

Present status of ab initio electronic structure calculations

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Part 2 @ the FHI 23.7
-1.8.01

Abstract

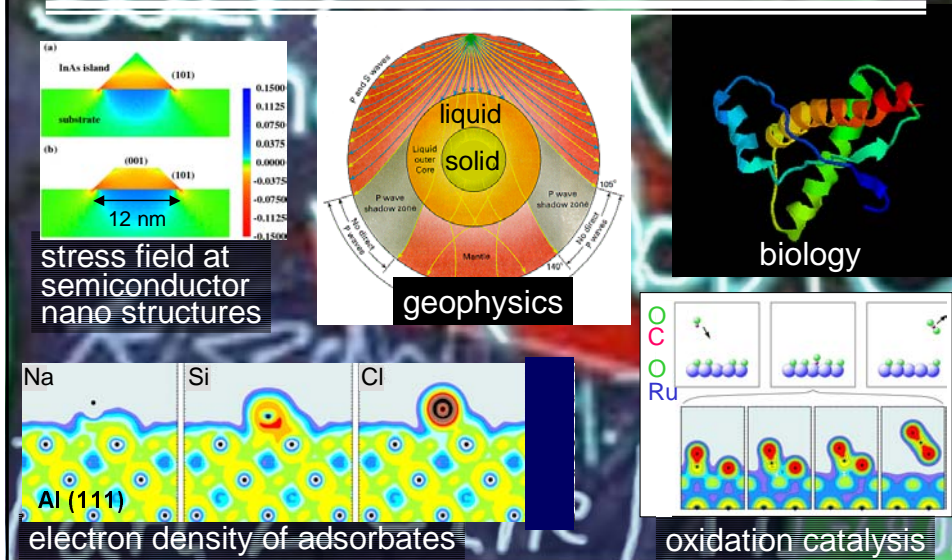
In this talk I will sketch the "ab initio line" of the field often called "Computational Sciences and Engineering", that at present is significantly increasing in importance and impact. To predict materials properties starting from the electronic structure and going all the way to technically relevant time and length scales, is a desire that has existed for many years in particular in condensed-matter physics, materials science, chemistry, and bio-chemistry. However, only recent, spectacular developments in methodology, and the availability of fast computers, enable us now to approach this goal.

I will emphasize how density-functional theory calculations can and should be combined with methods from statistical mechanics, meso-/macroscopic continuum mechanics, and thermo-dynamics, and I will demonstrate the value of such studies using selected examples. These examples include:

- phase transitions of crystals with relevance to mineralogy and geophysics,
- predictions and description of the function of materials surfaces (e.g., atomic structure, chemical activity),
- excited states, core-level spectroscopy (many-electron effects), etc.,
- crystal growth and self-assembly of nano structures,
- insights into the nature of interactions in biological molecules.

Emphasis will be placed on methodology, simple physical models, concepts, and the development of understanding.

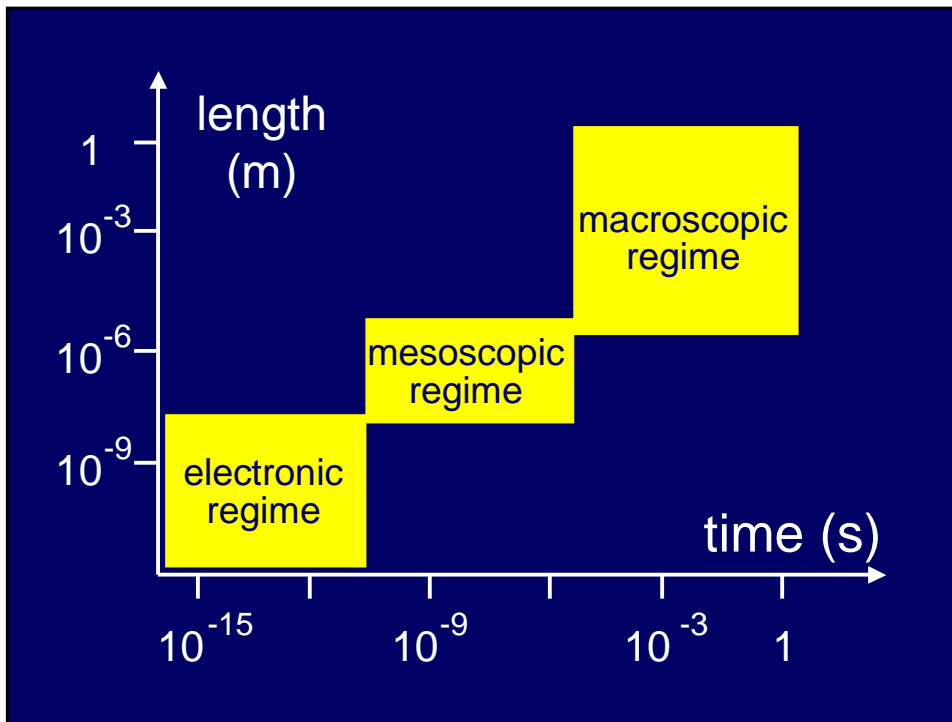
Present status of *ab initio* electronic structure calculations



Methods: schematic summary

Results, for example:

- crystal phase transitions and examples from mineralogy/geology
- thermal expansion of crystals in the bulk and at surfaces,
- crystal growth and self-organization of nano-scale structures.
- surface structure, chemical reactions, catalysis, thermal desorption,
- excited states, core-level spectroscopy (many-body effects), photo-chemistry, etc.
- nature of the interactions in bio-molecules



Density Functional Theory

The energy of the ground state of a many-electron system : $E_0(\{\mathbf{R}_I\}) = \text{Min}_{\Psi} \langle \Psi | H^e | \Psi \rangle$

Hohenberg and Kohn (1964): The functional

$$n(\mathbf{r}) = n[\Psi] = \langle \Psi | \sum_i \delta(\mathbf{r} - \mathbf{r}_i) | \Psi \rangle$$

can be inverted, *i.e.*,

$$\Psi(\mathbf{r}_1, \mathbf{r}_2, \dots, \mathbf{r}_N) = \Psi[n(\mathbf{r})] .$$

This implies:

$$E_0(\{\mathbf{R}_I\}) = \text{Min}_{n(\mathbf{r})} E_{\{\mathbf{R}\}}[n]$$

Kohn and Sham (1965):

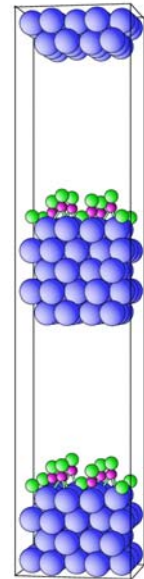
$$E_{\{\mathbf{R}_I\}}[n] = T_s[n] + \int d^3r v_{\{\mathbf{R}_I\}}^{\text{nuc}}(\mathbf{r})n(\mathbf{r}) + \frac{1}{2} \int \int d^3r d^3r' \frac{n(\mathbf{r})n(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} + E^{\text{xc}}[n]$$

with **local-density approximation**
or **generalized gradient approximation**

Accuracy of geometries is better than 0.1 Å. Accuracy of calculated energies (relative) is better than 0.2 eV [for special cases better than 0.01 eV].

Methods

- I {
 - **Density functional theory**
 - **ab initio** pseudopotentials
(the fhi98md - code ---
www.fhi-berlin.mpg.de/th/th.html)
 - **FP-LAPW**
(the WIEN - code by
P. Blaha, K. Schwarz, et al.;
M. Petersen et al., CPC 126 (2000))
- II {
 - **ab initio** Molecular Dynamics
 - **ab initio** Quantum Dynamics
 - **ab initio** Lattice Gas Hamiltonian
 - **ab initio** kinetic Monte Carlo



Methods: schematic summary

Results, for example:

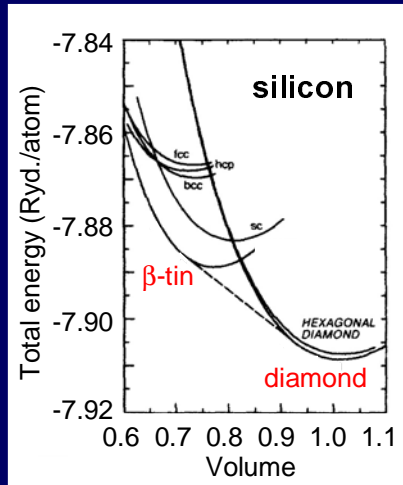
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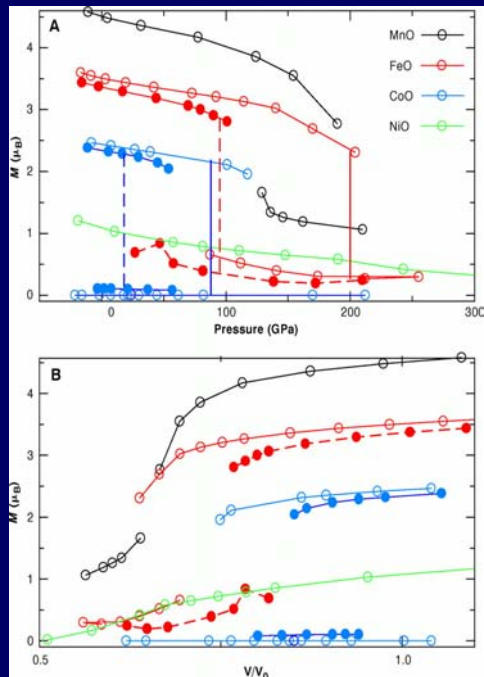
The First (Convincing) DFT Calculations: Stability of Crystals and Crystal Phase Transitions



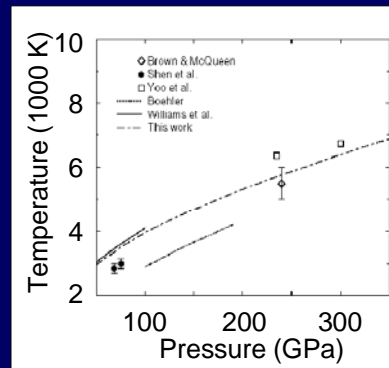
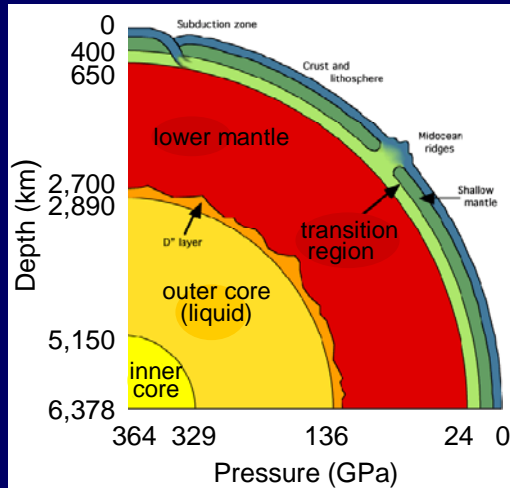
*M. T. Yin and
M. L. Cohen
PRB 26 (1982)*

Magnetic Collapse in Transition Metal Oxides at High Pressure: Implications for the Earth

R. Cohen, Mazin, Isaak,
Science 1997



Ab Initio Melting Curve of Fe as Function of Pressure



*D. Alfe, M. J. Gillan,
and G. D Price
NATURE 401 (1999)*

Methods: schematic summary

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- nature of the interactions in bio-molecules.

Thermal Expansion of Crystals

S. Biernacki, S. and M. Scheffler: Negative thermal expansion of diamond and zinc-blende semiconductors.

Phys. Rev. Lett. 63, 290 (1989).

J. Xie, S. de Gironcoli, S. Baroni, and M. Scheffler,

First-principles calculation of the thermal properties of silver.

Phys. Rev. B 59, 965 (1999)

and at Surfaces

J. Xie, S. de Gironcoli, S. Baroni, and M. Scheffler,

Temperature dependent surface relaxations of Ag(111).

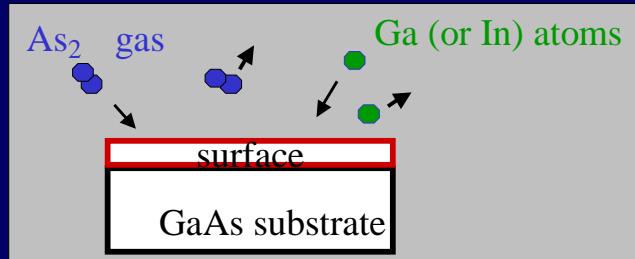
Phys. Rev. B 59, 970 (1999).

Methods: schematic summary

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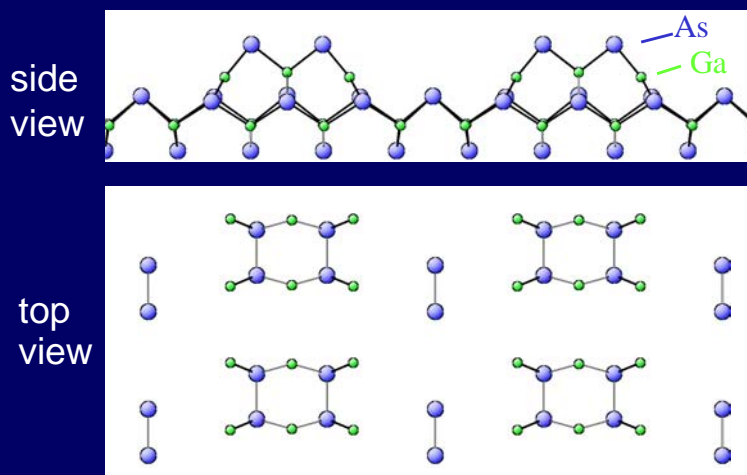
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Microscopic Processes Controlling the Growth -- Example: III-V Semiconductors



- | | |
|----------------------------|--------------------------------------|
| 1) deposition of Ga and As | 5) adsorption of As ₂ ? |
| 2) adsorption of Ga | 6) dissociation of As ₂ ? |
| 3) diffusion of Ga | 7) diffusion of As |
| 4) desorption of Ga | 8) desorption of As |
| | 9) island nucleation |
| | 10) growth |

β_2 Reconstruction of GaAs (001) (2x4) Unit Cell

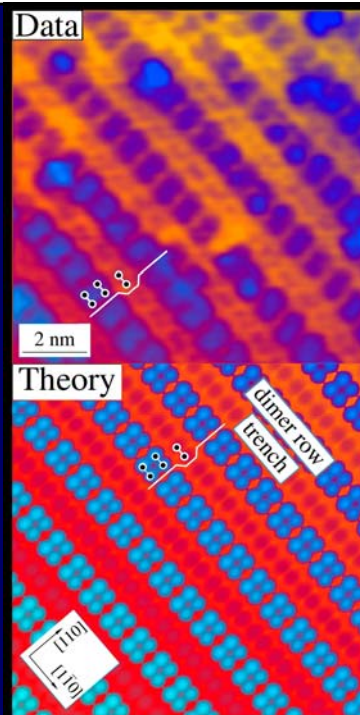


STM Imaging of GaAs(001)

measured filled state image
at $V_{\text{tip}} = -2.1 \text{ eV}$ →

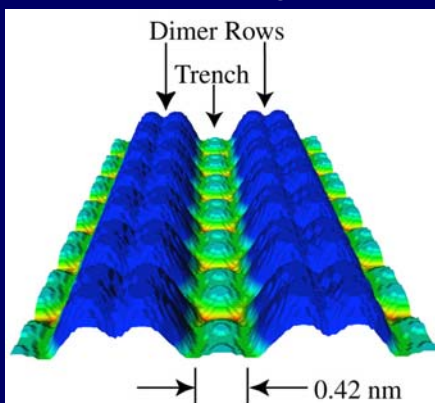
simulated image:
local density of states
integrated to 0.3 eV
below the valence
band maximum →

*LaBella, Yang, Bullock,
Thibado, Kratzer & Scheffler,
PRL 83, 2989 (1999).*

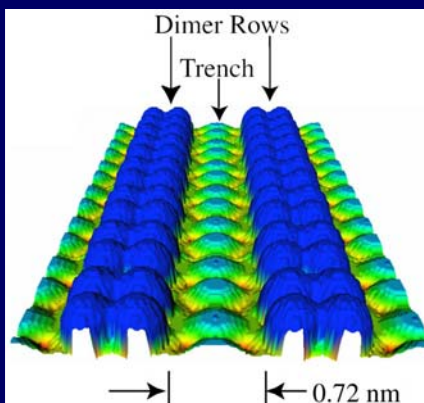


Voltage Dependence of the STM Current STM Simulation

-3.0 eV (high)

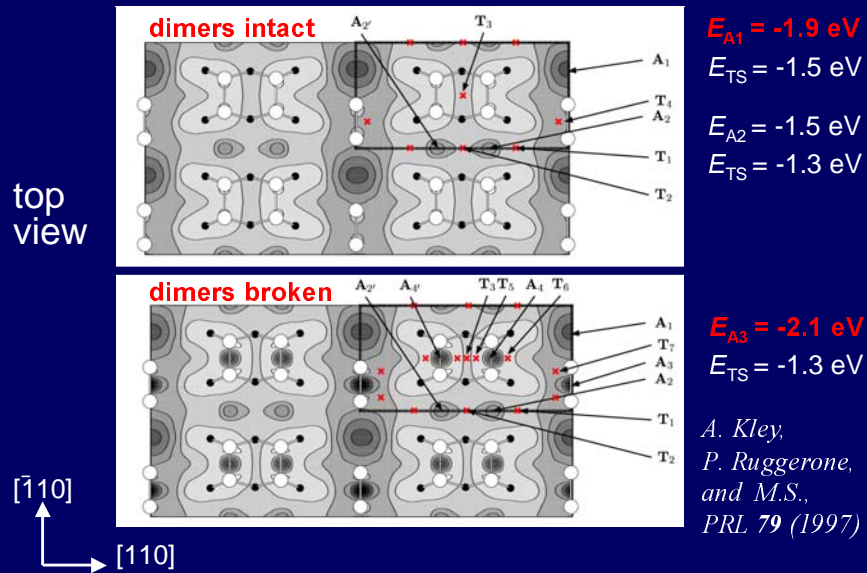


-2.1 eV (low)



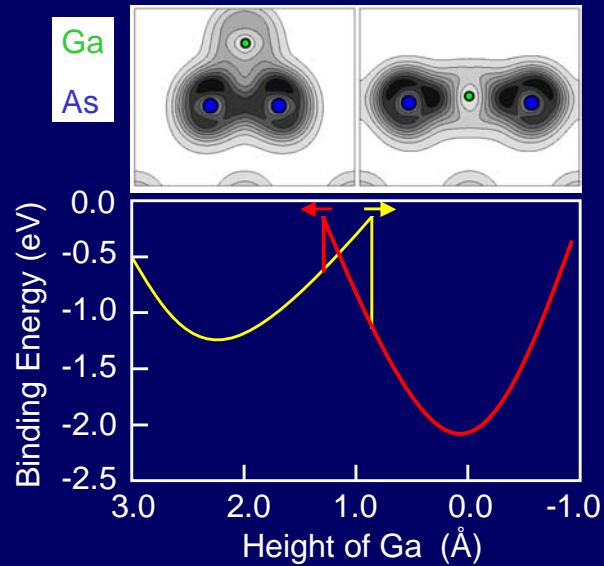
*LaBella, Yang, Bullock,
Thibado, Kratzer & Scheffler,
PRL 83, 2989 (1999).*

Total Energy of a Diffusing Ga Atom at GaAs (001)



Calculating such potential-energy surfaces is a significant part of modern *ab initio* electronic structure theory.

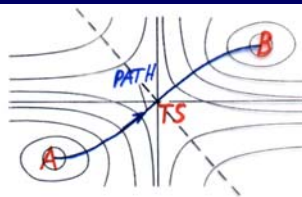
Unusually Stable Site for Ga Adatom Inside The Trench-Site As-Dimer



*A. Kley,
P. Ruggerone,
M.S.,
PRL 79 (1997)*

*P. Kratzer
& M. S.,
to be published*

Transition State Theory



Transition state theory

$$\Gamma = \frac{k_B T}{h} \exp\left(\frac{-\Delta F}{k_B T}\right)$$

$$\Delta F = -k_B T \ln Z_{TS} + k_B T \ln Z_A$$

$$\Gamma = \Gamma_0 \exp\left(\frac{-\Delta E}{k_B T}\right)$$

$$\Gamma_0 = \frac{k_B T}{h} \exp\left(\frac{\Delta S^{\text{vib}}}{k_B} - \frac{\Delta U^{\text{vib}}}{k_B T}\right)$$

Theory of the Kinetics of Growth

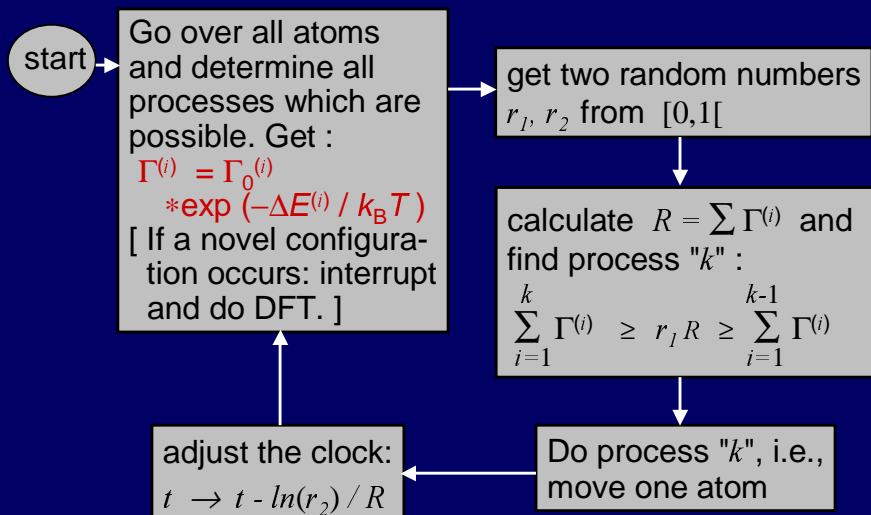
- 1) Analysis of all possibly relevant processes
- 2) Calculate the rates of all important processes

$$\Gamma^{(i)} = \Gamma_0^{(i)} \exp(-\Delta E^{(i)} / k_B T)$$

- 3) Statistical approach to describe
 - deposition
 - diffusion
 - nucleation
 - growth

➔ **kinetic Monte Carlo method**

Flowchart of kinetic Monte Carlo Simulation



➔ **unsuccessful processes are avoided**

Sketch of the kMC Approach

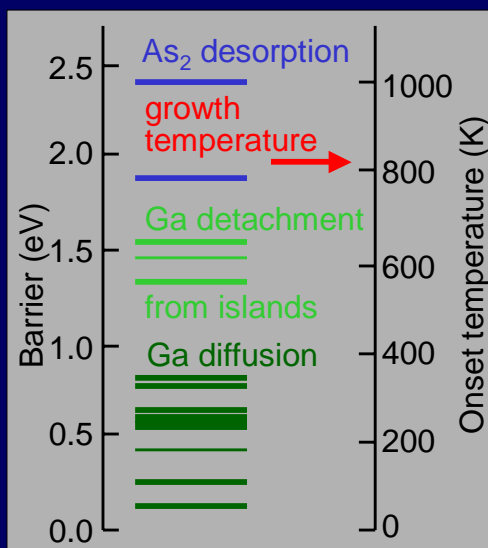


Graphical sketch of the statistics of the process that will be chosen in the kMC approach. Each bar corresponds to a certain atom. The color refers to the type of process, and the thickness to the rate.

for example: **yellow**: Ga diffusion in the trench;
light blue: Ga diffusion perpendicular to trench;
green: Ga enters an As dimer;
dark blue: Ga diffusion parallel to steps;
red: As₂ adsorption into the intermediate.

Kinetic Monte Carlo Simulations

- Reconstruction of surface considered
- 32 microscopically different Ga diffusion processes, and As₂ adsorption/desorption are included explicitly
- Computational challenge: very different time scales (10⁻¹² sec to 10 sec)

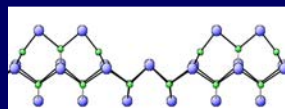


Kinetic Monte Carlo calculations are a very important add-on to modern *ab initio* electronic-structure theory.

Likely, this is the way to go in the future.

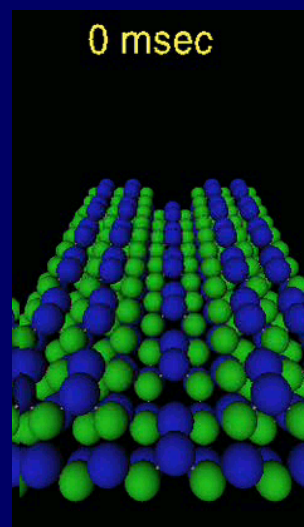
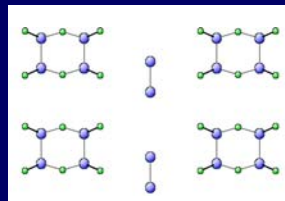
Adsorption, Diffusion, Island Nucleation, and Growth of GaAs

side view



Ga
As

top view

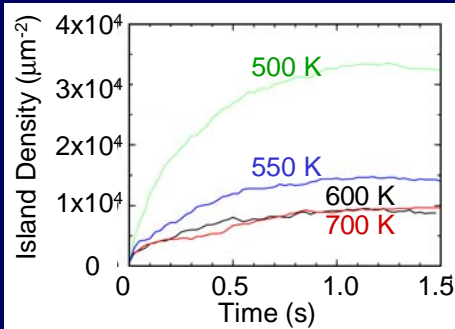


1/60 of the full simulation cell
As₂ pressure $\approx 1.33 \times 10^{-8}$ bar
Ga deposition rate = 0.1 ML/s
T = 700 K

P. Kratzer & M. S., submitted to PRL

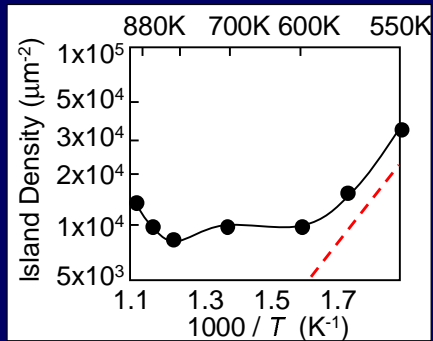
Island Density

*P. Kratzer & M. S.,
submitted to PRL*



As_2 pressure
 $\approx 1.33 \times 10^{-8}$ bar

Ga deposition rate
 $= 0.1$ ML/s



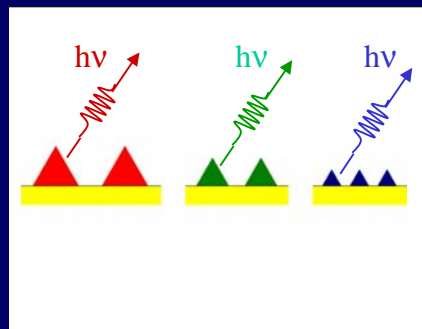
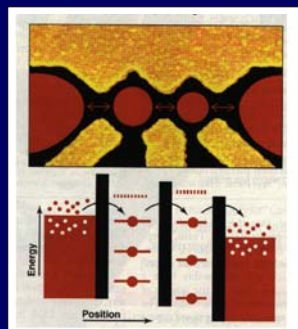
Log_{10} of island density
does not increase
linearly with $1/T$.

Self-Assembly of Nano-Scale Structures at Semiconductor Surfaces

Motivation:

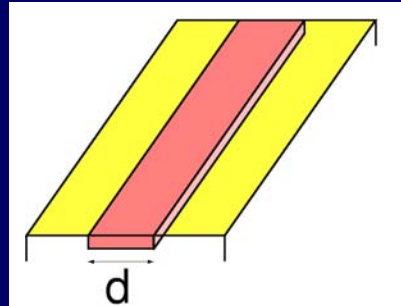
Single-electron transistor

LEDs and
laser diodes



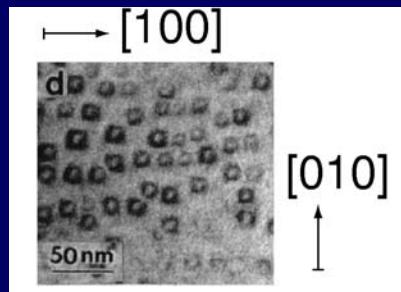
Required dimensions: 50 - 200 Å

Lithography ends
at $d > 200$ Å,
and has rough
edges

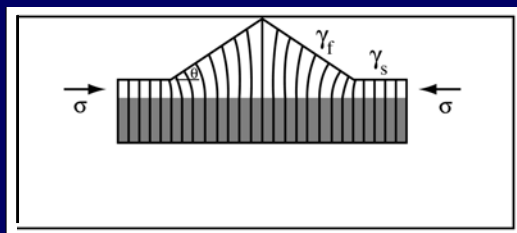
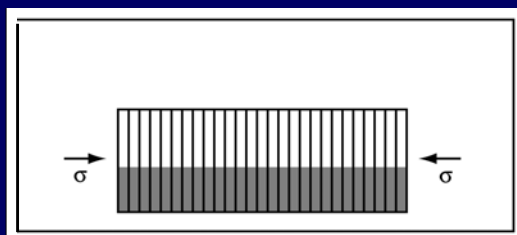


Self-assembly of
InAs quantum-
dots on
GaAs (100)

Ruvimov et al., PRB 51 (1995)



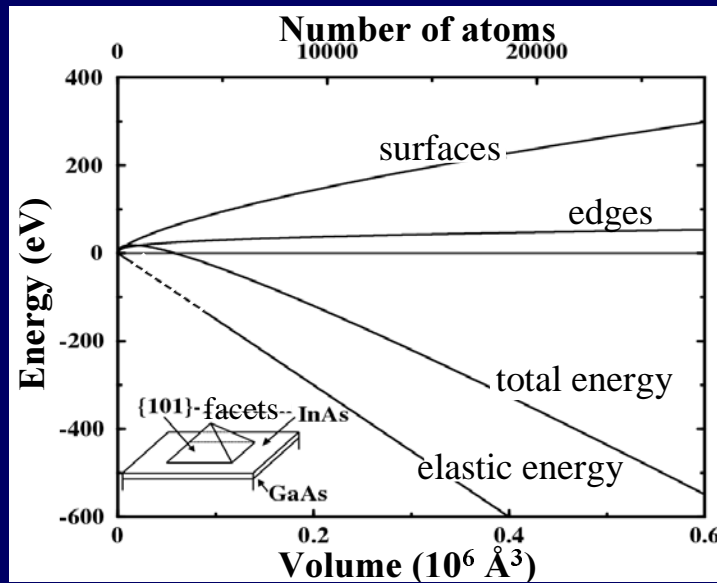
Stranski-Krastanow Morphology as one Way to reduce Misfit Strain Energy



e.g.
InAs on
GaAs

or
InP on
GaP

Equilibrium Shape of Quantum Dots

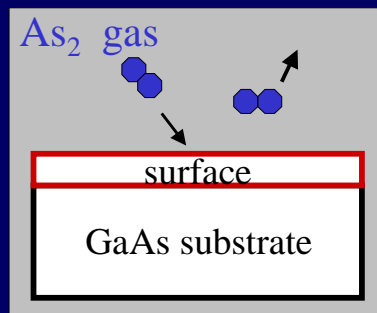


*L. Wang,
P. Kratzer,
and M.S.,
PRL 82
(1999)*

Stoichiometry and Structure of the Surface depend on the Environment

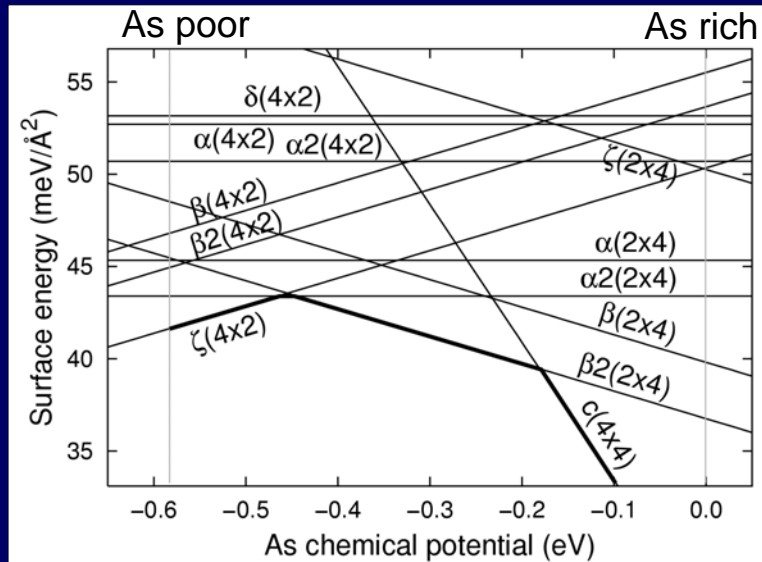
(atomic chemical potentials)

$$E_{\text{surface}} = \Delta E_{\text{tot}} - N_{\text{Ga}} \mu_{\text{Ga}} - N_{\text{As}} \mu_{\text{As}}$$



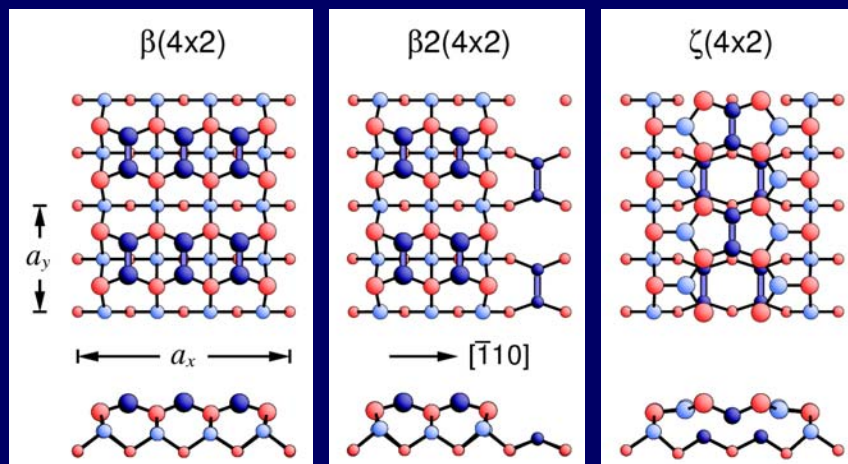
$$\mu_{\text{Ga}} + \mu_{\text{As}} = E^{\text{bulk}}(\text{GaAs})$$

Surface Energies of GaAs (001)



Sung-Hoon Lee, W. Moritz, and M.S., PRL **85**, 3890 (2000)

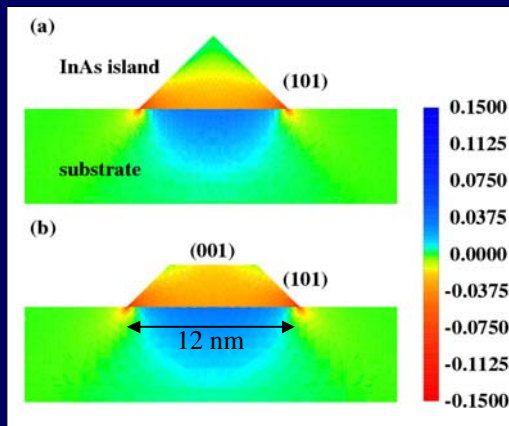
Surface Structures of GaAs (001)



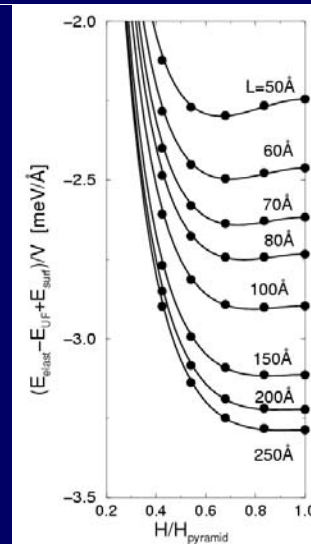
all these structures fulfill the electron counting rule

Sung-Hoon Lee, W. Moritz, and M.S., PRL **85**, 3890 (2000)

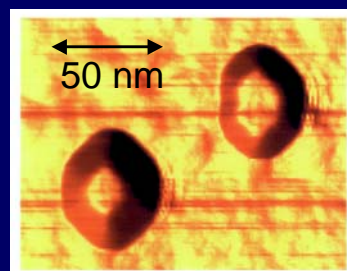
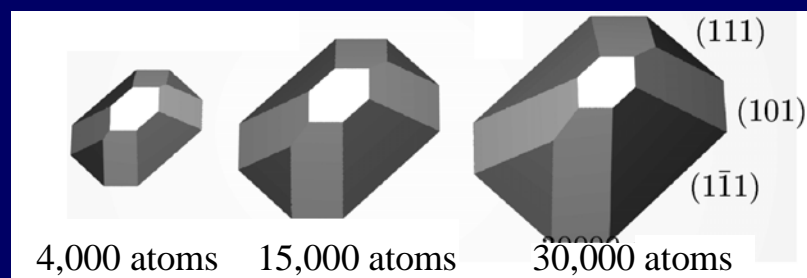
Stress Tensor at strained InAs Islands on GaAs



*N. Moll, M.S., and E. Pehlke,
PRB 58, 4566 (1998)*



InP Quantum Dots on GaP(001)



*Q. Liu, E. Pehlke,
N. Moll, and M.S.,
PRB 60 (1999)*

**InP islands on GaInP
grown by MOVPE**

Samuelson et al. (1996)

Oxidation Catalysis, e.g.:

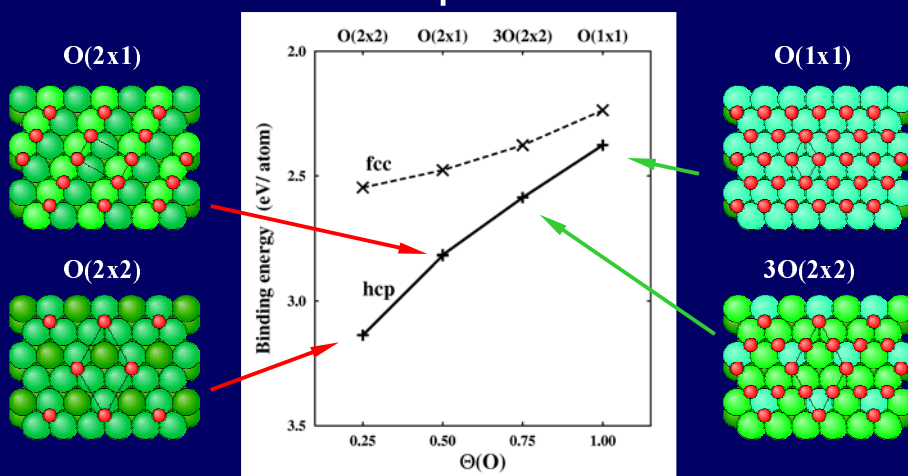


A "simple", prototypical surface chemical reaction

Oxygen Adsorption on Ru(0001)

UHV ■ ■ ■ ▶

High pressure



C. Stampfl and M. Scheffler, Phys. Rev. B 54, 2868 (1996)

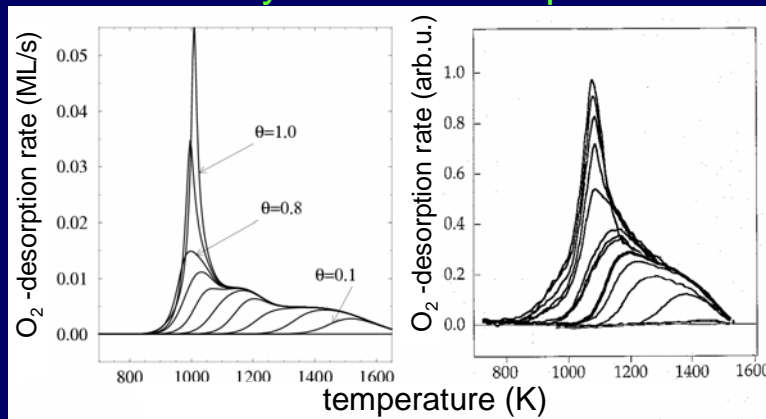
Building lattice-gas Hamiltonians from DFT calculations and evaluating the partition function is most helpful.

Also this is an important way to go in the future.

Temperature Programmed Desorption Spectra: O/Ru(0001)

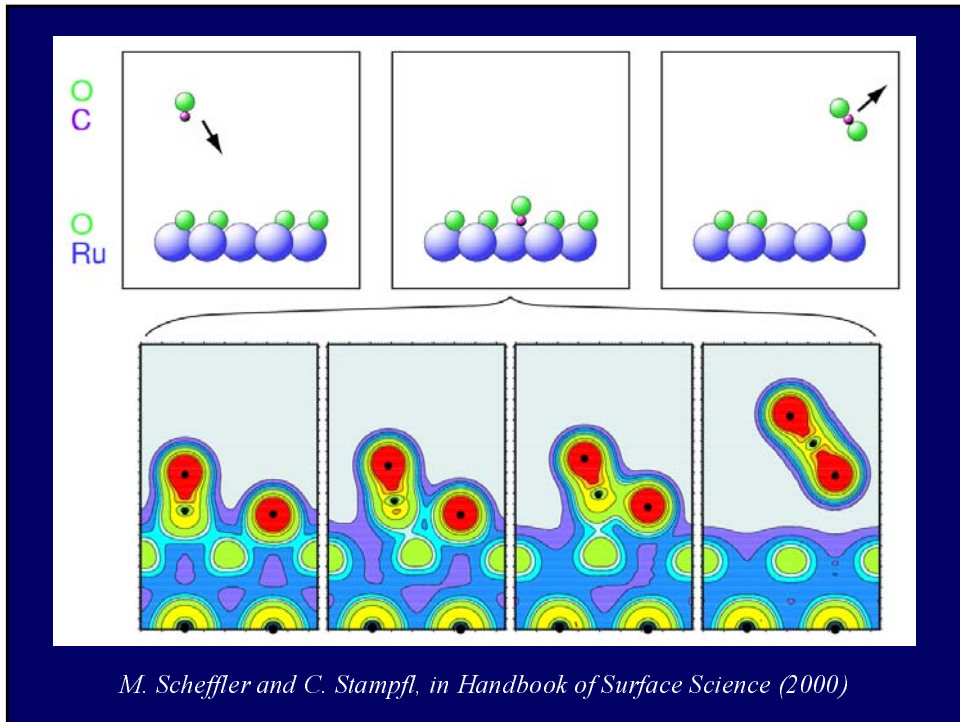
Theory

Experiment



Heating rate: 6 K/s

Stampfl et al.,
PRL 83 (1999)



Ru(0001) after Exposure to Oxygen: Atomic scale structure

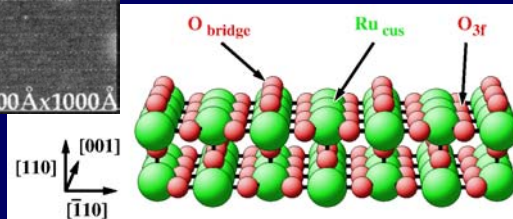
(1x1) O-Ru (0001)



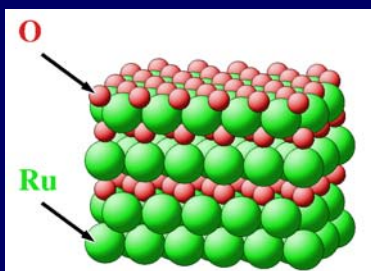
RuO₂ (110)

STM

H. Over, Y.D. Kim,
A.P. Seitsonen,
S. Wendt, A. Morgante,
E. Lundgren, M. Schmid,
P. Varga, and G. Ertl,
Science 287 (2000)

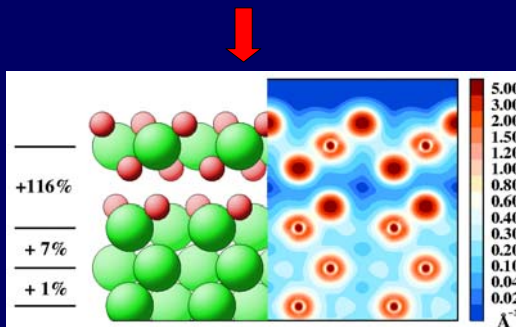


Increasing the Coverage Above Two Oxygen Layers



*K. Reuter et al.,
to be published*

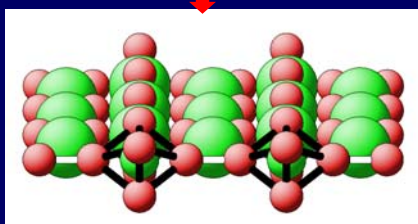
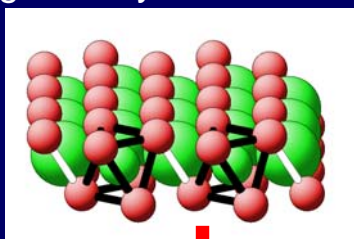
Test all 18 possible combinations



Additional O layer saturates the underlying Ru(0001) substrate and decouples the O-Ru-O trilayer.

Accordion Mechanism of Surface-Oxide Formation

geometry of O-Ru-O trilayer



geometry of RuO₂(110)



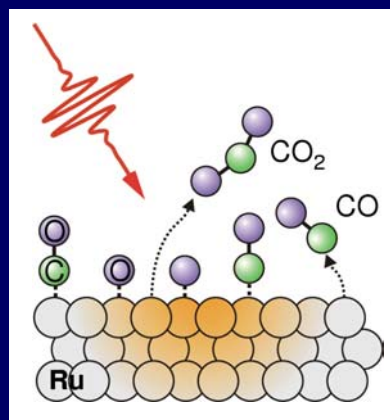
K. Reuter et al., to be published

Methods: schematic summary

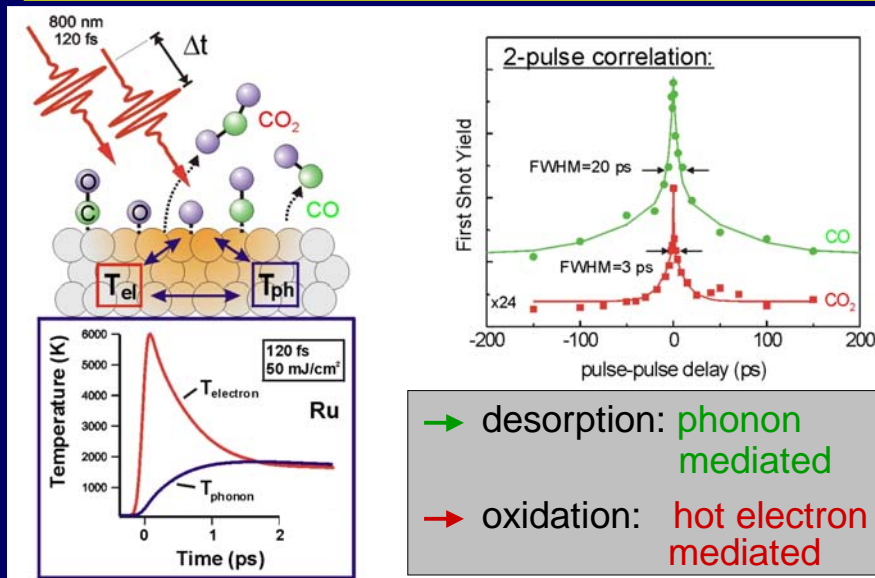
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Photochemistry: Phonon- versus electron-mediated surface reactions: Laser-induced desorption and oxidation of CO on Ru(0001)

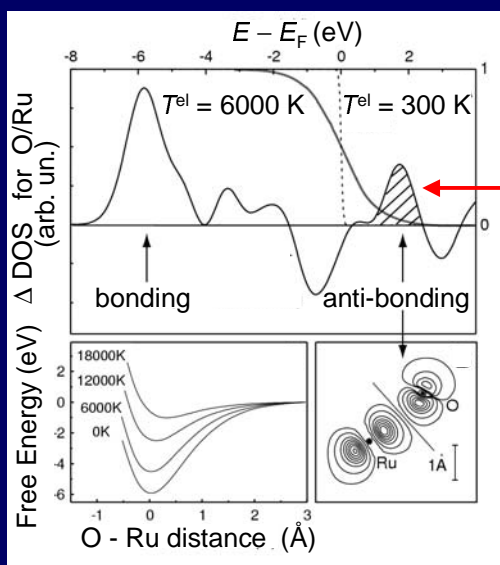


Femtosecond photooxidation of CO on Ru



M. Bonn et al., SCIENCE 285 (1999)

DFT calculations at finite temperature



O on Ru(0001)

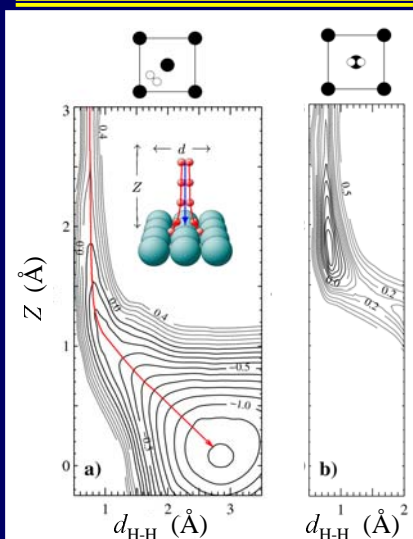
unoccupied anti-bonding orbital above E_F

increasing the electronic temperature leads to a weakening of the O-Ru bond

M. Bonn et al., SCIENCE 285 (1999)

Quantum-dynamics (and molecular dynamics) of H₂ dissociation at metal surfaces

"Divide-and-Conquer" strategy to determine the reaction dynamics



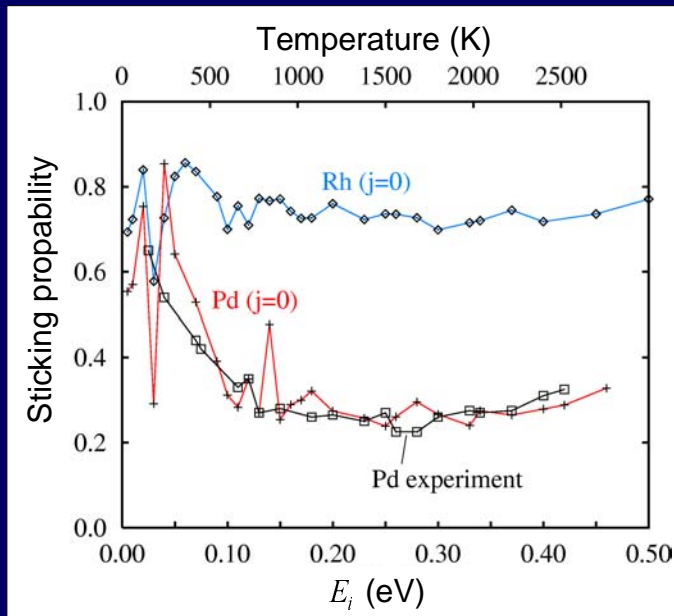
PES for H₂ -- Pd(100)

Three independent steps

- 1) Determine potential energy surface (PES) by DFT calculations (GGA for xc)
- 2) Analytical or numerical representation of this PES
- 3) Dynamics
 - a) Solve time-dependent or time-independent Schrödinger equation
 - b) solve Newton's equation of motion

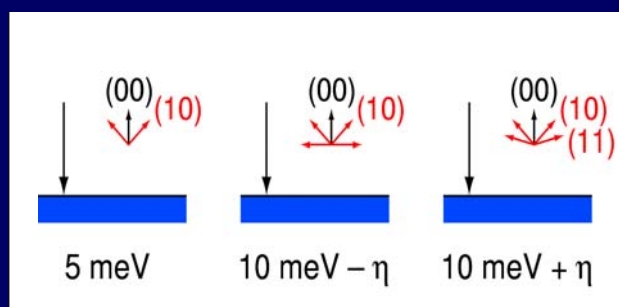
A. Gross and M.S., PRB 57 (1998)

Dissociative adsorption of H₂



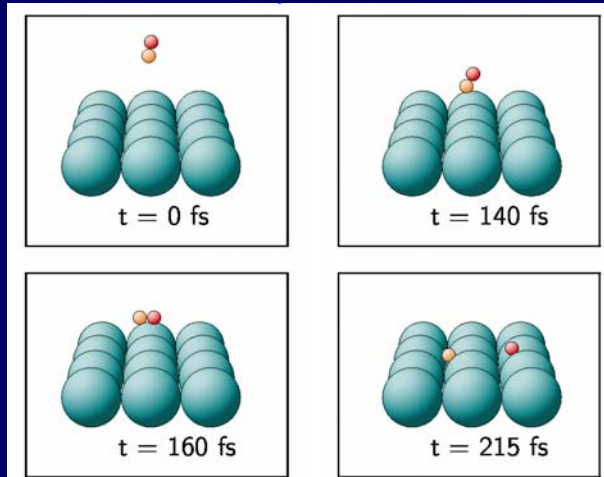
*A. Eichler
et al.,
PRB 59
(1999)*

Oscillations reflect the threshold,
e.g., exit conditions of reflected beams



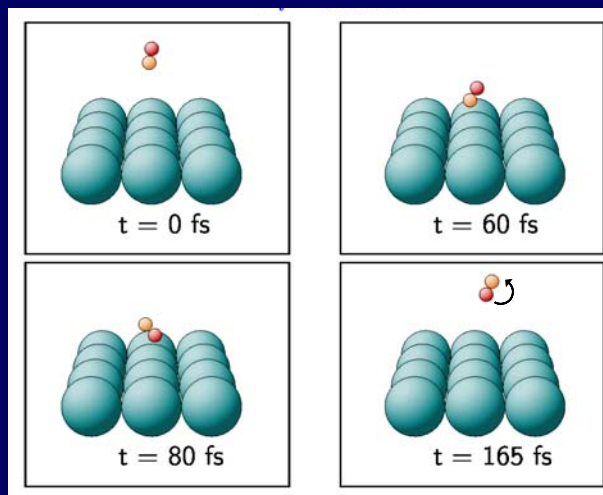
Molecular Dynamics Snapshots of $\text{H}_2 \rightarrow \text{Pd}(100)$

Steering effect: aligning the incoming molecule
(at low energies; here: $E_i = 0.01$ eV)



*A. Gross
and M.S.,
PRB 57,
(1998)*

No steering at higher energies; here: $E_i = 0.12$ eV



A. Gross and M.S., PRB 57 (1998)

Often the statistics of chemical reactions is very involved. Then a "divide and conquer" approach, as the one described here, is crucial for DFT-based molecular dynamics.

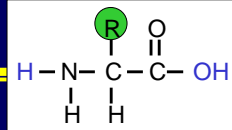
The difficult step is the representation (and interpolation) of the DFT total energies to represent the full potential-energy surface.

Methods: schematic summary

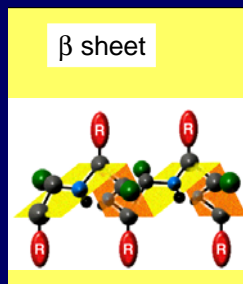
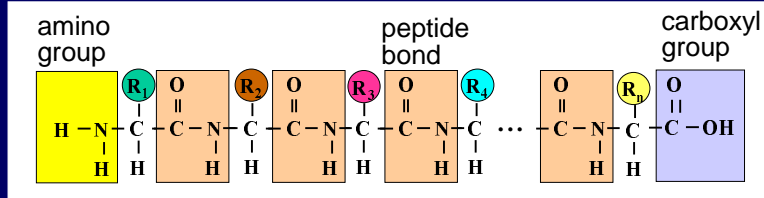
Results, for example:

- crystal phase transitions and examples from mineralogy/geology,
- thermal expansion of crystals in the bulk and at surfaces,
- crystal growth and self-organization of nano-scale structures,
- surface structure, chemical reactions, catalysis, thermal desorption,
- excited states, core-level spectroscopy (many-body effects), photo-chemistry, etc.,
- nature of the interactions in bio-molecules.

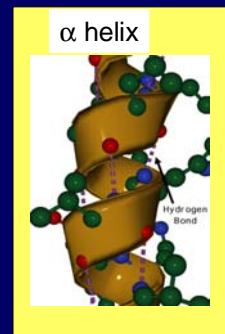
Stability of The α -Helix



structure of proteins (peptide chains):

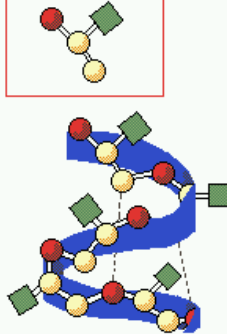


secondary structure

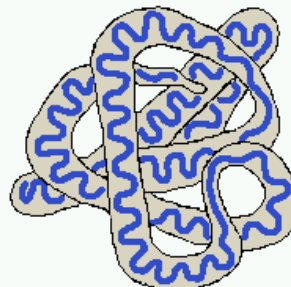


Structure of Proteins

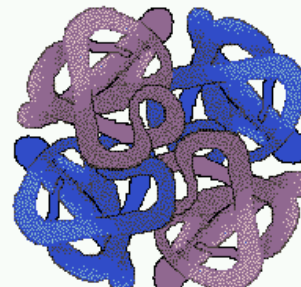
amino acid



secondary structure

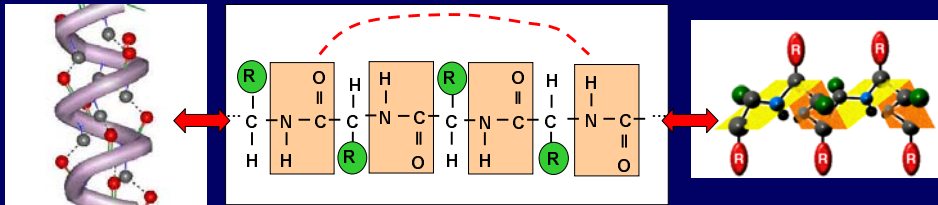


tertiary structure



quaternary structure

Enthalpy of Formation of a Hydrogen Bond



Enthalpy of formation: $\Delta H^N = E_{\sigma}^N - E_{\sigma}^{N-1} - \mu_{\text{peptide}}$

poly alanine (**R** = methyl group = CH_3)

sheet (extended structure) as "reservoir"

α -helix:

"back bone" costs : $\Delta H_{\text{bb}}^N \approx 6,5 \text{ kcal/mol}$ for all N

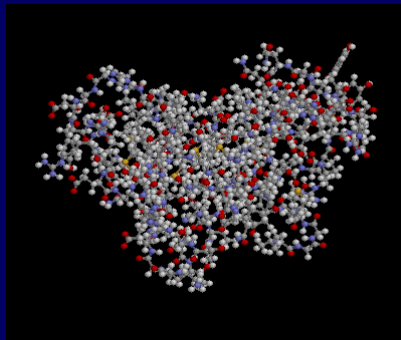
H-bond gain : $\Delta H_{\text{H-bond}}^N = -2,7 \text{ kcal/mol}$ for $N = 4$
 $= -9,5 \text{ kcal/mol}$ for $N = \infty$

→ hydrogen bond is strongly cooperative

J. Ireta, et al., submitted to PRL

Folding of the peptide chain

peptide chain in the bovine prion protein



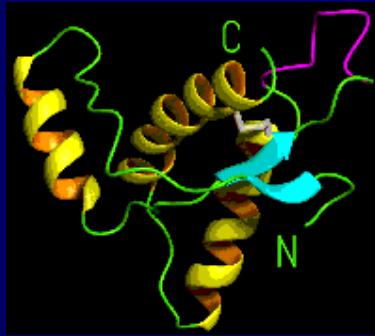
atomic structure



secondary and tertiary structure

<http://www.rcsb.org/> -- pdb-id: 1QM0

peptide chain in the bovine prion protein

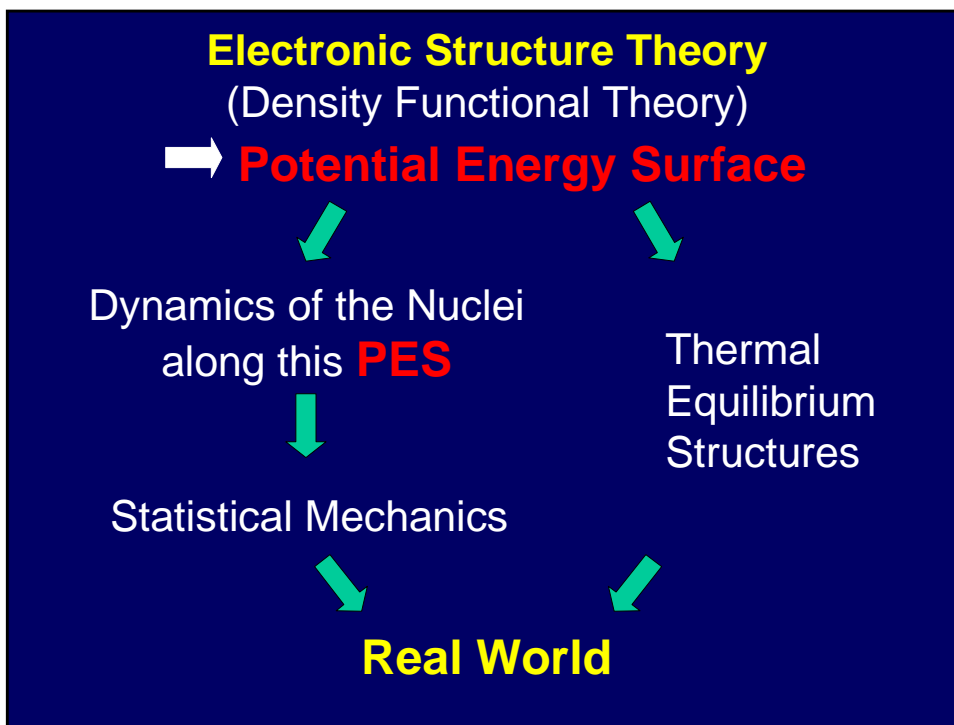


http://cyber-dyne.com/~tom/reverse_chron.html

peptide chain in the bovine prion protein



http://cyber-dyne.com/~tom/reverse_chron.html



The people behind the work:

Peter Kratzer Cathy Stampfl Karsten Reuter

Jörg Neugebauer Joel Ireta et al. ...