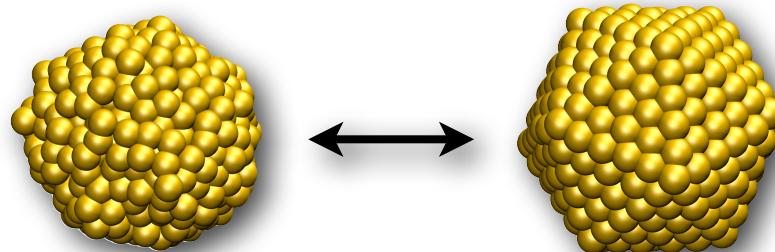


*Studying nucleation processes with computer simulations:*

# Neural networks for force calculation and structure recognition

**Christoph Dellago**  
Faculty of Physics  
University of Vienna  
Austria

Jörg Behler (U Bochum)  
Andreas Singraber (U Vienna)  
Philipp Geiger (U Vienna)  
Ernesto Borrero (U Vienna)

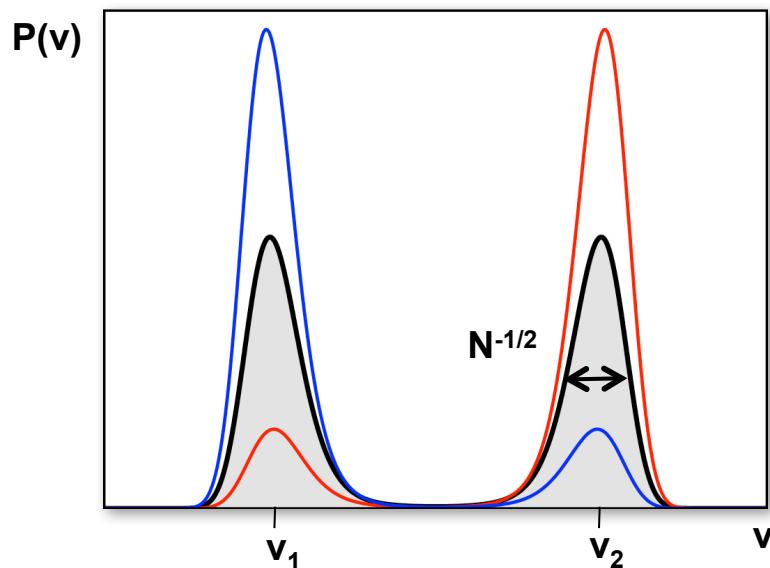
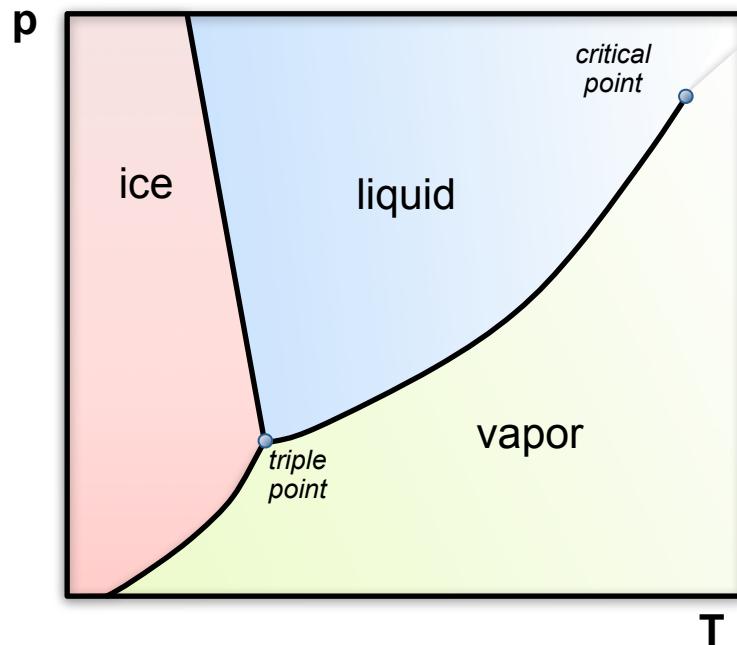


CENTER FOR  
COMPUTATIONAL  
MATERIALS  
SCIENCE



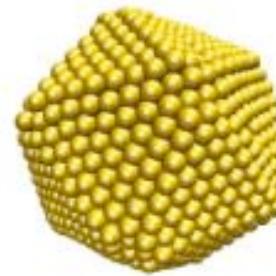
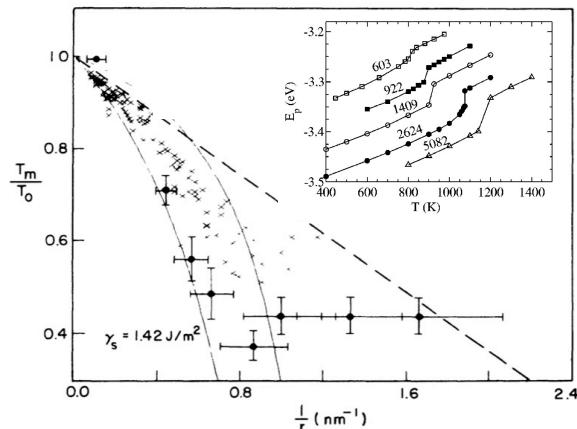
universität  
wien

# Rounding of phase transition



# Phase transitions in small systems

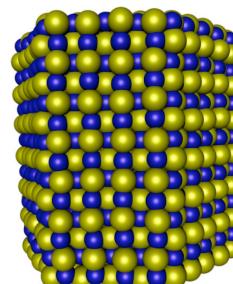
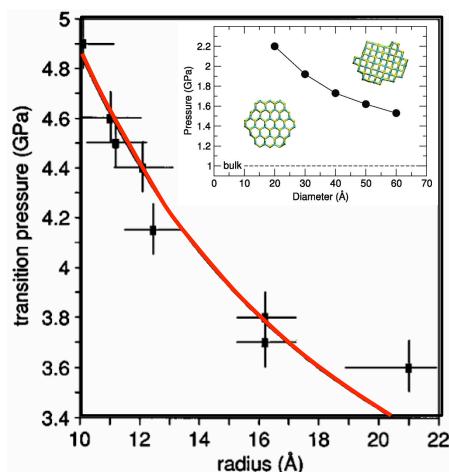
## Melting point depression of Au nanoparticles



$$\gamma_l < \gamma_s$$

Y. Wang, S. Teitel, and C. Dellago, JCP 122, 214722 (2005)  
T. Castro, R. Reifenberger, E. Choi, and R. P. Andres, PRB 42, 8548 (1990)

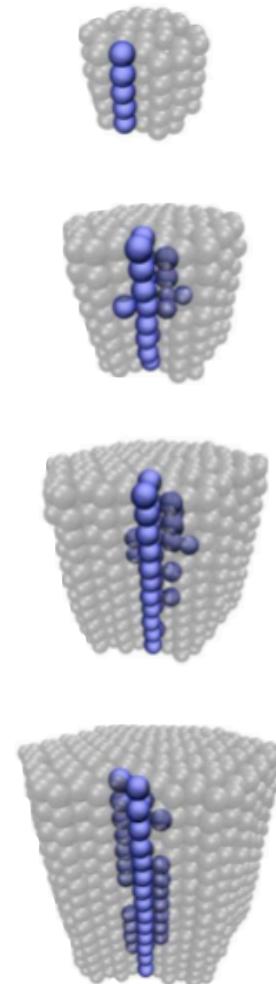
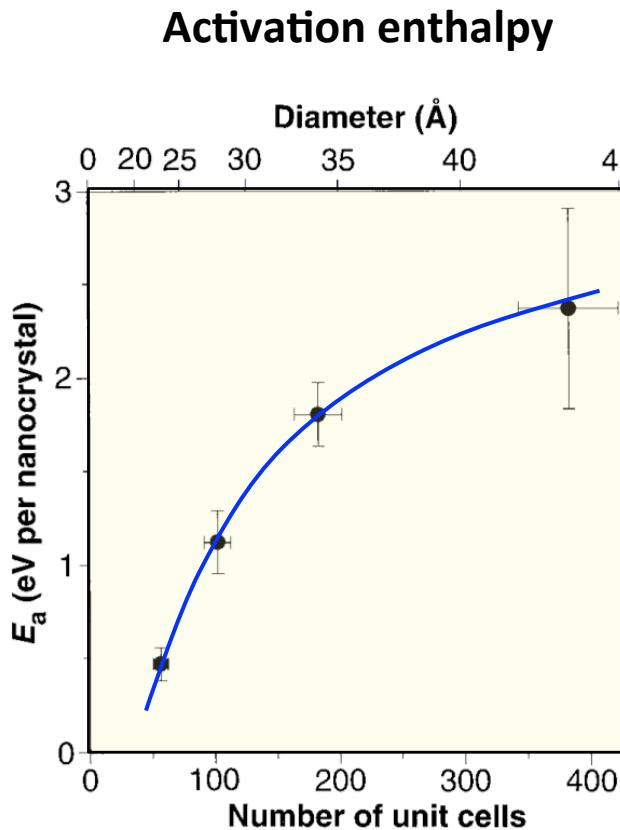
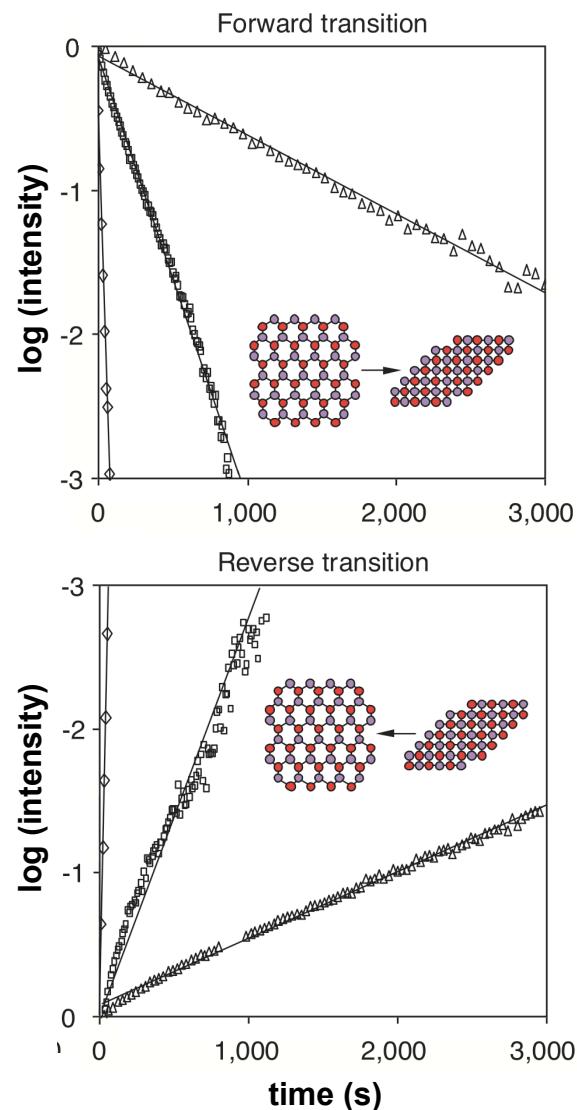
## Transition pressure elevation in CdSe nanocrystals



$$\gamma_{WZ} < \gamma_{RS}$$

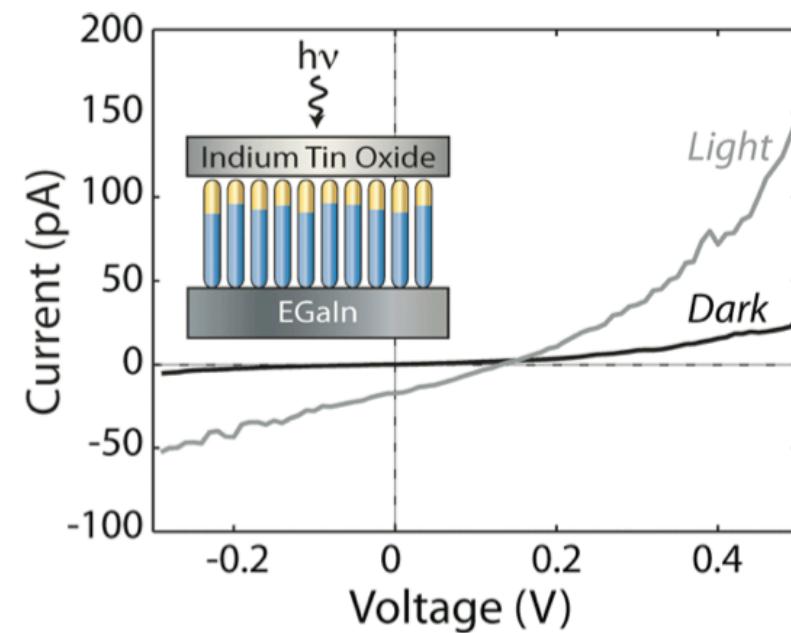
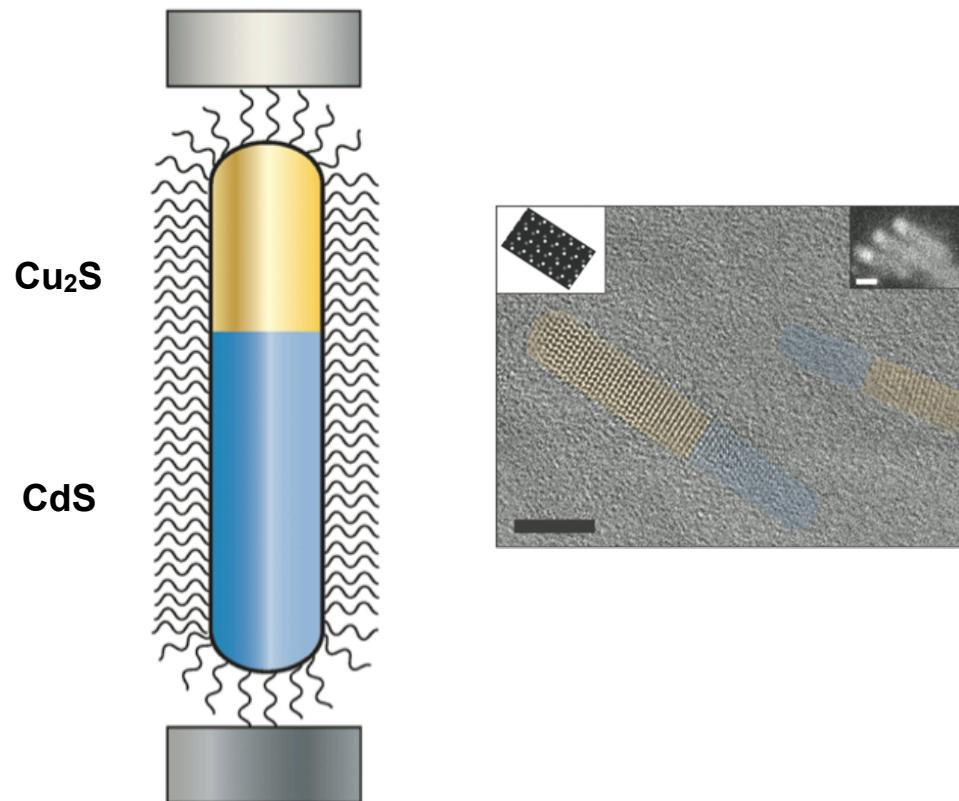
M. Grünwald and C. Dellago, Nano Letters 9, 2099 (2009)  
S. H. Tolbert and A. P. Alivisatos, Science 265, 373 (1994)

# Transformation kinetics and mechanism

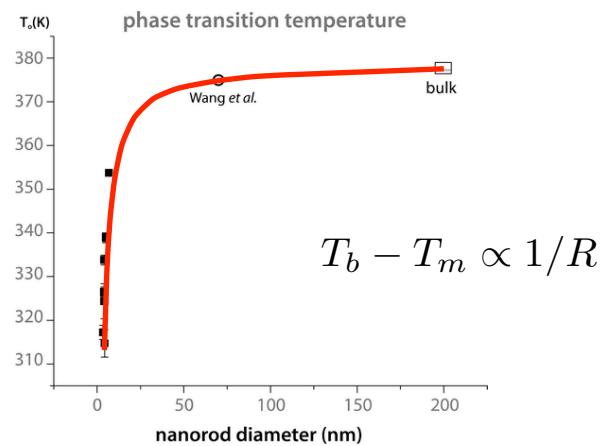
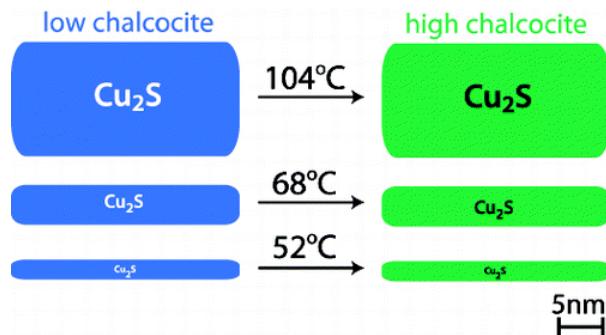
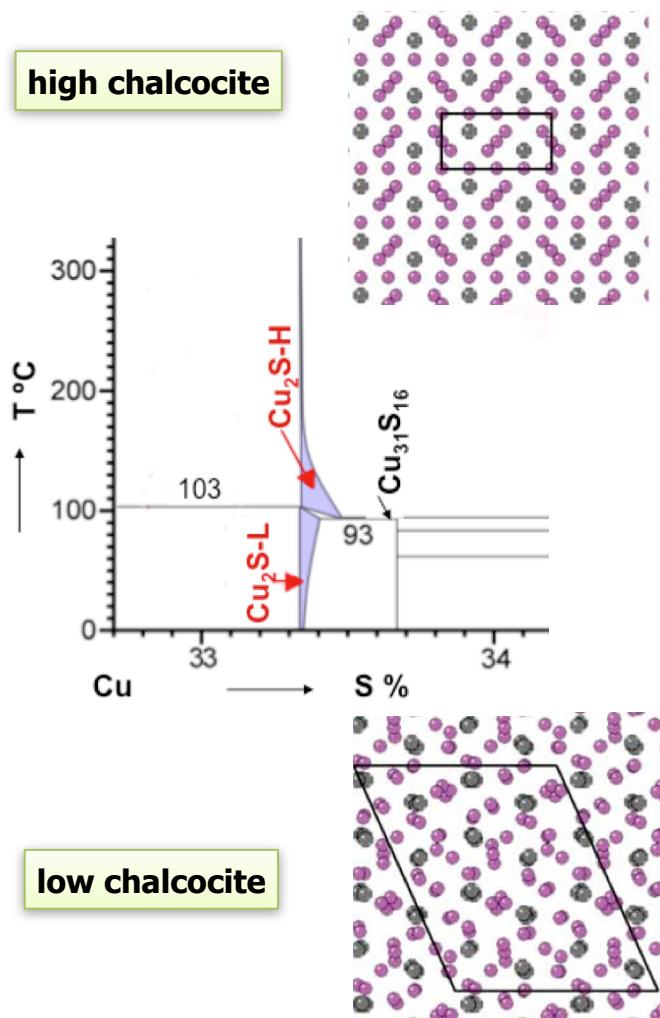


K. Jacobs, D. Zaziski, E. C. Scher, A. B. Herhold, and A. P. Alivisatos, Science 293, 1803 (2001)  
M. Grünwald and C. Dellago, Nano Letters 9, 2099 (2009)

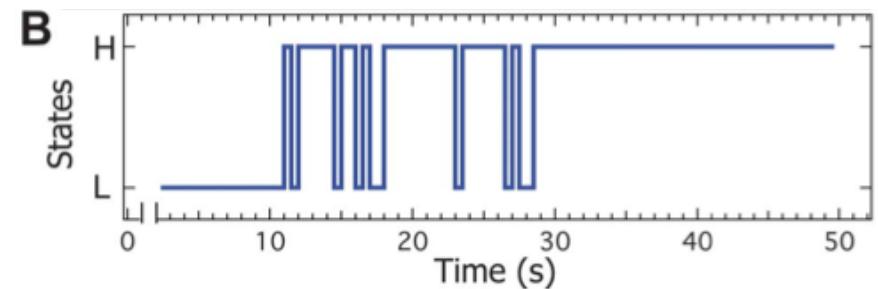
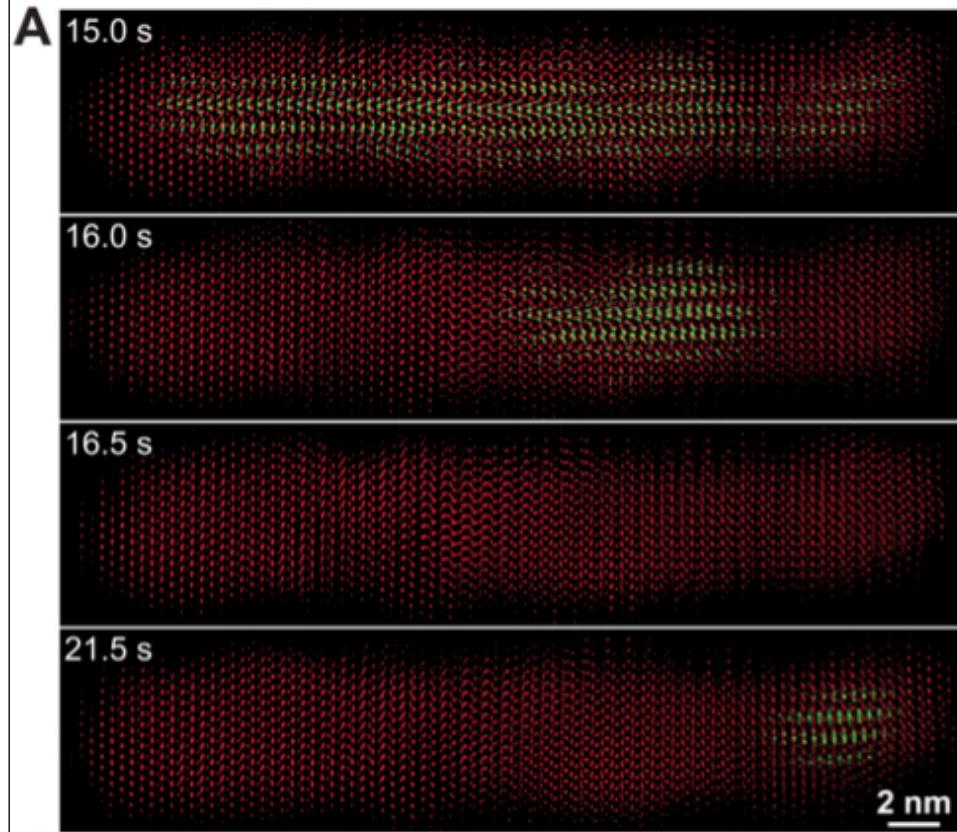
# $\text{Cu}_2\text{S}$ nanorods



# Low-to-high chalcocite transition in Cu<sub>2</sub>S nanorods



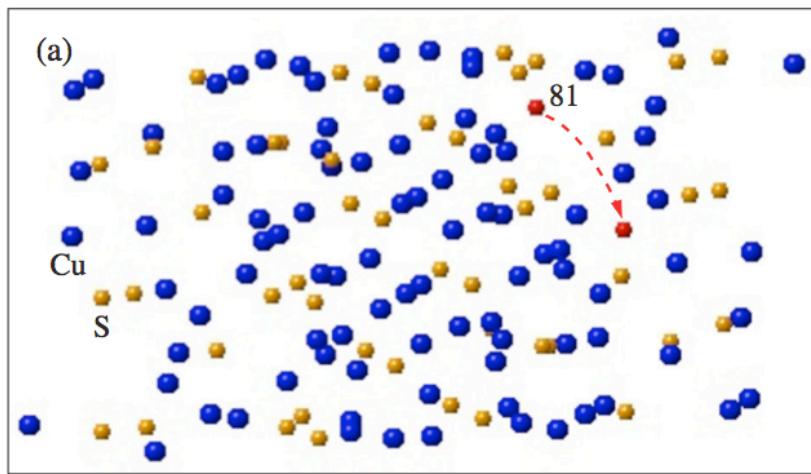
# Low- to high-chalcocite transition



$$P(\text{LC}) = \frac{1}{1 + \exp \left\{ \frac{\Delta E}{k_B T} \frac{(T - T_{\text{LH}})}{T_{\text{LH}}} \right\}}$$

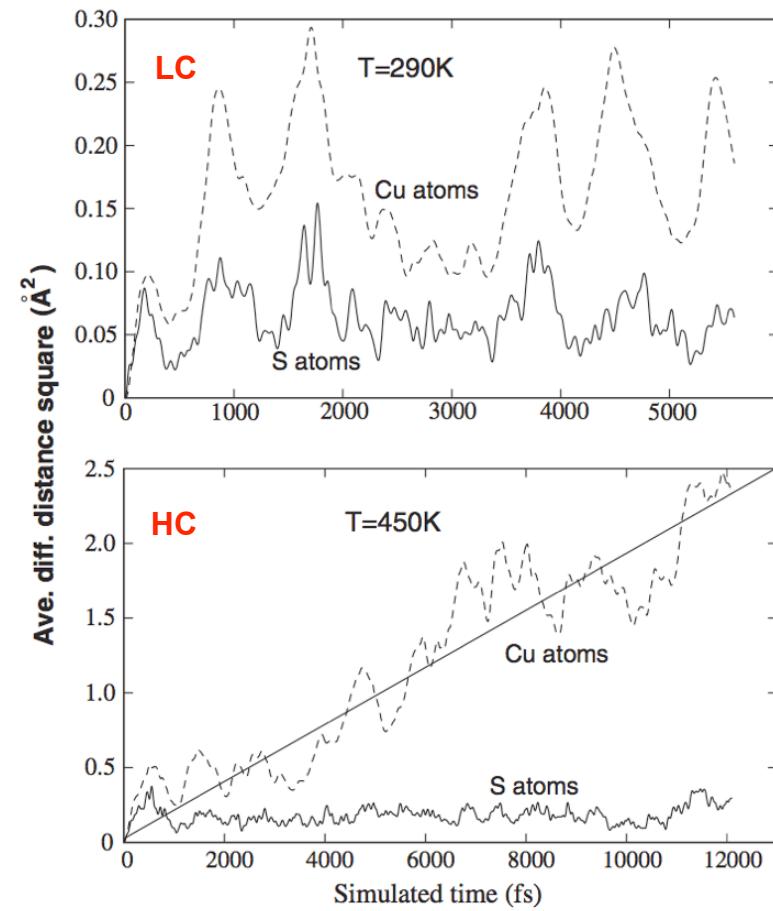
$$\Delta T = \frac{4k_B T_m^2}{\Delta E} \approx 0.2K$$

# $\text{Cu}_2\text{S}$ ab initio simulations



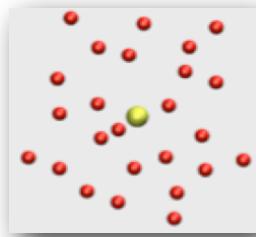
High chalcocite: N=48, 12 ps, T=450 K

Low chalcocite: N=144, 6 ps, T=290 K



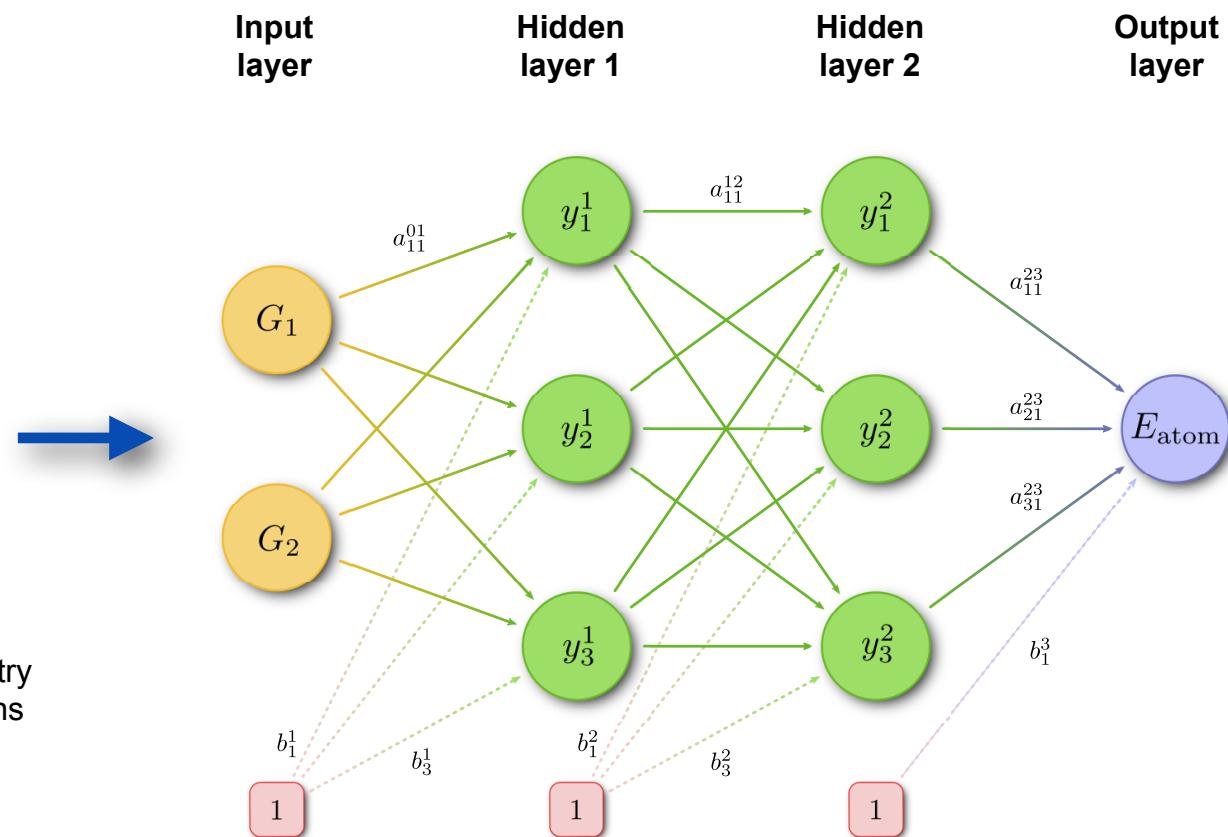
# Representing energy surfaces with neural networks

$$E = \sum_i E_i$$



Cartesian  
coordinates

$$\mathbf{R}_i \rightarrow G_j$$

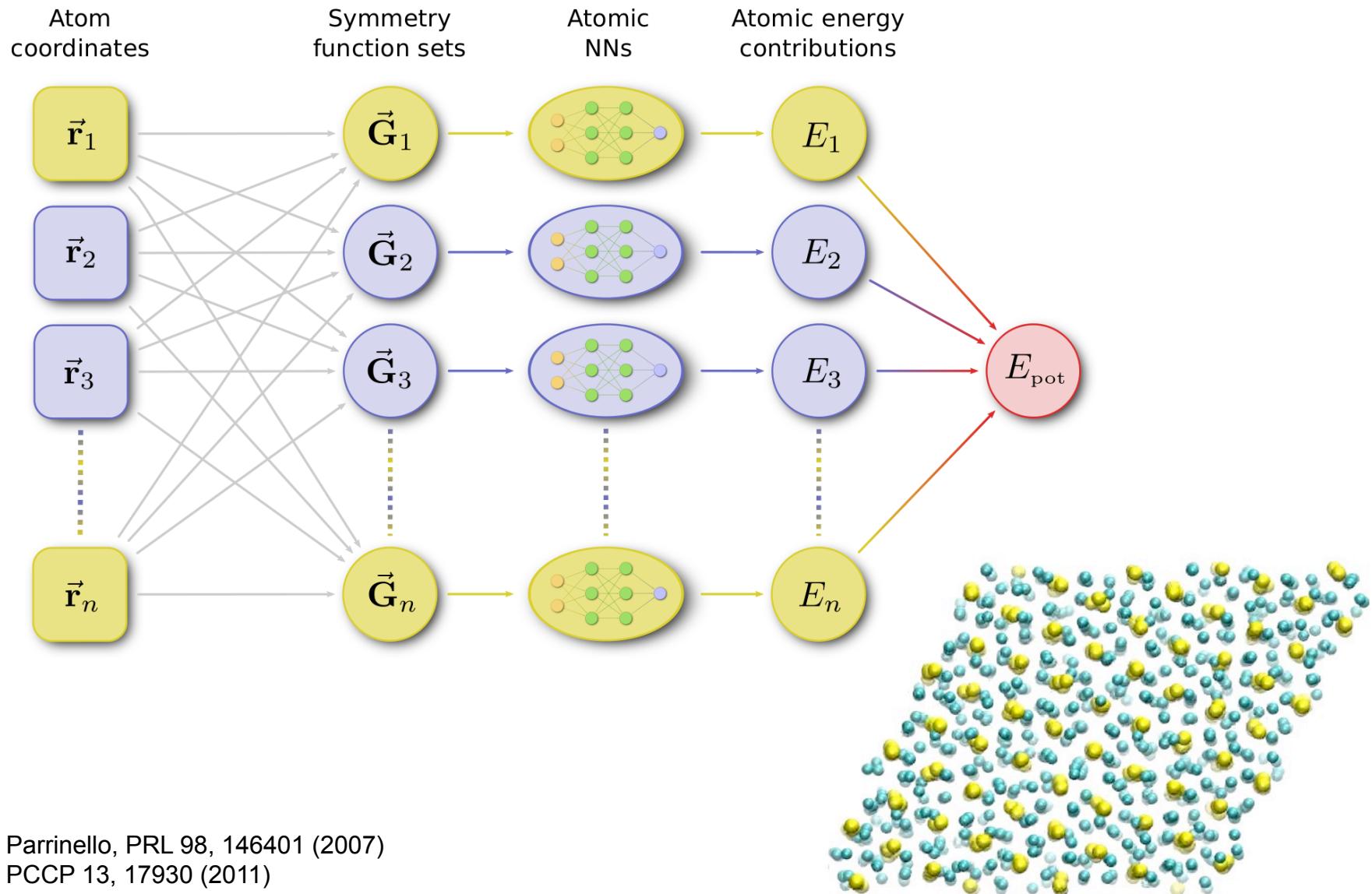


$$E_{\text{atom}} = \tilde{f} \left( b^3 + \sum_n a_n^{23} f \left( b_m^2 + \sum_m a_{mn}^{12} f \left( b_l^1 + \sum_l a_{lm}^{01} G_l \right) \right) \right)$$

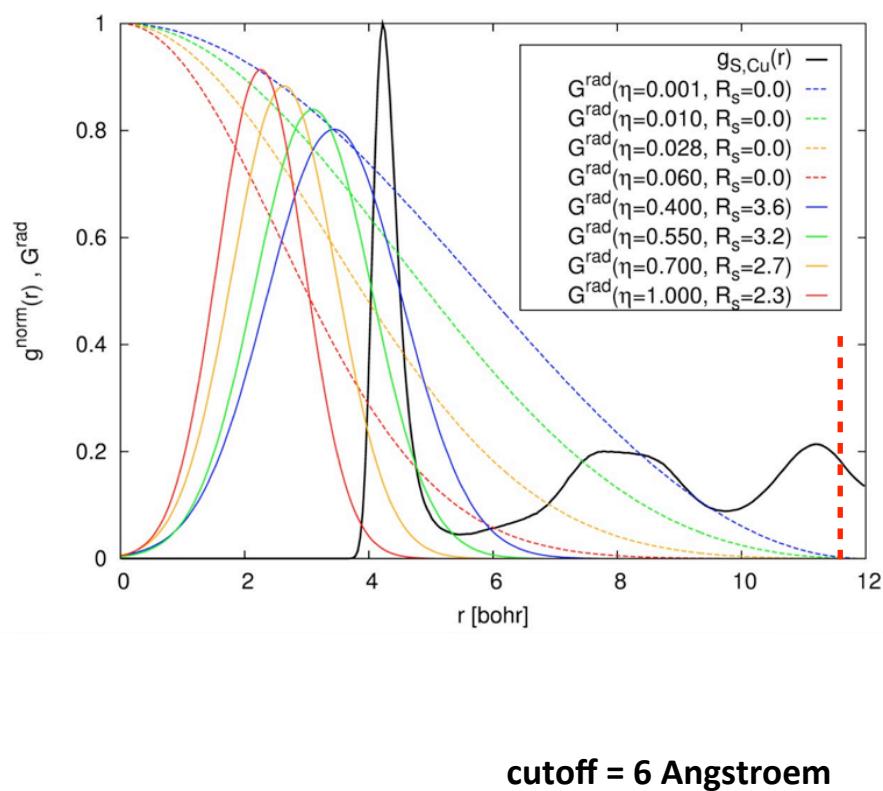
- Behler, Parrinello, PRL 98, 146401 (2007)  
Behler, Lorenz, Reuter, JCP 127, 014705 (2007)  
Behler, PCCP 13, 17930 (2011)

Al, Si, Na, Cu, ZnO, H<sub>2</sub>O

# One neural network per atom species



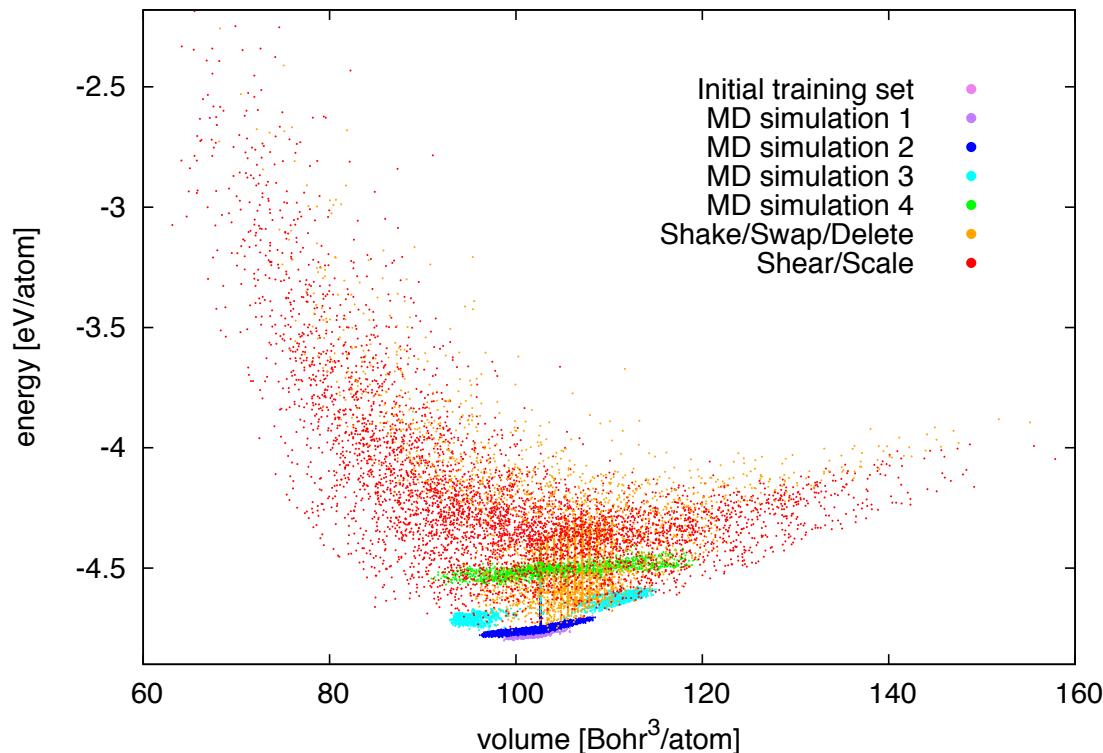
# Symmetry functions



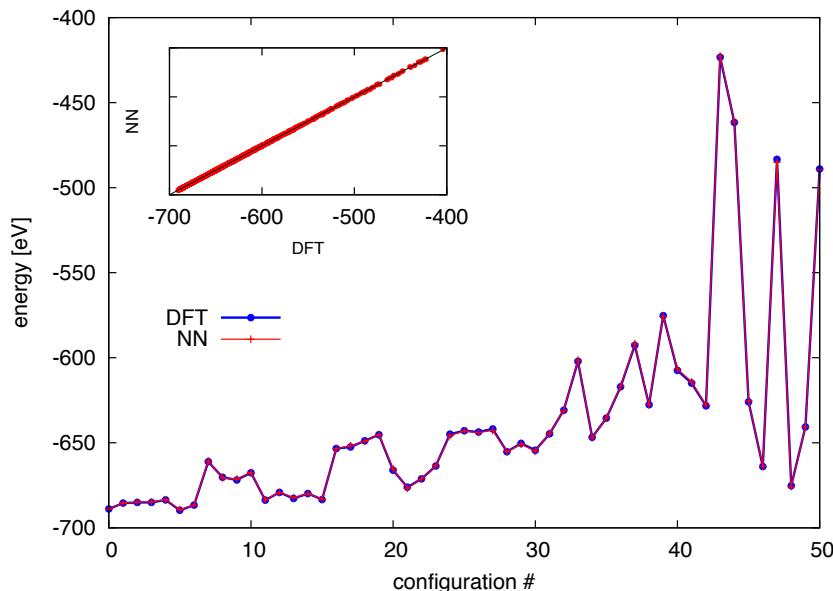
$$\begin{aligned}
 G_i^1 &= \sum_{j \neq i} f_c(R_{ij}), \\
 G_i^2 &= \sum_{j \neq i} e^{-\eta(R_{ij}-R_s)^2} f_c(R_{ij}), \\
 G_i^3 &= \sum_{j \neq i} \cos(\kappa R_{ij}) f_c(R_{ij}), \\
 G_i^4 &= 2^{-\zeta} \sum_{j,k \neq i} (1 + \lambda \cos \theta_{ijk})^\zeta \\
 &\quad \times e^{-\eta(R_{ij}^2 + R_{ik}^2 + R_{jk}^2)} f_c(R_{ij}) f_c(R_{ik}) f_c(R_{jk}), \\
 G_i^5 &= 2^{-\zeta} \sum_{j,k \neq i} (1 + \lambda \cos \theta_{ijk})^\zeta \\
 &\quad \times e^{-\eta(R_{ij}^2 + R_{ik}^2)} f_c(R_{ij}) f_c(R_{ik}), \\
 G_i^6 &= 2^{-\zeta} \sum_{j,k \neq i} (1 + \lambda \cos \theta_{ijk})^\zeta f_a(R_{ij}) f_a(R_{ik}), \\
 G_i^7 &= \frac{1}{2} \sum_{j,k \neq i} \sin[(\theta_{ijk} - \alpha)\eta] f_b(R_{ij}) f_b(R_{ik}),
 \end{aligned}$$

# Training set

- Perfect crystals (distorted)
- MD simulations NpT, NVT (different T and p)
- Shear + scale
- Displace + exchange + delete
- Reference energies and forces ab initio
- 21.000 training configurations

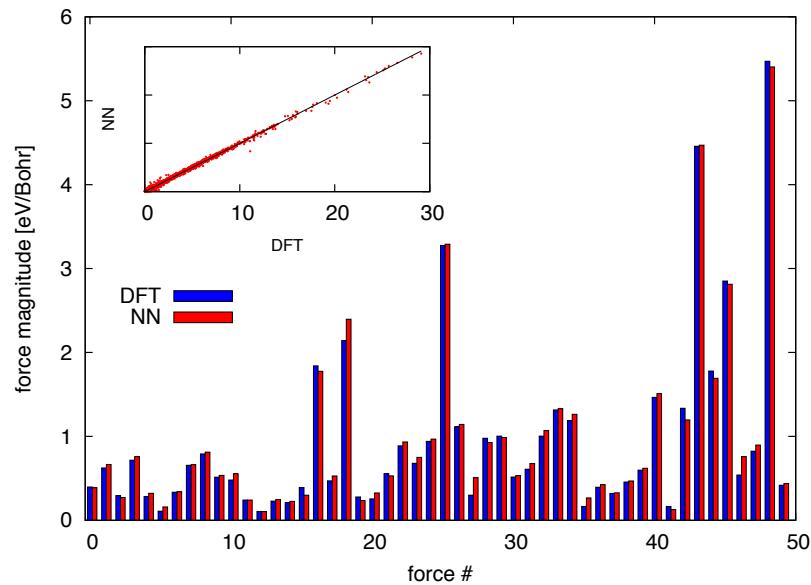


# Energy and force prediction from NN



## Neural Network

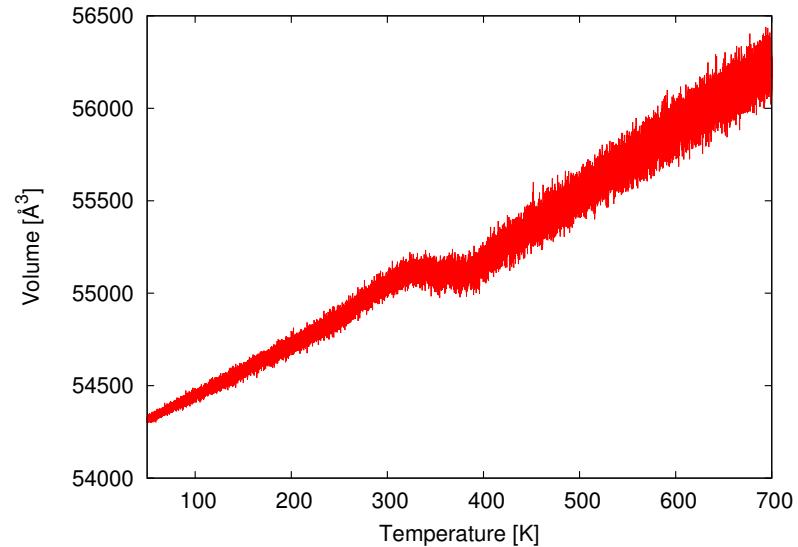
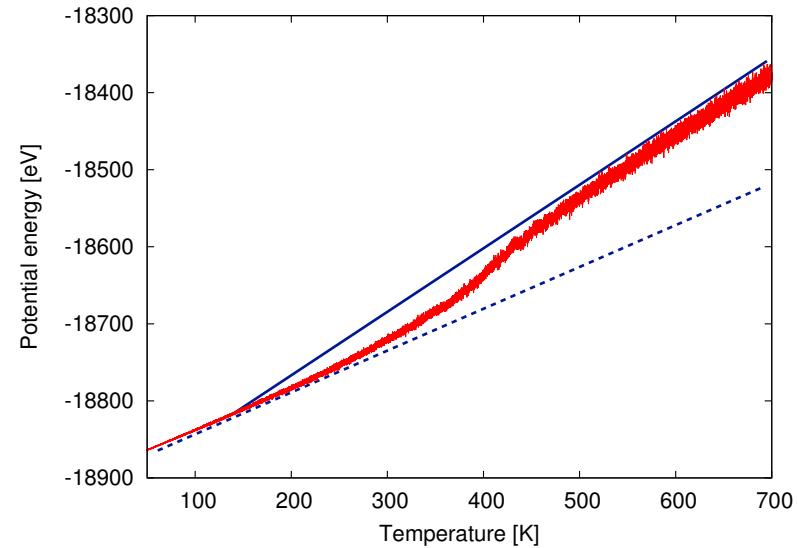
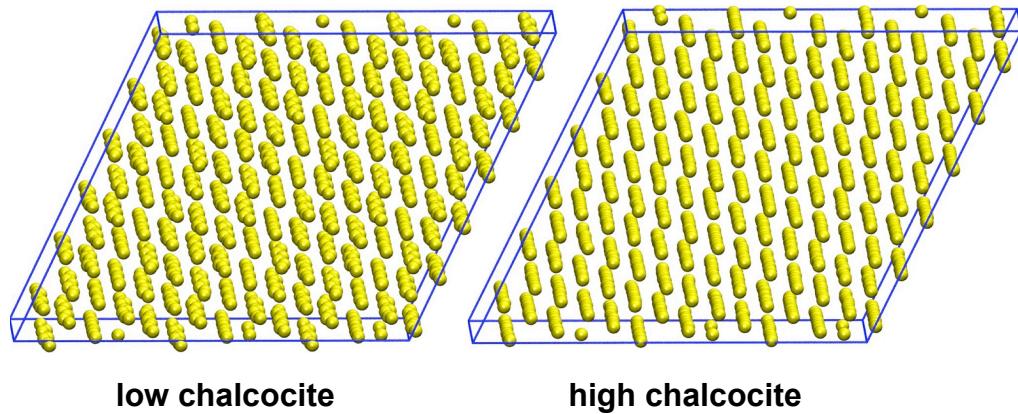
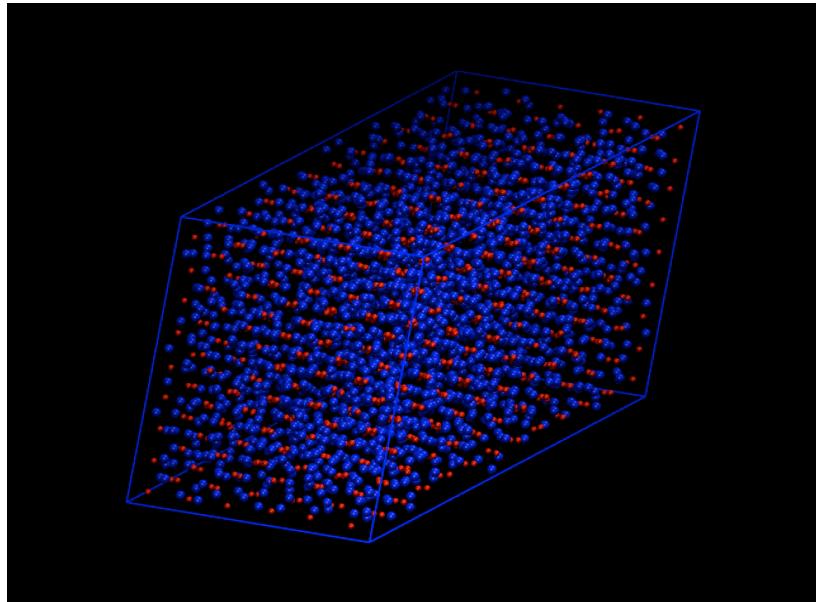
- 50 symmetry functions (Cu, S)
- 2000 weights
- 2 hidden layers 25 nodes
- cutoff: 6 Å
- training: Kalman filter
- minimize errors in forces and energies



## Errors

- Energies: 3 meV/atom
- Forces: 80 meV/Bohr

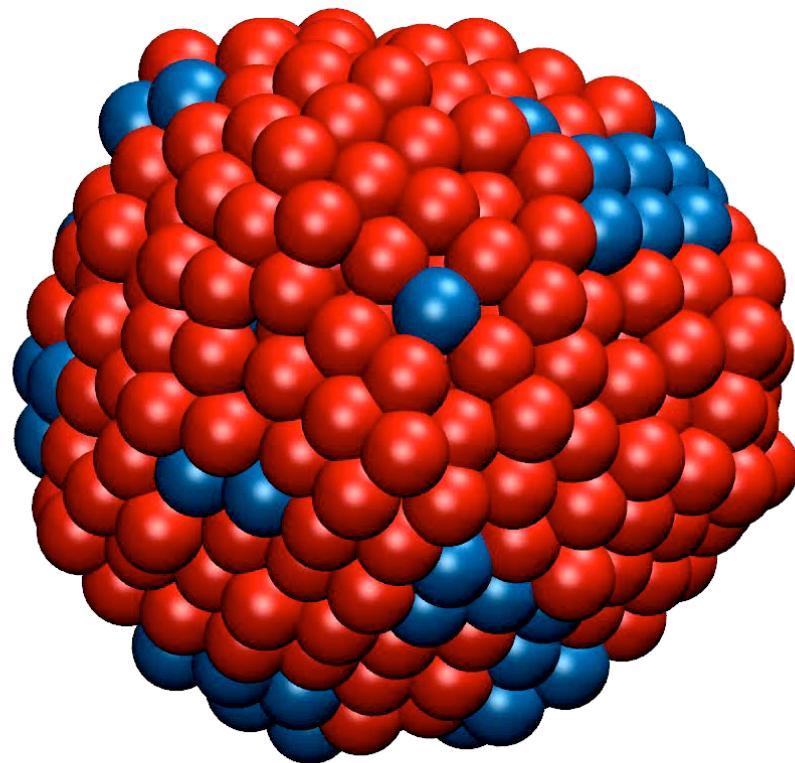
# Low- to high-chalocite transition in Cu<sub>2</sub>S



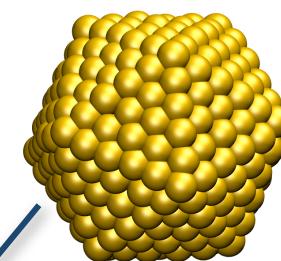
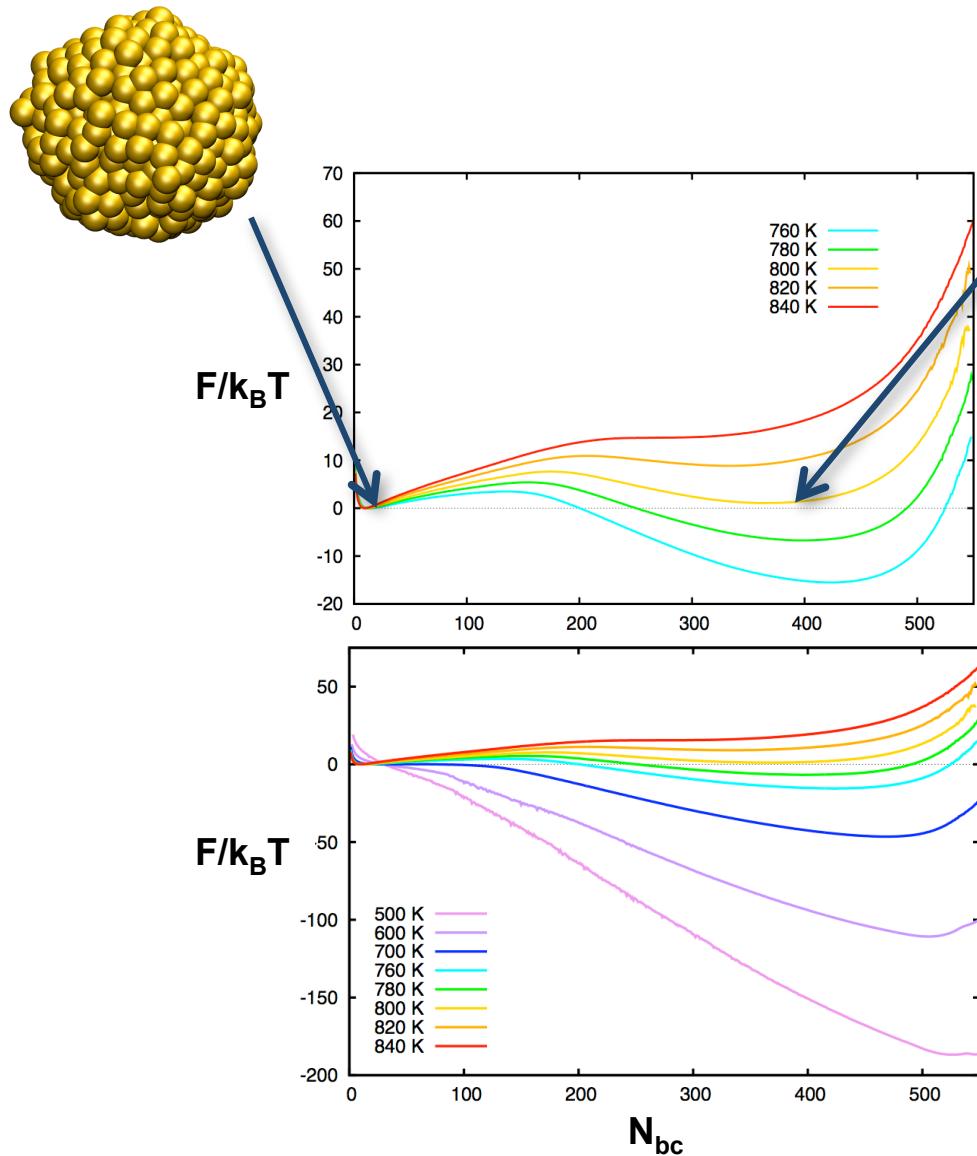
# Melting/freezing of Au nanoparticles

T=760 K, N=561

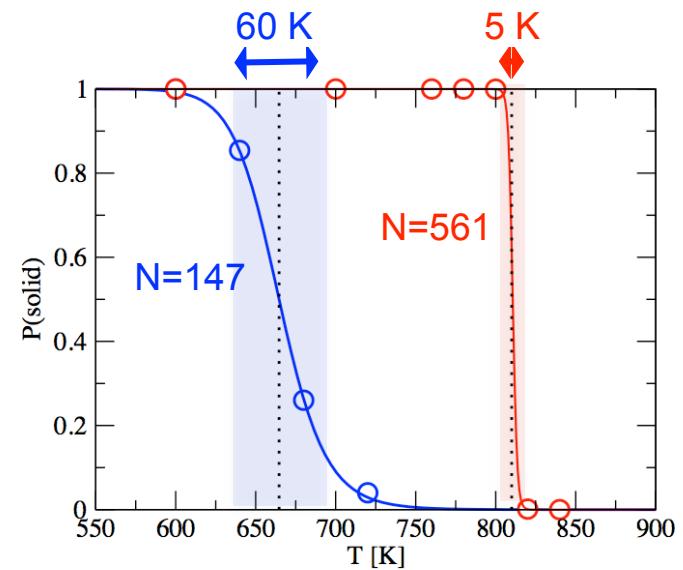
blue = crystalline  
red=liquid



# Free energy profiles



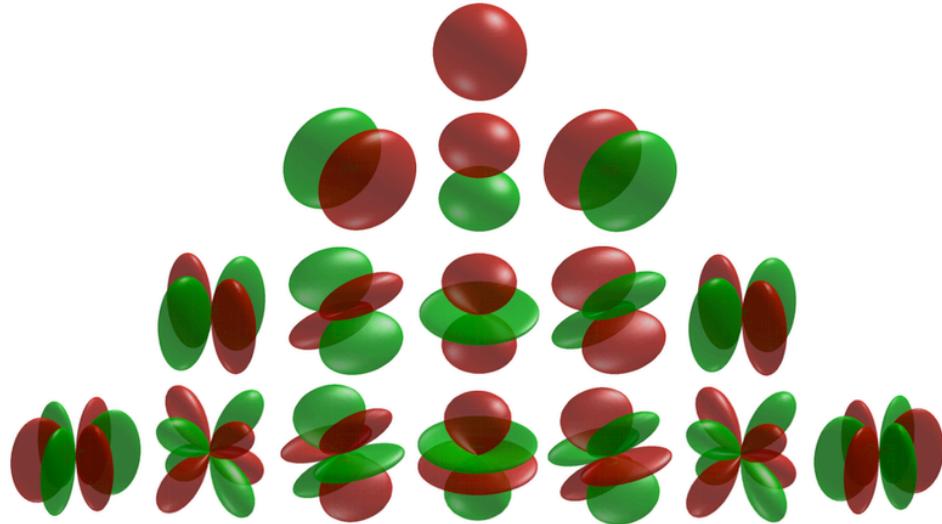
$$\Delta T = \frac{4k_B T_m^2}{\Delta E}$$



$$P(\text{solid}) = \frac{1}{1 + \exp \left\{ \frac{\Delta E}{k_B T_m} \frac{(T-T_m)}{T} \right\}}$$

# Local structure determination

## Steinhardt bond order parameters

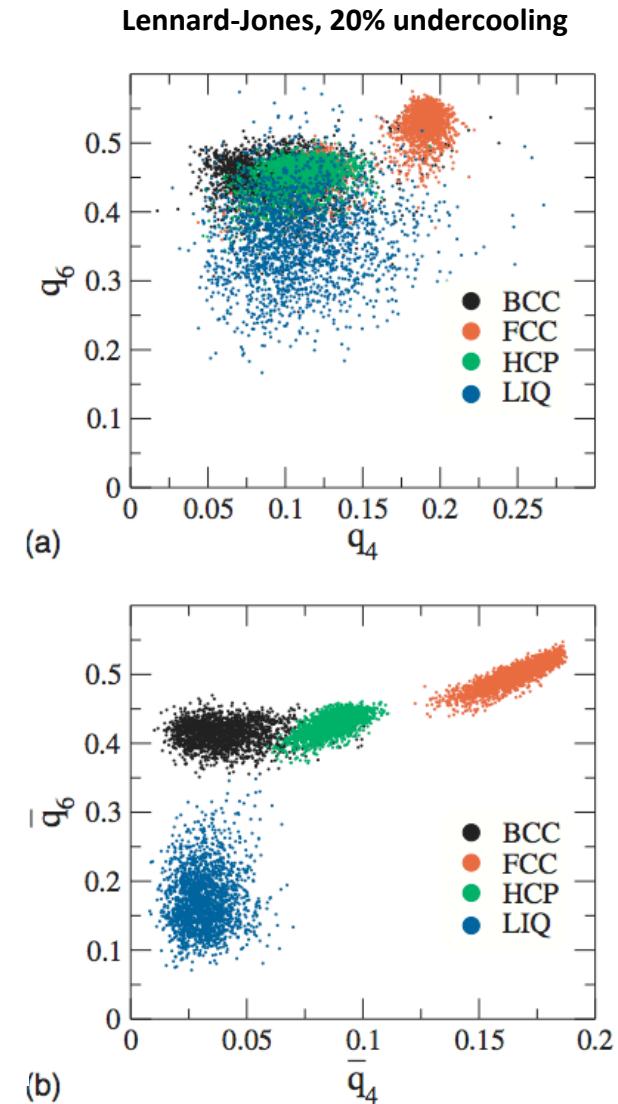


$$q_{lm}(i) = \frac{1}{N_b(i)} \sum_{j=1}^{N_b(i)} Y_{lm}(\mathbf{r}_{ij}) \quad q_l(i) = \sqrt{\frac{4\pi}{2l+1} \sum_{m=-l}^l |q_{lm}(i)|^2}$$

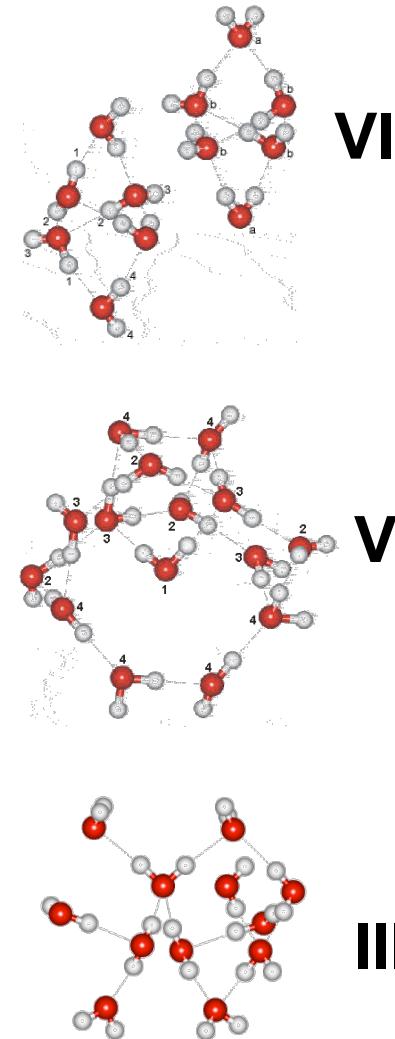
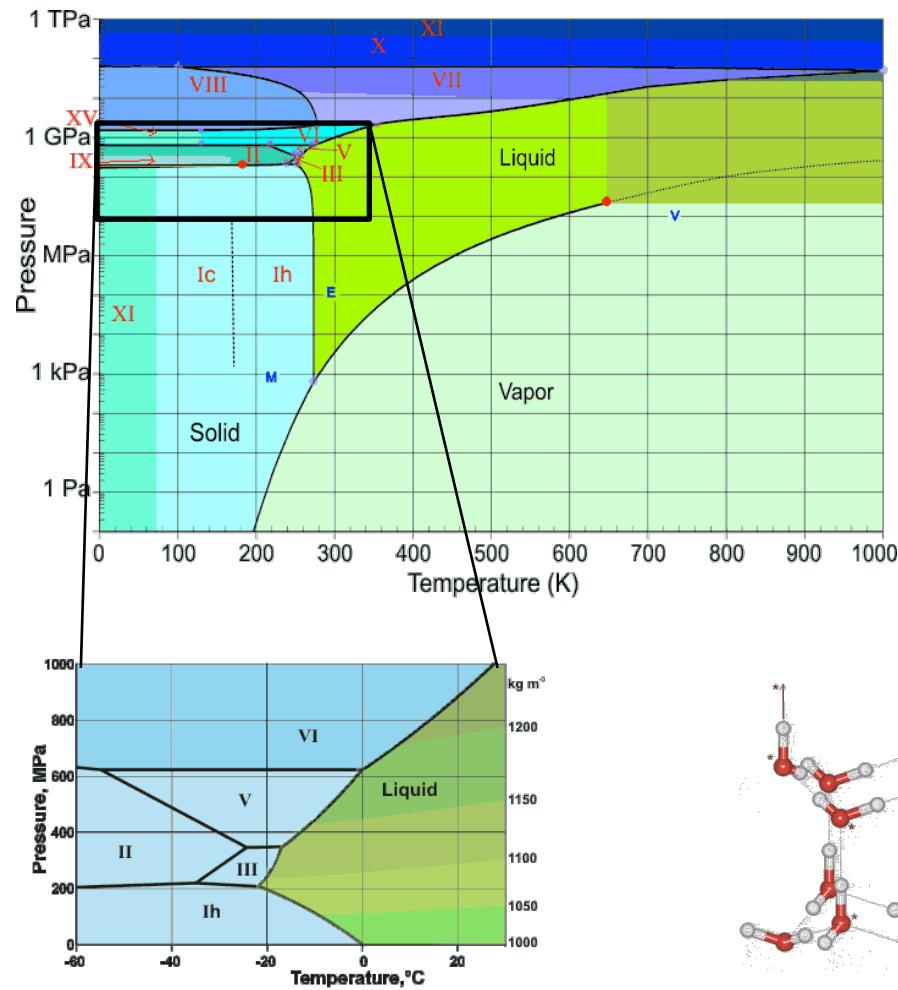
correlation  
sign of crystallinity

$$S_{ij} = \sum_{m=-6}^6 q_{6m}(i) q_{6m}^*(j)$$

P. Steinhardt, D. R. Nelson, and M. Ronchetti, PRB 28, 784 (1983)  
 Ten Wolde, Ruiz-Montero, Frenkel, JCP 110, 1591 (1999)  
 W. Lechner and C. Dellago, JCP 129, 114707 (2008)

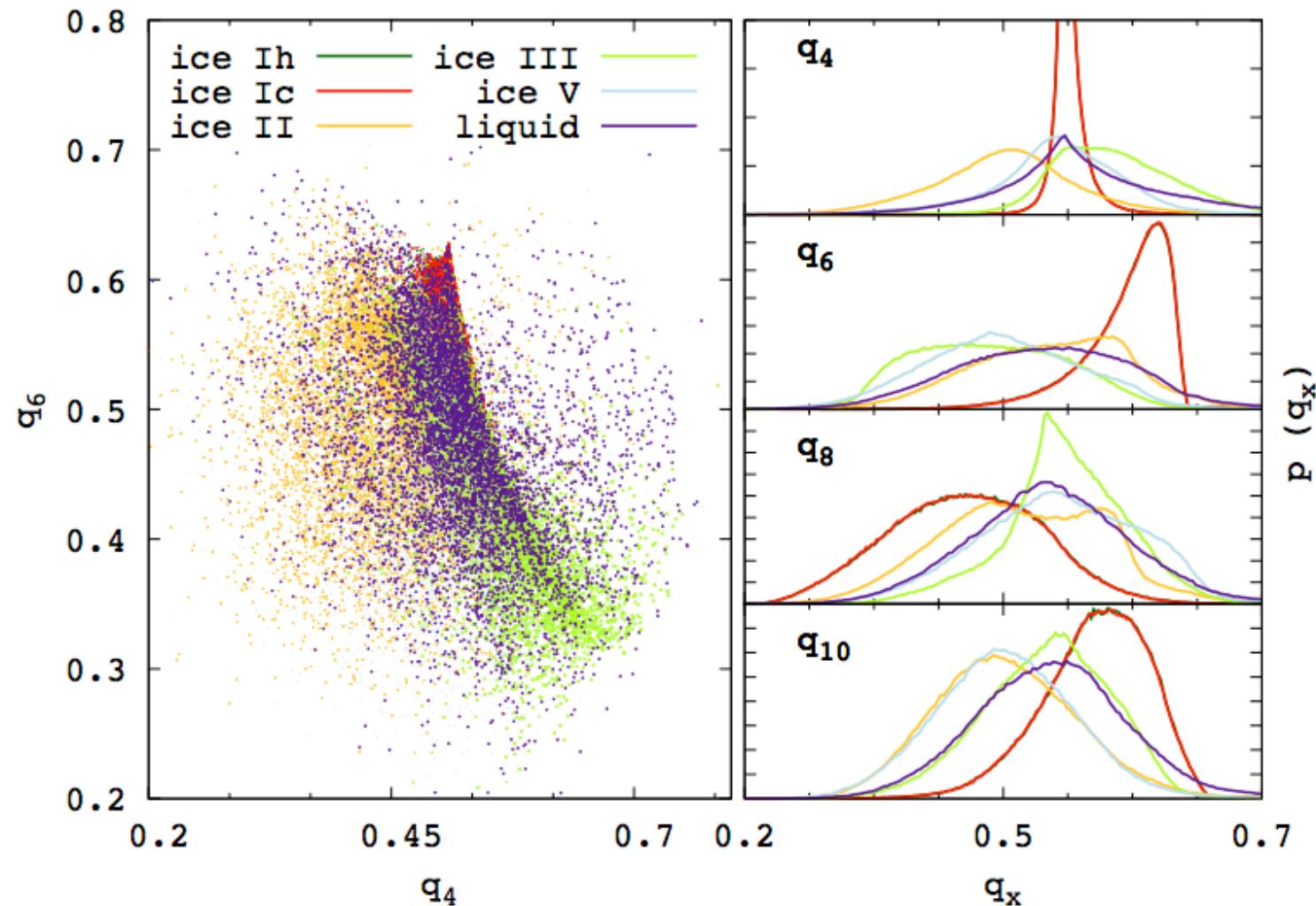


# Liquid water and ice



T. Lörting (Univ. Innsbruck)

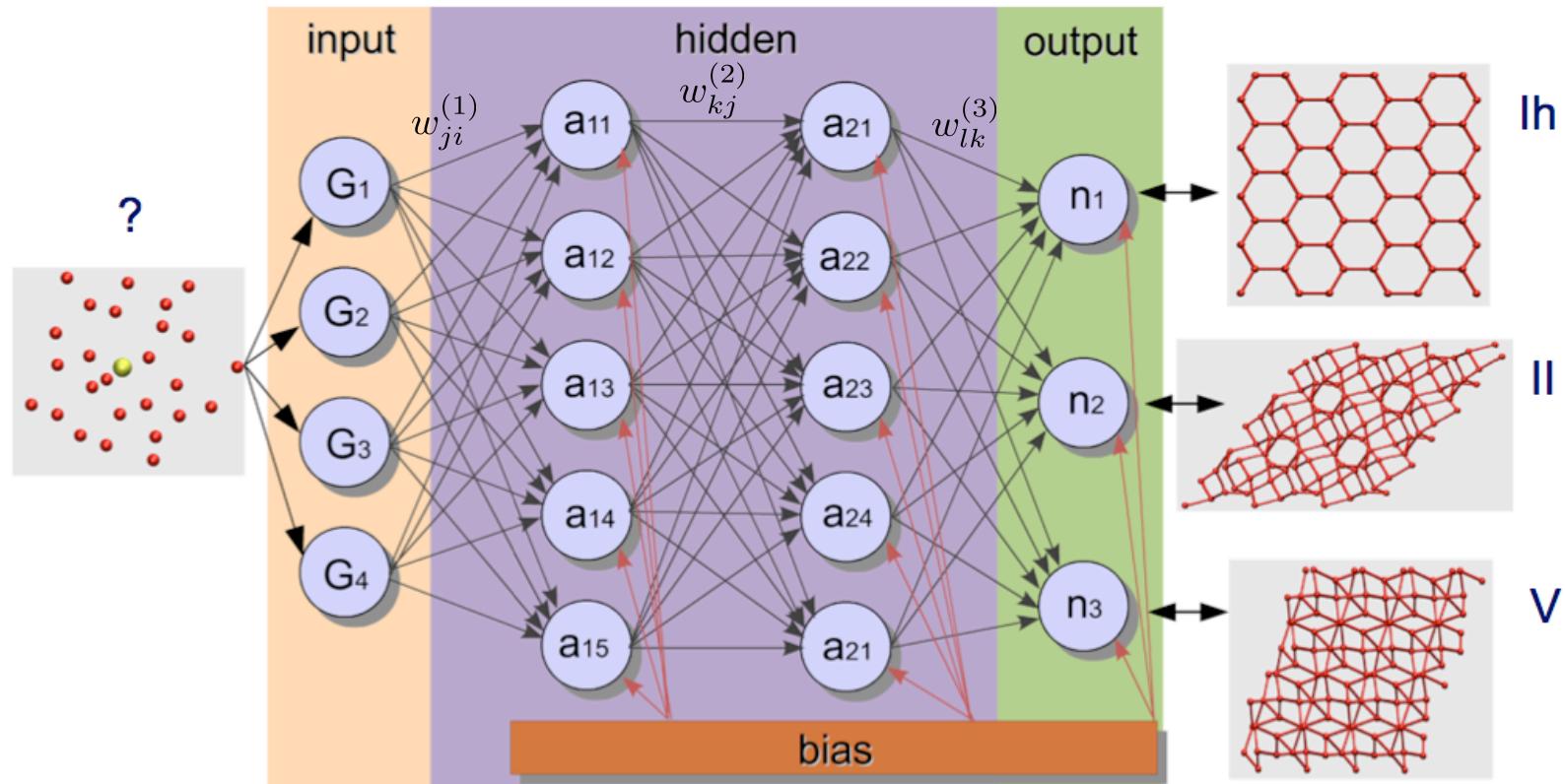
# Do bond order parameters work for ice?



# Ideal properties of order parameters

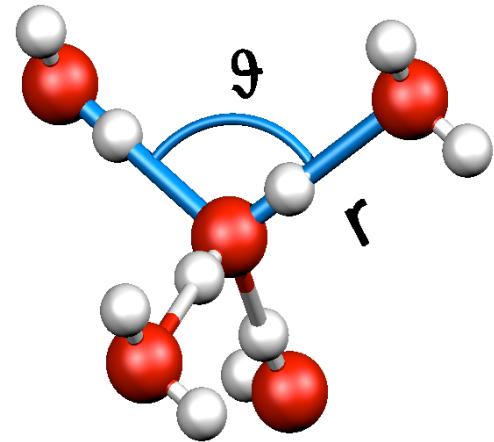
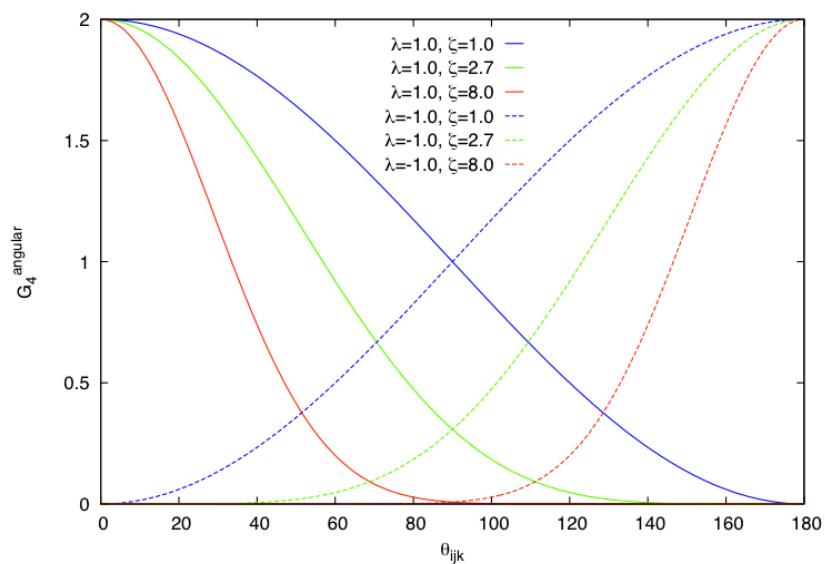
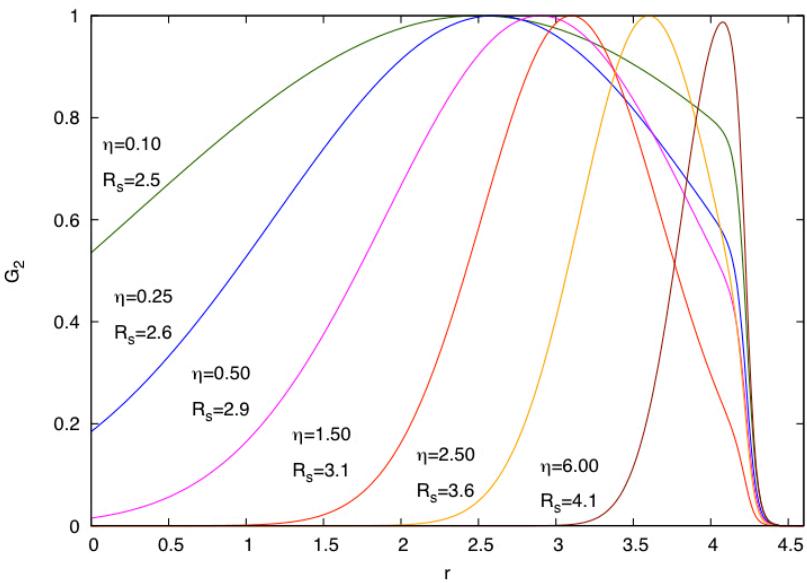
- ▶ Accurate assignment of structures
- ▶ Invariant with respect to rotations, translations, permutations
- ▶ Works over wide range of p & T
- ▶ Robust with respect to thermal fluctuations & elastic distortions
- ▶ Recognize defects and structures at interfaces
- ▶ Flexible
- ▶ Computationally inexpensive

# Neural network for structure recognition



- 30 symmetry functions, 2x30 hidden nodes, 5 output nodes
- $\approx 2000$  weights
- Training with global extended Kalman filter to find optimum weights
- Neural network training on GPU (takes a few days)
- Improvement of symmetry functions by sensitivity analysis

# Symmetry functions



$$G_i^1 = \sum_{j \neq i} f_c(R_{ij}),$$

$$G_i^2 = \sum_{j \neq i} e^{-\eta(R_{ij}-R_s)^2} f_c(R_{ij}),$$

$$G_i^3 = \sum_{j \neq i} \cos(\kappa R_{ij}) f_c(R_{ij}),$$

$$G_i^4 = 2^{-\zeta} \sum_{j,k \neq i} (1 + \lambda \cos \theta_{ijk})^\zeta \\ \times e^{-\eta(R_{ij}^2 + R_{ik}^2 + R_{jk}^2)} f_c(R_{ij}) f_c(R_{ik}) f_c(R_{jk}),$$

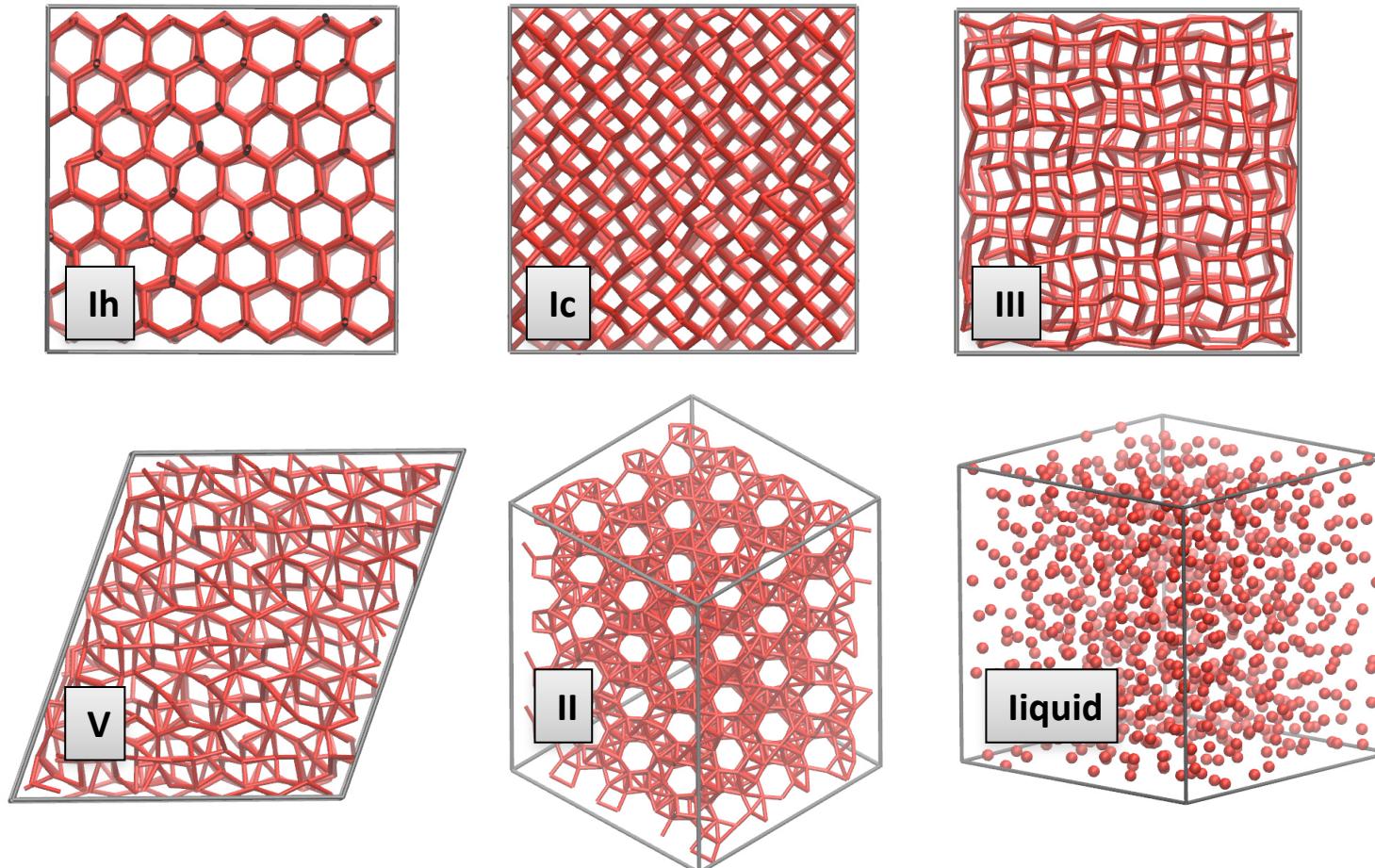
$$G_i^5 = 2^{-\zeta} \sum_{j,k \neq i} (1 + \lambda \cos \theta_{ijk})^\zeta \\ \times e^{-\eta(R_{ij}^2 + R_{ik}^2)} f_c(R_{ij}) f_c(R_{ik}),$$

$$G_i^6 = 2^{-\zeta} \sum_{j,k \neq i} (1 + \lambda \cos \theta_{ijk})^\zeta f_a(R_{ij}) f_a(R_{ik}),$$

$$G_i^7 = \frac{1}{2} \sum_{j,k \neq i} \sin[(\theta_{ijk} - \alpha)\eta] f_b(R_{ij}) f_b(R_{ik}),$$

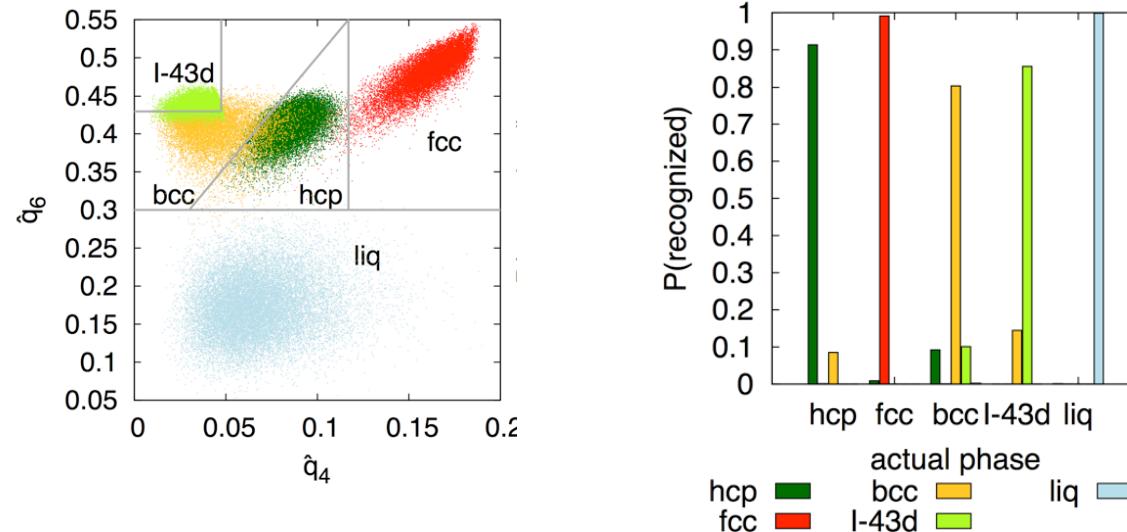
# Training set & test set

- Run MD simulations at **different p and T** for the **pure phases of H<sub>2</sub>O**
- Pick 30.000 **uncorrelated** local environments from MD configurations

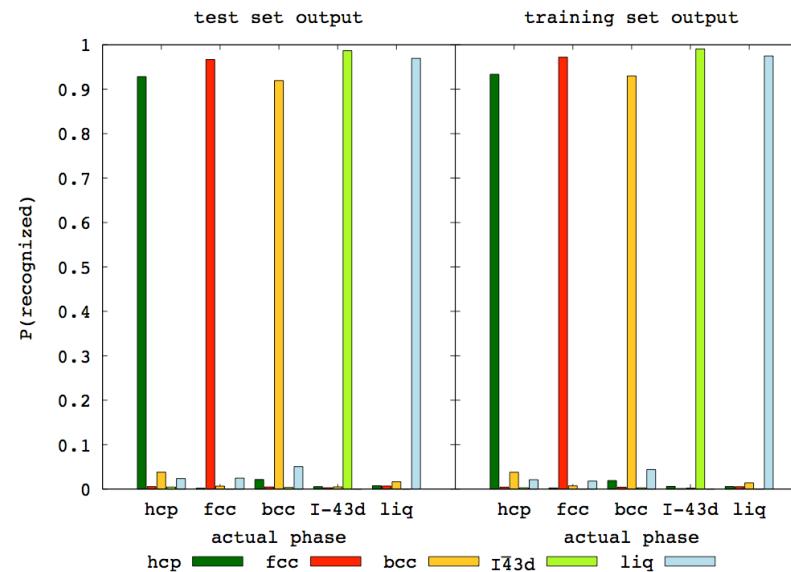


# Structure prediction for Lennard-Jonesium

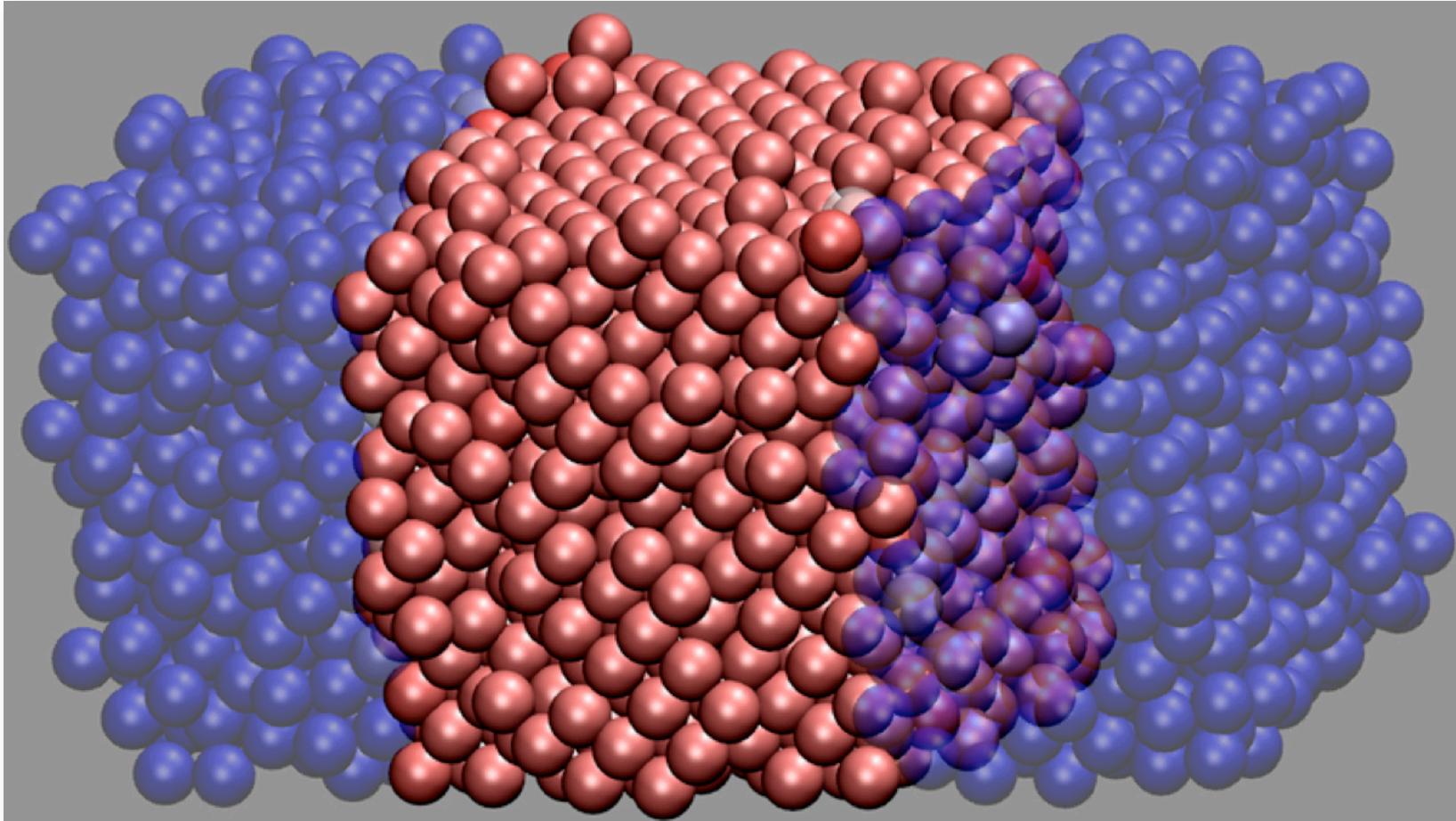
## bond order parameters



## neural network

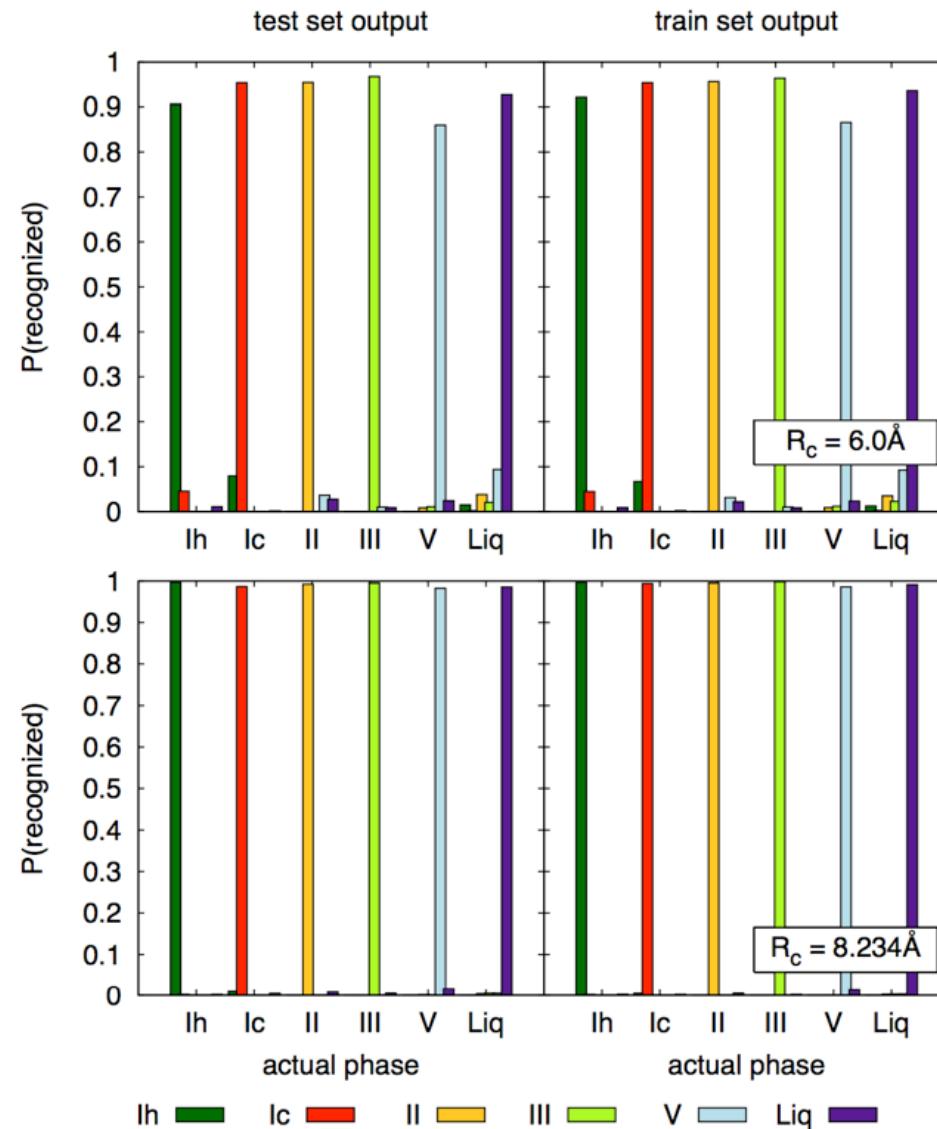


# fcc slab in Lennard-Jones liquid



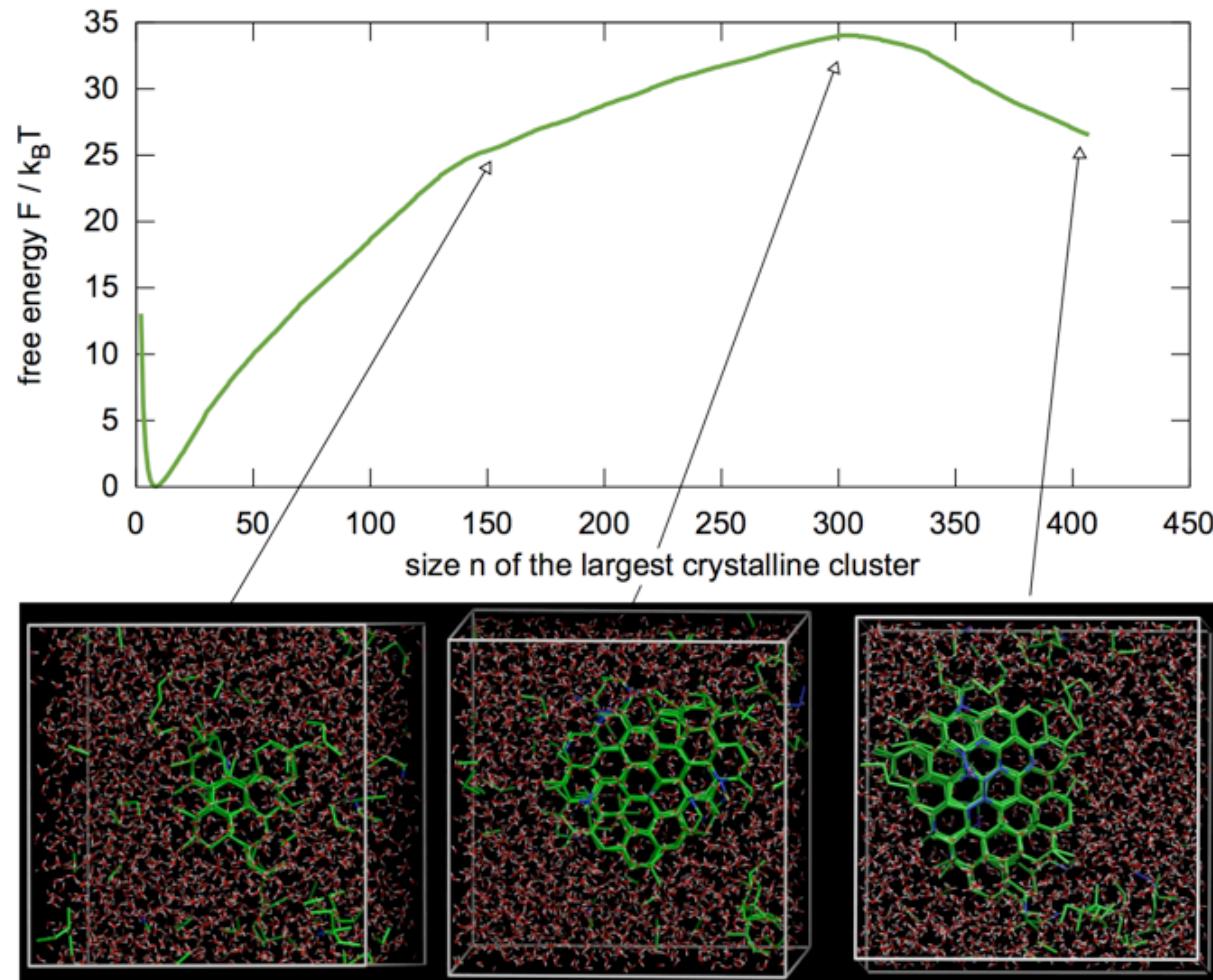
# Structure Prediction for Water and Ice

T=270K, various pressures, 30.000 training structures



# Nucleation of Ice

TIP4P/ice, T=235K, umbrella sampling + replica exchange



# Pro and cons of neural networks

- \* **Flexible, no assumptions**
- \* **Accuracy of DFT at fraction of cost**
- \* **Accurate structure recognition**
- \* **Need new NN for every new material**
- \* **Unclear where information resides in NN (black box)**