

Kinetic Monte Carlo modelling of semiconductor growth

UNIVERSITÄT
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ESSEN

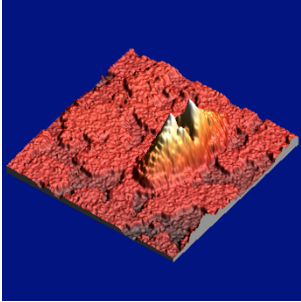


Peter Kratzer

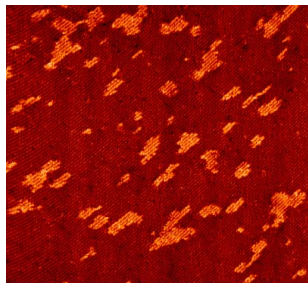
Faculty of Physics, University Duisburg-Essen, Germany

Time and length scales

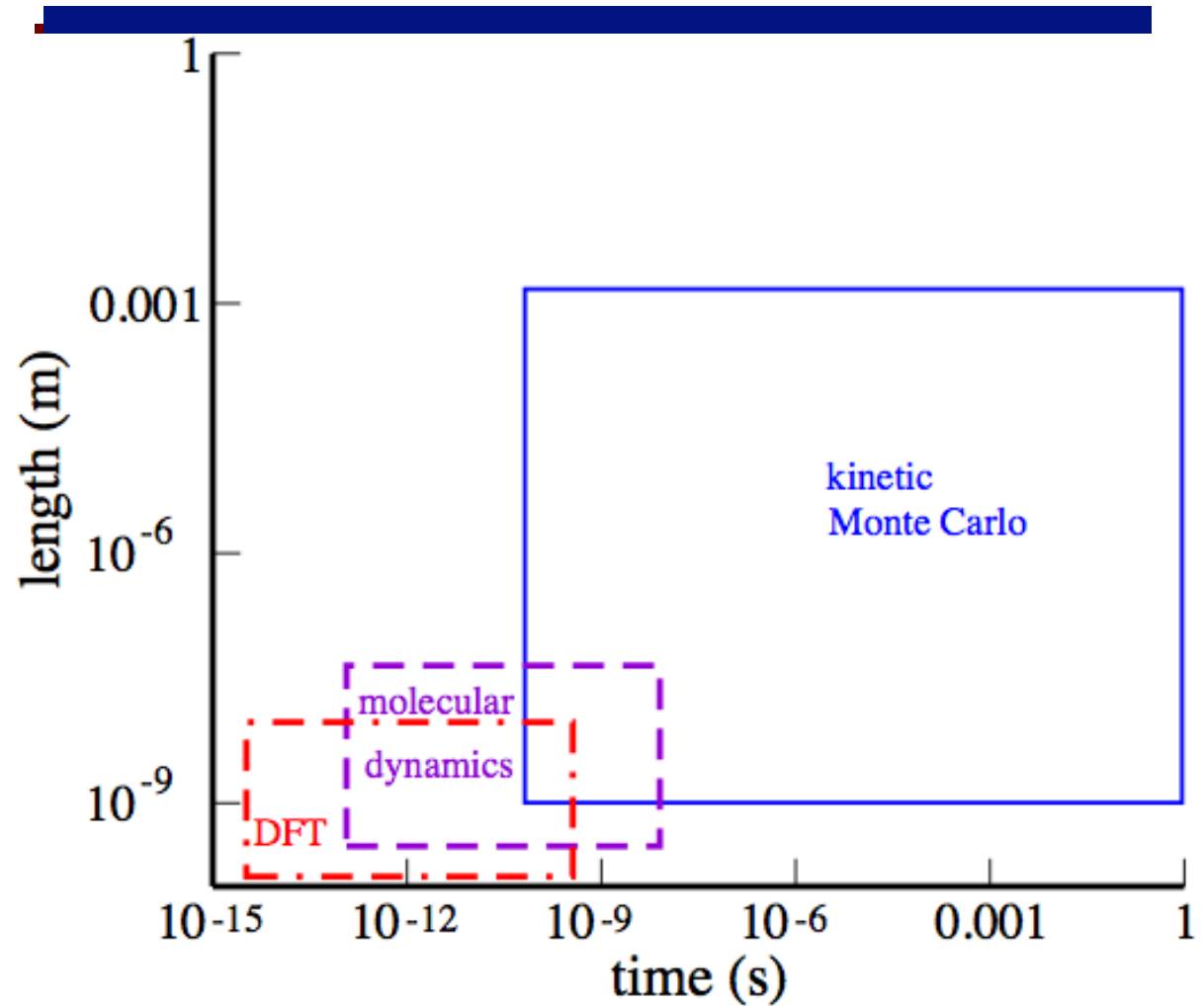
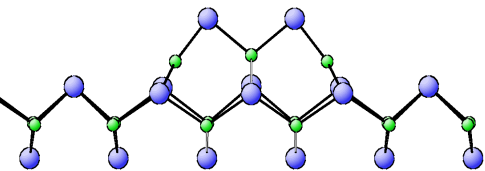
morphology



2D islands



surface reconstruction



Outline of this talk

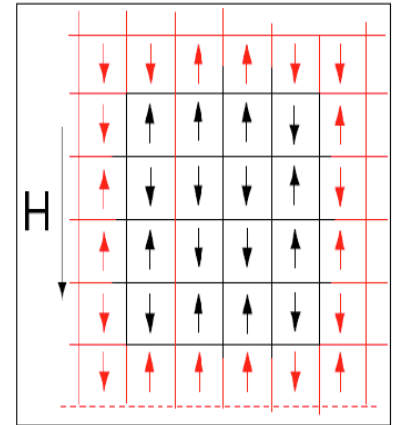
- Models in Statistical Physics, Monte Carlo (MC) methods, thermodynamic equilibrium
- How kinetic Monte Carlo exceeds over MC
- From molecular dynamics to kMC, how to make contact to DFT calculations
- Applications
- Summary

Methods of Statistical Physics

Discrete models in Statistical Physics

- Ising model (magnetism)

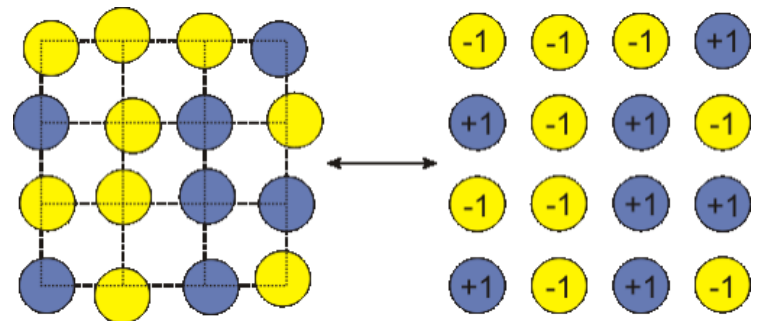
$$H(s) = -J_q \sum_i \sum_{j \in n(i)} s_i s_j - \mu_B B \sum_i s_i$$



- Lattice-gas interpretation $c_i = 0, 1$ $s_i = 2c_i - 1$

$$H = -4J_q \sum_i \sum_{j \in n(i)} c_i c_j + 2(qJ_q - \mu_B B) \sum_i c_i - N(qJ_q - \mu_B B)$$

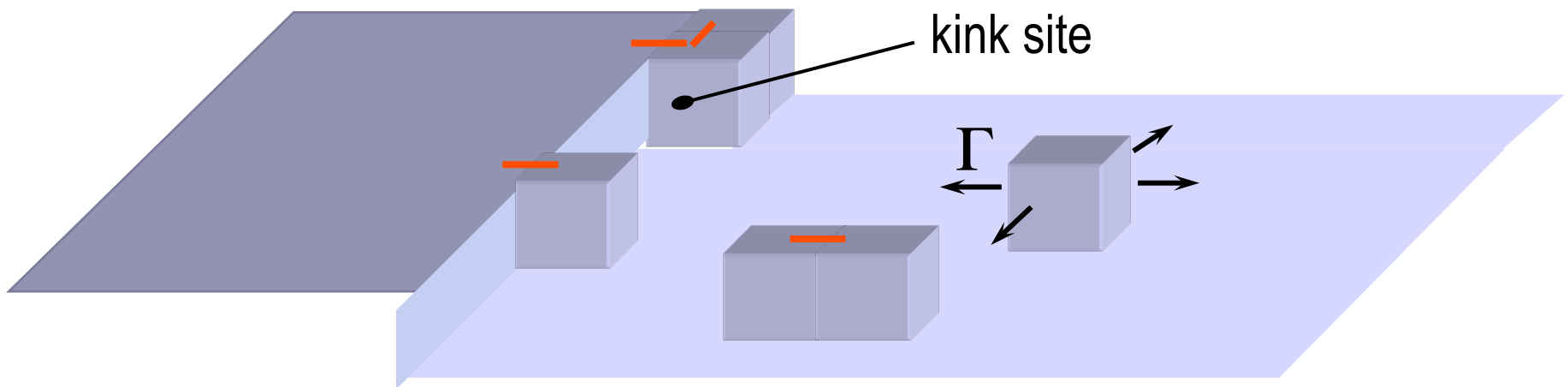
- Goal:**
Calculation of thermal averages



A discrete model for epitaxy: solid-on-solid (SOS) model

- Atoms are symbolized by little cubes placed on a lattice.
- The growth surface has no voids, no “overhangs”.
- Atoms move by discrete hops with rate $\Gamma = \exp(-E/kT)$.
- The binding energy is determined by the # of neighbors n

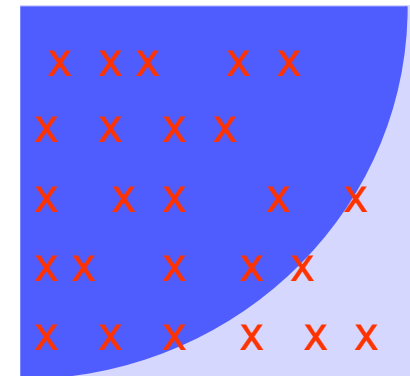
$$E = E_D + n E_B$$



Stochastic sampling

- Calculating thermal averages in many-particles systems requires evaluation of **high-dimensional integrals**.
- Choosing the sampling points in an (almost) **random** way is a good strategy, in particular in high dimensions !
- Even better: **importance sampling** -- density of sampling points proportional to local value of the integrand
- Idea: create a **stochastic process** that achieves importance sampling.

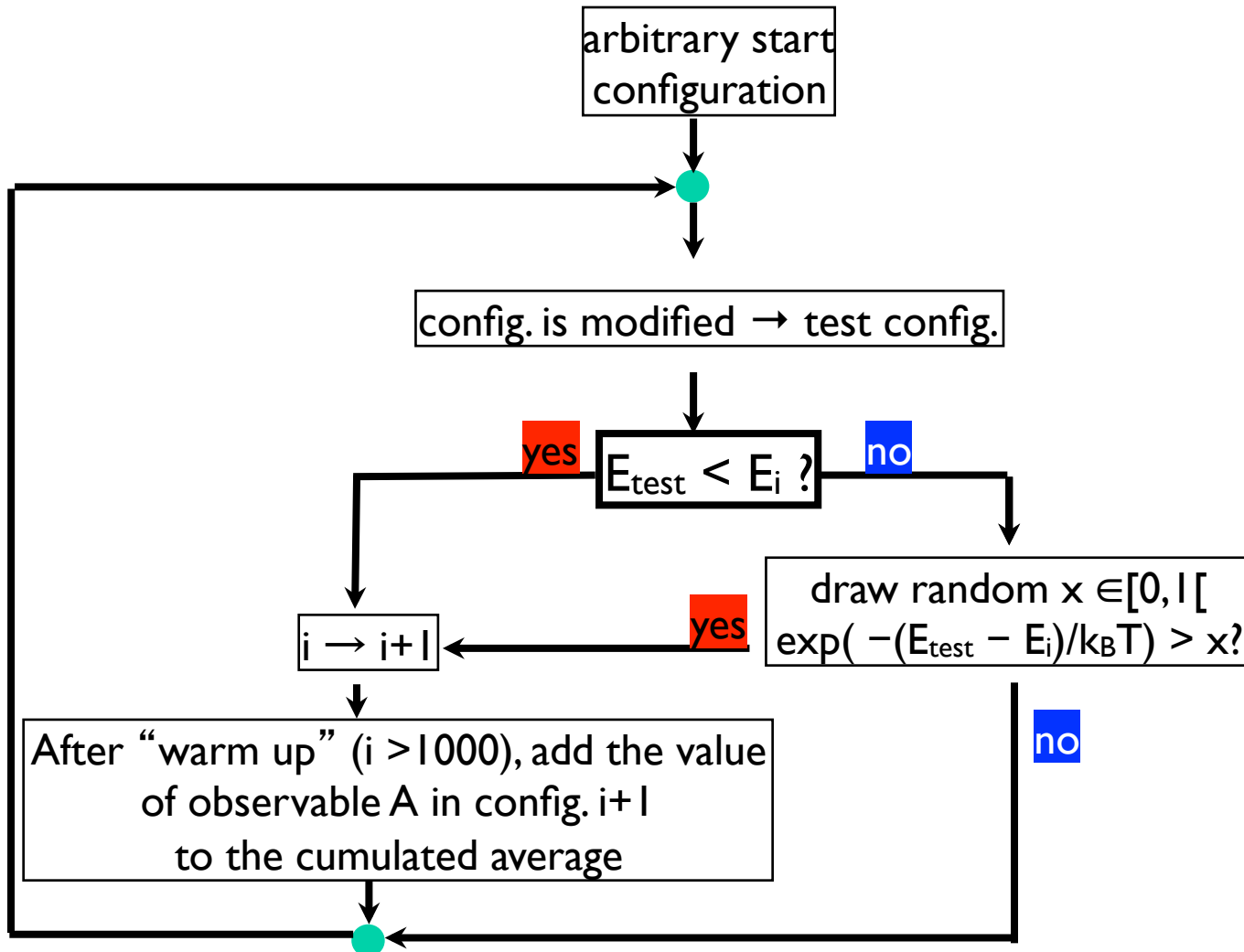
$$\pi/4 = 0.78 \dots \approx 20/25 = 0.8$$



Metropolis Sampling

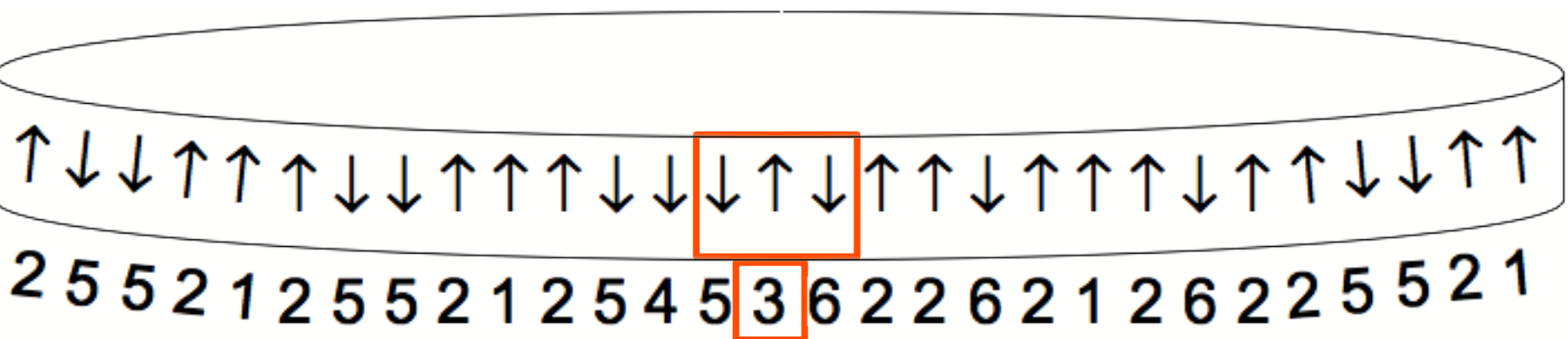
- **Solution:** *Importance Sampling* with $w(\mathbf{q}) = \frac{\exp(-V(\mathbf{q})/(k_B T))}{Z'}$
- Generate random support points, distributed according to $w(\mathbf{q})$, i.e., out of total K points, $k_i = K w(\mathbf{q}_i)$ in the unit volume around \mathbf{q}_i
- The expectation value of an observable is calculated as $\langle A \rangle \approx \frac{1}{K} \sum_{i=1}^K k_i A(\mathbf{q}_i)$
- The Metropolis algorithm generates, starting from \mathbf{q}_0 , successively a sequence of K configurations \mathbf{q}_i , distributed according to $w(\mathbf{q})$.
- Even though we don't know Z' , this is possible, because it is just the correct **relative** probabilities that matter:
 - accept new config. \mathbf{q}_{i+1} , if $\exp\left(-\frac{V(\mathbf{q}_{i+1}) - V(\mathbf{q}_i)}{k_B T}\right) > \text{rnd}$
 - else reject. $\text{rnd} \in [0,1[$
- This assures that $\frac{w(\mathbf{q}_{i+1})}{w(\mathbf{q}_i)} = \exp\left(-\frac{V(\mathbf{q}_{i+1}) - V(\mathbf{q}_i)}{k_B T}\right)$

Metropolis algorithm



From MC to kMC: the *N*-fold way

Classification of spins according to their neighborhood

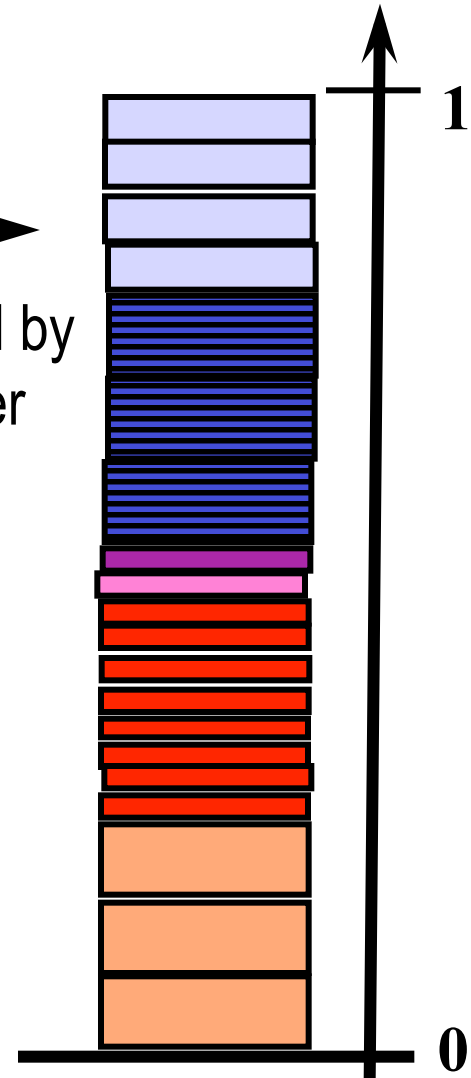


class	central spin	neighbors	class members n_i
1	↑	↑, ↑	4
2	↑	↑, ↓	12
3	↑	↓, ↓	1
4	↓	↓, ↓	1
5	↓	↑, ↓	8
6	↓	↑, ↑	3

The N -fold way algorithm in MC

- processes are chosen with a probability proportional to their rates
- no discarded attempts (in contrast to Metropolis)

→ pointer steered by random number



class	central spin	neighbors	class members n_i
1	↑	↑, ↑	4
2	↑	↑, ↓	12
3	↑	↓, ↓	1
4	↓	↓, ↓	1
5	↓	↑, ↓	8
6	↓	↑, ↑	3

Simulations of non-equilibrium processes: kinetic MC

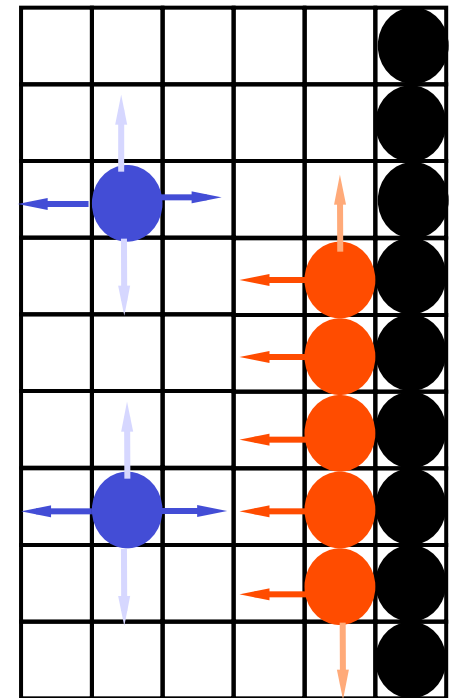
- While being aware of all processes possible at an instant of time, we need a way of (randomly) selecting one process with the **appropriate** relative probability.
- An internal **clock** keeps track of the advancement of **physical** time.
 - If the processes are clearly separated in time, i.e. processes are uncorrelated on the time scale *during which* the processes takes place, the waiting time for each individual process has Poissonian distribution.
(K. A. Fichthorn and W.H. Weinberg, J. Chem. Phys. **95**, 1090 (1991))
- We need to update the list of **all** possible processes according to the new situation after the move.

Specific algorithms:

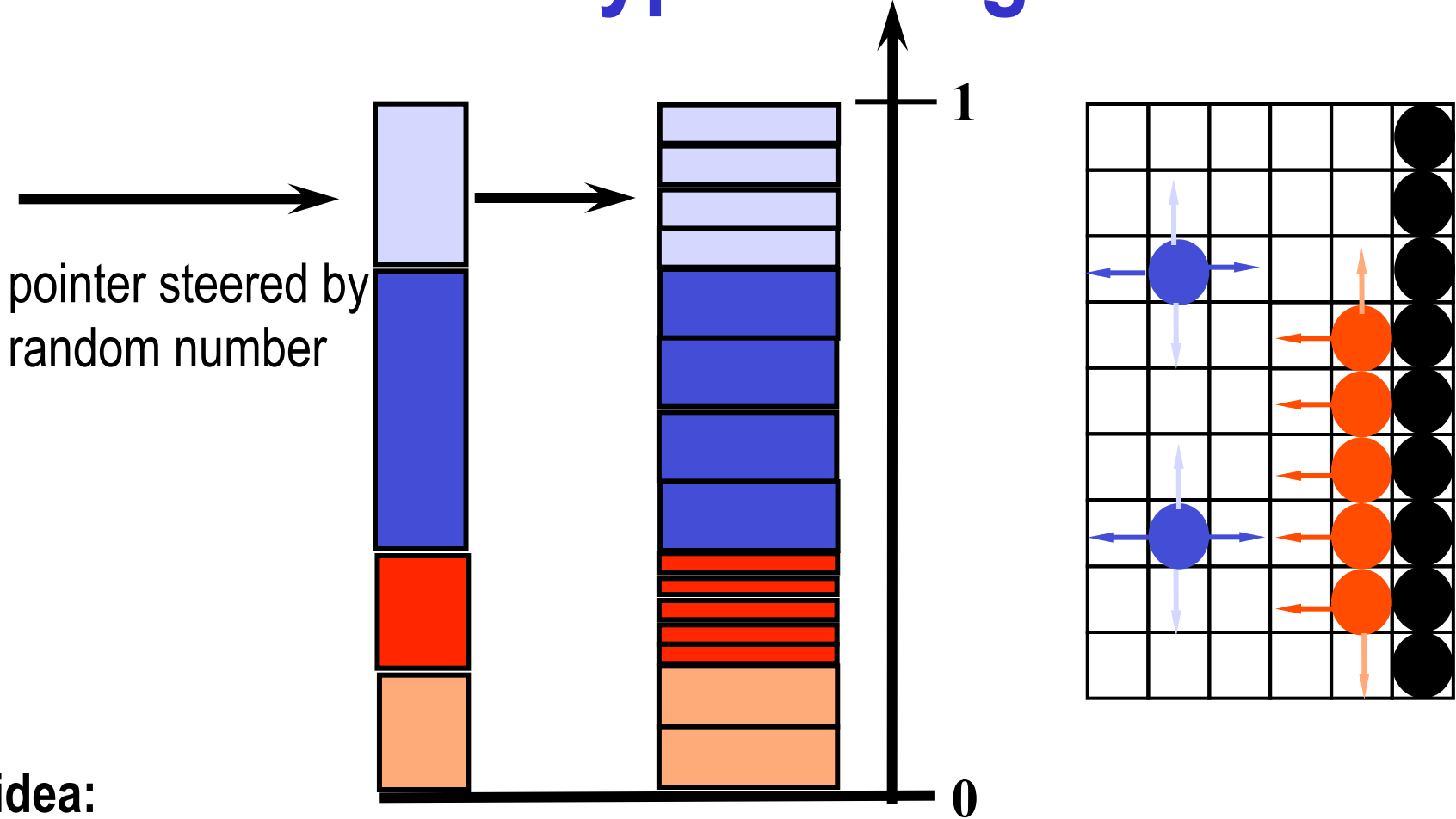
- process-type list algorithm
- binary-tree algorithm
- time-ordered-list algorithm

Application to a lattice-gas model

- example: lattice $L_x \times L_y$
- ~~foo's algorithm: first select one particle, then select one move of that particle~~
- **the correct solution:** cumulated partial rates
$$r_k = \sum_{i=1}^k \Gamma_i$$
, normalized to the total rate $R = r_N$
- **selection process:** draw a random number ρ and compare it to all the r_k/R sequentially; as soon as ρ exceeds r_k/R , execute process k
- **problem:** we need to compare ρ to many (in the worst case all) of the r_k/R
- **note:** Selecting a process with the right probability requires that we **can enumerate** all N processes.

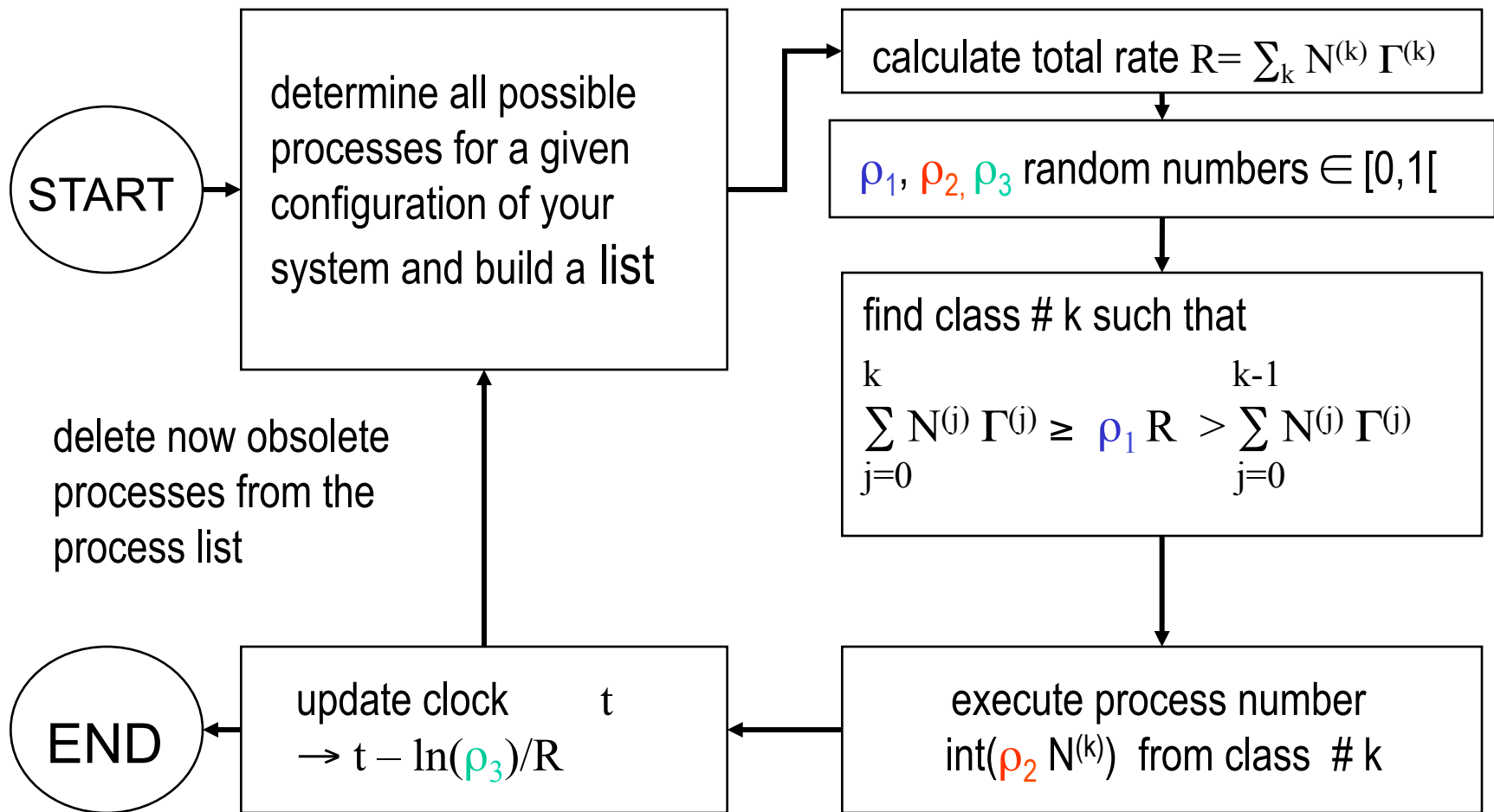


Process-type-list algorithm



for p process types, we need to compare only to the p numbers $N^{(k)} \Gamma^{(k)}$, $k=1,p$, rather than to **all** r_k/R (which are much more numerous)

flow chart for a kMC algorithm



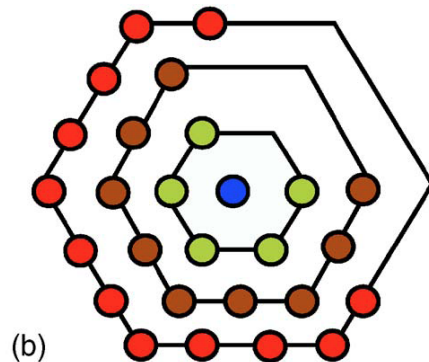
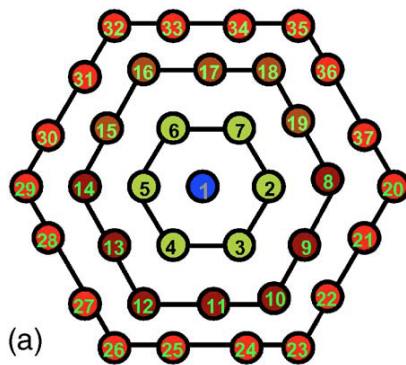
Time-ordered list algorithm

1. assign a random waiting time t_i to each individual process
 2. sort all processes according to ascending waiting time (requires only $\log(N)$ comparisons, if done in a way similar to the binary tree)
 3. always select the **first** process and execute it
 4. advance the clock by $t \rightarrow t + t_i$
 5. Update the list and return to 1.
- This algorithm requires many exponentially distributed random numbers; thus it's advisable to use specially a designed random number generator.

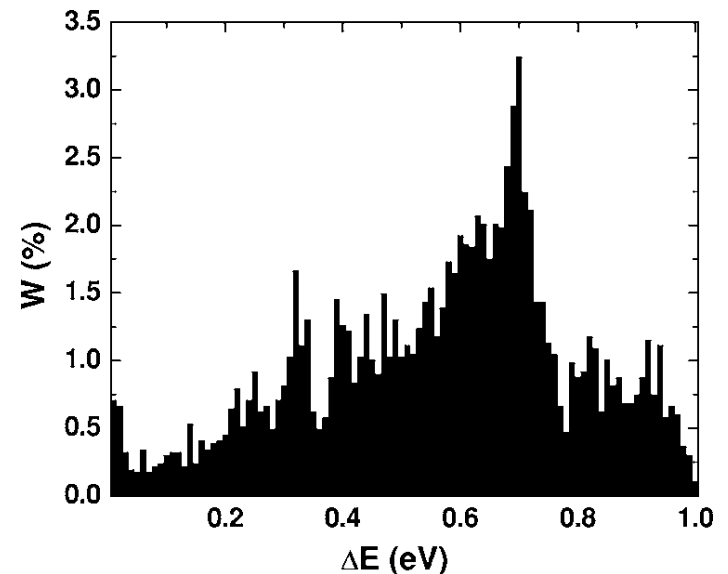
B. Lehner, M. Hohage & P. Zeppenfeld, Chem. Phys. Lett. **336**, 123 (2001)

Self-learning kMC

- **Idea:** build up a database of rates on the fly
- If a certain environment/certain process is missing in the database, spawn a calculation of the barrier for this process.
- All environments **on a lattice** can be classified by the occupancy of neighbor shells.



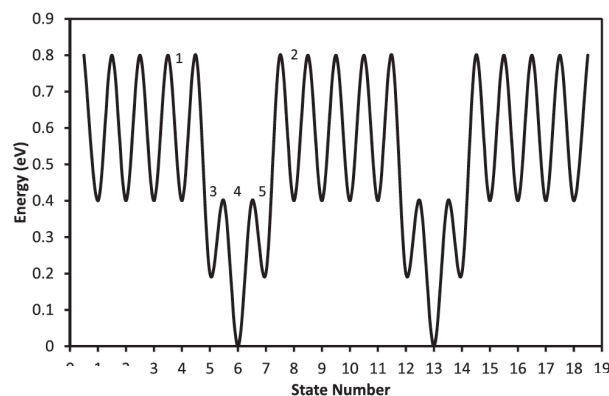
Shell	Base 2	Base 10
1	111110	31
2	11111111000	511
3	001111111111110000	16380



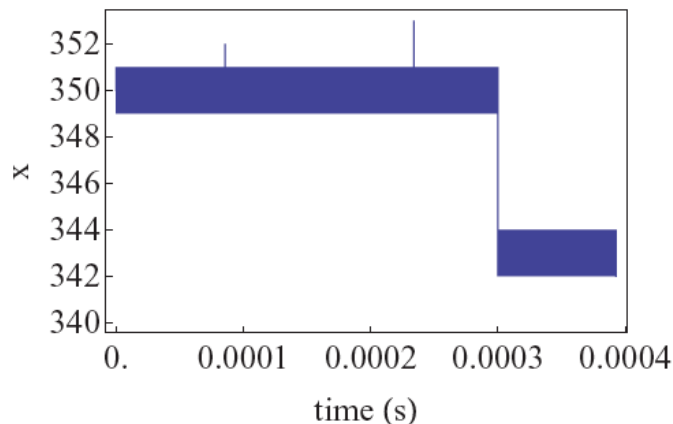
superbasin algorithm

- If “fast” hops occur, consolidate them into a superbasin
- several exits with analytically calculated partial probabilities
- various models for exit time distributions available
- superbasins can be created or dismantled "on the fly"

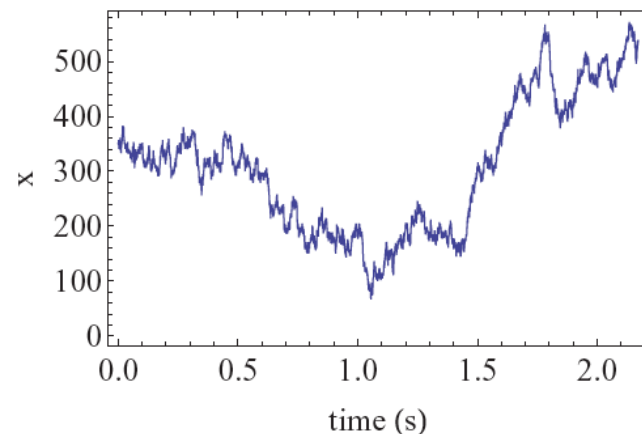
example:
1D potential



conventional kMC



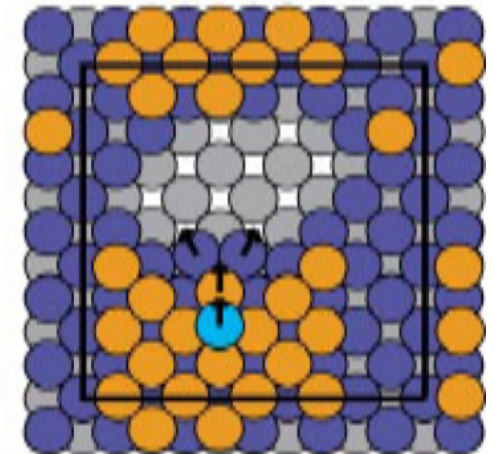
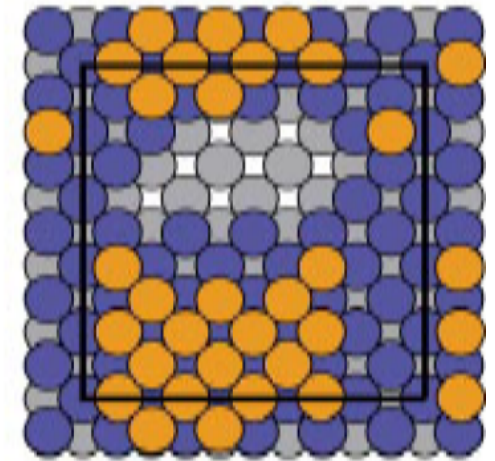
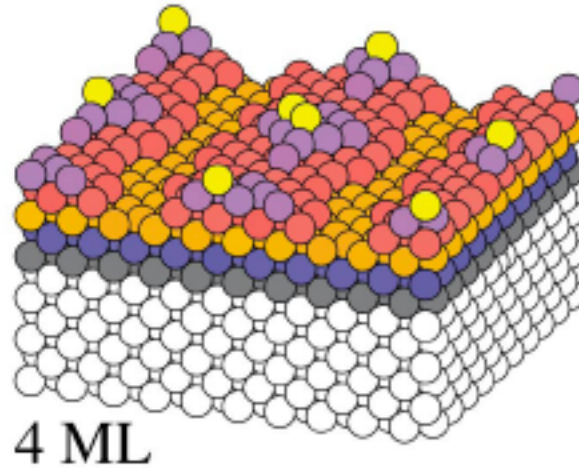
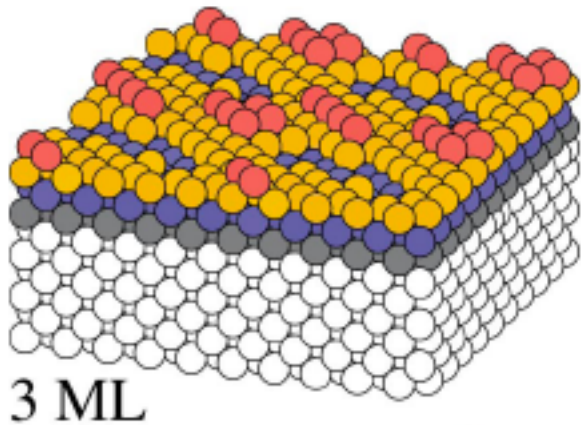
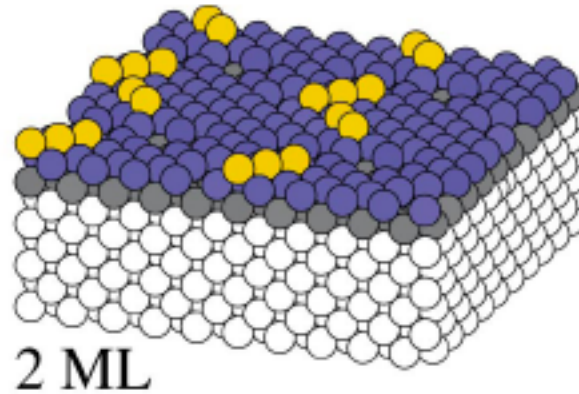
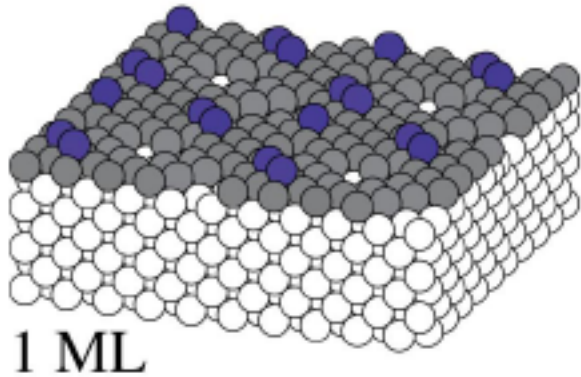
with superbasins



K. Fichthorn & Y.
Lin, J. Chem.
Phys. **138**,
164104 (2013)

**From molecular
dynamics to kinetic
Monte Carlo**

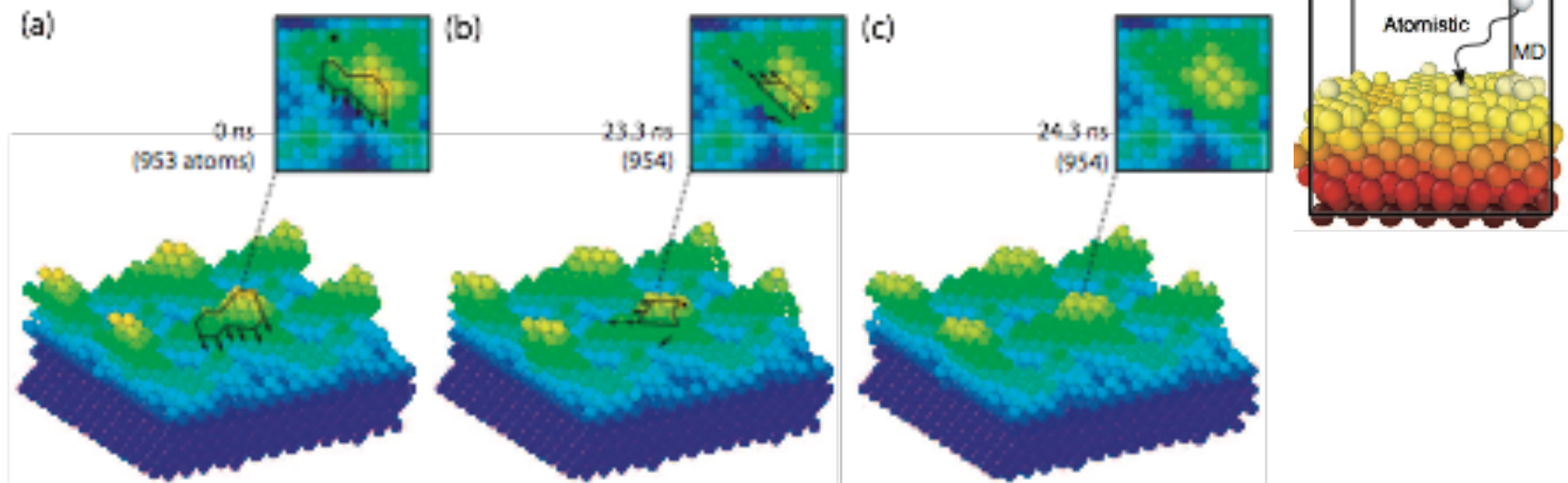
Collective processes



F. Montalenti, M.R. Sørensen and A.F. Voter,
Phys. Rev. Lett. **87**, 126101 (2001)

Counter-example: liquid-solid epitaxy

- Molecular dynamics (MD) may be unavoidable in cases when the atoms **not** are sitting on lattice sites
- possibly use some **accelerated** MD
- here: only the solid phase is treated atomistically



Transition State Theory (1-dim)

- Kramer's rate theory

$$\Gamma = \frac{\lambda}{\omega_b} \left(\frac{\omega_0}{2\pi} \exp\left(-E_b/kT\right) \right) \quad \lambda = \left(\gamma^2/4 + \omega_b^2 \right)^{1/2} - \frac{\gamma}{2}$$

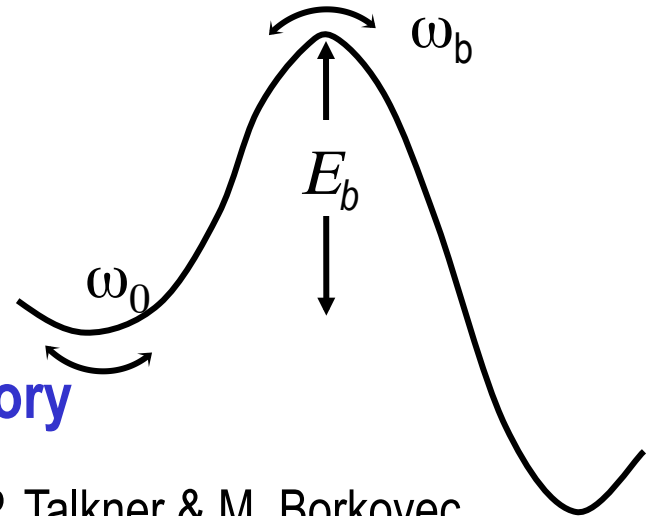
γ : friction due to coupling to the heat bath

- high-friction limit

$$\Gamma = \frac{\omega_0 \omega_b}{2\pi\gamma} \exp\left(-E_b/kT\right)$$

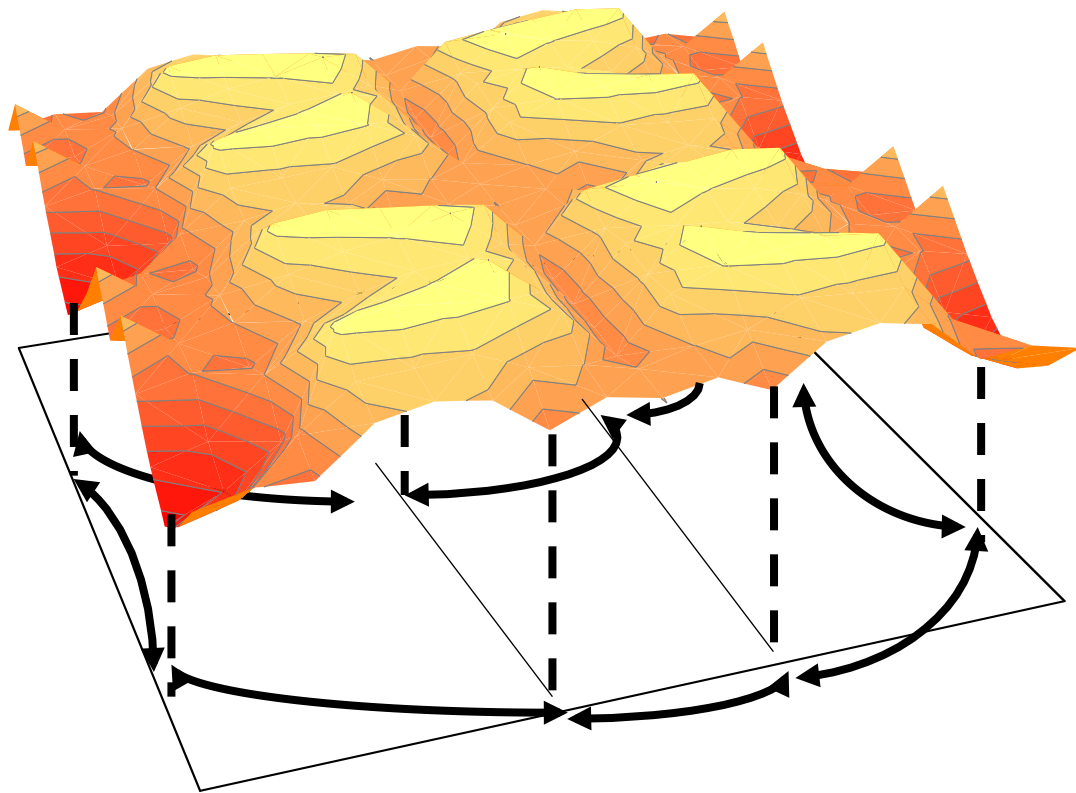
- 'medium' friction \rightarrow **transition state theory**

$$\Gamma = \frac{\omega_0}{2\pi} \exp\left(-E_b/kT\right)$$



P. Hänggi, P. Talkner & M. Borkovec,
Rev. Mod. Phys. **62**, 251 (1990)

From the PES to rate constants Γ (multi-dimensional)



idea:

associate minima with the nodes, hops with the interconnects in a network

hopping rates derived from the PES

$$E(x_i, y_i) = \min_{z_i, c_\alpha} E_{\text{tot}}(x_i, y_i, z_i, c_\alpha)$$

$$\Gamma = kT/h \frac{Z_{\text{TS}}}{Z_i} = \text{(harmonic \& classical approximation)} = \prod_N \nu_{k,i} / \prod_{N-1} \nu_{k,\text{TS}} \exp(-\Delta E/kT)$$

How accurate is Transition State Theory ?

Three levels of approximation:

- 1 direct molecular dynamics
- 2 TST with **thermodynamic integration** of partition functions from restricted molecular dynamics at the ‘ridge’ (‘blue-moon-ensemble’)
- 3 TST within harmonic approximation

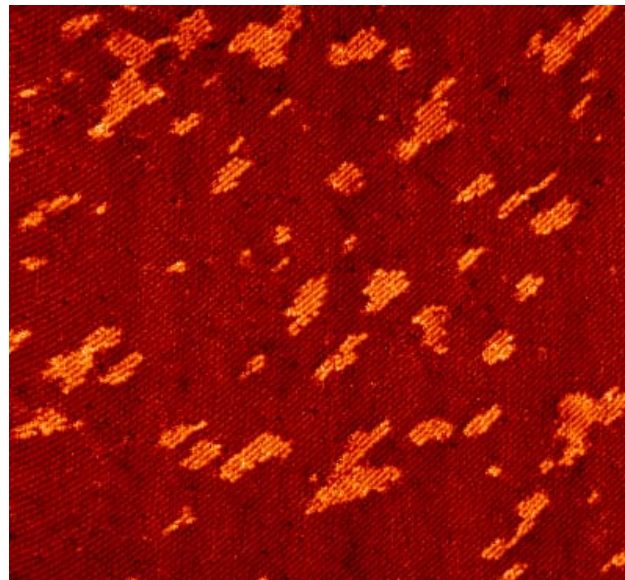
Cu/Cu(100): good agreement between method 1) and 2)

	$\ln\Gamma_0$ [THz]		ΔE [eV]		static
	TI	MD	TI	MD	
hop	2.9 ± 0.2	3.0 ± 0.2	0.51 ± 0.02	0.49 ± 0.01	0.50
exchange	6.5 ± 0.6	6.1 ± 0.7	0.74 ± 0.02	0.70 ± 0.04	0.73

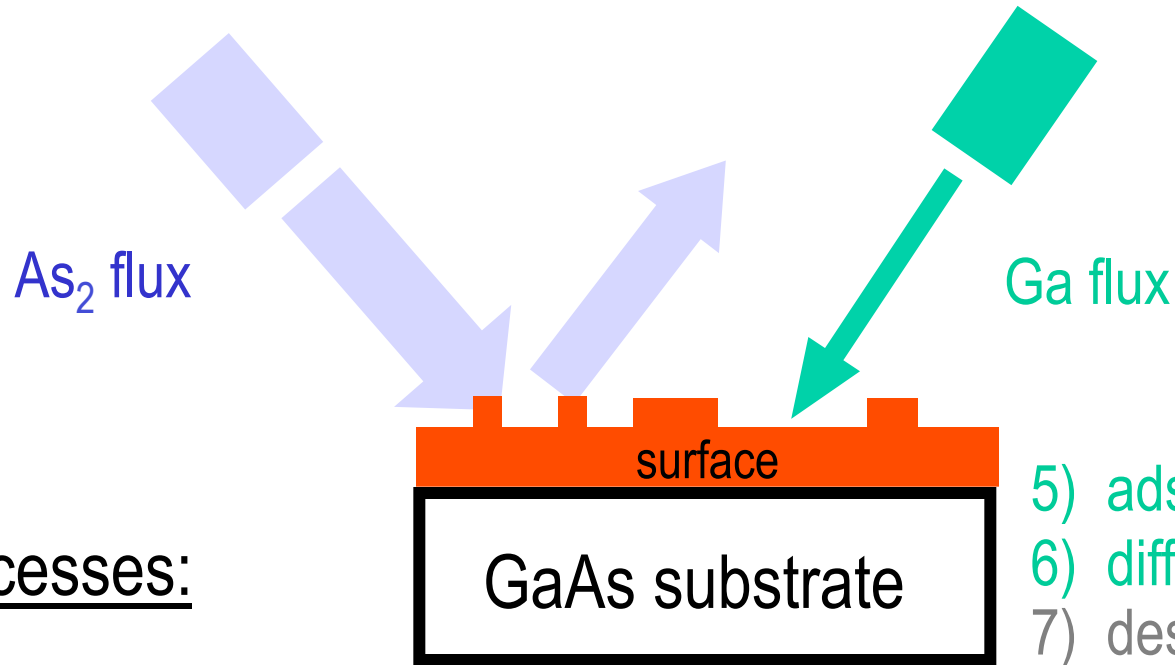
G. Boisvert, N. Mousseau & L.J. Lewis, PRB **58**, 12667 (1998)

Application I:

Molecular beam epitaxy on
GaAs(001) $\beta 2(2 \times 4)$



Molecular beam epitaxy of III-V semiconductors



Processes:

- 1) adsorption of As₂
- 2) dissociation of As₂
- 3) diffusion of As
- 4) desorption of As₂

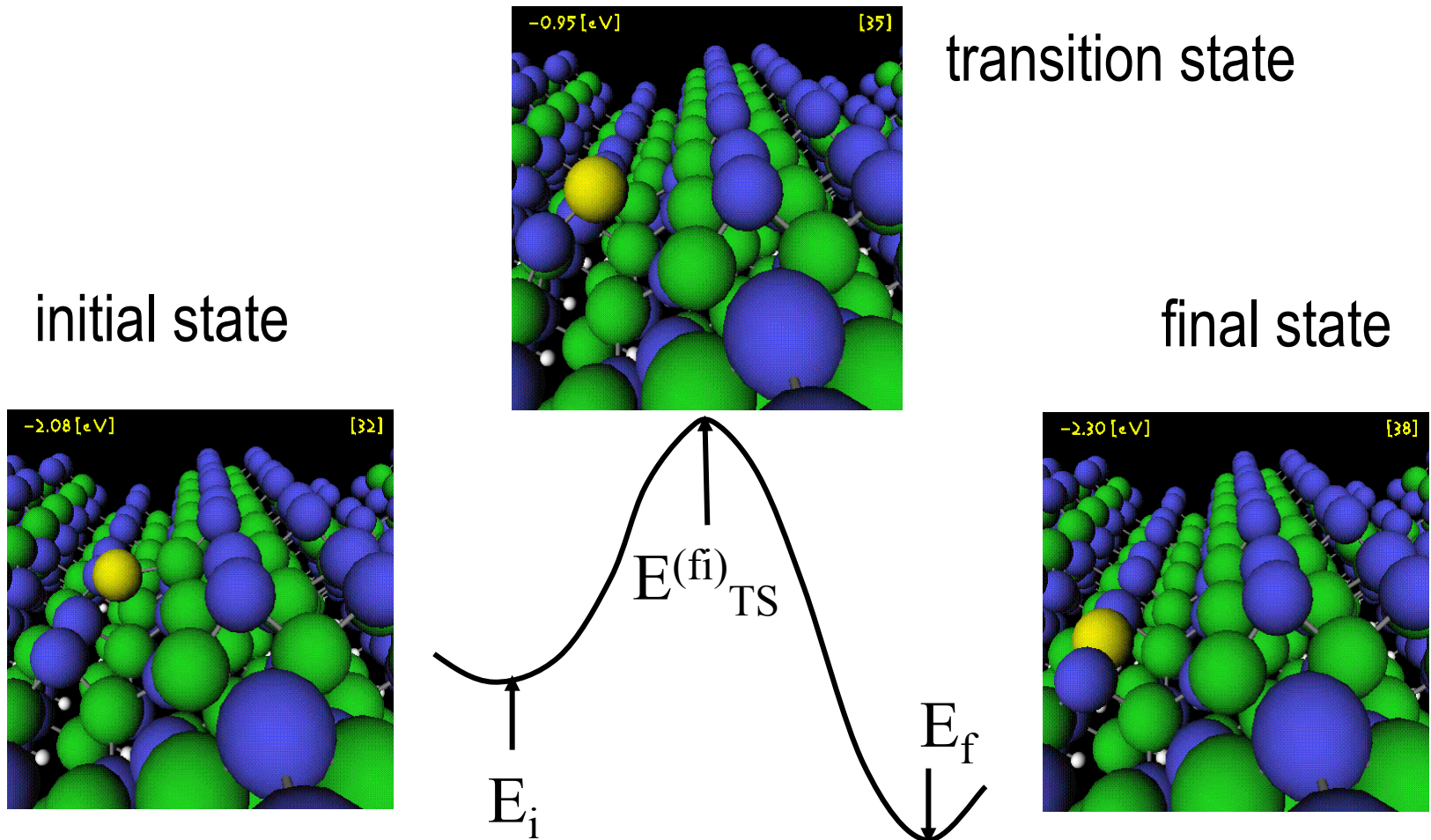
- 5) adsorption of Ga
- 6) diffusion of Ga
- 7) desorption of Ga

- 8) island nucleation
- 9) growth

What is the interplay of these processes for a given temperature and flux ?

Rates from first-principles calculations

$$\Gamma^{(k)} = W(f,i) = \Gamma^{(fi)}_0 \exp(- (E^{(fi)}_{TS} - E_i) / kT)$$

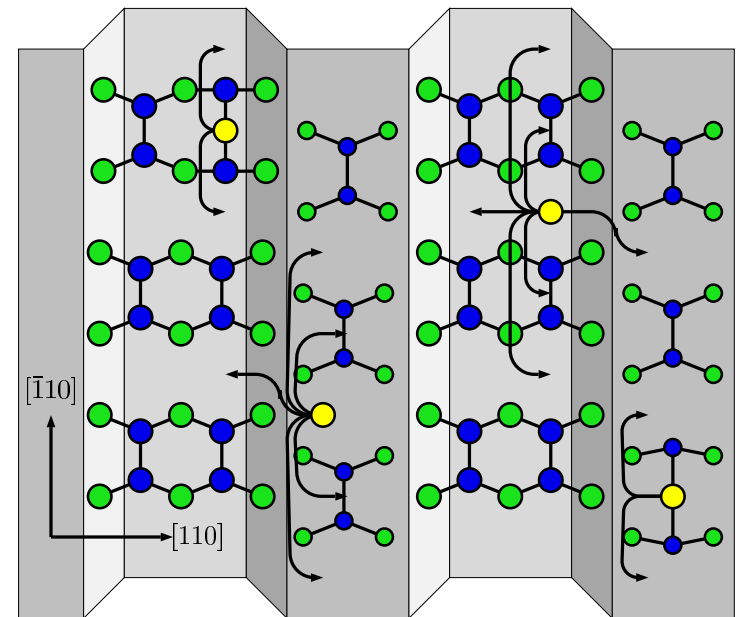
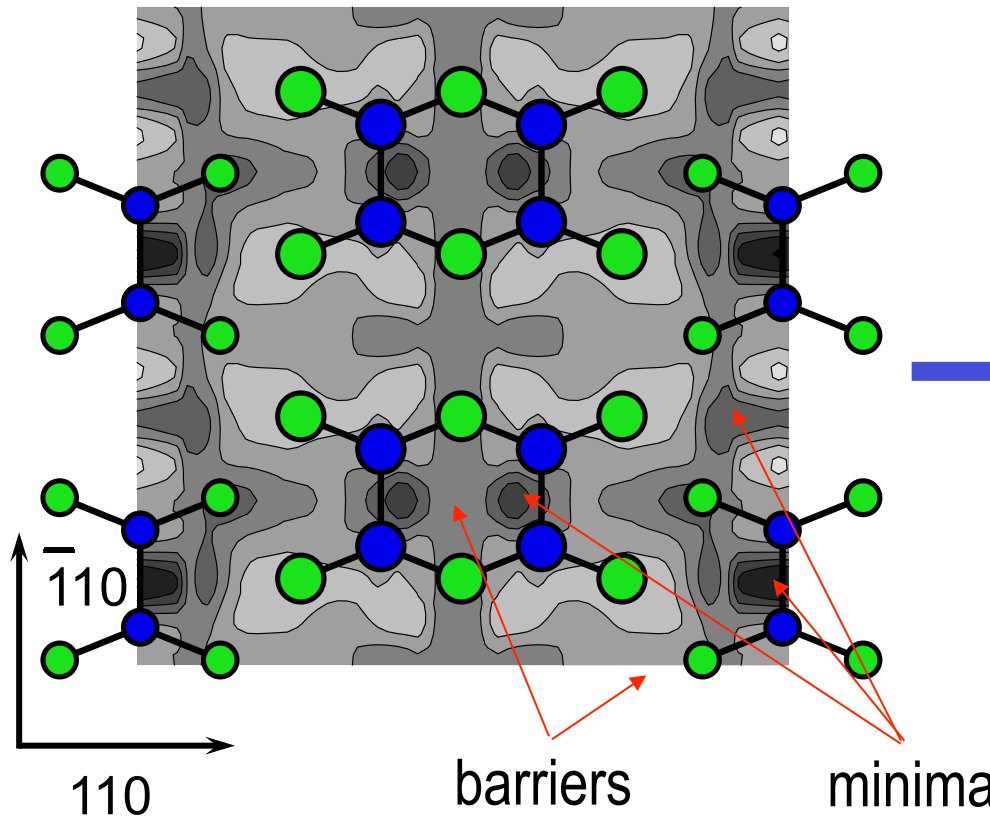


Surface diffusion on GaAs(001): mapping of PES to network graph

PES from DFT calculations

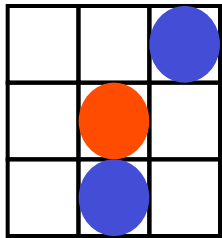


network of hops



kMC with explicit list of process types

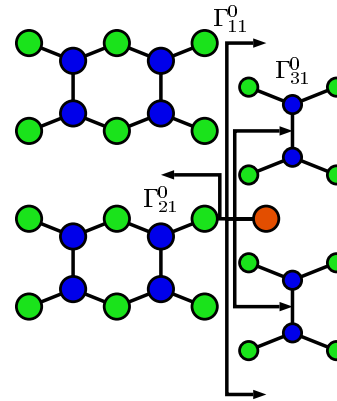
Voter's lattice kMC:



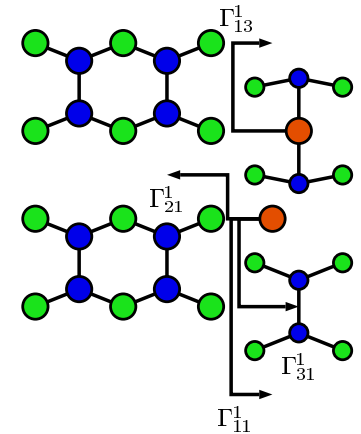
A.F. Voter PRB **34**,
6819 (1986)

- simulation on a lattice
- group possible transitions $\Gamma(f,i)$ from i to f into **classes**, each class is characterized by a rate
- classification of initial and final state by 'atomic neighborhoods' e.g., the **number and relative position of neighbors** define a process type

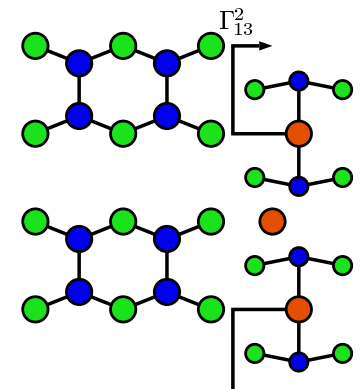
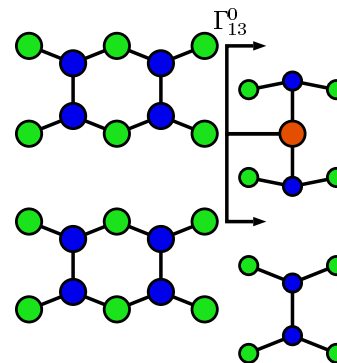
DFT-based kMC:



possible hops
in the trench...

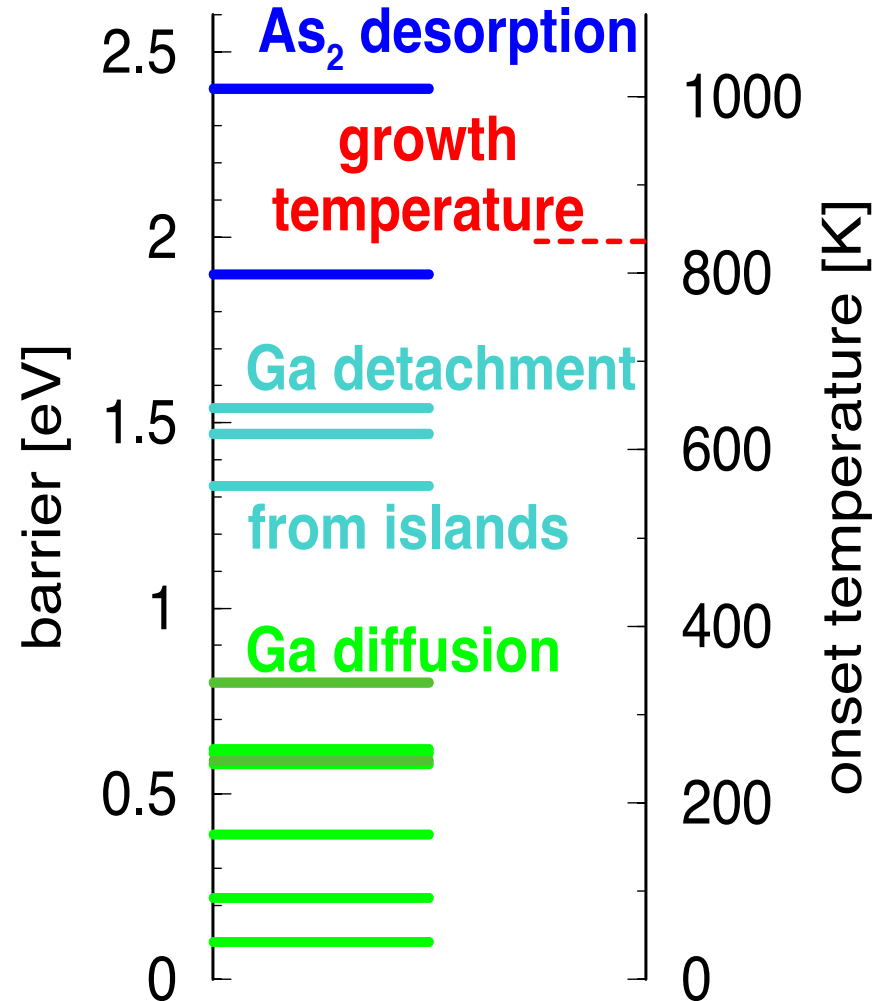


.. modified rates
due to neighbors.



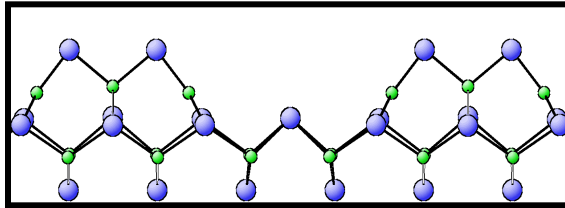
kinetic Monte Carlo simulations for GaAs epitaxy

- 32 microscopically different Ga diffusion processes, and As₂ adsorption/desorption are included explicitly
- computational challenge: widely different time scales (10⁻¹² sec to 10 sec)
- simulation cell 160 x 320 sites (64 nm x 128 nm)



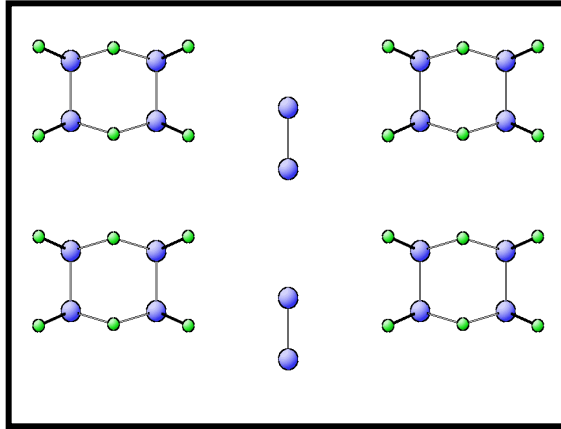
kinetics of island nucleation and growth

side
view

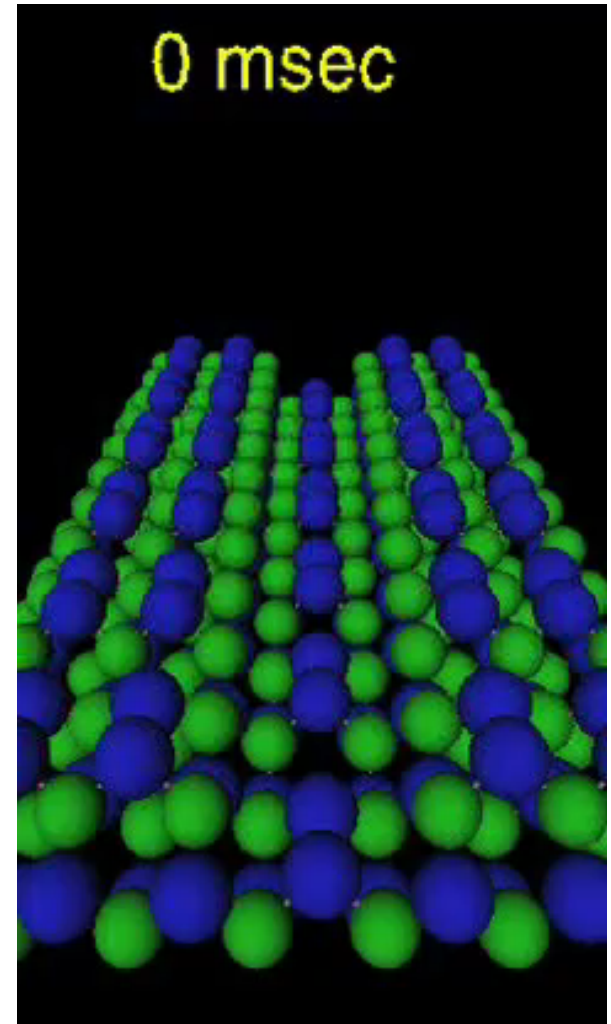


Ga
As

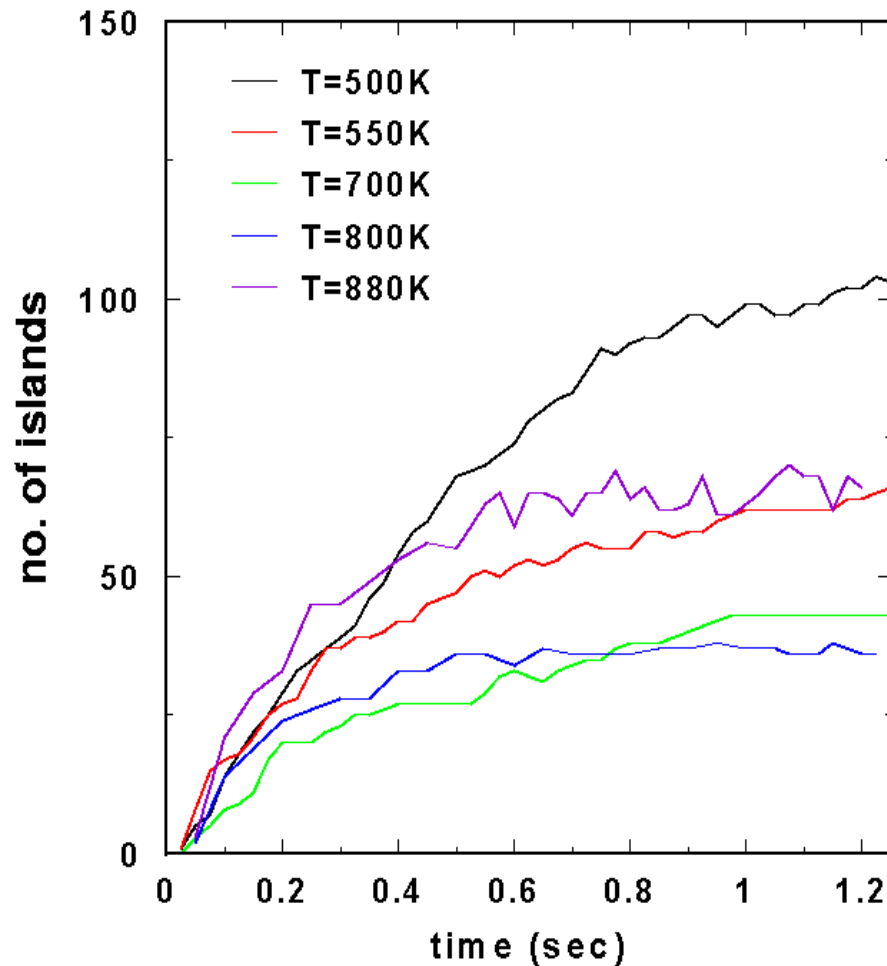
top
view



1/60 of the full simulation cell
As₂ pressure = 0.85×10^{-8} bar
Ga deposition rate = 0.1 ML/s
 $T = 700$ K

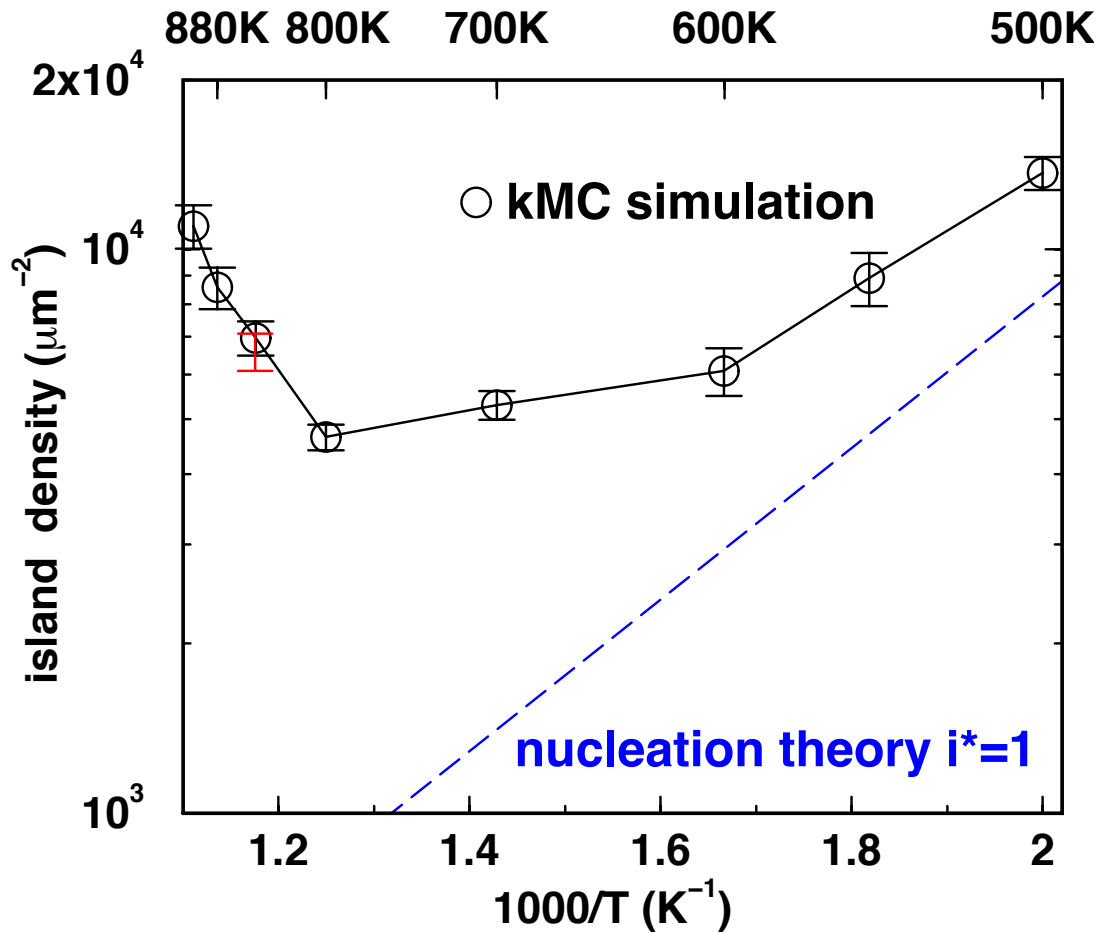


island density



deposition rate
0.1 ML Ga per second, III/V
ratio 1:1000, T=700K

scaling with temperature ?



‘conventional’
nucleation theory

$$N_{\text{is}} = \eta (R/D)^{i^*/(i^*+2)}$$

N_{is} island density

D diffusion constant

R deposition flux

η numerical const.

i^* critical nucleus

simulation: P. Kratzer and M. Scheffler,
Phys. Rev. Lett. **88**, 036102 (2002)

experiment: G.R. Bell et al.,
Surf. Sci. **423**, L280 (1999)

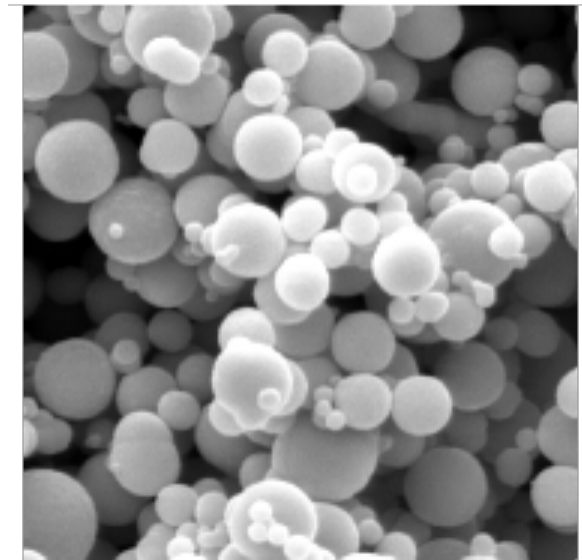
Application II:

kinetics of sintering

Sintering in materials synthesis

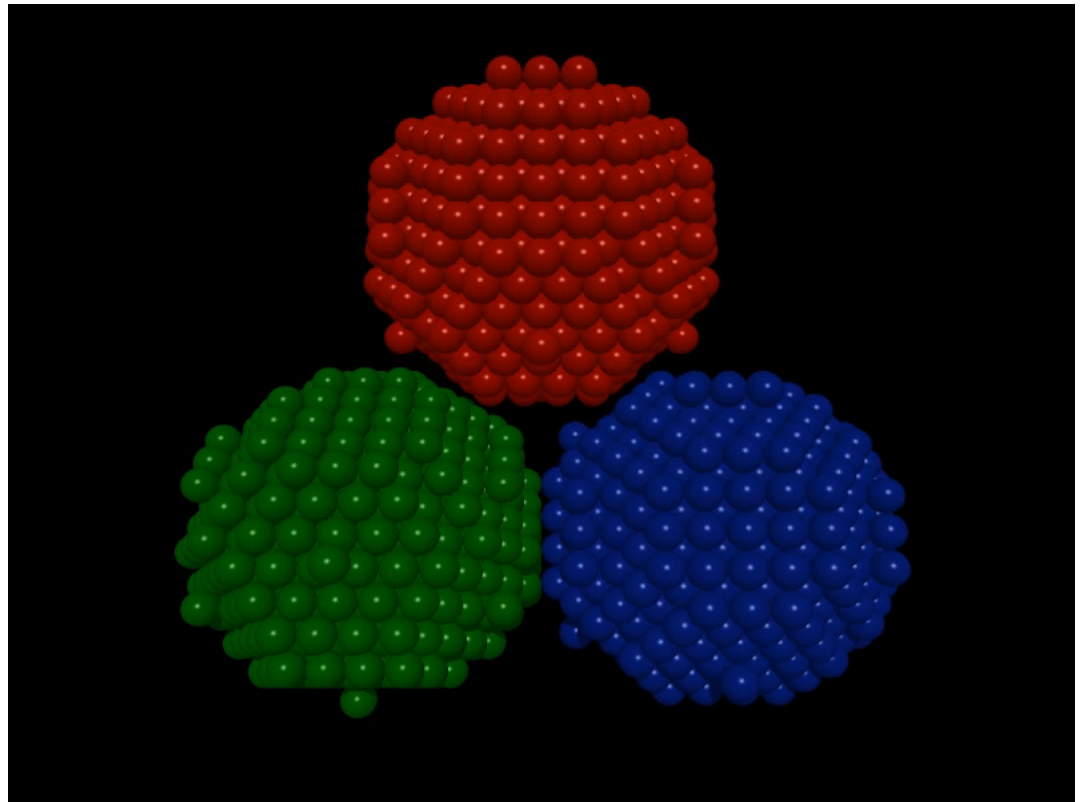
- For thermodynamics reasons, some materials (e.g. alloys) cannot be grown from solution
- polycrystalline samples may be obtained by synthesising small particles and compaction, followed by a temperature and/or heat treatment
- large crystals grow on the expense of smaller ones and may enforce re-orientation of neighbouring crystallites

Carbonyl iron
powder (electron
microscopy image)



Hybrid simulation

- particles treated as rigid bodies, using molecular dynamics with few collective variable
- contact dynamics for touching particles
- surface diffusion and growth treated by self-learning kMC

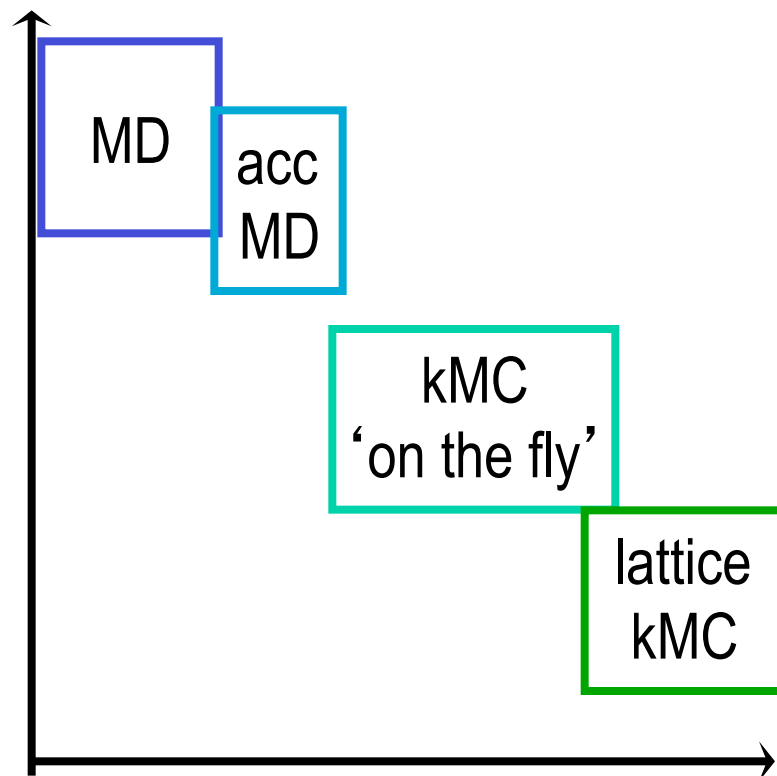


L. Brendel & D.E. Wolf,
University Duisburg-Essen

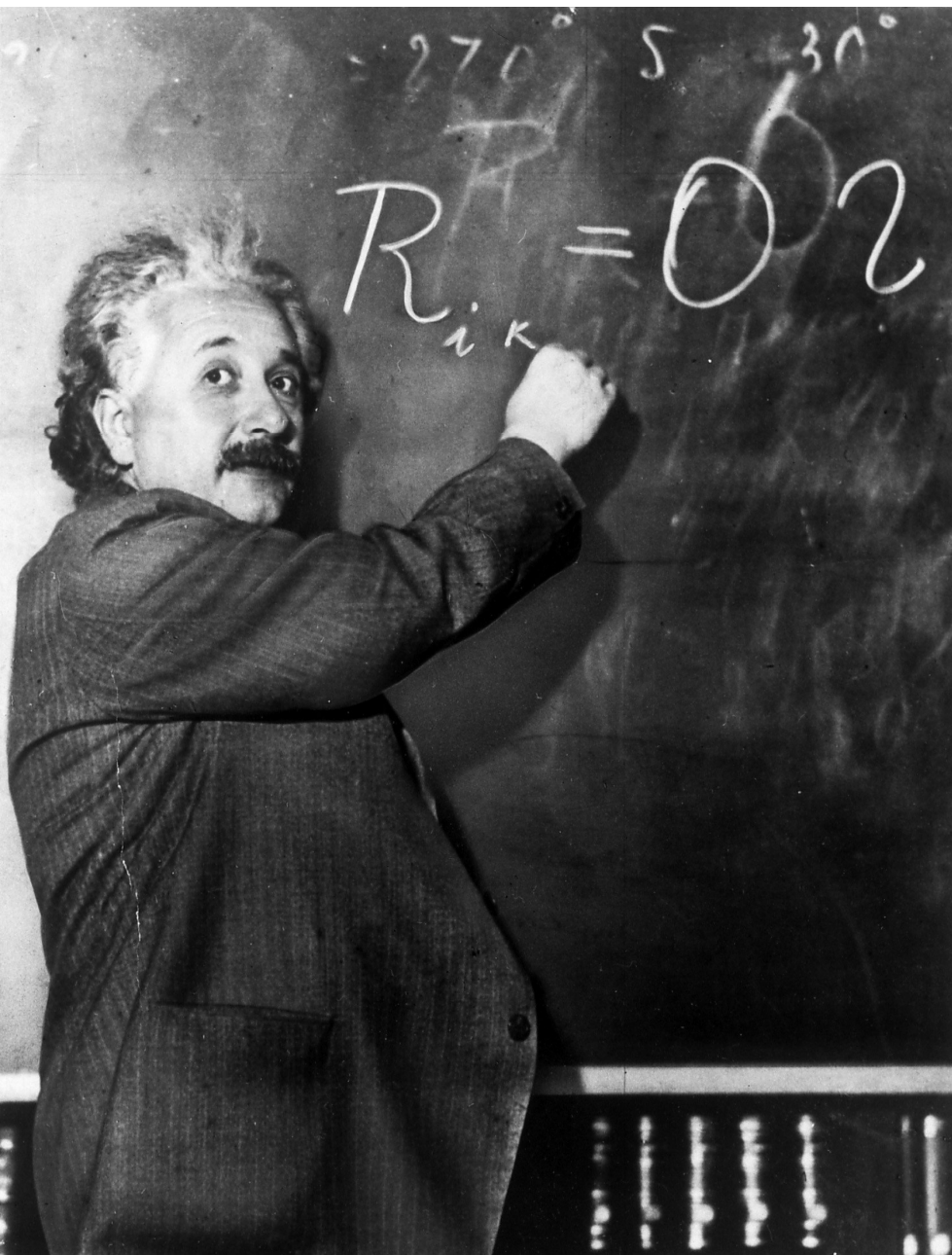
Summary: Bridging the time-scale gap

- molecular dynamics (Car-Parrinello method)
- accelerated molecular dynamics
 - using a boost potential (Voter, Fichthorn,...)
 - temperature-accelerated MD (Montalenti et al. PRL **87**, 126101 (2001))
- kinetic Monte Carlo with transition state search on the fly (avoids both lattice approximation and pre-defined rate table)
- lattice kinetic Monte Carlo, N -fold way (Voter PRB **34**, 6819 (1986))

computational effort



... more and more schematic,
risk of oversimplification



“Keep things as simple as possible,
but not simpler ..”

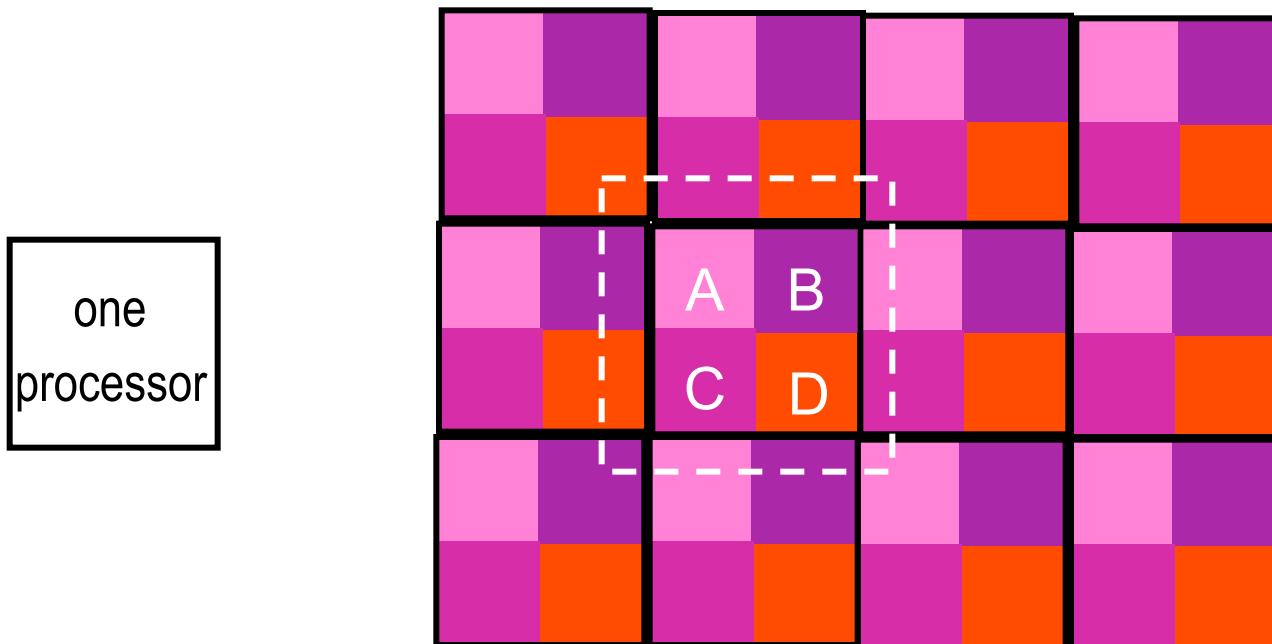
**Thank you for your
attention !**

Summary: [arXiv:0904.2556](https://arxiv.org/abs/0904.2556)

Parallelization of kMC

semi-rigorous synchronous sublattice algorithm

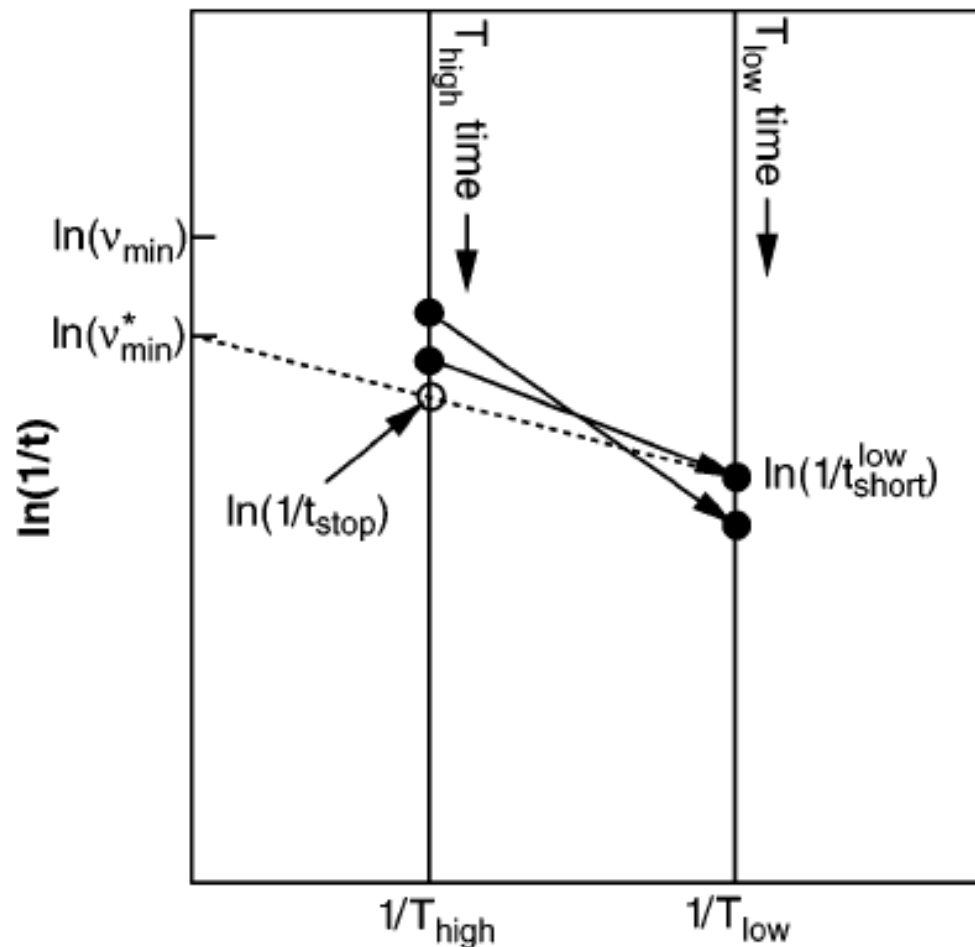
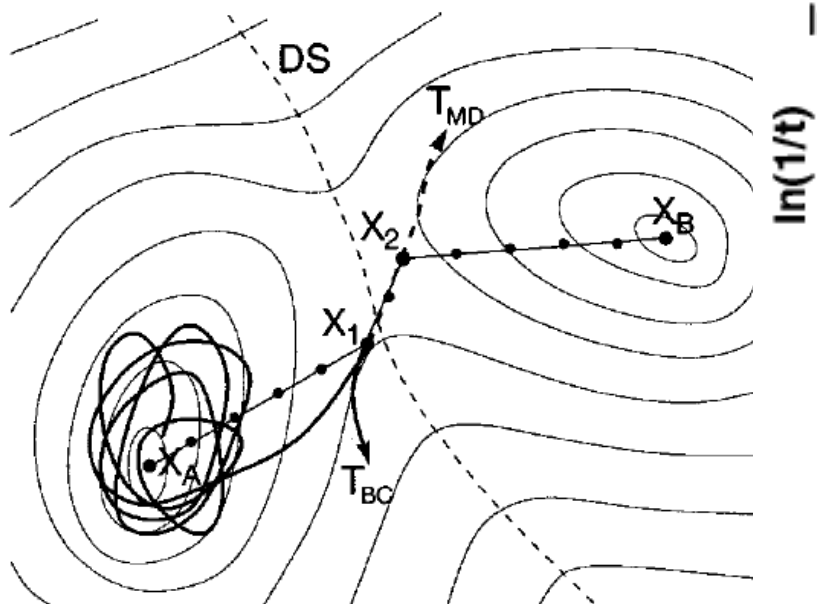
[Y. Shim and J.G. Amar, PRB 71, 115436 (2005)]



A-B-C-D-A-B-C-D- ...

Temperature-accelerated dynamics (TAD)

Event is observed at T_{high} , but its rate is extrapolated to T_{low} (using the TST rate law).



“Speculative” TAD

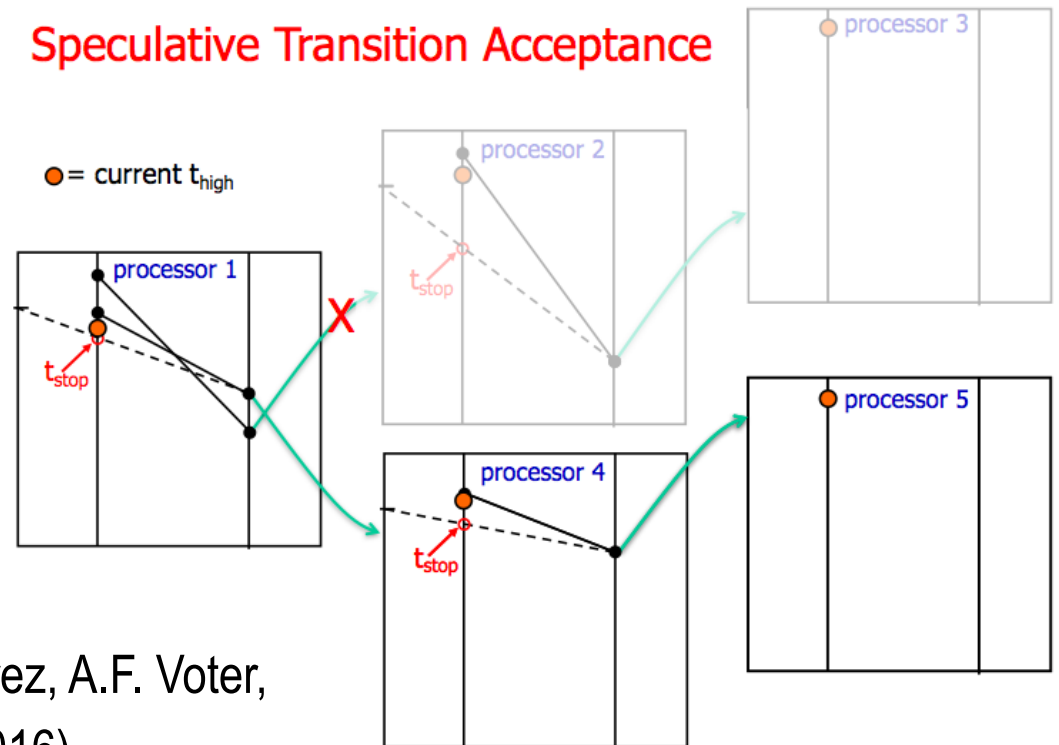
A way to use computational **parallelism** in kinetic simulations ..

If you have many processor cores available,

- spawn a new TAD sub-simulation as soon as a transition is seen
- use the retro-diction from T_{high} to T_{low} to assign a time when to expect this event
- branching continues

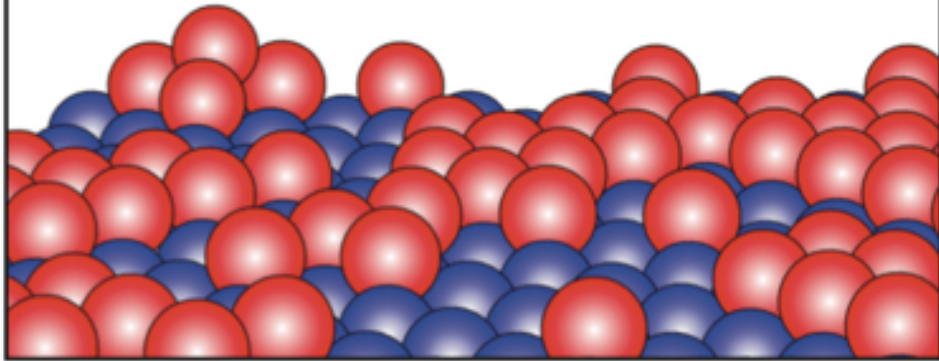
until/unless it has become clear that this transition is **not** the one to be accepted (at T_{low}).

Speculative Transition Acceptance

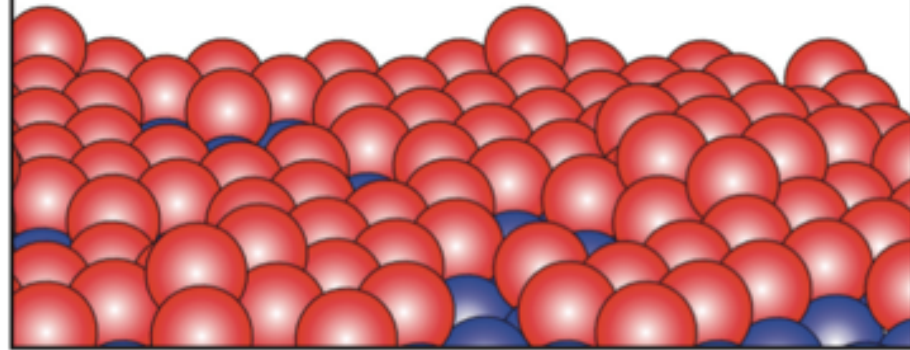


Example: Vapor-phase epitaxy of Cu on Ag(100)

Simulated time: 11.6 s (0.5 ML)
Wall clock time: 0.75 h (Boost $\approx 7 \times 10^9$)



Simulated time: 25.6 s (1.0 ML)
Wall clock time: 8.8 h (Boost $\approx 1 \times 10^9$)



It took ~ 1 year to grow 1.5 ML with serial TAD.

(Sprague et al, Phys. Rev. B 66, 205415 (2002))