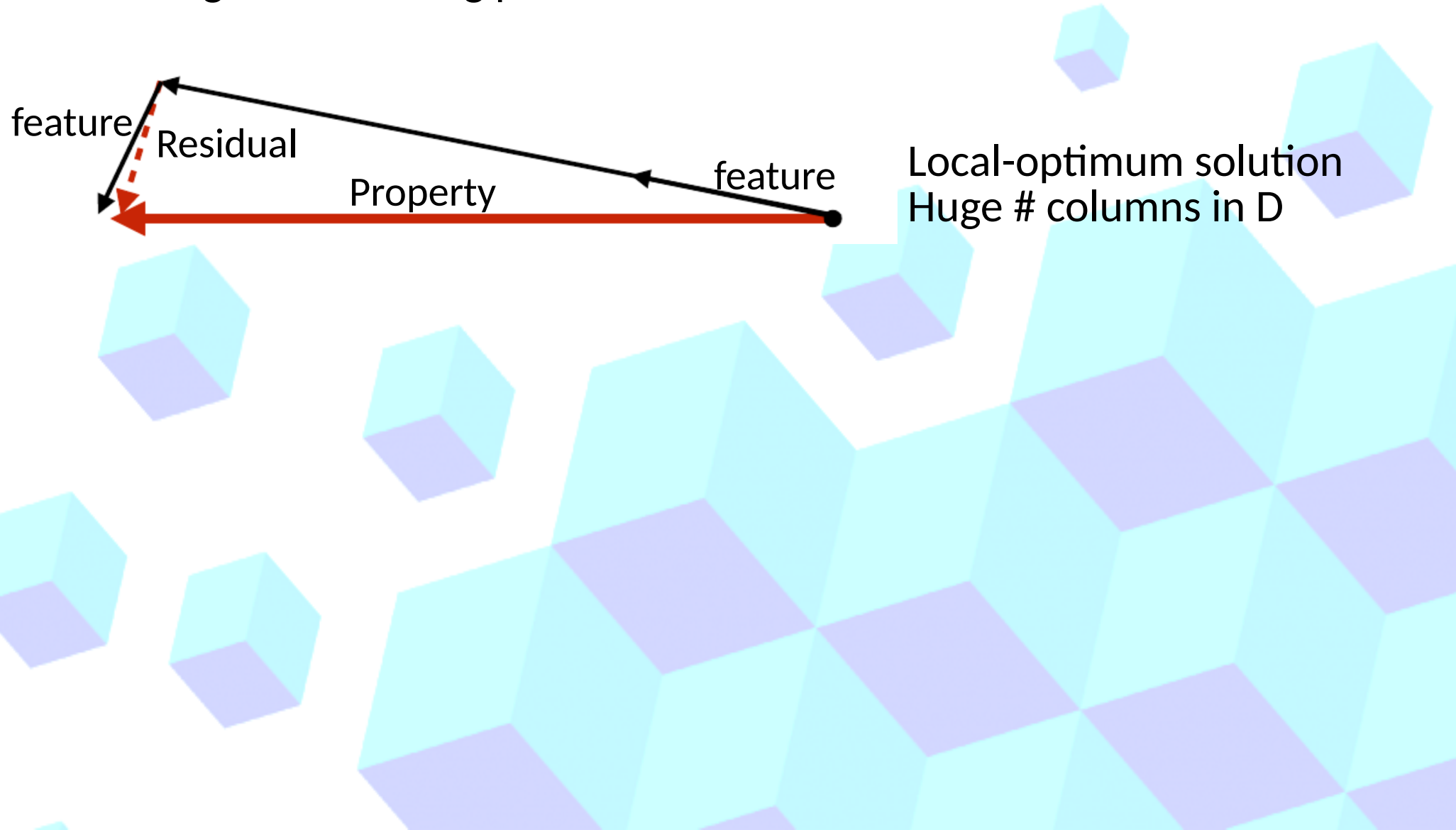
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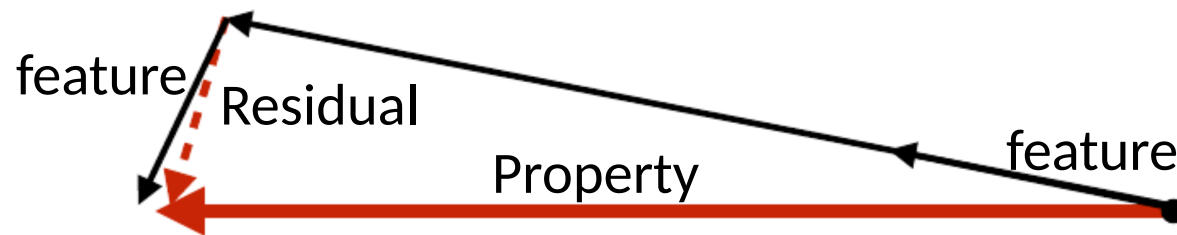
Compressed sensing: the quest for descriptors and predictive models

From orthogonal matching pursuit



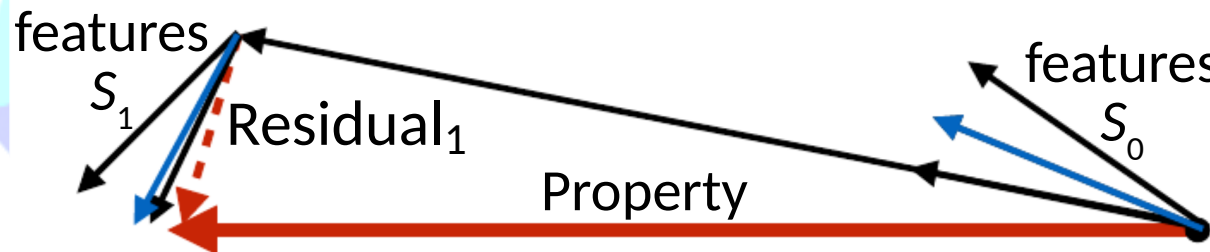
Compressed sensing: the quest for descriptors and predictive models

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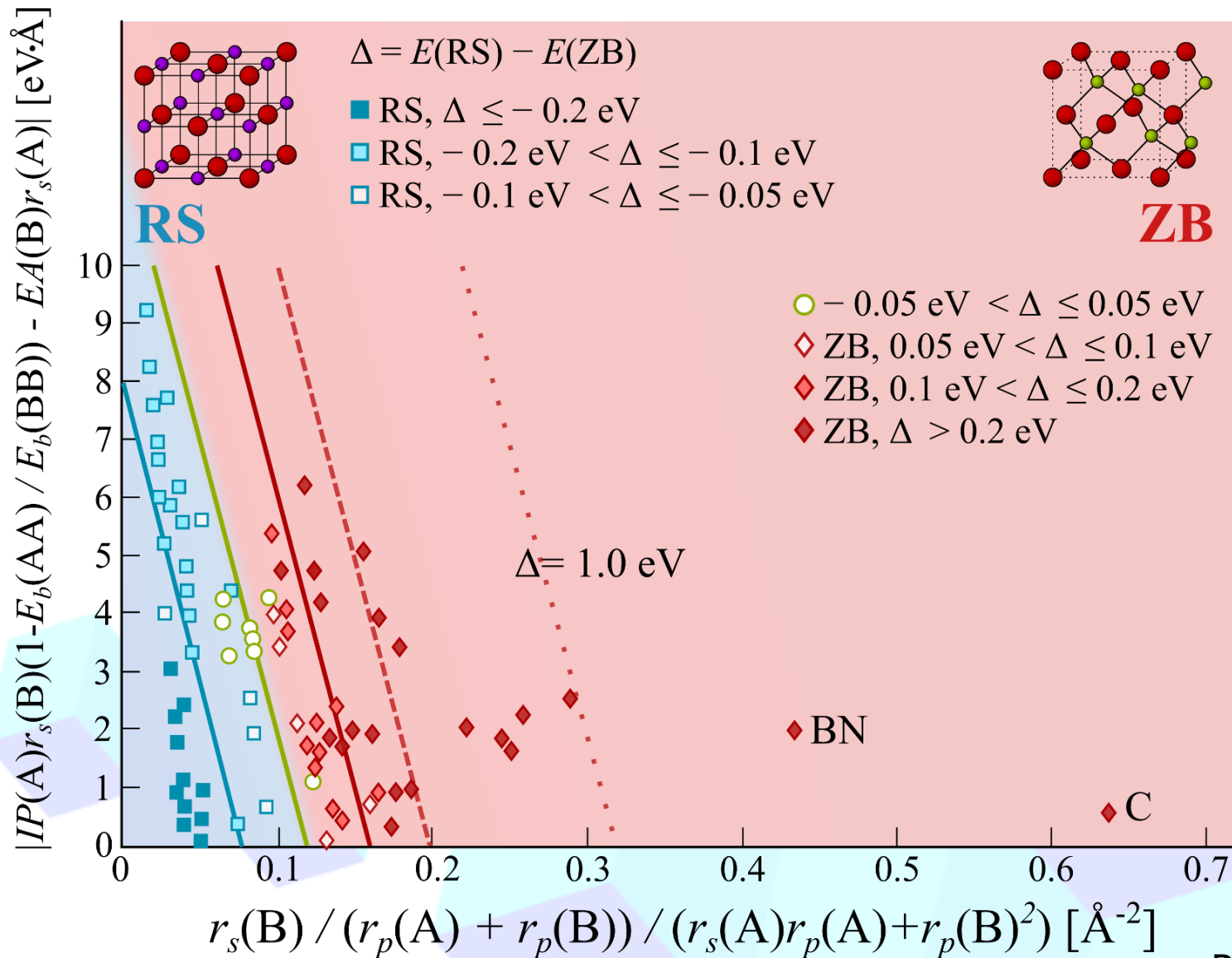
Local-optimum solution
Huge # columns in D

... to Sure Independence Screening + Sparsifying Operator (SISSO)



Proxy of
global-optimum solution
Huge # columns in D

Compressed sensing: the quest for descriptors and predictive models



Structure map with SISSO, starting from 7 atomic + 6 dimer features
 Feature space: 10^{11} features

Compressed sensing: the quest for descriptors and predictive models

$$\arg \min_{\mathbf{c}} (\|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2 + \lambda\|\mathbf{c}\|_0)$$

Compressed-sensing-based model identification:
Shares concepts with

Regularized regression. But: Massive sparsification.

Dimensionality reduction. But supervised, and yielding sparse, “inspectable” descriptors

Feature/Basis-set selection/extraction. But: non-greedy solver.

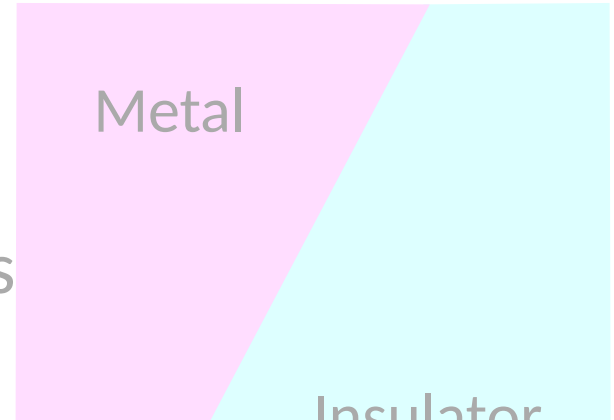
Symbolic regression. But: deterministic solver.

Charts/maps of materials

$$\operatorname{argmin}_{\mathbf{c} \in \mathbb{R}^M} (\underbrace{\|\mathbf{P} - \mathbf{D}\mathbf{c}\|_2^2}_{\text{overlap of convex domains}} + \lambda \|\mathbf{c}\|_0)$$

New cost function to be minimized:
overlap of *convex* domains

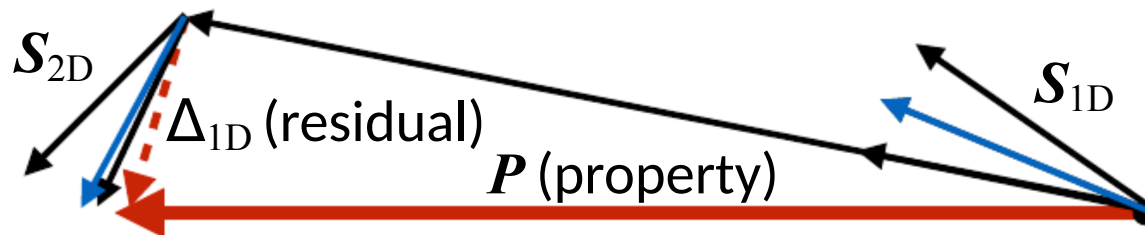
d'_2
 $A_x B_y$
binaries



1. # points in the *convex* overlap domain
2. Area of the domain overlap
3. Distance between domains

Good also for multi-categorical problems
(see A. F. Bialon *et al.*, Chem. Mater. **28**, 2550 (2016))

Iterative generation of feature subspaces



Insulator

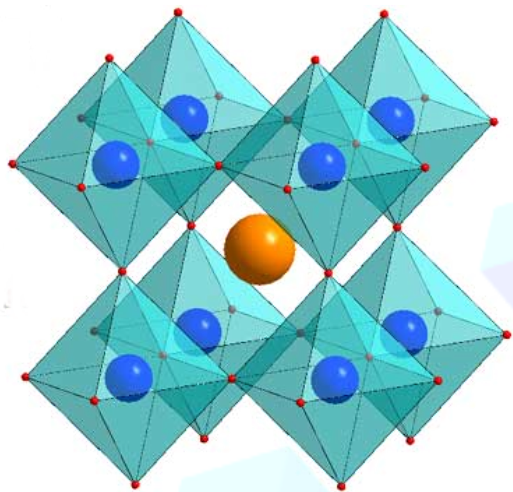
d'_1

Materials

Topological
Insulator

d''_1

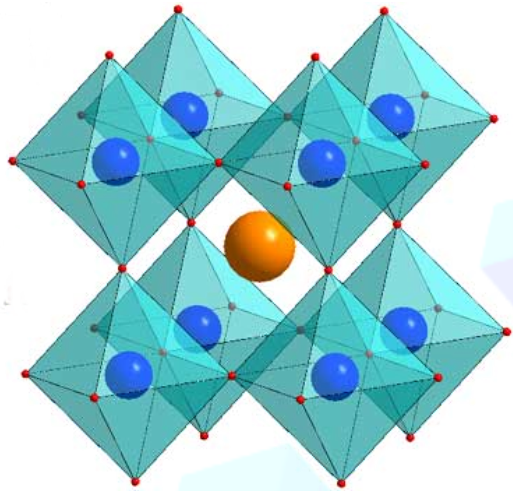
Perovskites' stability: an improved Goldschmidt Tolerance Factor



$$t = \frac{r_A + r_X}{\sqrt{2}(r_B + r_X)} \longrightarrow \text{Ionic radius}$$

Goldschmidt* stable perovskites: $0.825 < t < 1.059$, accuracy 79%

Perovskites' stability: an improved Goldschmidt Tolerance Factor



ABX_3

$$t = \frac{r_A + r_X}{\sqrt{2}(r_B + r_X)}$$

Ionic radius

$$\tau = \frac{r_X}{r_B} - n_A \left(n_A - \frac{r_A/r_B}{\ln(r_A/r_B)} \right)$$

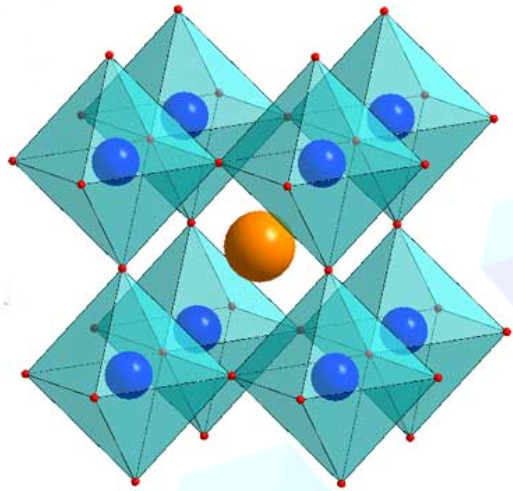
Oxidation state

$1/\mu =$ Octahedral factor

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Our stable perovskites: $\tau < 4.18$, accuracy 92%

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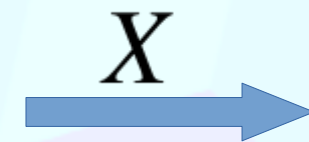
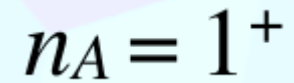
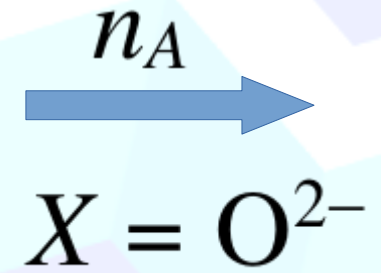
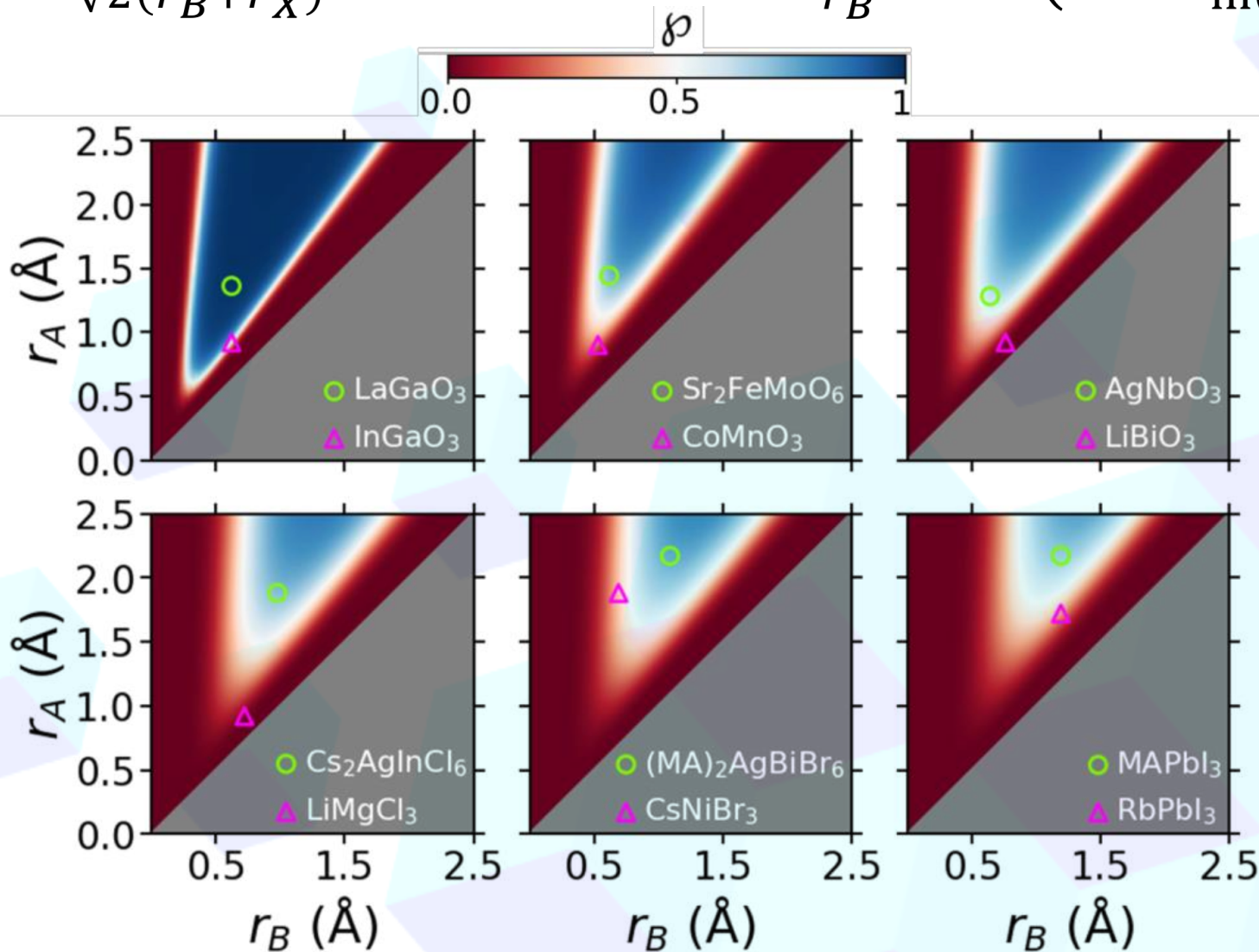
Our stable perovskites: $\tau < 4.18$, accuracy 92%

$\tau < 3.31$ or $\tau > 5.92$, 99% accuracy (1/3 of the training data)

$\tau < 3.31$ or $\tau > 12.08$, 100% accuracy (1/4 of the training data)

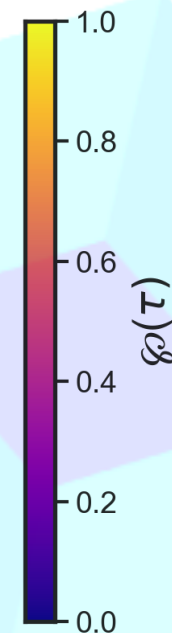
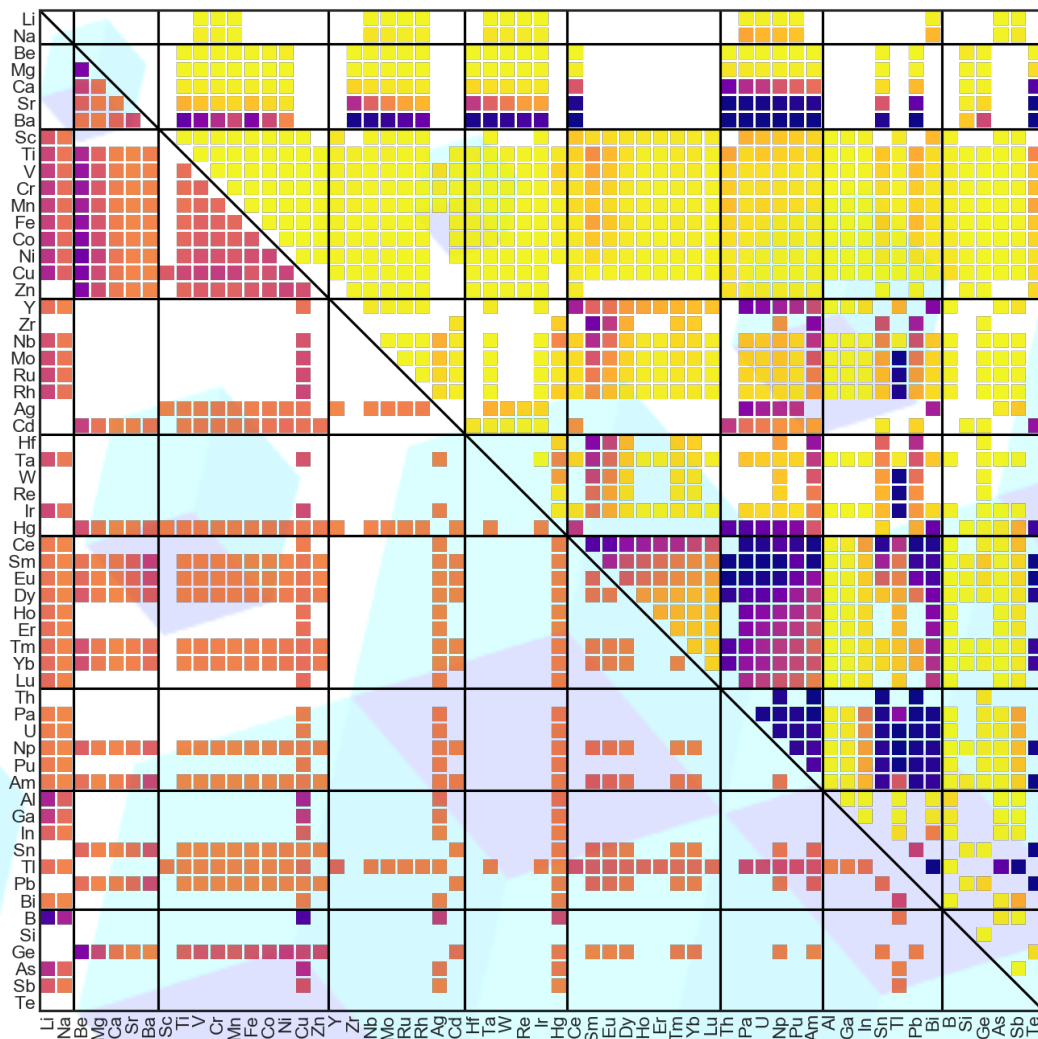
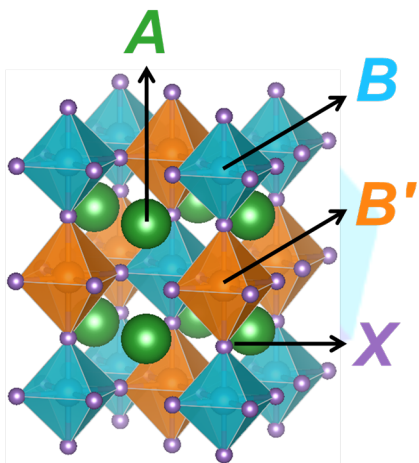
Improved Goldschmidt Tolerance Factor: Materials design

$$t = \frac{r_A + r_X}{\sqrt{2}(r_B + r_X)} \longrightarrow \tau = \frac{r_X}{r_B} - n_A \left(n_A - \frac{r_A/r_B}{\ln(r_A/r_B)} \right)$$



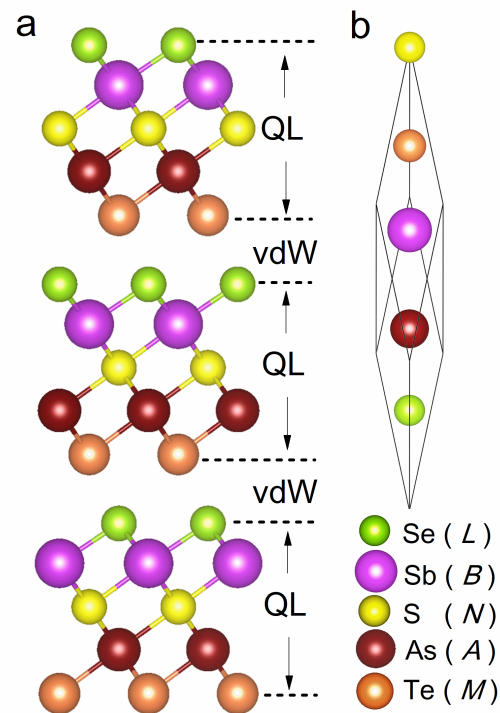
Improved Goldschmidt Tolerance Factor: Extension of the materials space

$$t = \frac{r_A + r_X}{\sqrt{2}(r_B + r_X)} \longrightarrow \tau = \frac{r_X}{r_B} - n_A \left(n_A - \frac{r_A/r_B}{\ln(r_A/r_B)} \right)$$



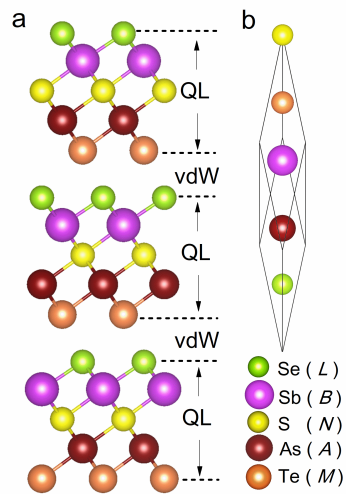
SISSO: predicting new tetradymite topological insulators

Prototype formula:
 $AB-LNM$
 $AB = \{As, Sb, Bi\}$
 $LNM = \{S, Se, Te\}$



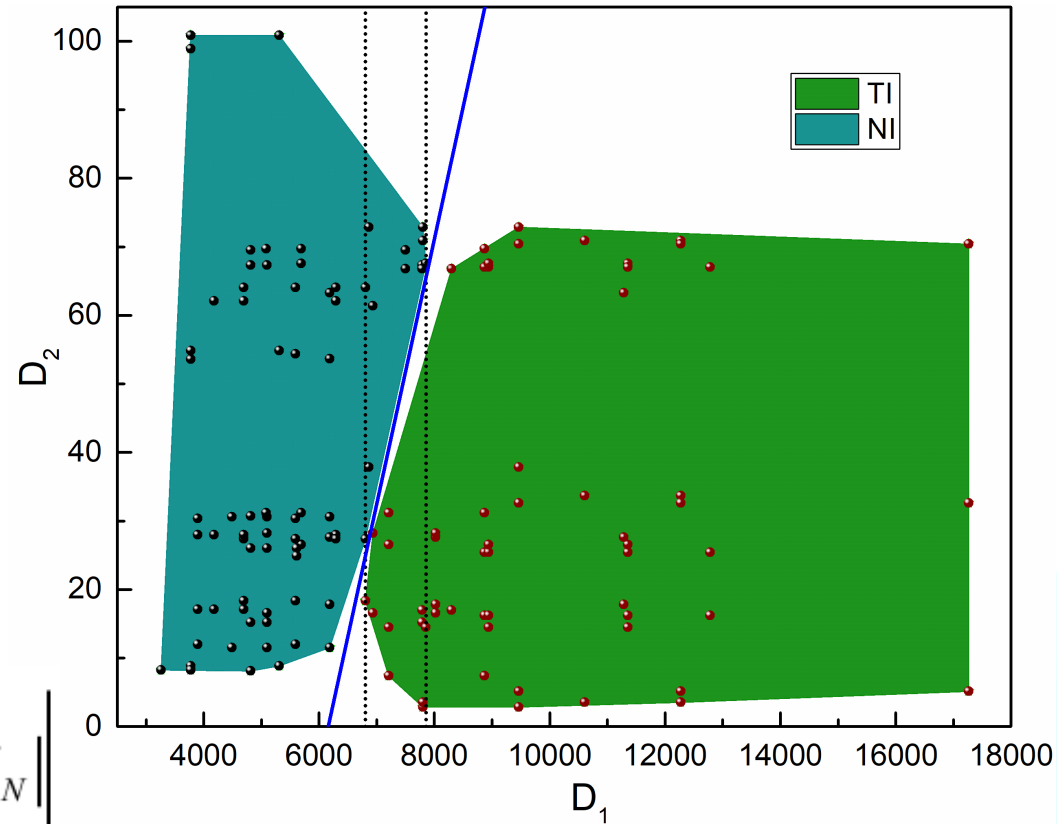
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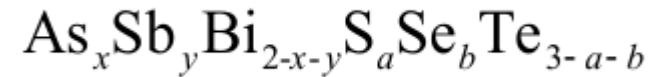
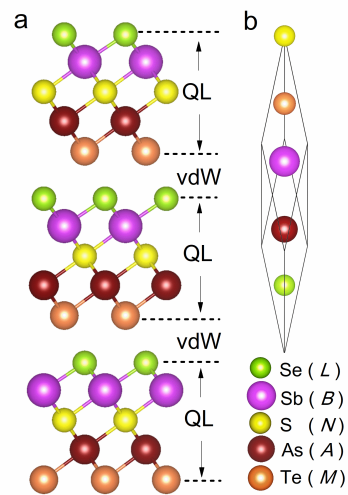
$$D_1 = (Z_A + Z_B) \cdot (Z_L + Z_M) - |Z_A Z_M - Z_B Z_L|$$

$$D_2 = \left| \frac{(\chi_M + \chi_N) \cdot Z_E}{\chi_A} - (Z_M + Z_N) - |Z_M - Z_N| \right|$$



SISSO: predicting new tetradymite topological insulators

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 $AB-LNM$
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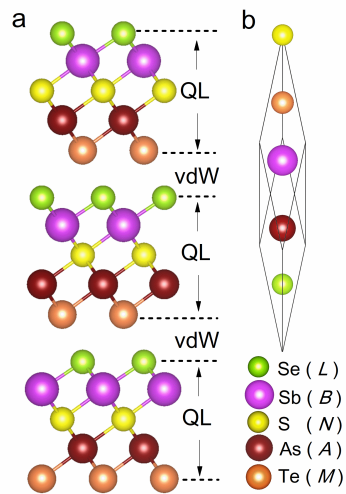


$$D_1 = (Z_A + Z_B) \cdot (Z_L + Z_M) - |Z_A Z_M - Z_B Z_L|$$

$$D_2 = \left| \frac{(\chi_M + \chi_N) \cdot Z_E}{\chi_A} - (Z_M + Z_N) - |Z_M - Z_N| \right|$$

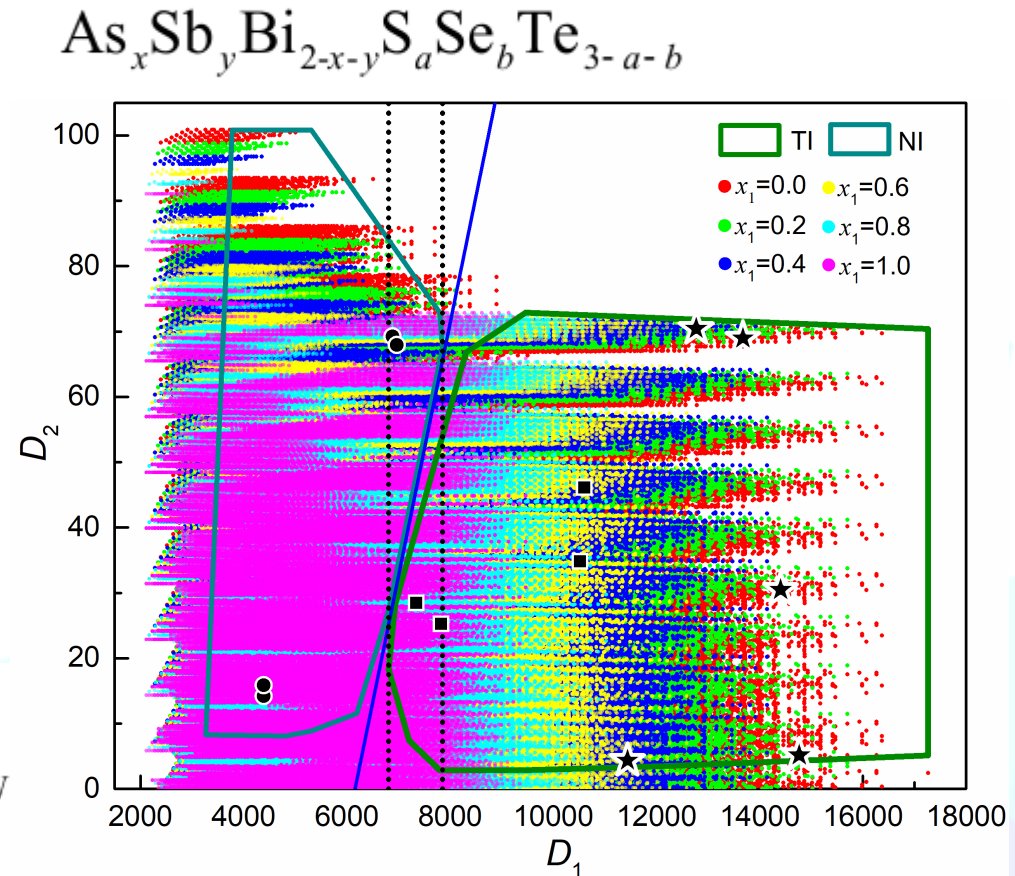
SISSO: predicting new tetradymite topological insulators

Prototype formula:
 $AB-LNM$
 $AB = \{As, Sb, Bi\}$
 $LNM = \{S, Se, Te\}$



$$D_1 = (Z_A + Z_B) \cdot (Z_L + Z_M) - |Z_A Z_M - Z_B Z_L|$$

$$D_2 = \left| \frac{(\chi_M + \chi_N) \cdot Z_E}{\chi_A} - (Z_M + Z_N) - |Z_M - Z_N| \right|$$



Acknowledgements

Compressed sensing and SISSO

Jan Vybiral, Runhai Ouyang, Emre Ahmetcik, Stefano Curtarolo, Christian Carbogno, Sergey Levchenko, Claudia Draxl

Application of SISSO to perovskites

Christopher J. Bartel, Christopher Sutton, Bryan R. Goldsmith, Runhai Ouyang, Charles B. Musgrave

Application of SISSO to topological insulators

Guohua Cao, Runhai Ouyang, Huijun Liu, Zizhen Zhou, Zhenyu Zhang, Christian Carbogno

And
Matthias Scheffler



NOVEL MATERIALS DISCOVERY



NOMAD has received funding from the European Union's Horizon 2020 research and innovation programme under grant agreement No. 676580.