
State of FHI-aims

Volker Blum

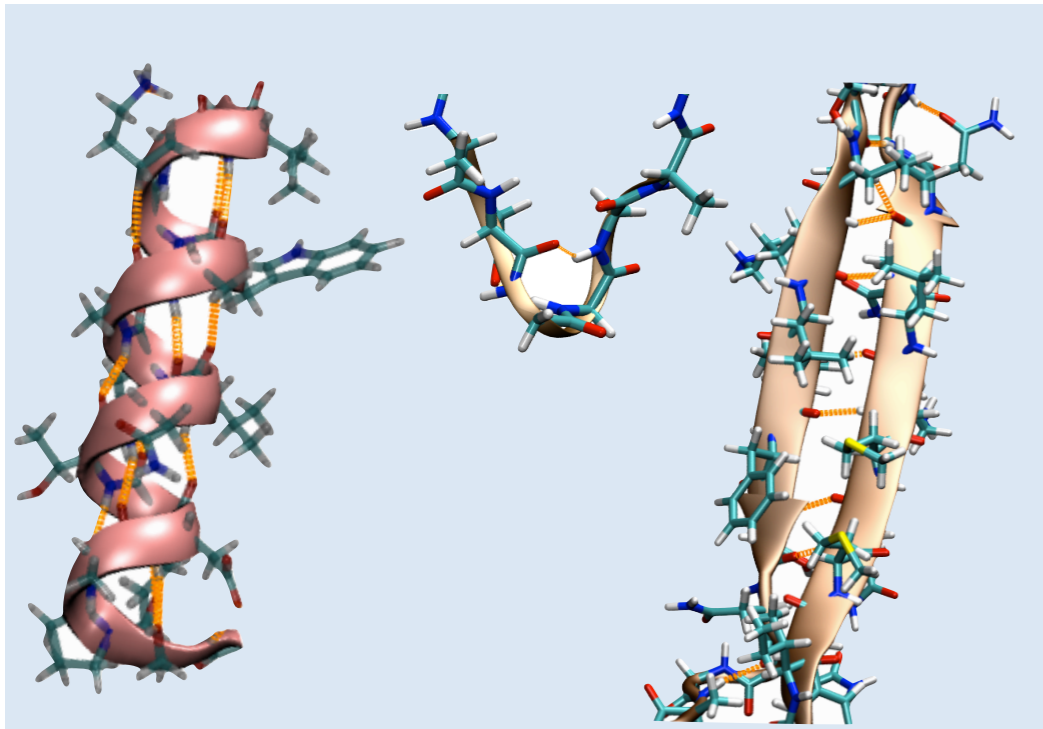


Objective: Electronic structure theory today

Enormous successes:

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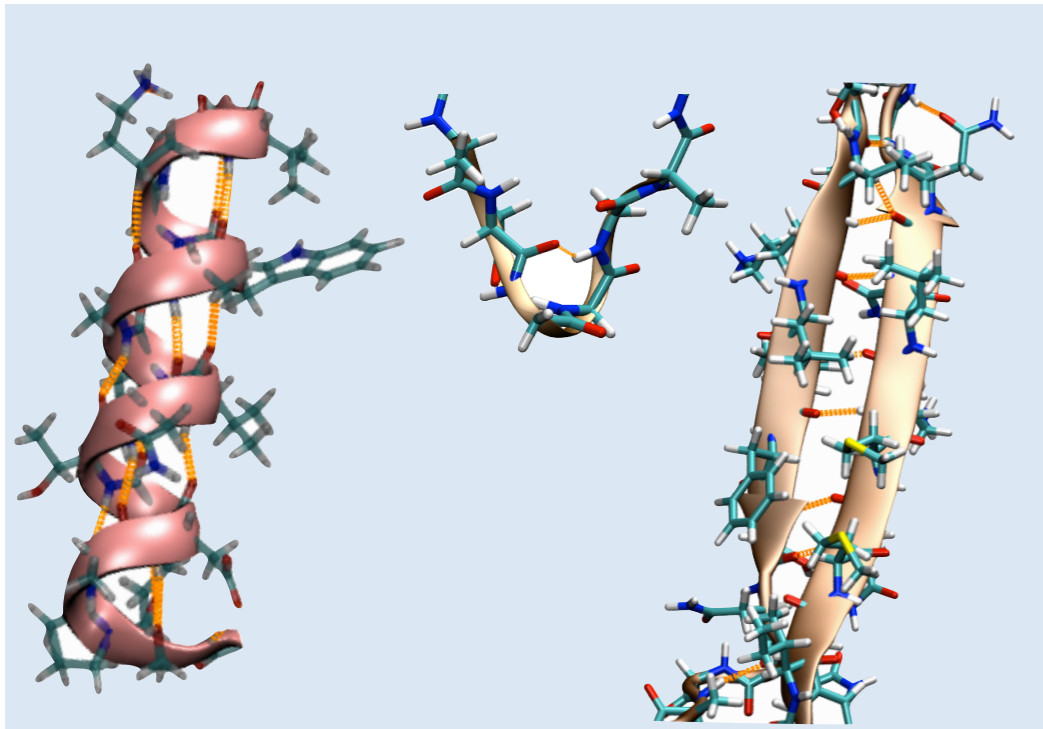


(Bio)molecular matter

- Structural complexity
- statistical averages & dynamics
- “weak” interactions critical

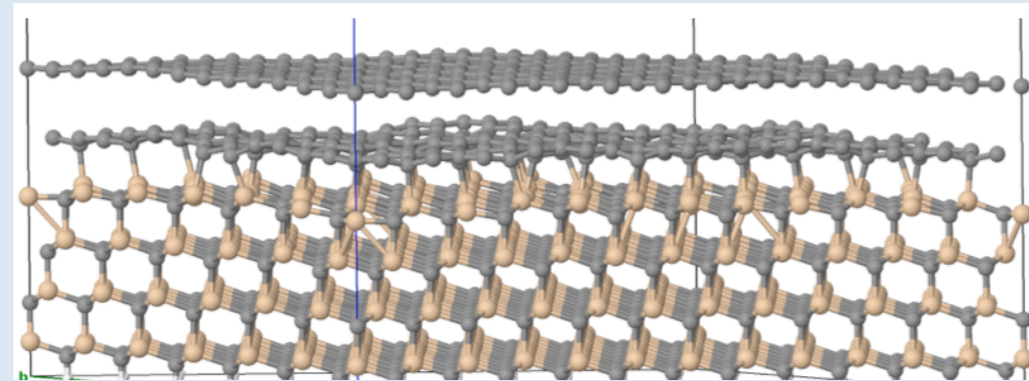
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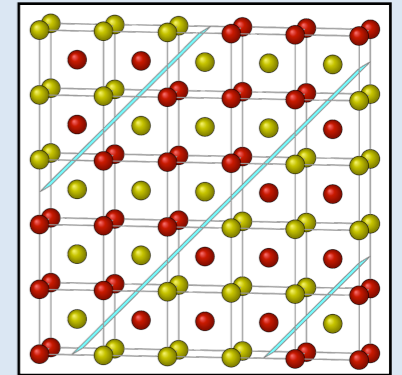


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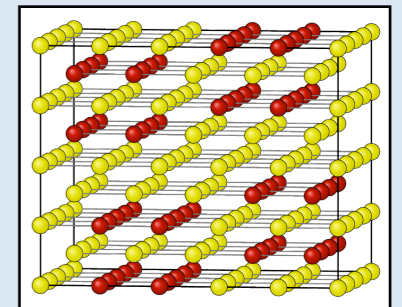
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Graphene / SiC



Ta₃W₃



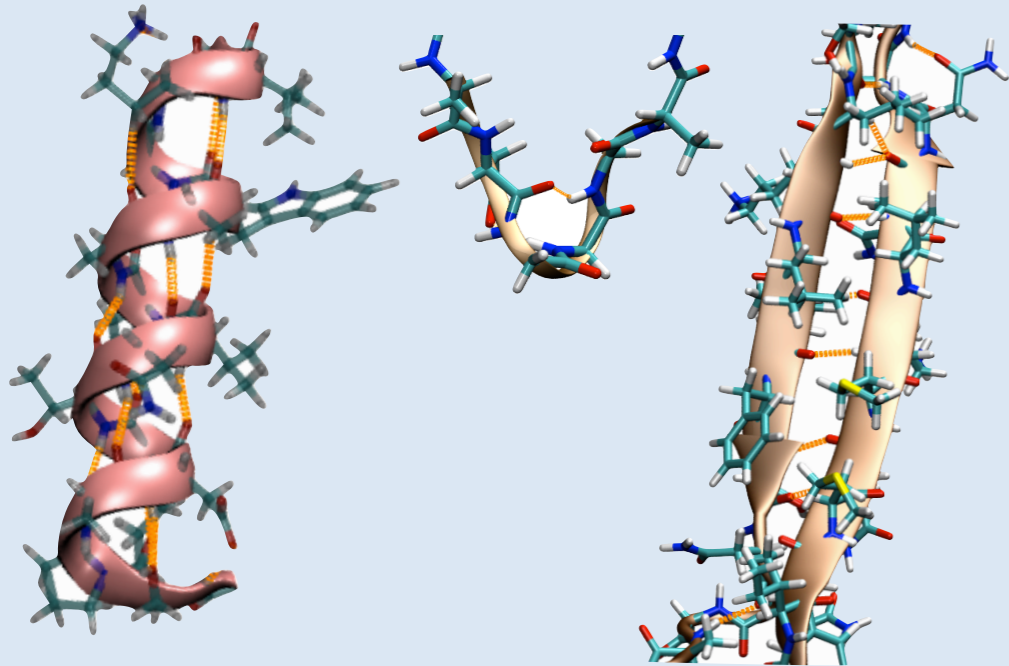
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Condensed phases

(solids, surfaces, ...)

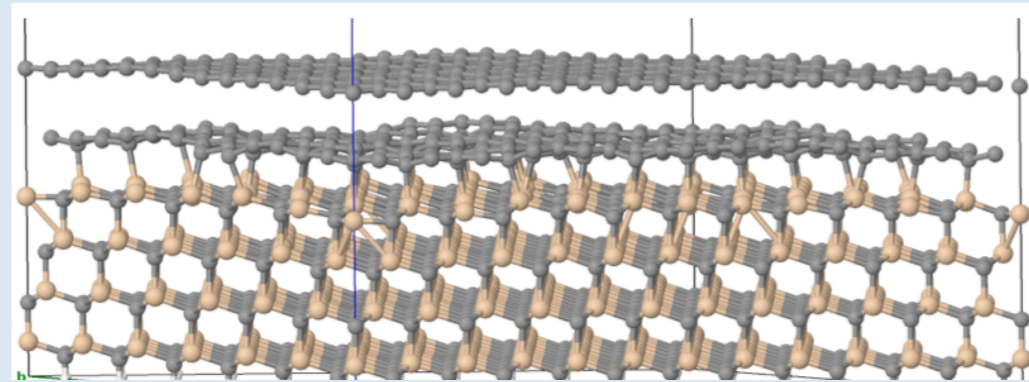
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- Stability, free energies
- electronic, mechanical, optical, ... properties

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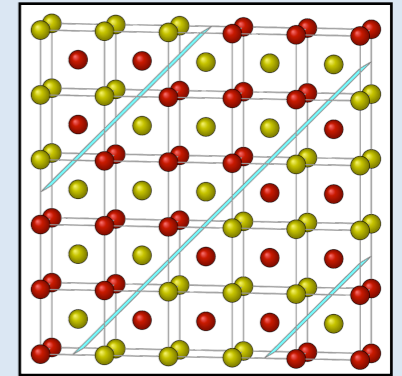


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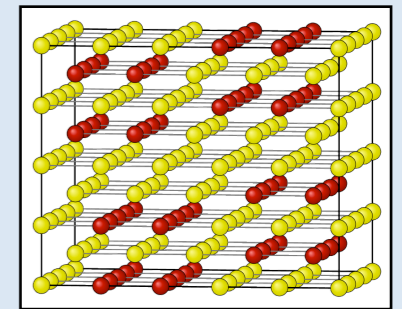
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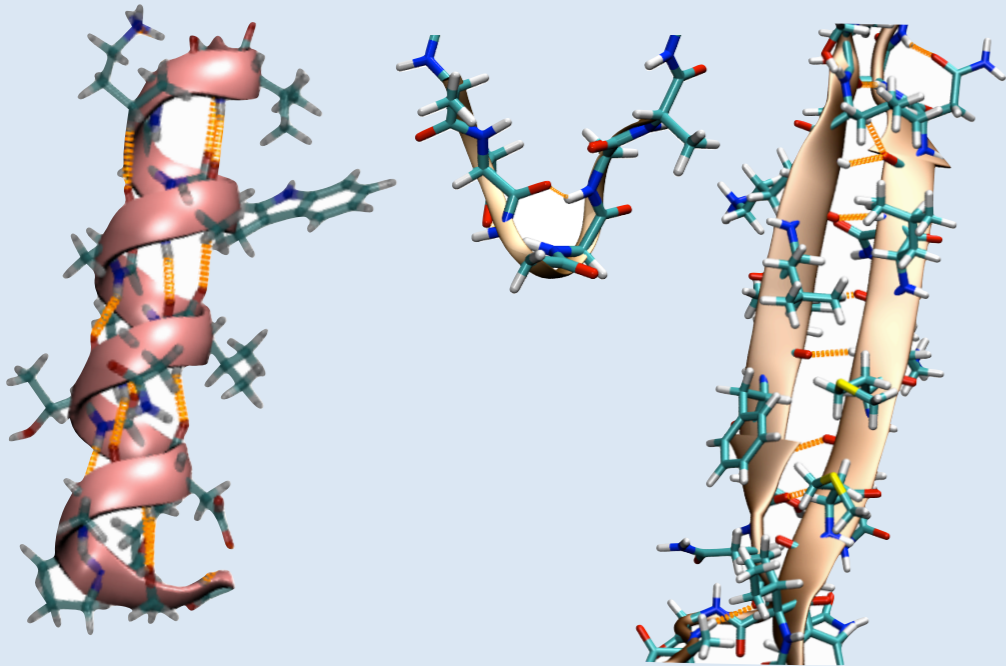
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Matter at extreme conditions

- “electron gas + protons”; high-pressure compounds, transitions
- (Born-Oppenheimer) molecular dynamics, classical nuclei
- Quantum nuclei? (PIMD)

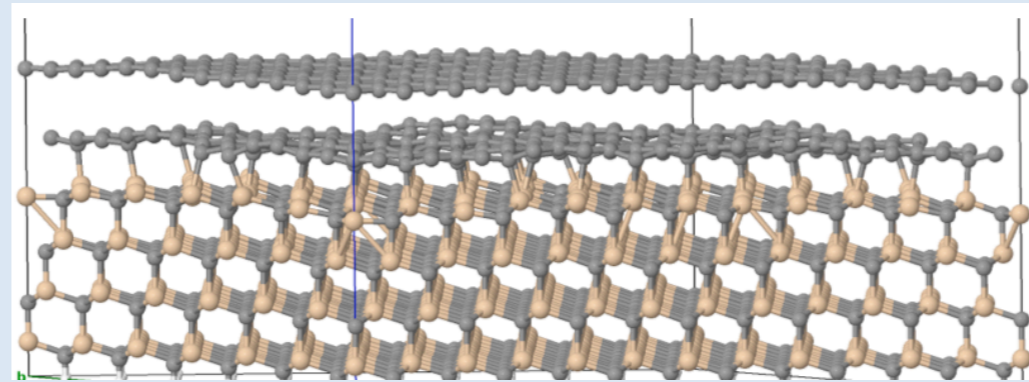


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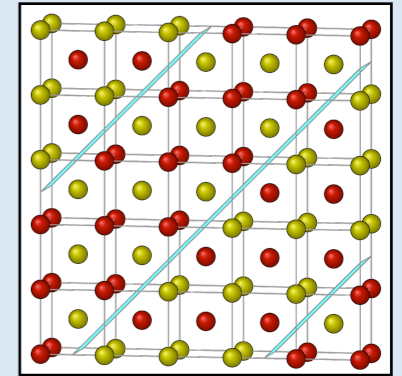


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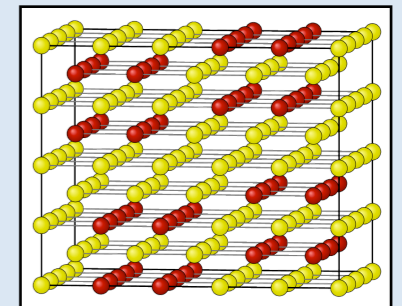
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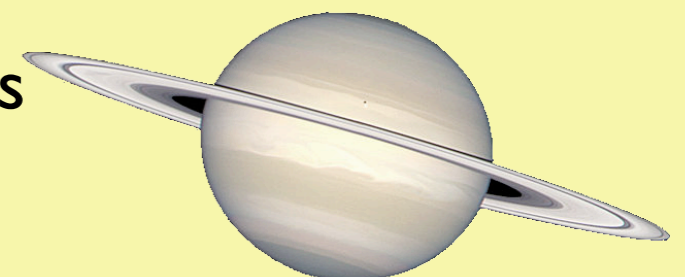
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- Structure!
- Stability, free energies
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Today: “Mostly density-functional theory”, plenty of flavors

Matter at extreme conditions

- “electron gas + protons”; high-pressure compounds, transitions
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Electronic structure theory for real materials

... but also some significant challenges:

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- **Are we computing the right thing?**
 - ▶ Current DFT (LDA/GGA *and beyond*) may qualitatively fail with *or without* warning for much of the interesting space, even for “structure”
 - ▶ Other numerical approximations? (grids, cutoffs, core vs. valence, ...)
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- **Can we compute the right thing?**

- ▶ Realistically sized systems to capture “reality”
- ▶ Statistical averages, dynamics, combinatorial complexity of “structure”?
- ▶ Simply, hardware vs. software - utilize *available* hardware effectively

Outline



- ▶ FHI-aims: Some (very) few basics
- ▶ Significant new developments:
 - Periodic Hartree-Fock and hybrid functionals
 - Unit cell relaxation & analytical stress tensor
 - Many-body perturbation theory: scGW and rPT2
 - “Quantum nuclei”
 - ... many others! (transport, visualization, ...)
- ▶ and how far can we push? Large-scale surface reconstruction
- ▶ “What would be good to have” (future?)

People



Fritz Haber Institute, Berlin

[Richard-Willstätter-Haus]

People



Matthias
Scheffler



Fritz Haber Institute, Berlin

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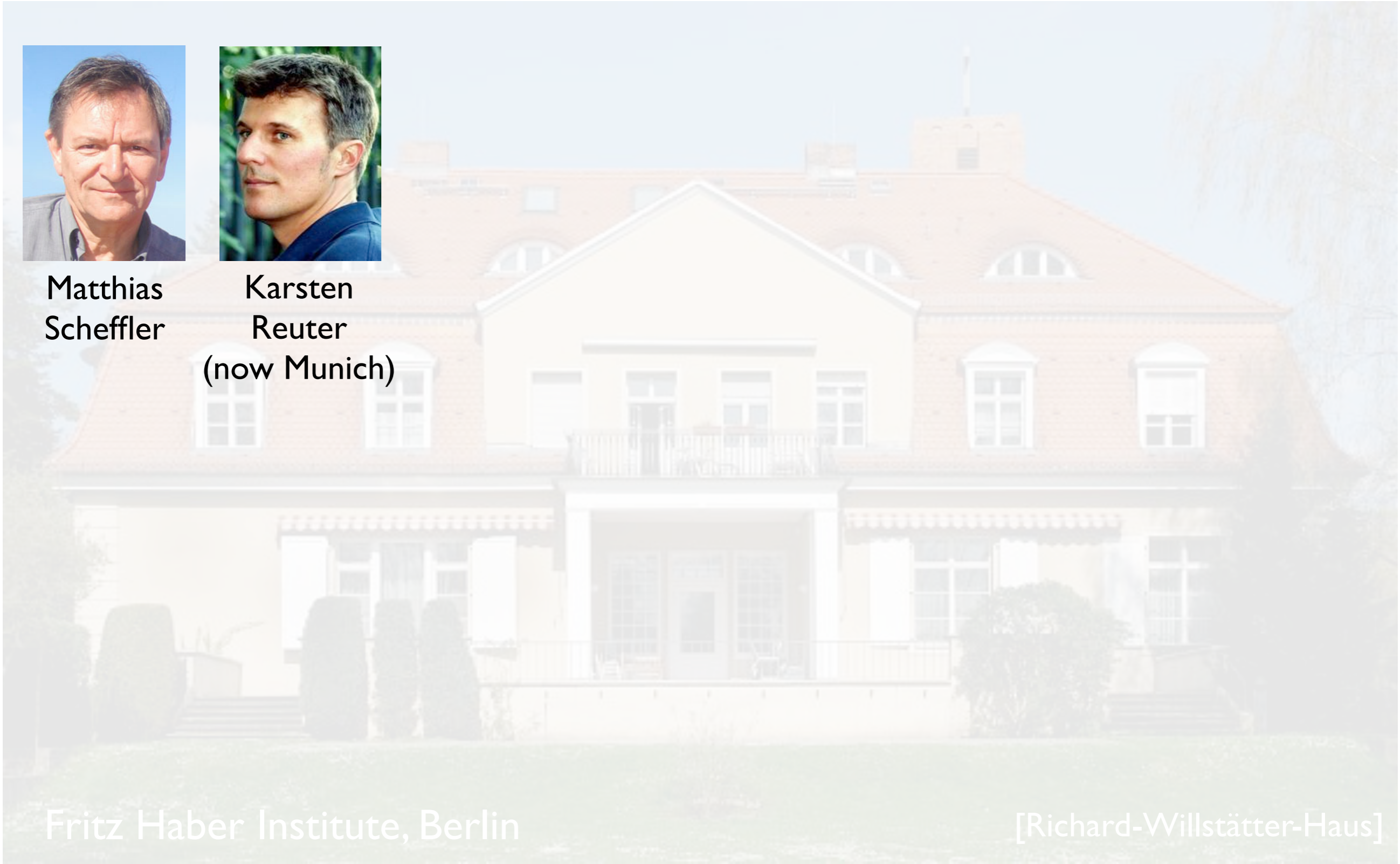
People



Matthias
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Karsten
Reuter
(now Munich)



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Patrick
Rinke

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People



Matthias
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Patrick
Rinke

... FHI-aims - MANY contributors:

[Xinguo Ren](#), Ville Havu, Paula Havu, Ralf Gehrke, Rainer Johanni, Andreas Dolfen, Felix Hanke, Stefan Gutzeit, Andrea Sanfilippo, Luca Ghiringhelli, Mariana Rossi, Alex Tkatchenko, Sergey Levchenko, Matthias Gramzow, Benedikt Biedermann, Aloysius Soon, Mina Yoon, Jörg Meyer, Christian Carbogno, Norbert Nemec, Fabio Caruso, Sucismita Chutia, Franziska Schubert, Jürgen Wieferink, Simiam Ghan, Viktor Atalla, Matti Ropo, Ferdinand Evers, Alexej Bagrets, Fabio Della Sala, Eduardo Fabiano, Heiko Appel, Daniel Berger, Oliver Hofmann, Yong Xu, Marco Casadei, Klaus Reuter, Andreas Marek, Werner Jürgens, Igor Ying Zhang, Jan Kloppenburg, Franz Knuth, Xin-Zheng Li, ...

Dr. Rainer Johanni (1959-2012)



Dr. Rainer Johanni

* 20. 6. 1959 † 5. 6. 2012

Reminder - what did we want from a new code

Cover essentially the entirety of materials / chemistry

- ◆ Light main-group elements (H, C, N, O, ...)
- ◆ 3*d* transition metals & compounds
- ◆ 4*d* / 5*d* transition metals
- ◆ *f* elements
- ◆ ...

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Periodic and non-periodic systems on equal footing

All-electron method

Reliable production methods (DFT-LDA, -GGA, hybrids)

Validation “beyond DFT” (RPA, GW, Hartree-Fock+MP2, ...)

Scalable from laptop to massively parallel platforms

Efficient (1000s of atoms), but do not sacrifice accuracy!

Key choice: Numeric atom-centered basis functions

$$\varphi_{i[lm]}(\mathbf{r}) = \frac{u_i(r)}{r} \cdot Y_{lm}(\Omega)$$

- $u_i(r)$: Flexible choice - “Anything you like.”

Many popular implementations:
DMol³ (Delley), FPLO (Eschrig *et al.*), PLATO (Horsfield *et al.*),
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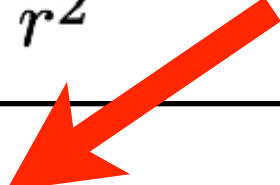
$$\left[-\frac{1}{2} \frac{d^2}{dr^2} + \frac{l(l+1)}{r^2} + v_i(r) + v_{\text{cut}}(r) \right] u_i(r) = \epsilon_i u_i(r)$$

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- free-atom like: $v_i(r) = v_{\text{free atom}}^{\text{DFT}}(r)$
- Hydrogen-like: $v_i(r) = z/r$
- free ions, harm. osc. (Gaussians), ...

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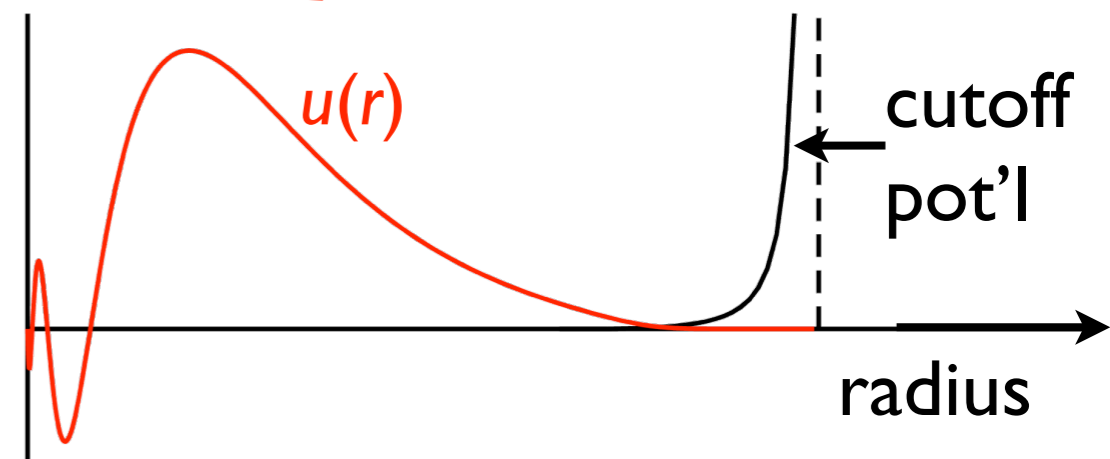
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- $u_i(r)$: Flexible choice - “Anything you like.”
 - Localized; “naturally” all-electron
 - The choice of efficient and of enough radial functions is obviously important
 - We have a basis set library for all elements (1-102), from fast qualitative to meV-converged (total energy, LDA/GGA) calculations - efficient and accurate approach

V. Blum, R. Gehrke, F. Hanke, P. Havu, V. Havu, X. Ren, K. Reuter and M. Scheffler,
“Ab Initio Molecular Simulations with Numeric Atom-Centered Orbitals”,
Computer Physics Communications **180**, 2175-2196 (2009)

Basis set “language” in FHI-aims

	H	C	O	Au
minimal	1s	[He]+2s2p	[He]+2s2p	[Xe]+6s5d4f
Tier 1	H(2s,2.1)	H(2p,1.7)	H(2p,1.8)	Au ²⁺ (6p)
	H(2p,3.5)	H(3d,6.0)	H(3d,7.6)	H(4f,7.4)
		H(2s,4.9)	H(3s,6.4)	Au ²⁺ (6s)
				H(5g,10)
				H(6h,12.8)
				H(3d,2.5)
Tier 2	H(1s,0.85)	H(4f,9.8)	H(4f,11.6)	H(5f,14.8)
	H(2p,3.7)	H(3p,5.2)	H(3p,6.2)	H(4d,3.9)
	H(2s,1.2)	H(3s,4.3)	H(3d,5.6)	H(3p,3.3)
	H(3d,7.0)	H(5g,14.4)	H(5g,17.6)	H(1s,0.45)
		H(3d,6.2)	H(1s,0.75)	H(5g,16.4)
				H(6h,13.6)
Tier 3	H(4f,11.2)	H(2p,5.6)	O ²⁺ (2p)	H(4f,5.2)*
	H(3p,4.8)	H(2s,1.4)	H(4f,10.8)	H(4d,5.0)

Systematic hierarchy of basis (sub)sets, iterative *automated* construction based on *dimers*

“First tier”

“Second tier”

“Third tier”

...

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Systematic hierarchy of basis (sub)sets, iterative *automated* construction based on *dimers*

“First tier”

“Second tier”

... bundled together in “light”, “tight”, really_tight” standard settings with grids, Hartree potential, etc. - but every piece verifiable “by hand.”

H(3p,4.8) H(2s,1.4) H(4f,10.8) H(4d,5.0)

...

...

...

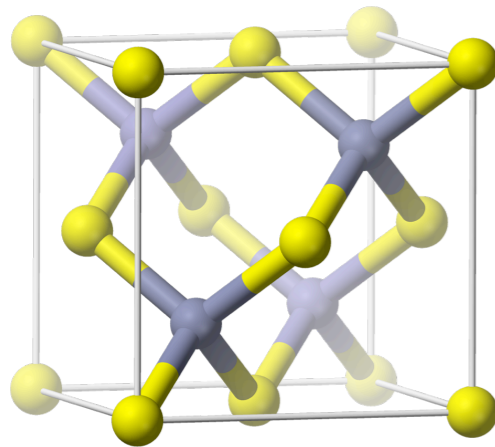
...

“Third tier”

...

Accuracy: Periodic hybrid functionals

Cohesive properties, bulk semiconductors



Zincblende GaAs

Si			
PBE0	a [Å]	B_0 [Mbar]	E_{coh} [eV]
FHI-aims, <i>tight</i>	5.439	0.99	4.553
Ref. [1]	5.433	1.00	4.555
HSE06			
FHI-aims, <i>tight</i>	5.446	0.98	4.527
Ref. [2]	5.435	0.98	4.582
GaAs			
HSE06			
FHI-aims, <i>tight</i>	5.695	0.71	3.150
Ref. [2]	5.687	0.71	3.149
Ge			
HSE06			
FHI-aims, <i>tight</i>	5.700	0.71	3.761
Ref. [3]	5.703	0.73	n/a

Essentially linear scaling
exchange operator:
Levchenko, Ren, Wieferink,
Johanni, Blum, Rinke,
Scheffler 2012

[1] J. Paier *et al.*, J. Chem. Phys. **124**, 154709 (2006).

[2] J. Paier *et al.*, J. Chem. Phys. **125**, 249901 (2006).

[3] A. Stroppa *et al.*, PRB **83**, 085201 (2011).

State of periodic hybrid functionals in FHI-aims

Levchenko, Ren, Wieferink, Johanni, Blum, Rinke, Scheffler 2012

$$E_{\text{xc}}[n] = (1 - \alpha)E_{\text{x}}^{\text{loc}}[n] + \alpha E_{\text{x}}^{\text{HF}} + E_{\text{c}}^{\text{loc}}[n]$$

Becke,
Burke,
Perdew,
Ernzerhoff,
others

- Long-standing “most wanted” feature in the code
- Now stable (single-point geometries) in an effective $O(N)$ implementation

State of periodic hybrid functionals in FHI-aims

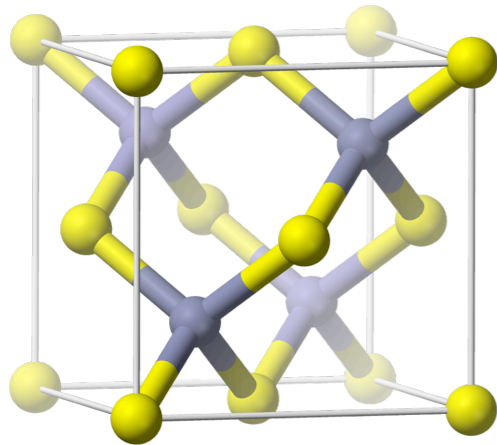
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HSE06 - GaAs, tight, 48 CPUs, **no symmetry use yet(!)**



Zincblende GaAs

Time for K_{ij} :

2 atom cell, k-grid 8x8x8

63 s

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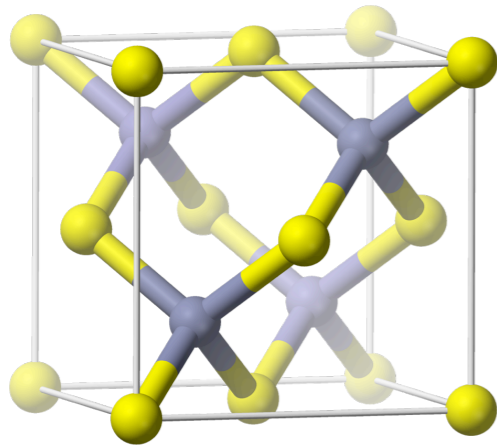
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	<u>Time for K_{ij}:</u>	<u>Scaling exponent</u>
2 atom cell, k-grid 8x8x8	63 s	
16 atom cell, k-grid 4x4x4	311 s	0.77
128 atom cell, k-grid 2x2x2	3629 s	1.18

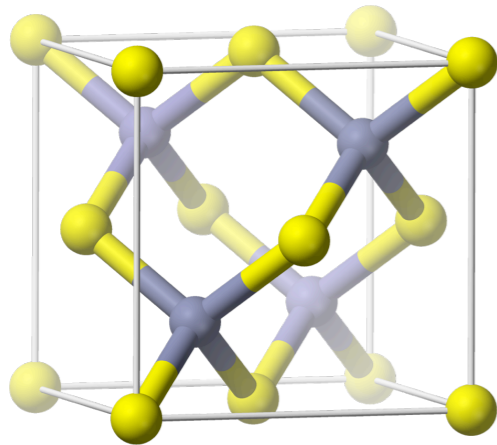
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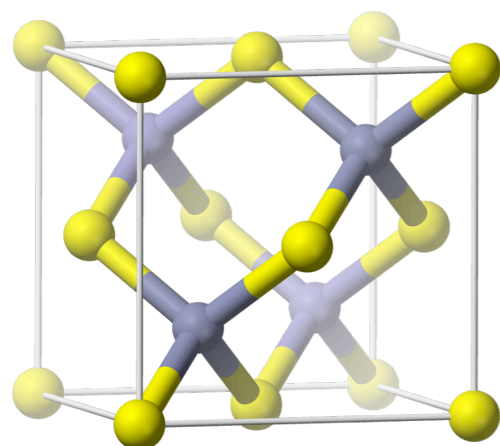
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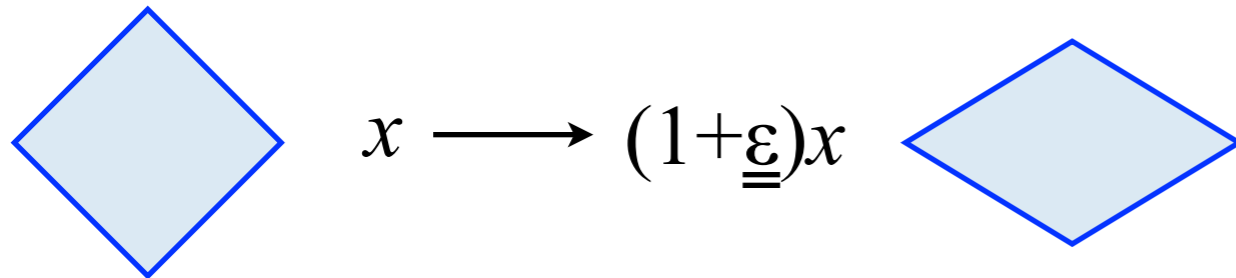
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Another “most wanted” feature: Analytical stress tensor

Global spatial distortion:



→ Strain derivative (stress tensor):

$$\sigma^{\mu\nu} = \frac{1}{V} \left. \frac{\partial E}{\partial \epsilon_{\mu\nu}} \right|_{\epsilon_{\mu\nu}=0}$$

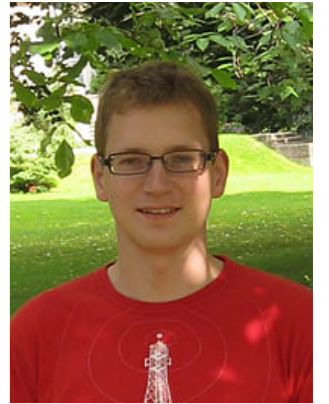
- Standard energy derivative for unit cell shape optimization, pressure
- Finite-difference implementation, cell shape optimization exist but costly
- Analytical implementation: (Somewhat) faster, but unfortunately a lot of terms



*Christian
Carbogno*



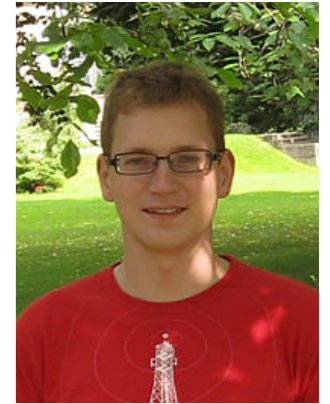
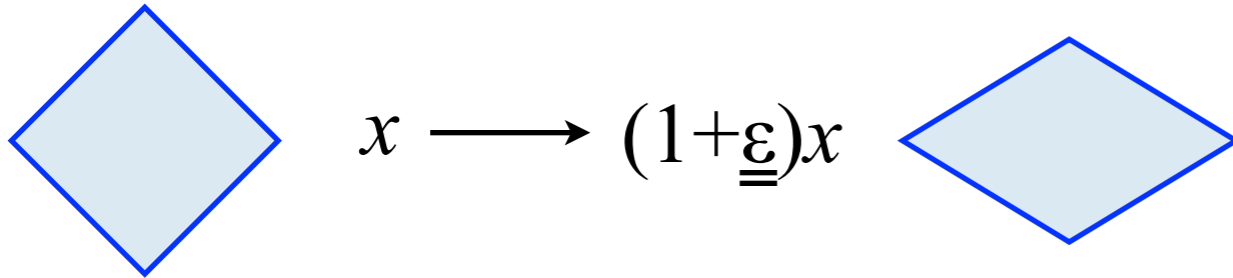
*Viktor
Atalla*



*Franz
Knuth*

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→ Strain derivative (stress

$$\sigma^{\mu\nu} = \frac{1}{V} \left. \frac{\partial E}{\partial \varepsilon_{\mu\nu}} \right|_{\varepsilon_{\mu\nu}}$$

- Standard energy derivative for shape optimization, pressure
- Finite-difference implementations for shape optimization exist but
- Analytical implementation: (faster, but unfortunately a lot

Formulas for analytical stress

$$\sigma_{ij} = \sigma_{ij}^{\text{HF}} + \sigma_{ij}^{\text{MP}} + \sigma_{ij}^{\text{Pulay}} + \sigma_{ij}^{\text{kin}} + \sigma_{ij}^{\text{Jac}}$$

$$\sigma_{ij}^{\text{HF}} = \frac{1}{2V} \sum_{\alpha, \beta \neq \alpha} \frac{\partial v_{\beta}^{\text{es,tot}}(|\mathbf{R}_{\alpha} - \mathbf{R}_{\beta}|)}{\partial R_i^{\alpha}} (\mathbf{R}_{\alpha} - \mathbf{R}_{\beta})_j$$

$$\begin{aligned} \sigma_{ij}^{\text{MP}} = & \frac{1}{V} \sum_{\alpha} \int_{\text{UC}} d\mathbf{r} \left[n(\mathbf{r}) - \frac{1}{2} n_{\text{MP}}(\mathbf{r}) \right] \frac{\partial v_{\alpha}^{\text{es,tot}}(|\mathbf{r} - \mathbf{R}_{\alpha}|)}{\partial r_i} (\mathbf{r} - \mathbf{R}_{\alpha})_j \\ & - \frac{1}{2V} \sum_{\alpha} \int_{\text{UC}} d\mathbf{r} \frac{\partial n_{\alpha}^{\text{MP}}(\mathbf{r} - \mathbf{R}_{\alpha})}{\partial r_i} (\mathbf{r} - \mathbf{R}_{\alpha})_j v_{\text{es,tot}}(\mathbf{r}) \end{aligned}$$

$$\sigma_{ij}^{\text{Pulay}} = \frac{2}{V} \sum_k \sum_{\alpha, l(\alpha)} \sum_{\beta, m(\beta)} f_k c_{kl} c_{km} \int_{\text{UC}} d\mathbf{r} \frac{\partial \varphi_l(\mathbf{r} - \mathbf{R}_{\alpha})}{\partial r_i} (\mathbf{r} - \mathbf{R}_{\alpha})_j [\hat{h}_{\text{KS}} - \varepsilon_k] \varphi_m(\mathbf{r} - \mathbf{R}_{\beta})$$

$$\sigma_{ij}^{\text{kin}} = \frac{1}{V} \sum_k \sum_{\alpha, l(\alpha)} \sum_{\beta, m(\beta)} f_k c_{kl} c_{km} \int_{\text{UC}} d\mathbf{r} \varphi_l(\mathbf{r} - \mathbf{R}_{\alpha}) (\mathbf{r} - \mathbf{R}_{\alpha})_j \left[\frac{\partial}{\partial r_i} \frac{\partial}{\partial r_j} \varphi_m(\mathbf{r} - \mathbf{R}_{\beta}) \right]$$

$$\sigma_{ij}^{\text{Jac}} = \frac{1}{V} \delta_{ij} \left[E_{\text{xc}}[n] - \int d\mathbf{r} n(\mathbf{r}) v_{\text{xc}}(\mathbf{r}) - \frac{1}{2} \int d\mathbf{r} n_{\text{MP}}(\mathbf{r}) v_{\text{es,tot}}(\mathbf{r}) \right]$$

Another “most wanted” feature: Analytical stress tensor

→ Strain derivative (stress tensor):

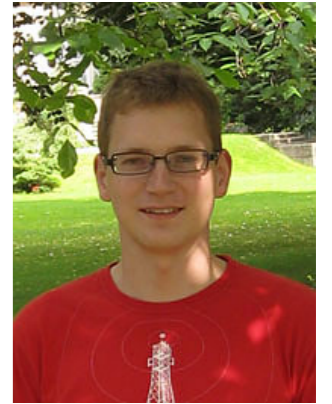
$$\sigma^{\mu\nu} = \frac{1}{V} \frac{\partial E}{\partial \varepsilon_{\mu\nu}} \Big|_{\varepsilon_{\mu\nu}=0}$$



*Christian
Carbogno*



*Viktor
Atalla*



*Franz
Knuth*

The analytical stress tensor is now here ...

→ Strain derivative (stress tensor):

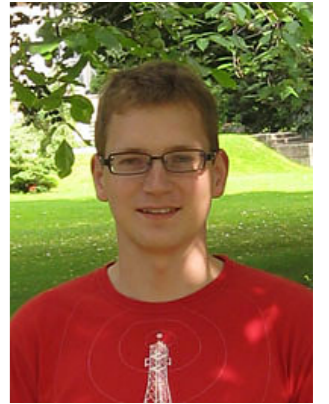
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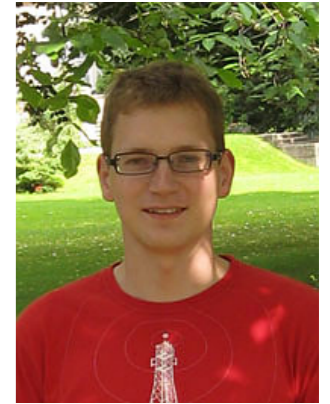
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Christian Bogno



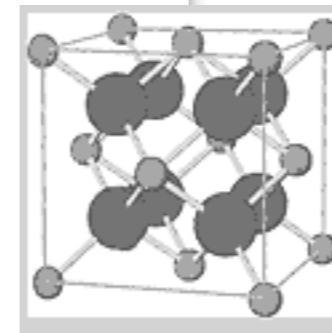
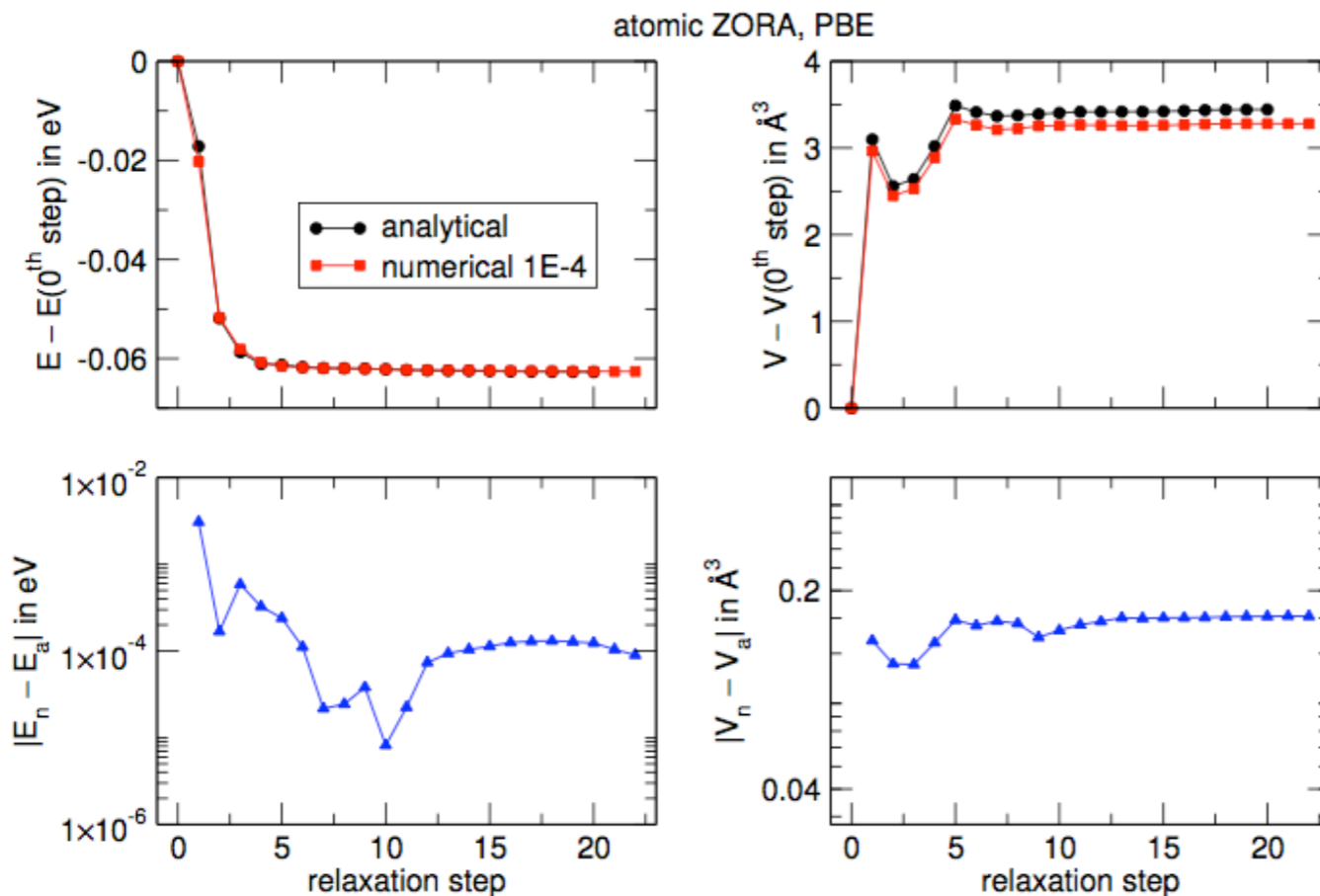
Viktor Atalla



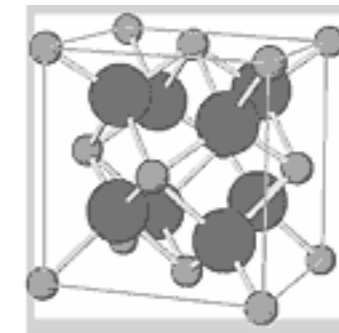
Franz Knuth

2) Relaxation of ZrO₂ monoclinic – 12 atoms

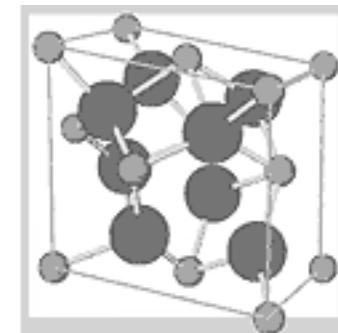
Starting geometry is near equilibrium



ZrO₂
cubic



ZrO₂
tetragonal



ZrO₂
monoclinic

<http://www.keramverband.de>

... and next?

→ Strain derivative (stress tensor):

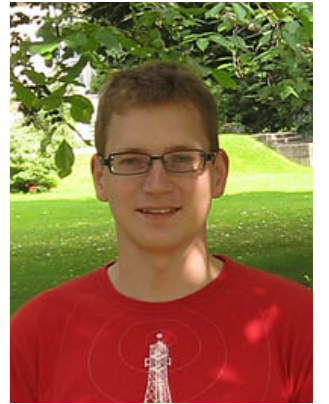
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*Christian
Carbogno*



*Viktor
Atalla*



*Franz
Knuth*

- ... works for LDA, GGA[+vdW]
- Unit cell shape relaxation, constant pressure thermostats?
- Next: Hybrid functionals (so far, stress by finite differences)
- Numerical improvements? (speed?)
- ... in general, second energy derivatives are needed.

“Beyond LDA / GGA / mGGA / hybrids”

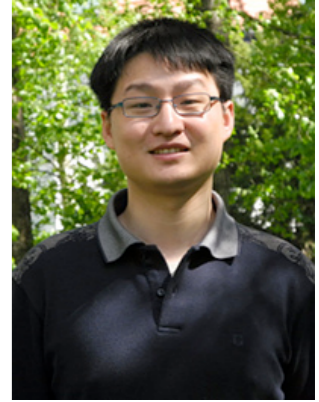
Current DFT may fail with or without warning,
even qualitatively (for structure).
How to go beyond?



*Xinguo
Ren*



*Fabio
Caruso*



*Igor
Zhang*

“Beyond LDA / GGA / mGGA / hybrids”

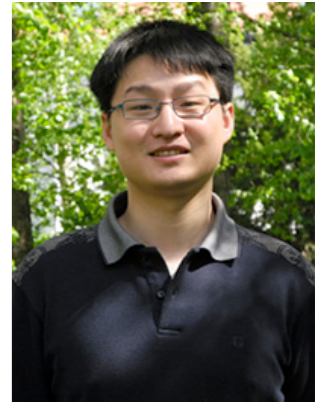
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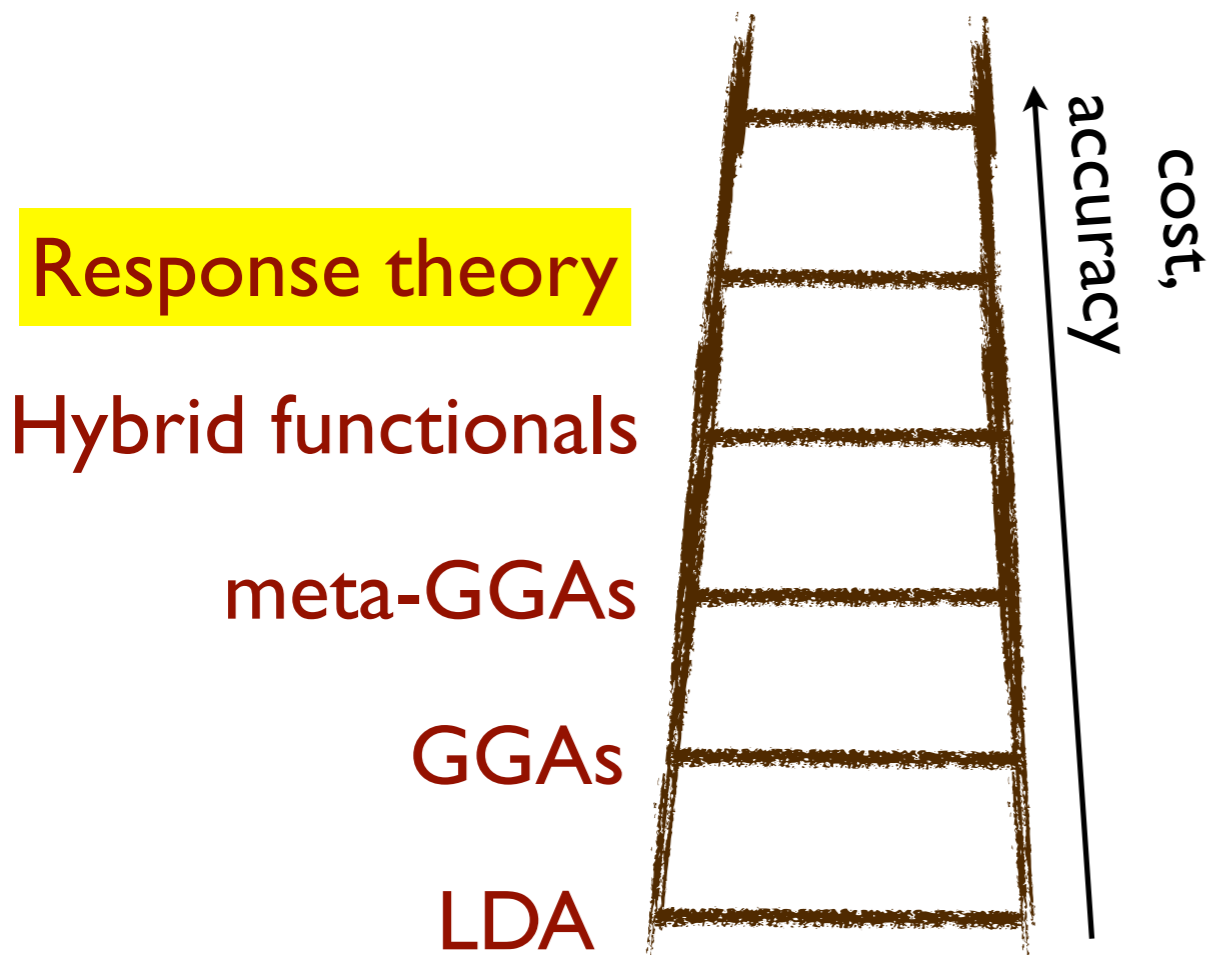


*Fabio
Caruso*



*Igor
Zhang*

“Perdew’s ladder”
to exact solution



“Beyond LDA / GGA / mGGA / hybrids”

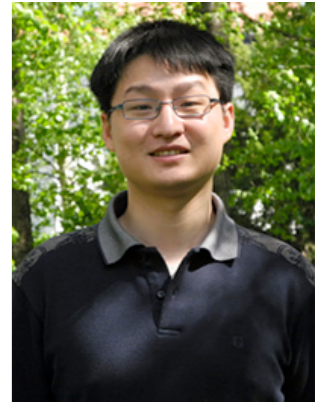
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Xinguo Ren



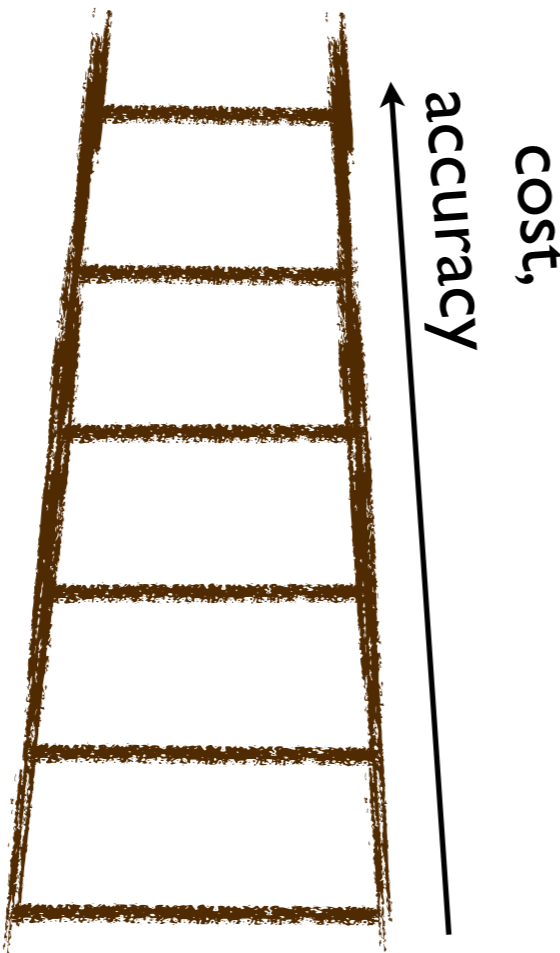
Fabio Caruso



Igor Zhang

“Perdew’s ladder”
to exact solution

Response theory
Hybrid functionals
meta-GGAs
GGAs
LDA



- Renormalized second-order perturbation theory (“rPT2”): RPA+rSE+SOSEX

Xinguo Ren, Wed. 11:15 h

- Self-consistent GW

Fabio Caruso, Wed. 11:50 h

- Doubly-hybrid functionals

Igor Zhang, Fri. 09:35 h

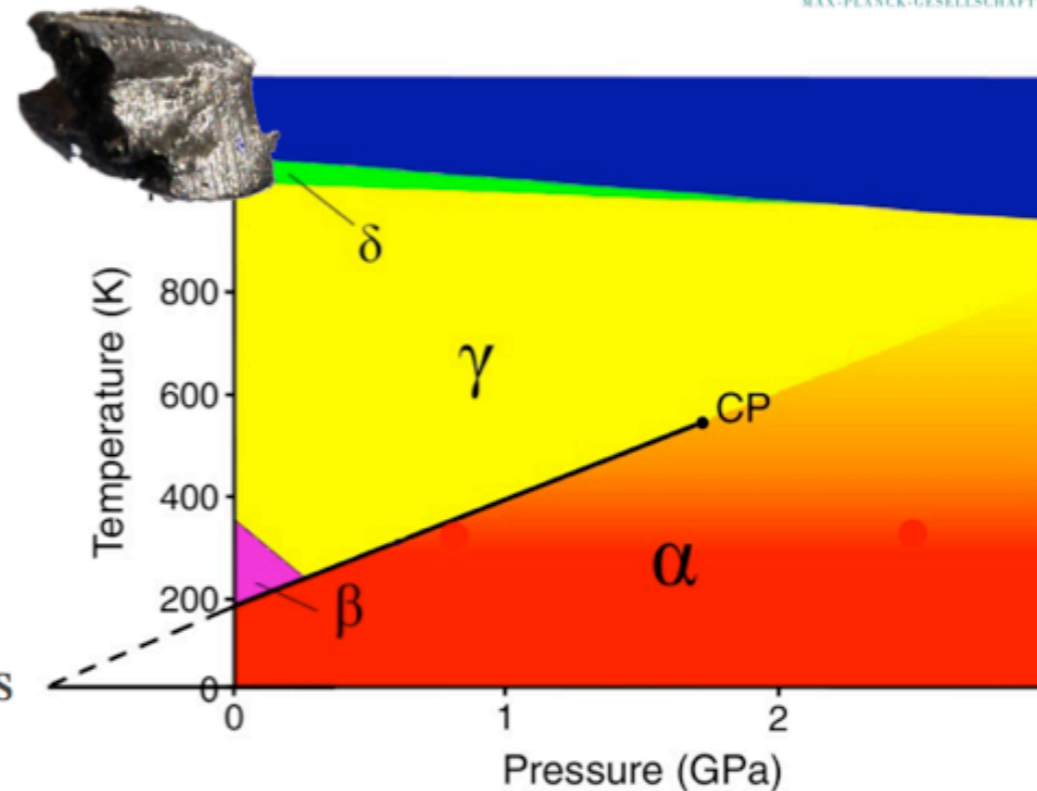
Where MBPT makes a difference: Ce

Casadei, Ren, Rinke, Rubio, Scheffler, PRL 2012

Cerium

- › First of the lanthanide elements
- › 58 electrons, one electron in the $4f$ states
- › Very abundant in the earth crust
- › Growing interest for application

- › Isostructural (fcc \rightarrow fcc) α - γ phase transition
- › Accompanied by volume collapse of 15-17% at room temperature
 - › High volume γ phase \rightarrow magnetic moments
 - › Low volume α phase \rightarrow paramagnetic



* Krisch M. et al., Proc. Natl. Acad. Sci. USA. 108, 9342 (2011)

- › LDA, GGA \rightarrow capable of describing only the α -phase, $4f$ electrons are always at the Fermi level (self-interaction problem)
- › SIC-LSD, LDA+U \rightarrow add non local potential to localized electrons, capable of describing only the γ -phase
- › LDA+DMFT \rightarrow results for high temperature



MAX-PLANCK-GESELLSCHAFT



Marco Casadei

Where MBPT makes a difference: Ce

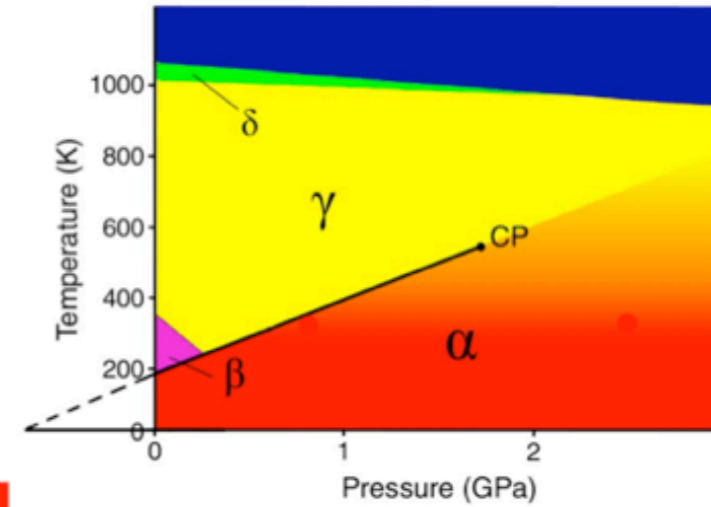
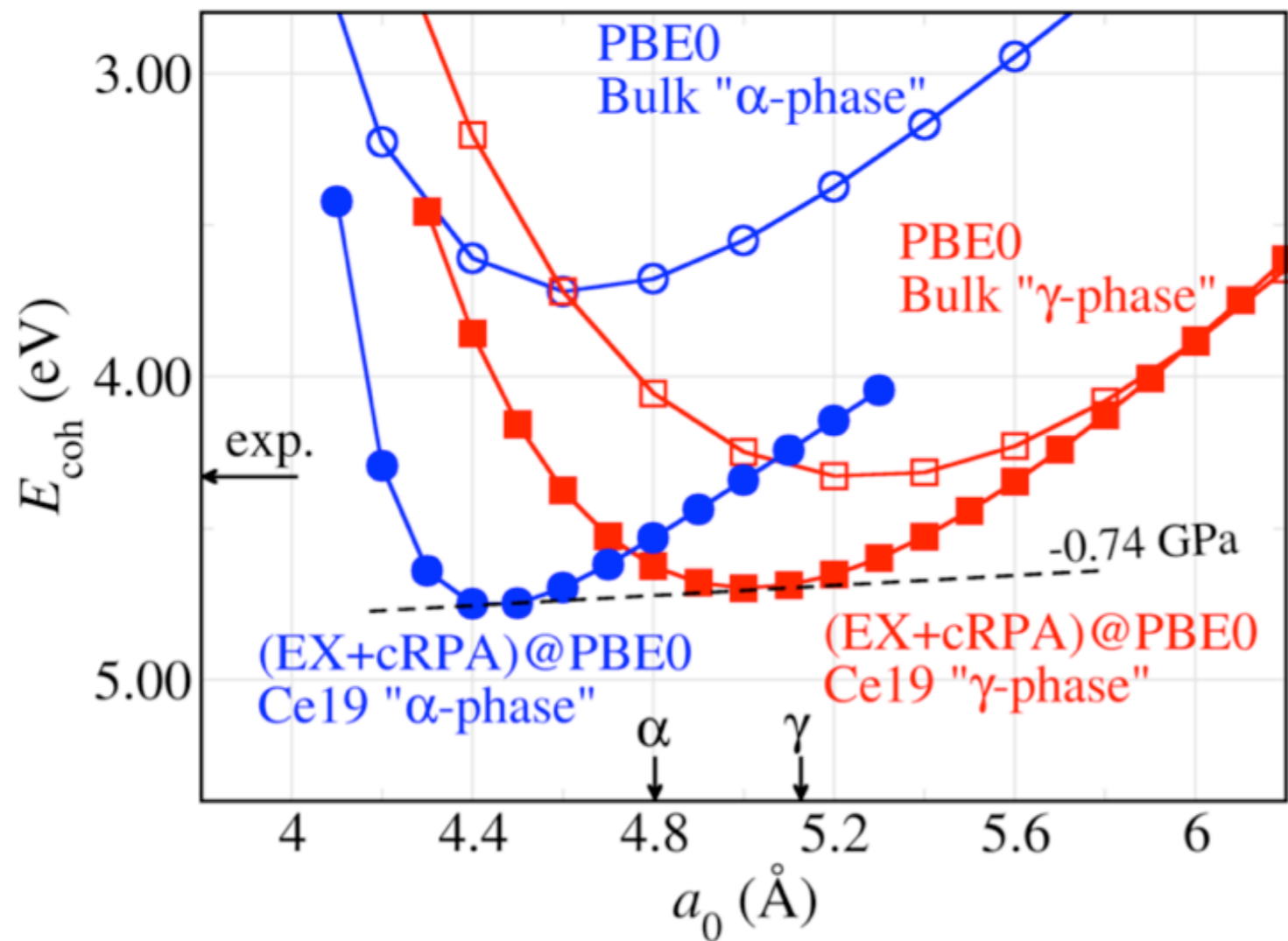
Casadei, Ren, Rinke, Rubio, Scheffler, PRL 2012

EX+cRPA@PBE0



Marco Casadei

› Correct phase ordering

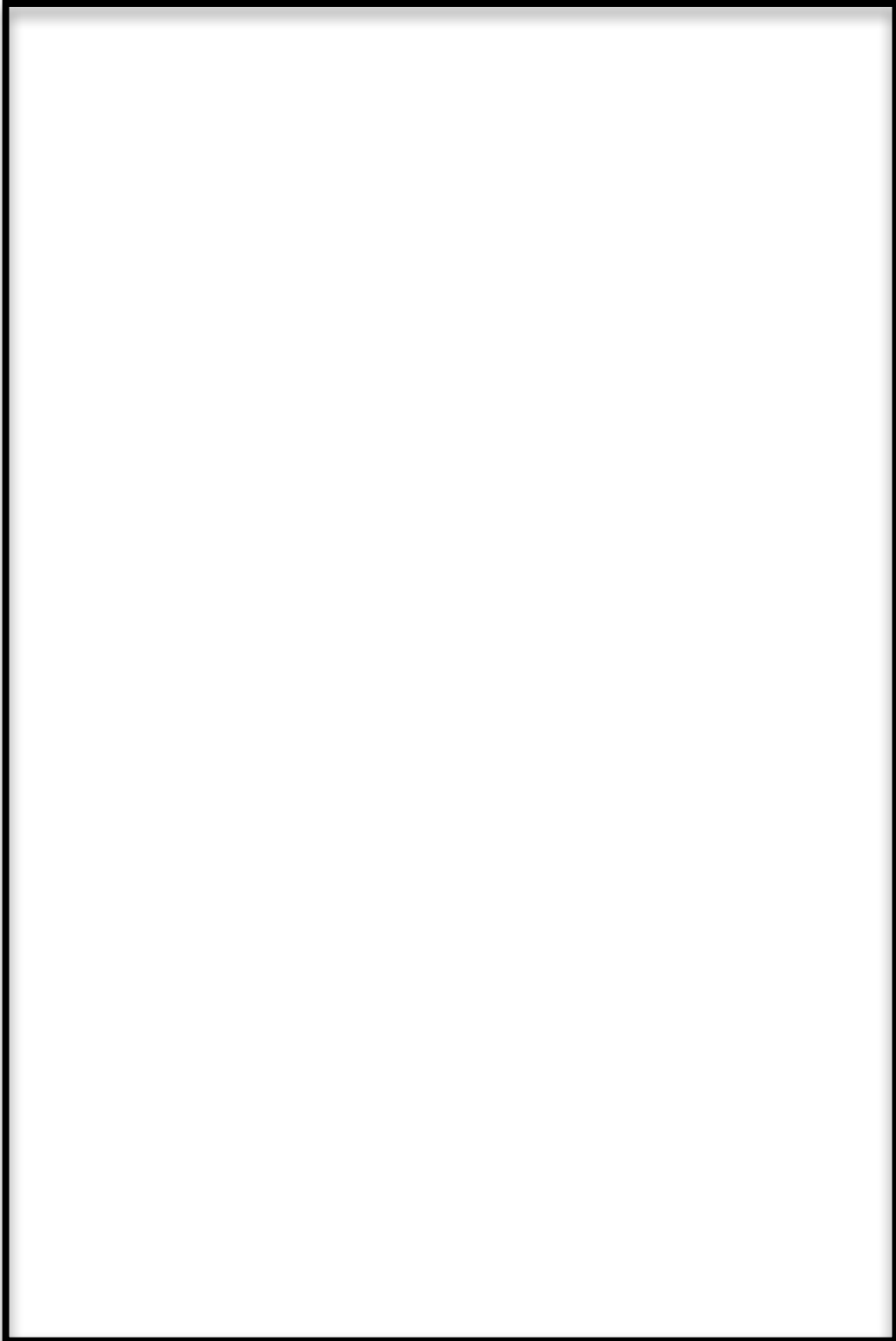


... and “beyond electronic structure”?

25 ps Born-Oppenheimer molecular dynamics, “tight”(!), DFT-PBE+vdW

Here:

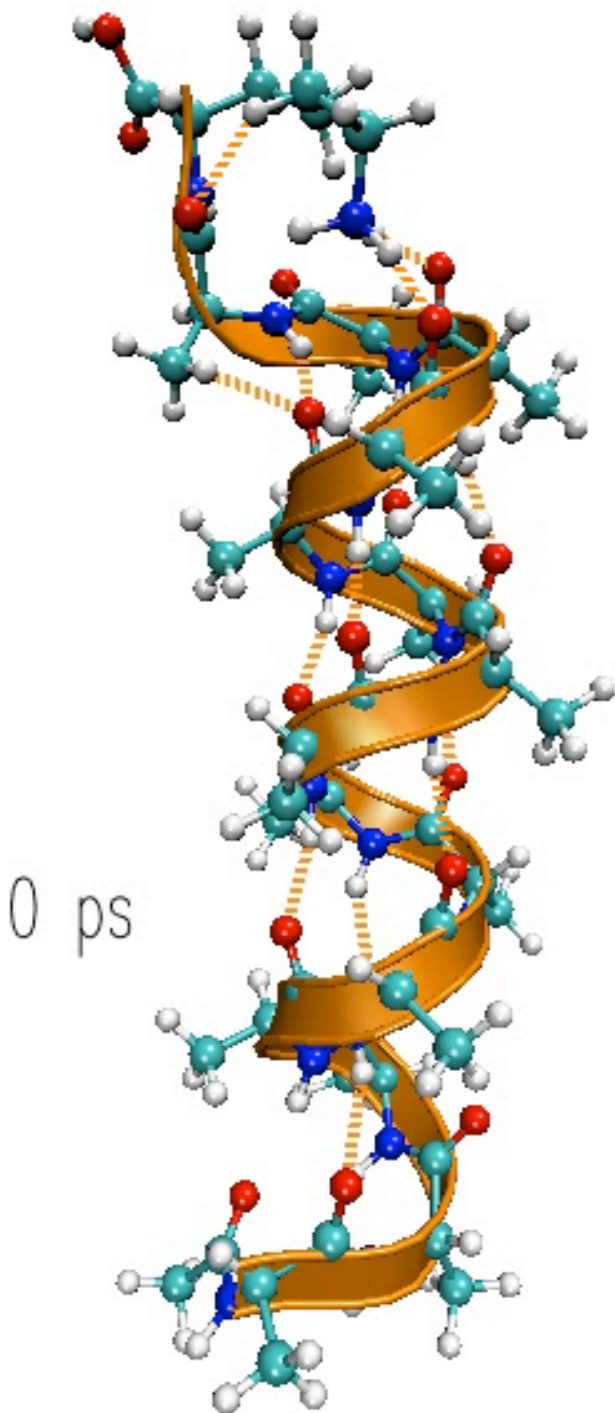
Infrared multiphoton dissociation
spectroscopy, FELIX free electron laser
Room temperature


$$I(\omega) \propto \omega^2 \int_{-\infty}^{\infty} dt \underbrace{\langle \vec{M}(t) \cdot \vec{M}(0) \rangle}_{\text{dipole-dipole time correlation function}} e^{i\omega t}$$

dipole-dipole time correlation function
see, e.g., *M.-P. Gaigeot, others*

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Rossi, **Blum**, Kupser, von Helden, Bierau, Pagel, Meijer, Scheffler, *J. Phys. Chem. Lett.* **1**, 3465 (2010)

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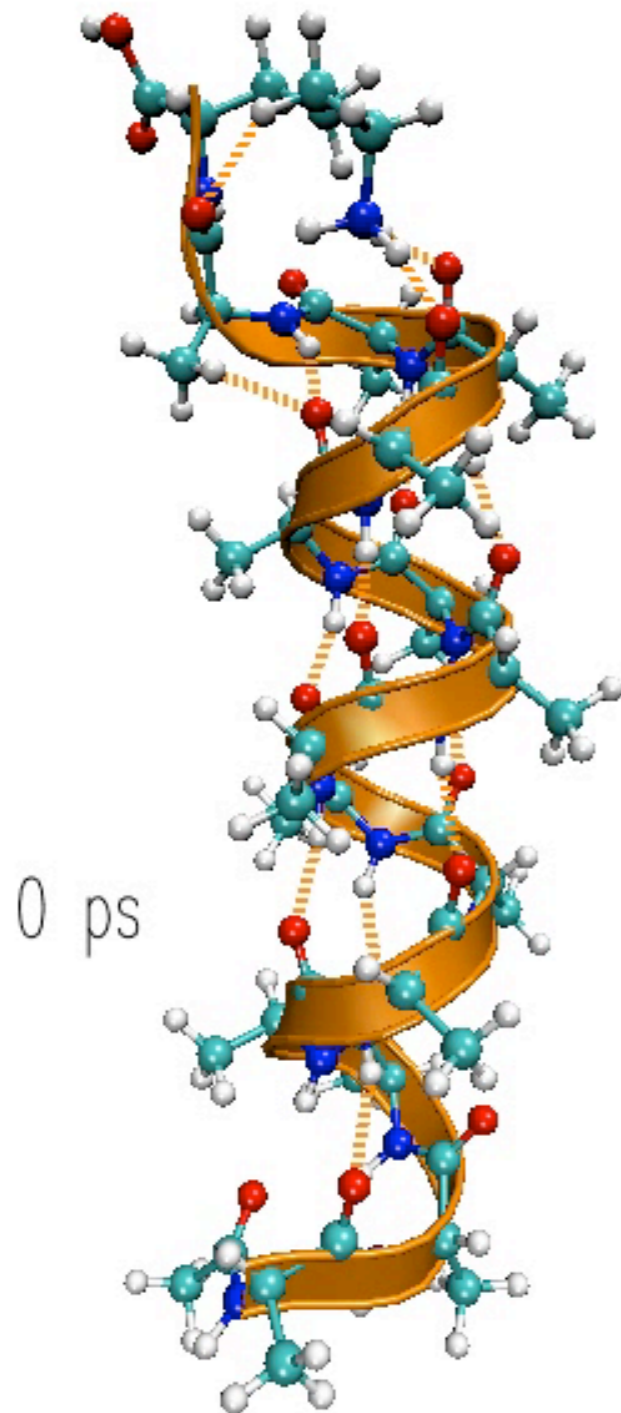
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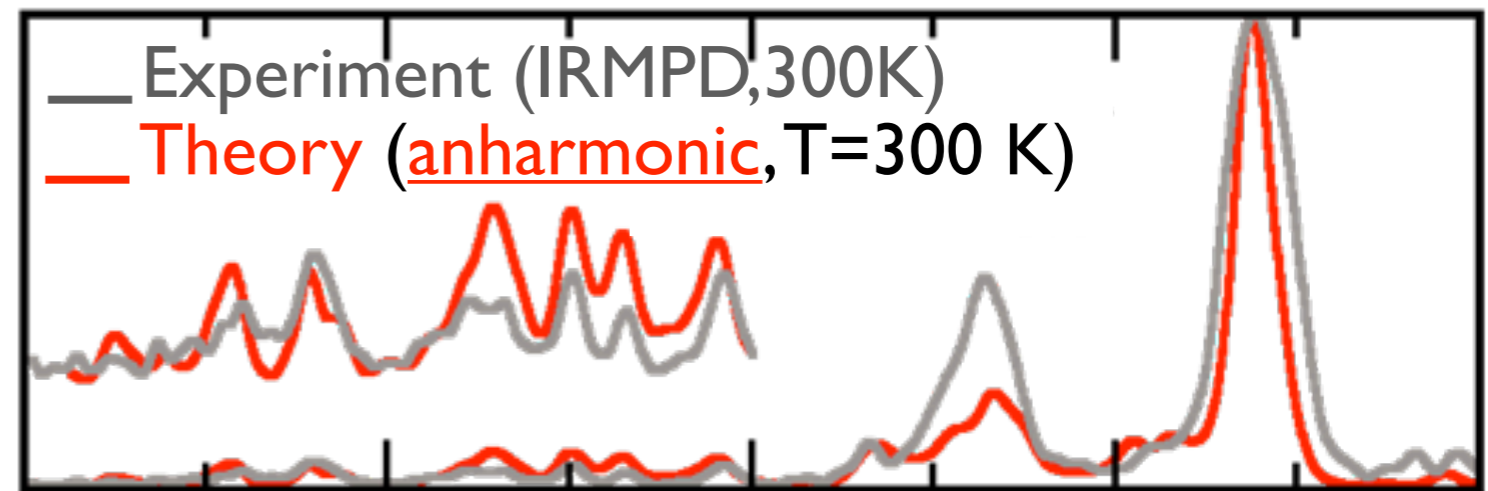
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Rossi, **Blum**, Kupser, von Helden, Bierau, Pagel, Meijer, Scheffler, *J. Phys. Chem. Lett.* **1**, 3465 (2010)

Theory: PBE+vdW, shifted, not scaled



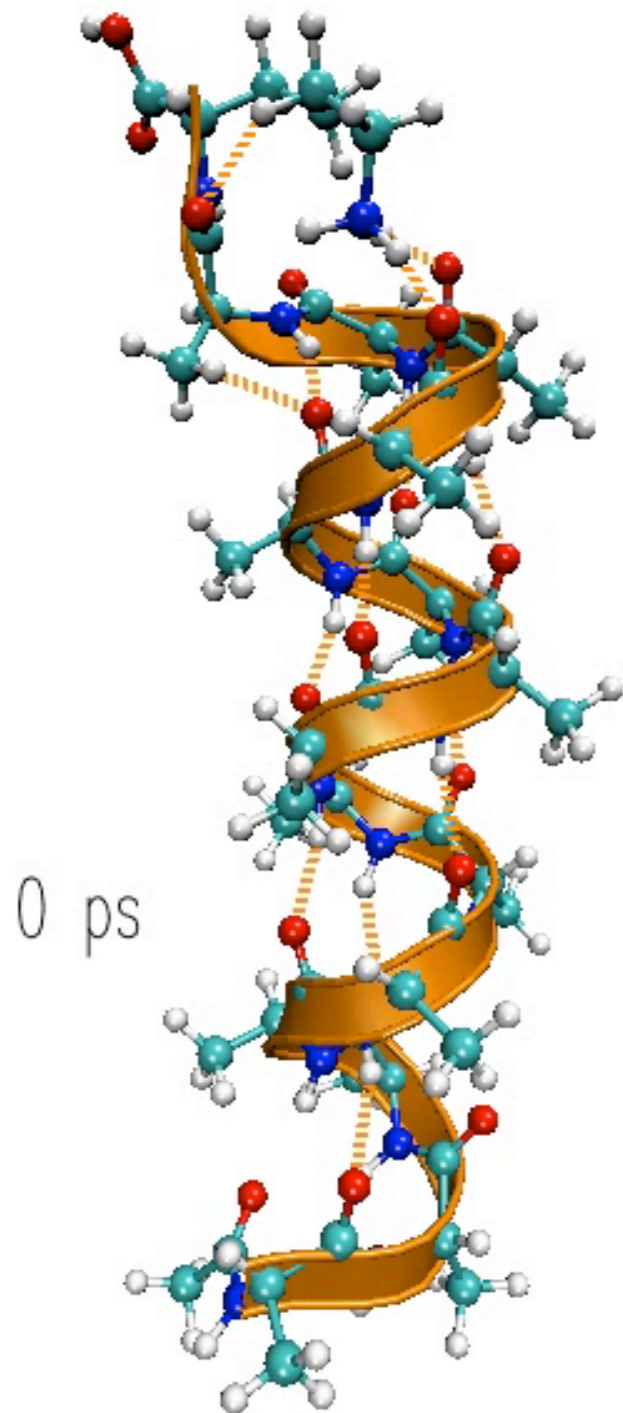
1000

wave number [cm⁻¹]

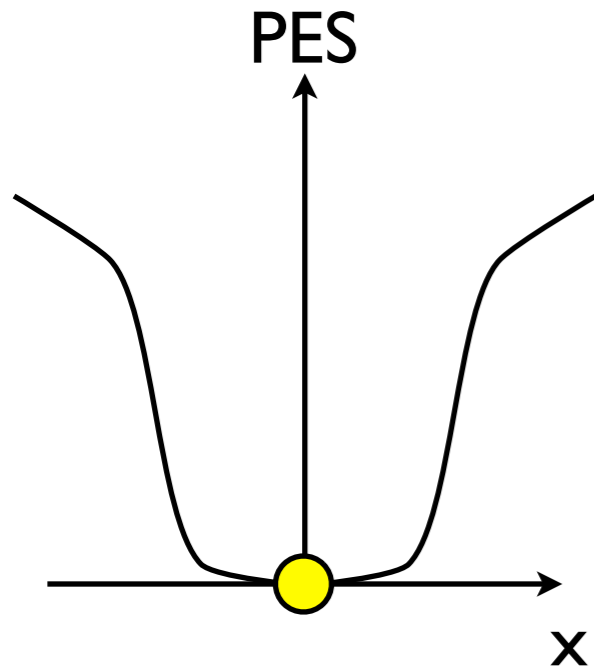
1800

$$I(\omega) \propto \omega^2 \int_{-\infty}^{\infty} dt \underbrace{\langle \vec{M}(t) \cdot \vec{M}(0) \rangle}_{\text{dipole-dipole time correlation function}} e^{i\omega t}$$

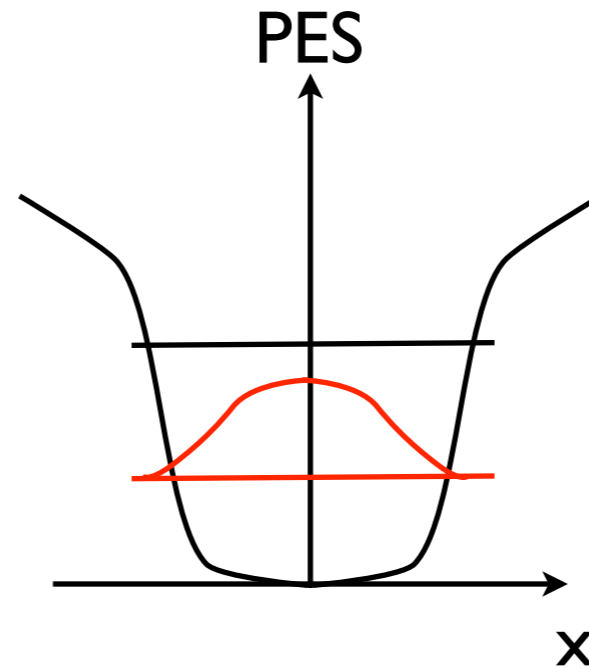
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see, e.g., M.-P. Gaigeot, others



But nuclei are not classical particles!



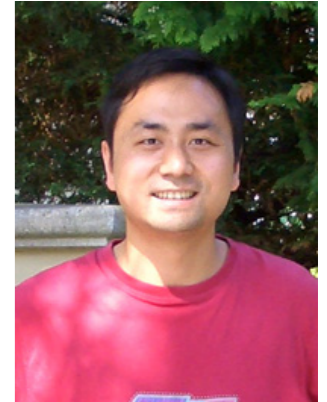
Classical system, $T=0$:
 $\Delta x=0, \Delta p=0$



Quantum system, $T=0$:
 $\Delta x \neq 0, \Delta p \neq 0$



Mariana
Rossi

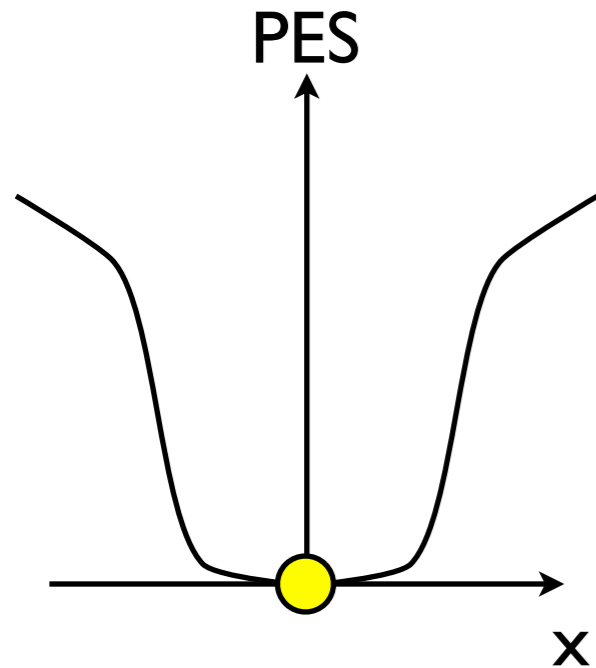


Xin-Zheng
Li

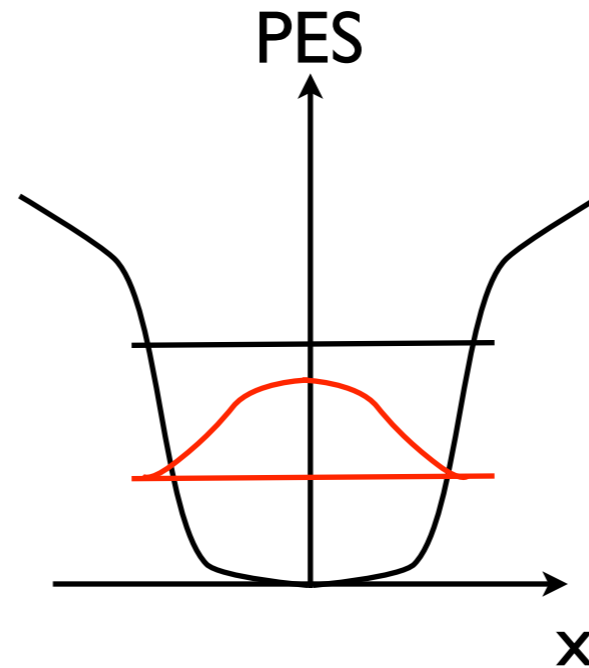
Especially affected: Hydrogen-bonded systems (protons!)

- Finite-temperature effects?
- Statistical averages? (free energy?)
- Dynamical quantities?

But nuclei are not classical particles!



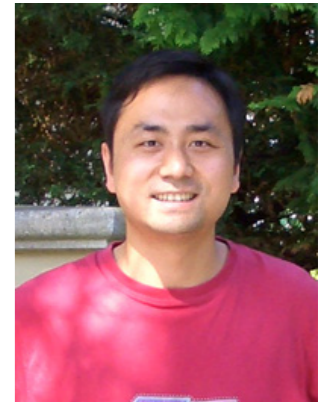
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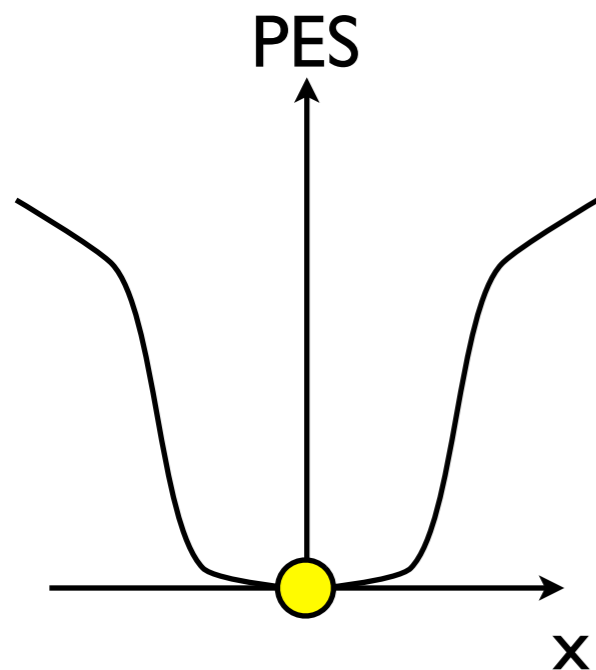
Xin-Zheng
Li

Two recent additions (any $T \neq 0$):

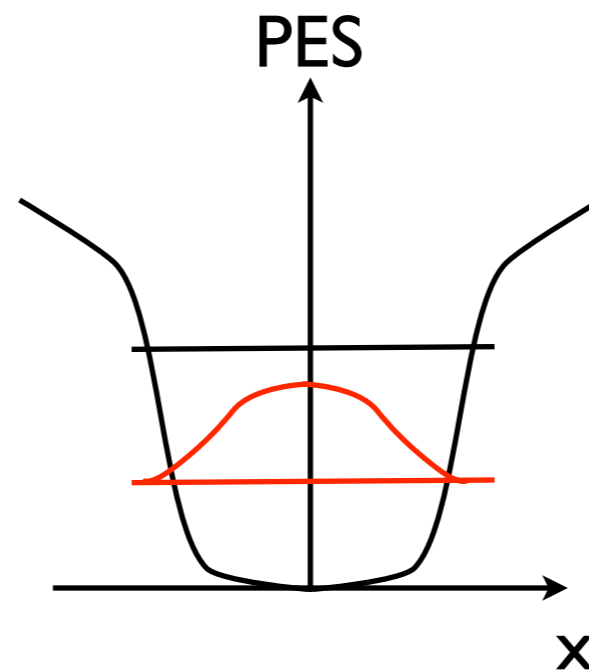
- 1) Colored-noise thermostat, keep quantum nuclear momentum distribution
(Parrinello, Ceriotti, Bussi)

Mariana Rossi, Thu. 09:35 h

But nuclei are not classical particles!



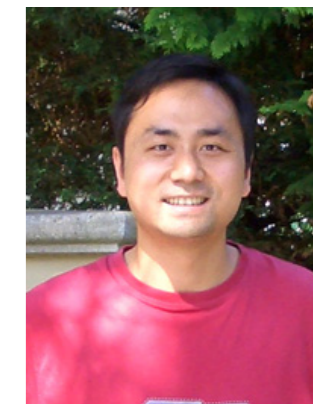
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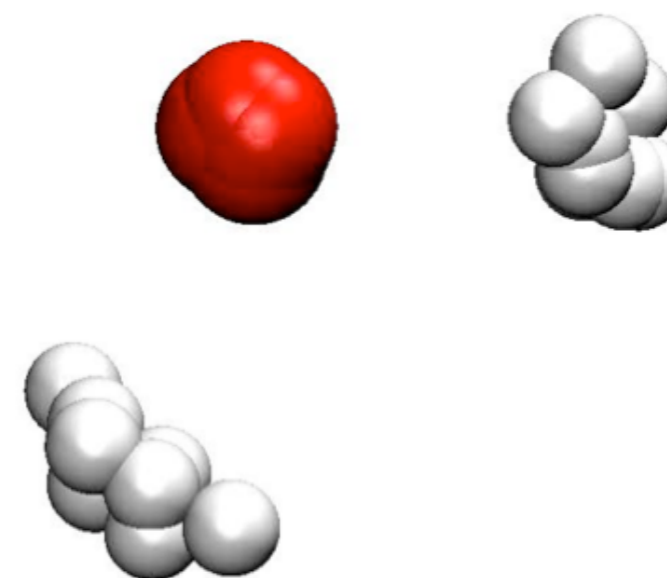
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Mariana Rossi, Thu. 09:35 h

- 2) Path-integral molecular dynamics

Xin-Zheng Li, Poster



Large-scale surface reconstruction: Graphene on SiC

Even for conceptually simple materials or molecules, the *relevant* structures can be uncomfortably large.



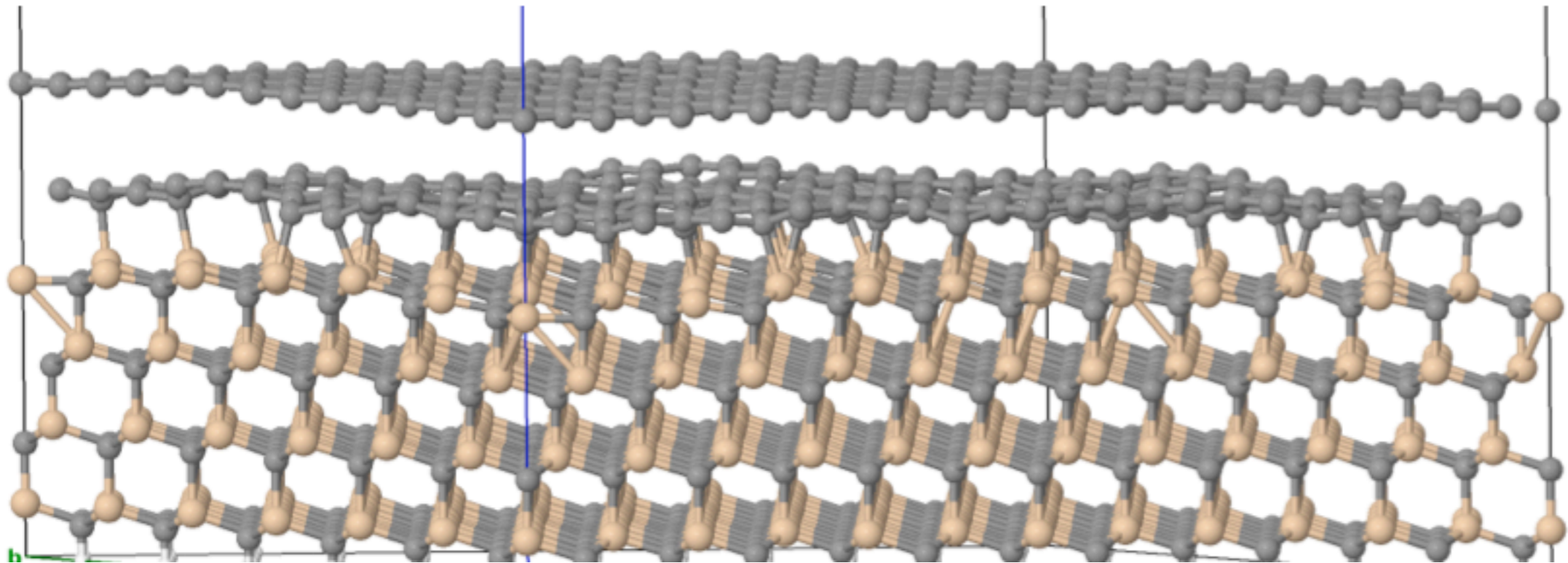
Lydia
Nemec

Large-scale surface reconstruction: Graphene on SiC

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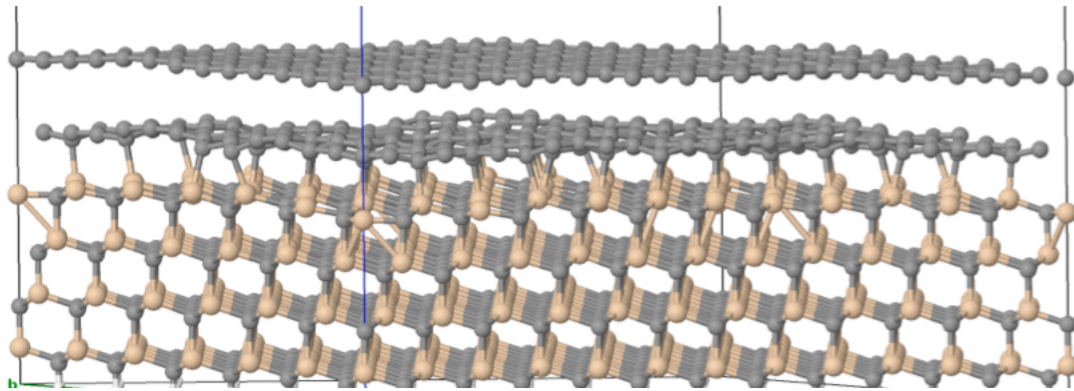
Lydia
Nemec



Graphene growth on SiC(0001)

Large-scale surface reconstruction: Graphene on SiC

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Graphene growth on SiC(0001)



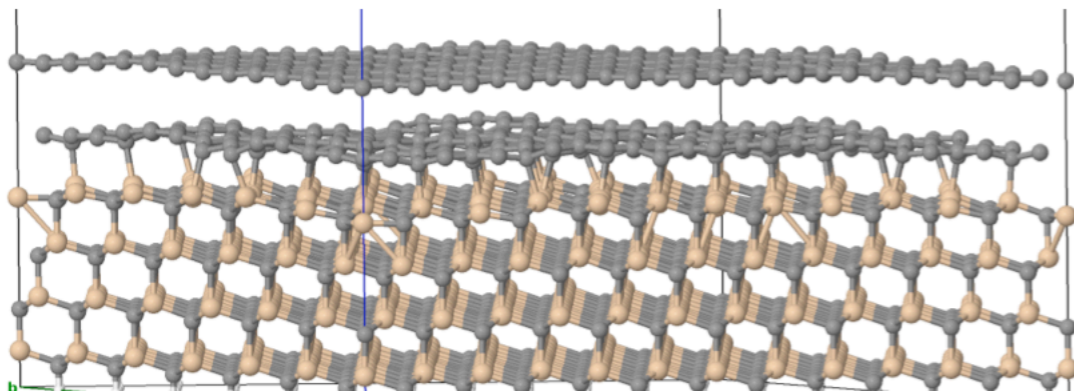
Lydia
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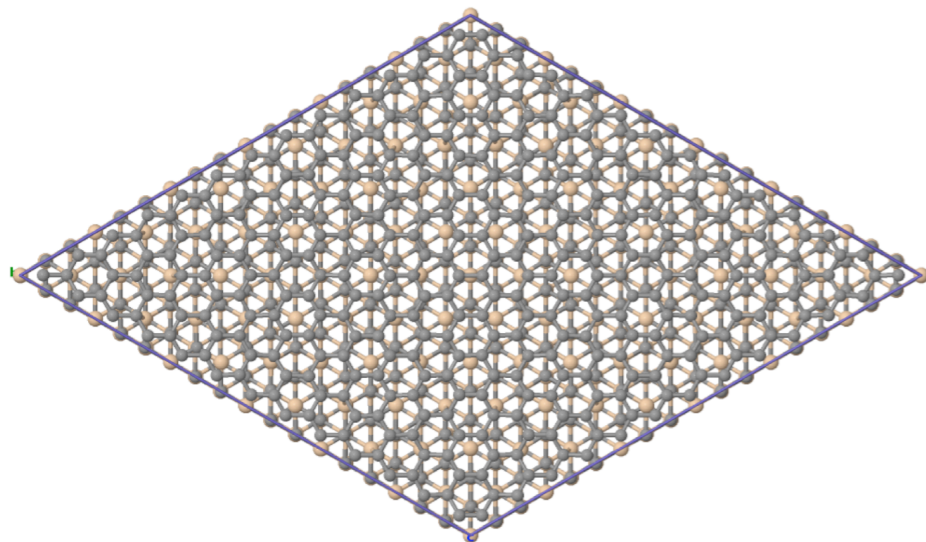


Lydia
Nemec



Graphene growth on SiC(0001)

Commensurate phase:
(13×13) graphene
on $(6\sqrt{3}\times 6\sqrt{3})\text{-R}30^\circ$ SiC

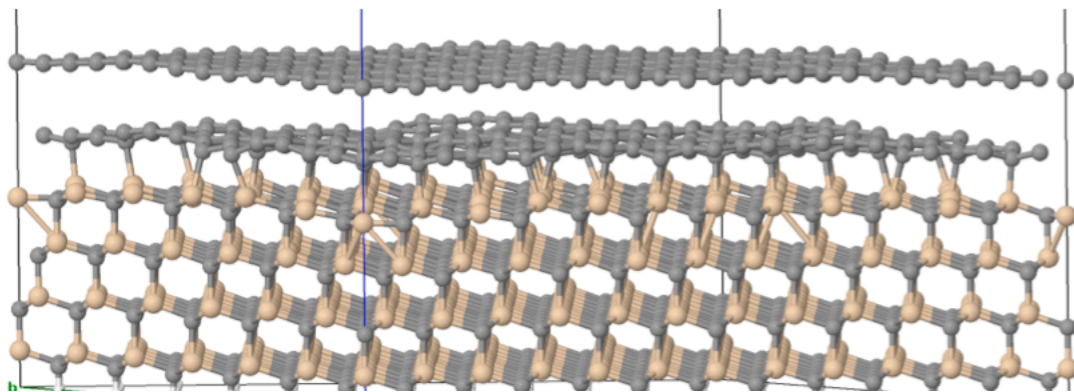


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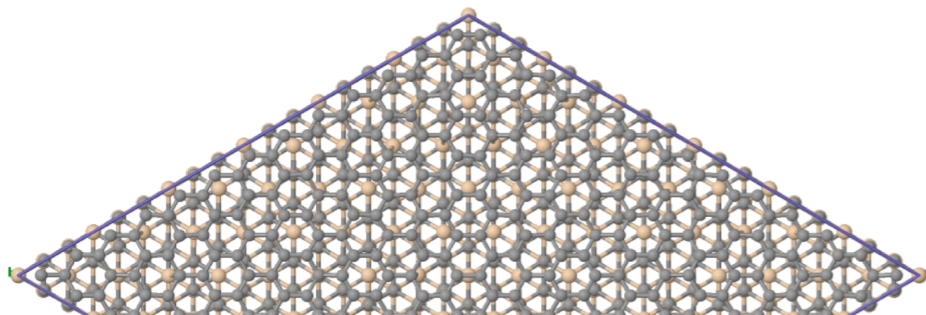


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Graphene growth on SiC(0001)

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Surface energy? Strain?
Electronic effect of interface?
van der Waals?

Epitaxial graphene on a semiconducting substrate: SiC

Many ways to grow graphene: Exfoliation, growth on metals ...

Among the oldest: *(van Bommel, Crombeen, van Tooren 1975)*

High-temperature sublimation of Si from SiC.

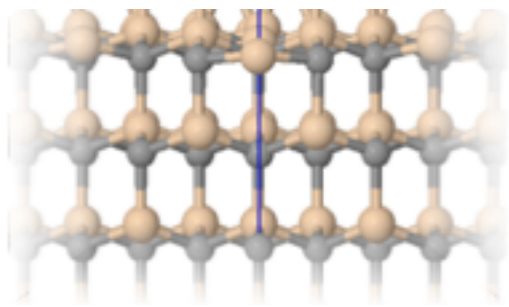
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On Si-side SiC:



SiC
(Si-side
surface)

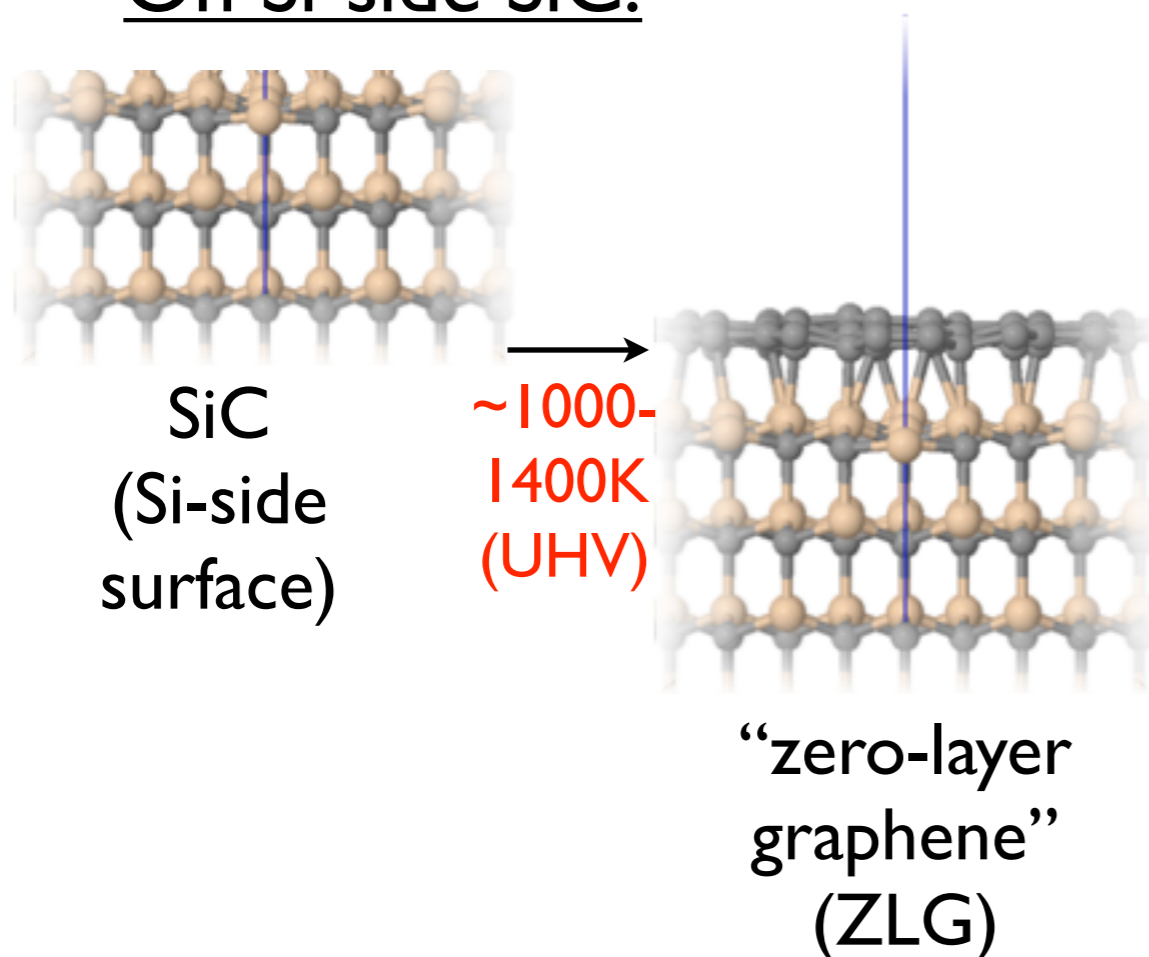
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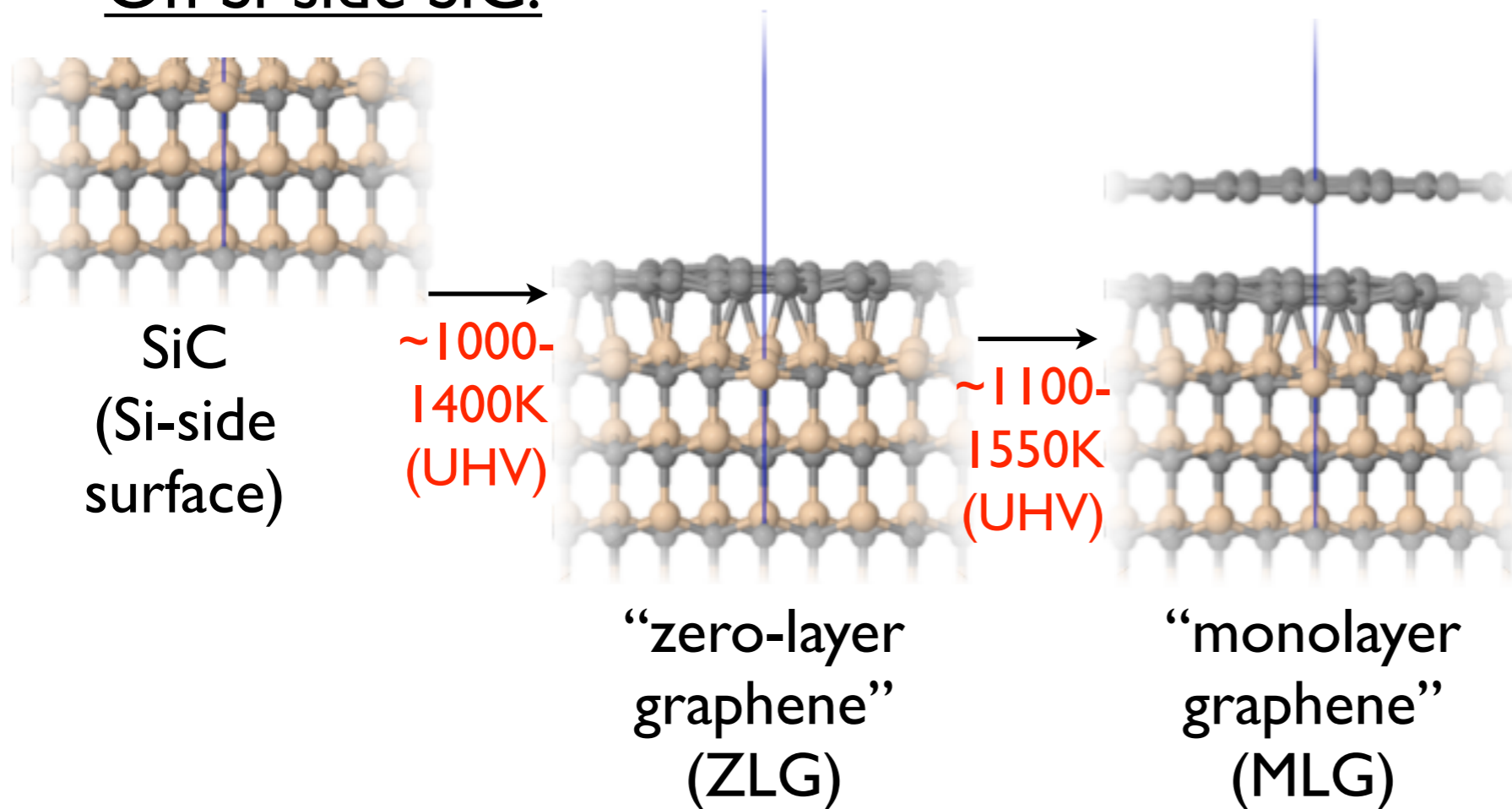
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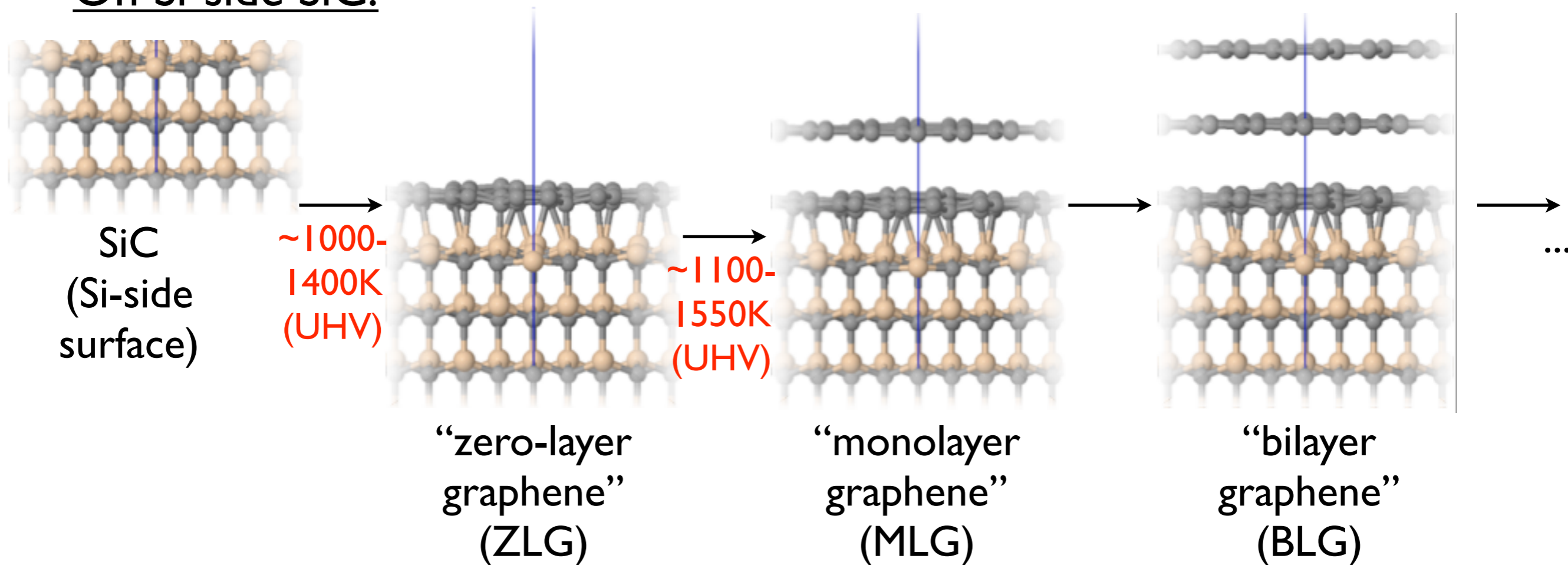
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... but how to understand the growth conditions?

- UHV: small terrace sizes, high defect densities
(e.g., de Heer et al., *PNAS* 108, 16900 (2011), many other groups)

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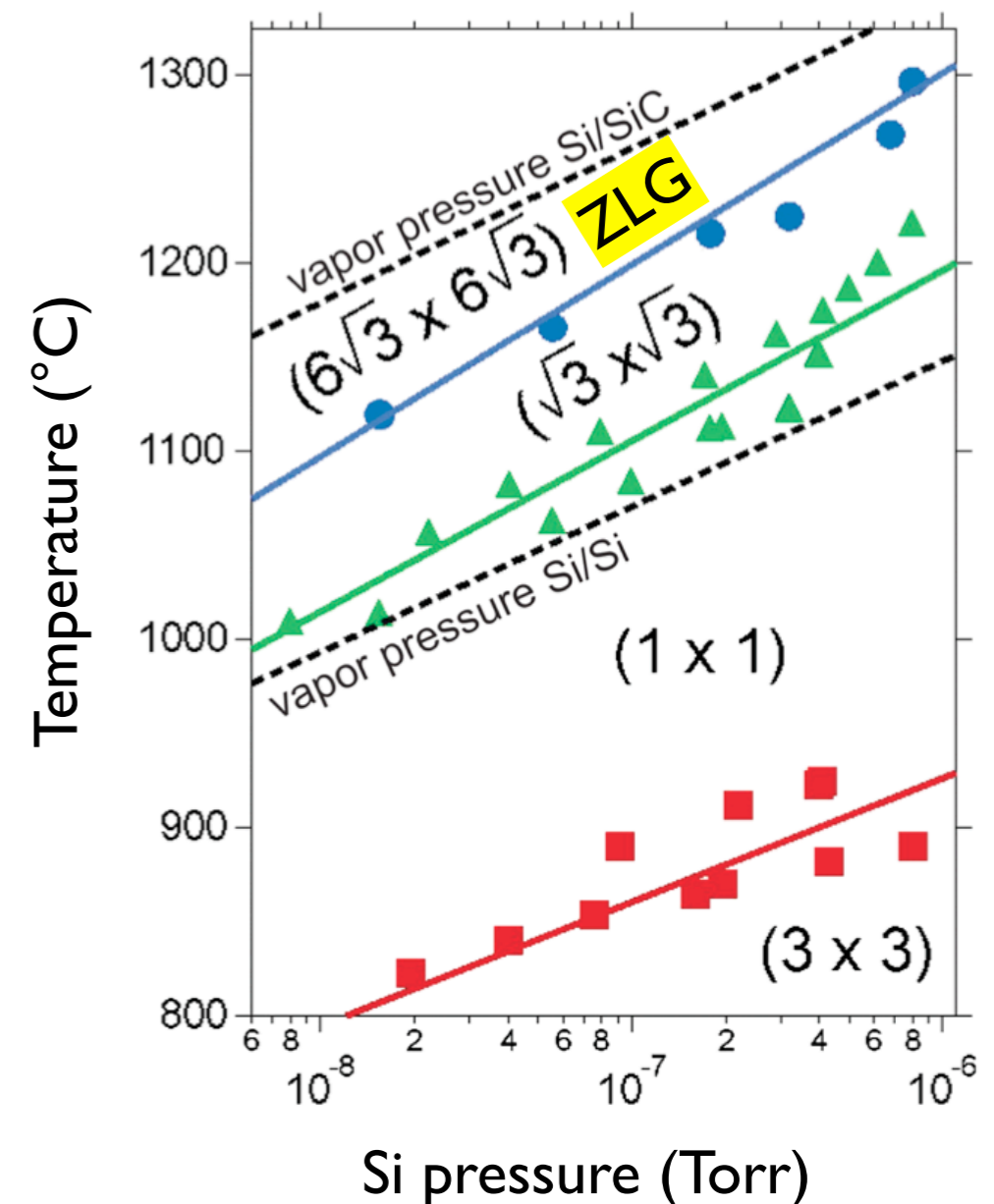
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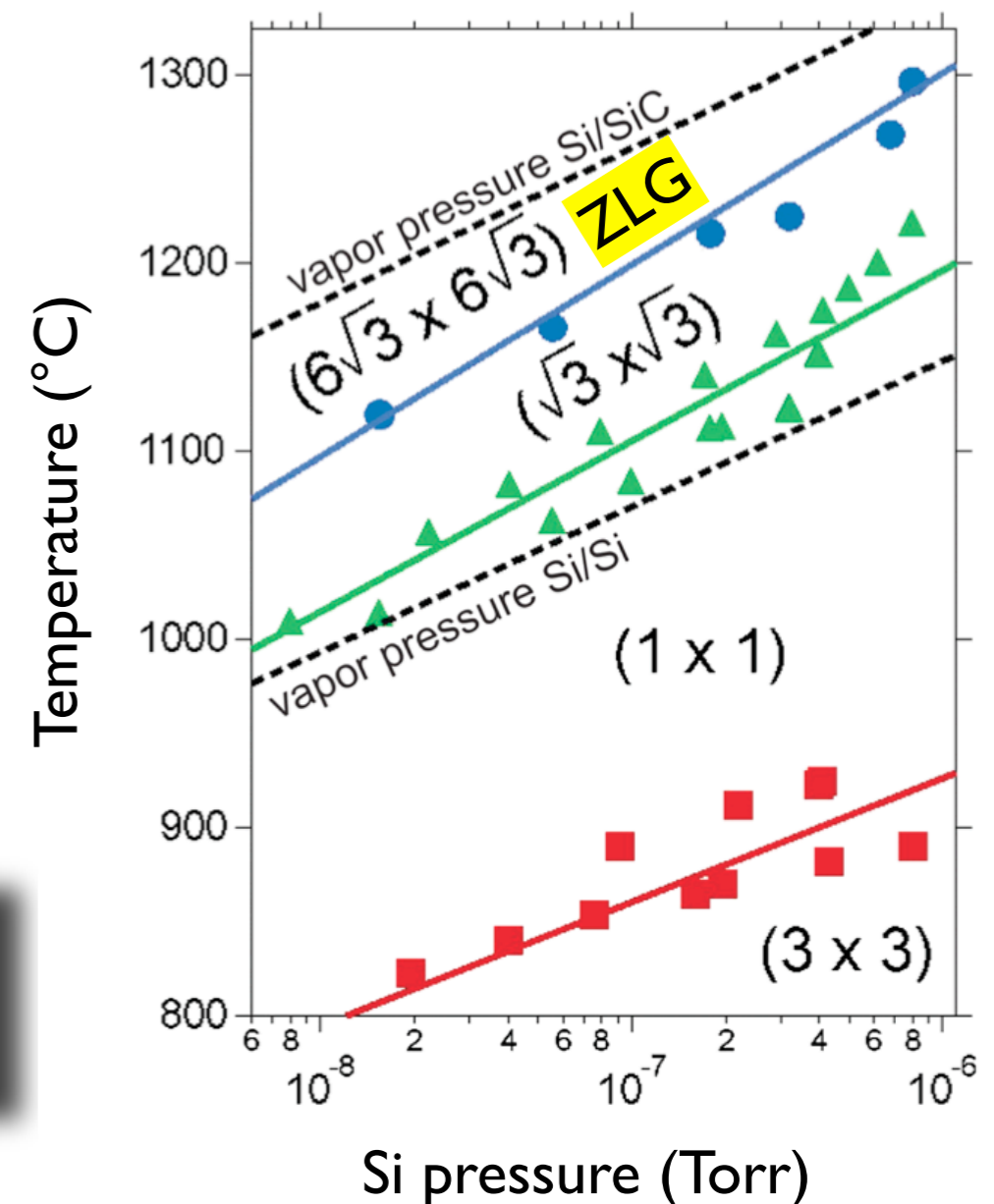
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- Reversible equilibrium conditions at least for ZLG!
Tromp, Hannon, Phys. Rev. Lett. 102, 106104 (2009)



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- UHV: small terrace sizes, high defect densities
(e.g., de Heer et al., PNAS 108, 16900 (2011), many other groups)
- Significantly improved morphology, higher growth T in Ar atmosphere
Emtsev et al., Nat. Mater. 8, 203 (2009)
- Large ordered areas from confined cavity at high T , Si background
de Heer et al., PNAS 108, 16900 (2011)
- Reversible equilibrium conditions at least for ZLG!
Tromp, Hannon, Phys. Rev. Lett. 102, 106104 (2009)

How close are MLG and BLG on SiC(111) to equilibrium phase growth?



Ab initio thermodynamics for Si-side graphene/SiC

Thermodynamic stability criterion for competing surface phases:

$$E_{\text{surf}} = \frac{1}{A} [E_{\text{slab}} - N_{\text{Si}}\mu_{\text{Si}} - N_{\text{C}}\mu_{\text{C}}]; \quad E_{\text{SiC}} = \mu_{\text{Si}} + \mu_{\text{C}}$$

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Total energies, full relaxation from first principles:

- six-bilayer SiC slabs + surface planes
- full relaxation, “tight” numerical settings (C: tier 2, Si: tier 1+gd)
- **Density functional: “PBE+vdW” [1]**

[1] Tkatchenko, Scheffler, *Phys. Rev. Lett.* 102, 073005 (2009)

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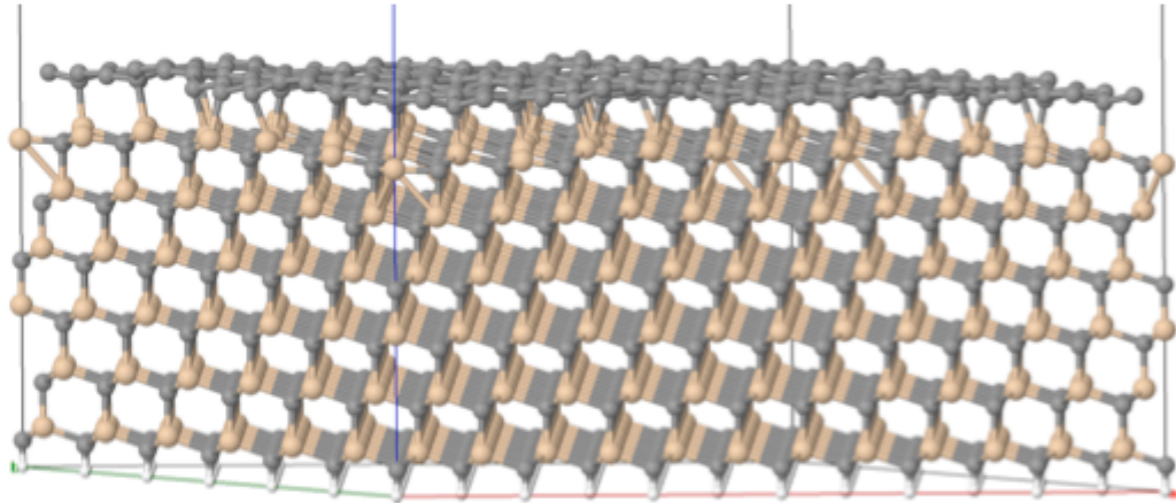
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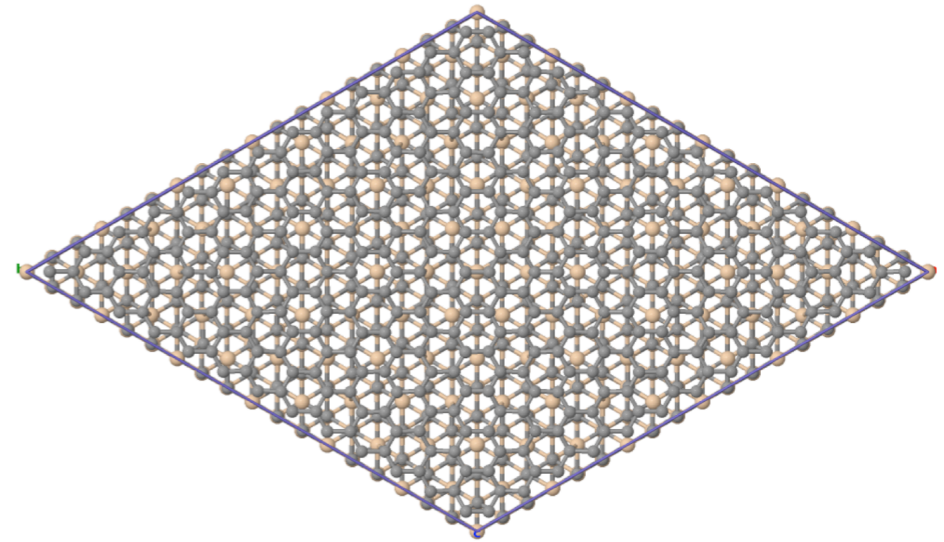
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Ab initio thermodynamics for Si-side graphene/SiC

$(6\sqrt{3}\times 6\sqrt{3})$ SiC(111) + (13×13) graphene:



ZLG, side view



Commensurate growth -
nearly strain-free (0.2%), but large:
1742 atoms (ZLG) - 2756 atoms (3LG)

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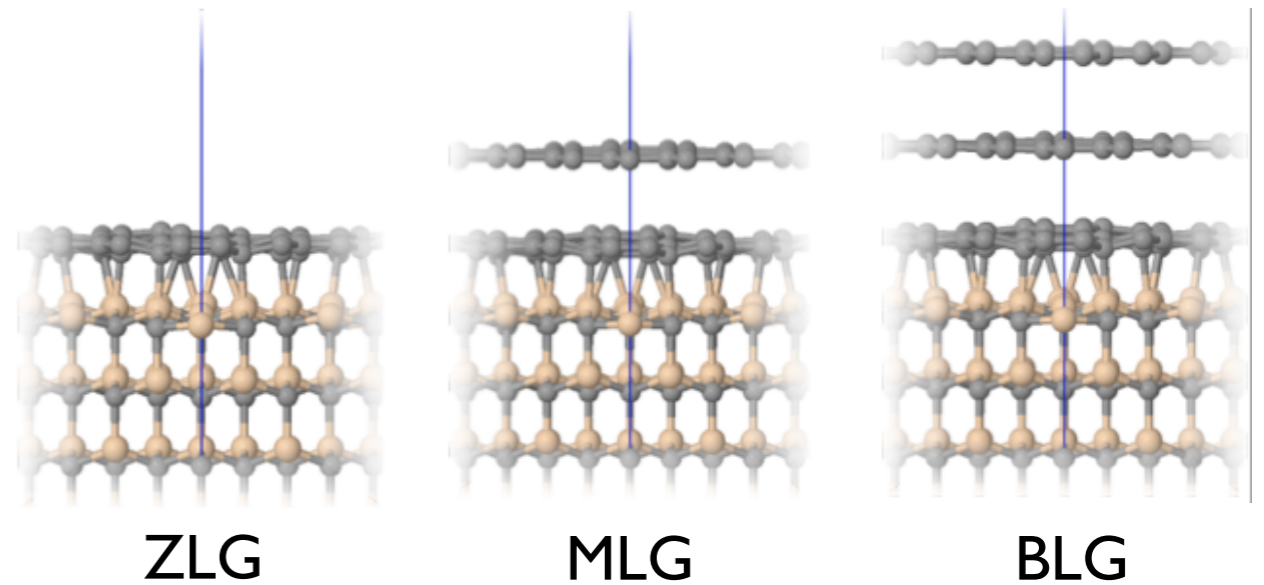
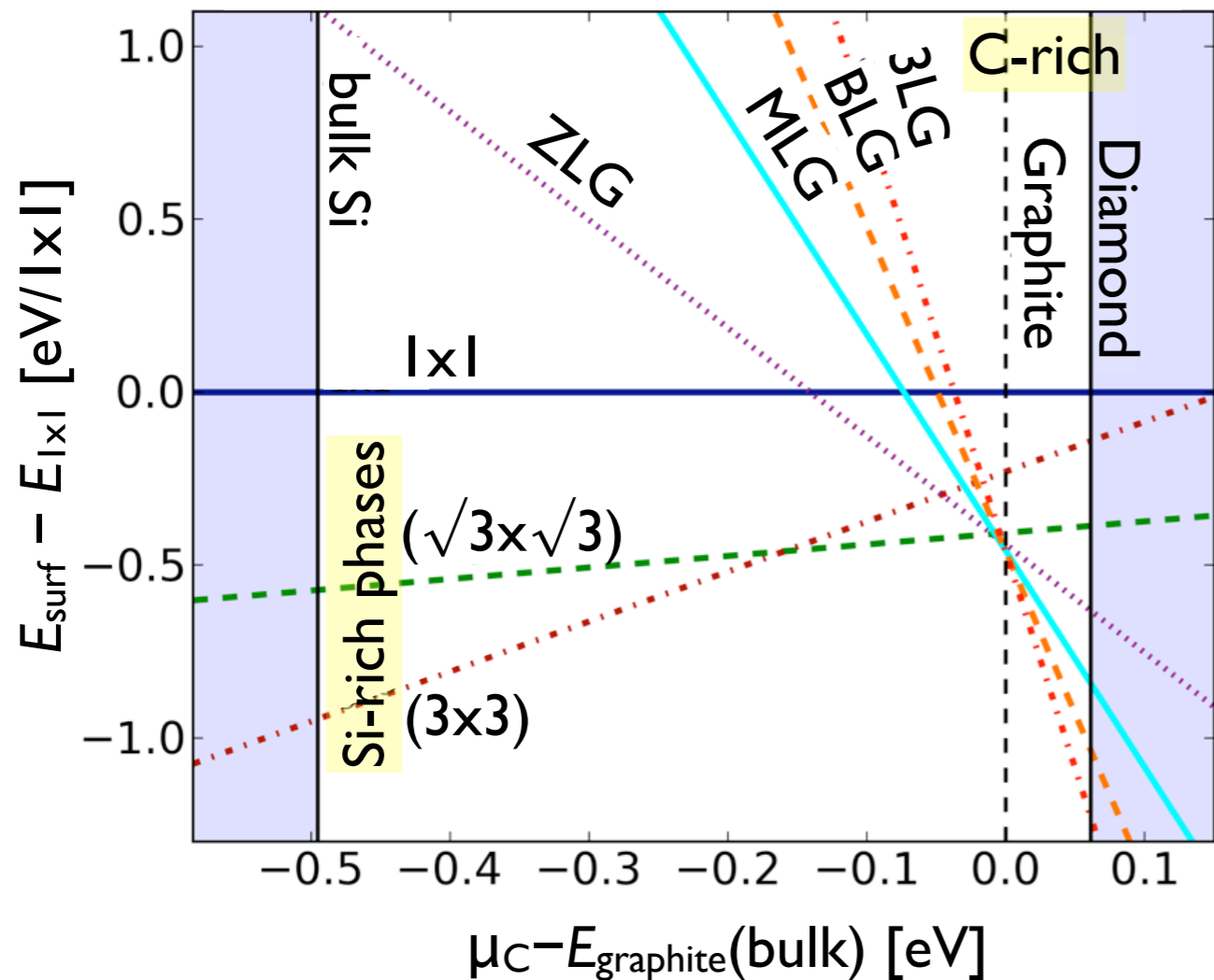
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Stability of surface phases: PBE+vdW

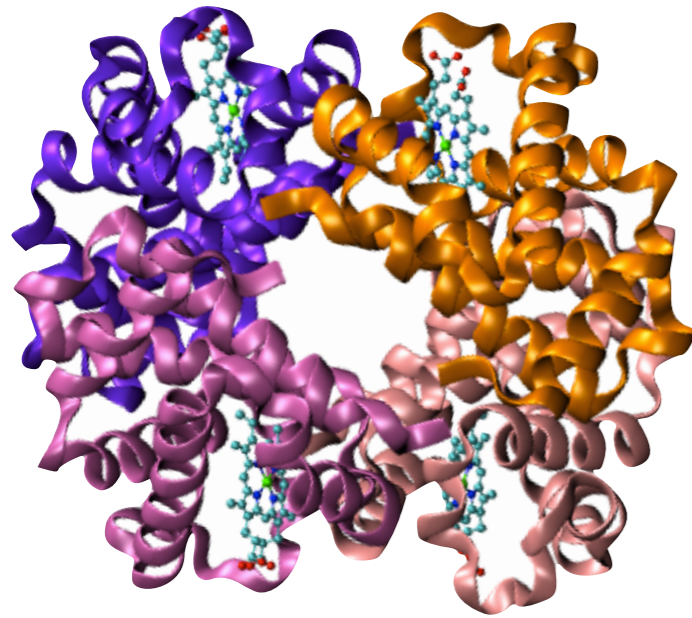
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Surface energy hierarchy: 3C-SiC(111)

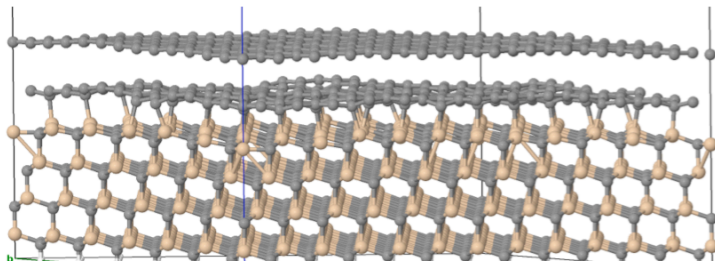


So which (computational) challenges are next?

Molecular world

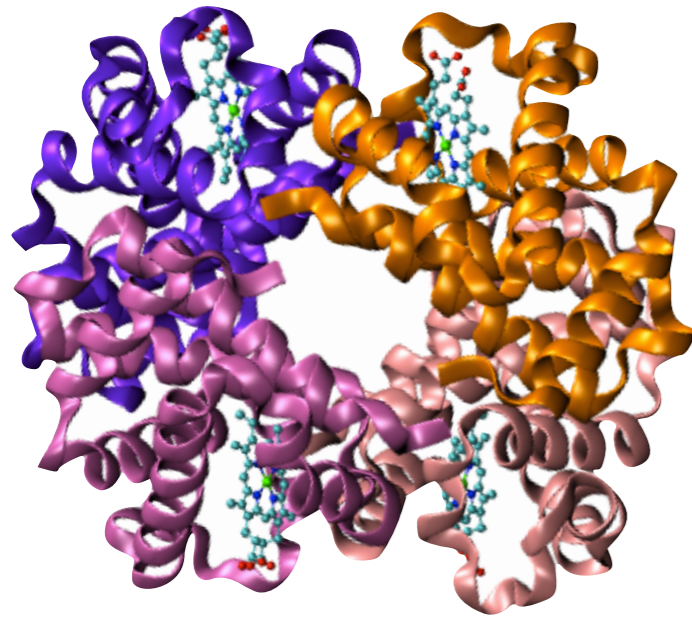


Materials
world



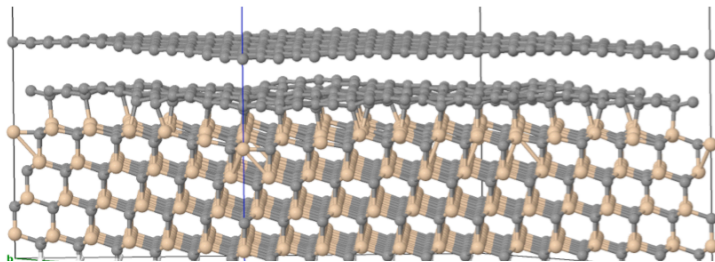
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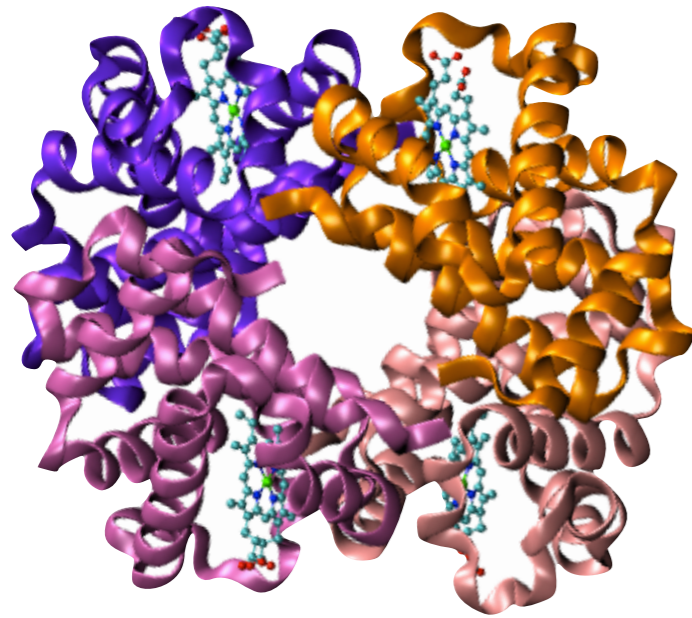
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~100 picoseconds, ~1000 atoms still at low end.
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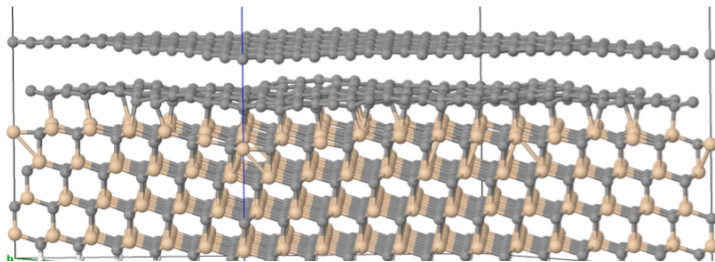
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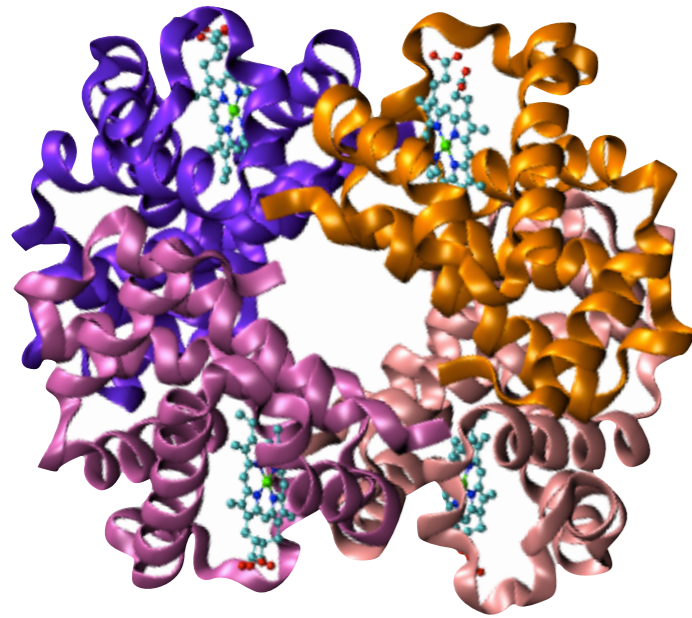
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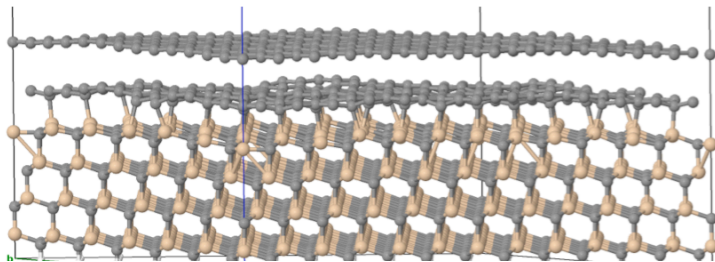


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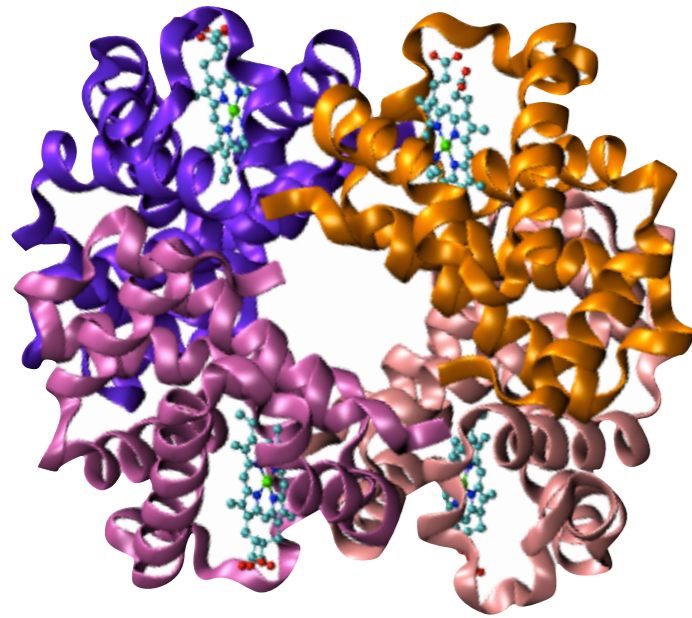
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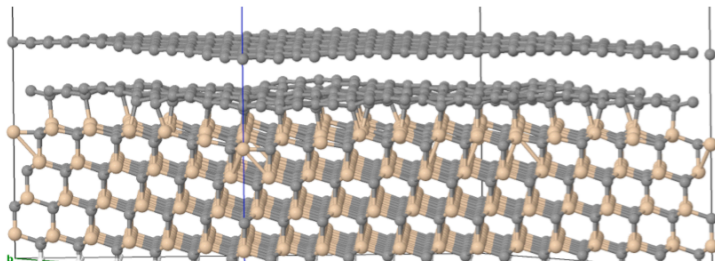


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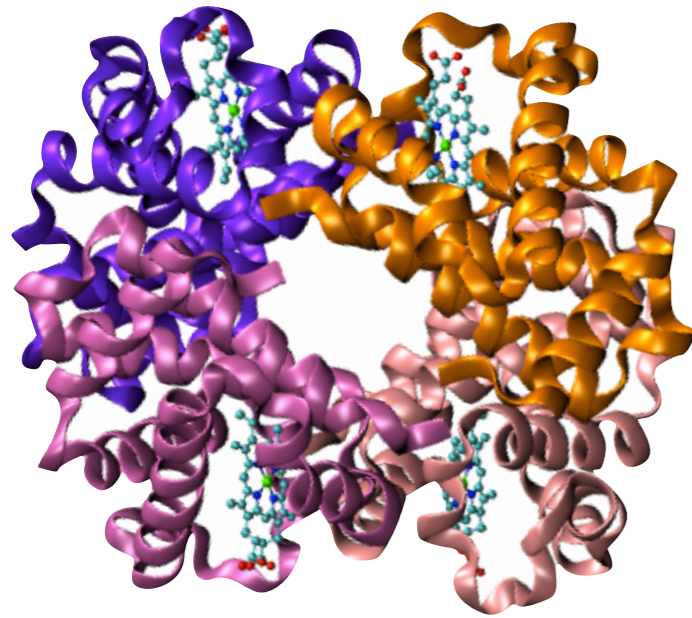
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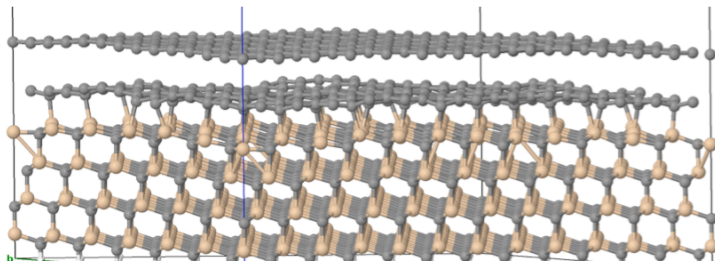


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Efficient, accurate all-electron “DFT and beyond”: FHI-aims

- All-electron “DFT and beyond” based on numeric atom-centered basis sets
- Hierarchical, preconstructed basis sets for elements 1-102, from fast qualitative to meV-level converged total energies
→ high throughput up to gold standard accuracy within one framework
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Efficient, scalable eigenvalue solver library ELPA
- Efficient structure optimization, Born-Oppenheimer *ab initio* molecular dynamics incl. current thermostats (Bussi-Donadio-Parrinello), massively parallel replica exchange
- “Properties and function”:
Vibrations, phonons, harmonic free energies, anharmonic free energies by thermodynamic integration or by interface to plumed, IR spectra, connection to Karlsruhe single-molecule transport library, path integral MD, many more



The Fritz Haber Institute ab initio molecular simulations (FHI-aims) package
V. Blum, R. Gehrke, F. Hanke, P. Havu, V. Havu, X. Ren, K. Reuter and M. Scheffler,
Computer Physics Communications **180**, 2175-2196 (2009) - <http://aims.fhi-berlin.mpg.de/>

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Thank you!