

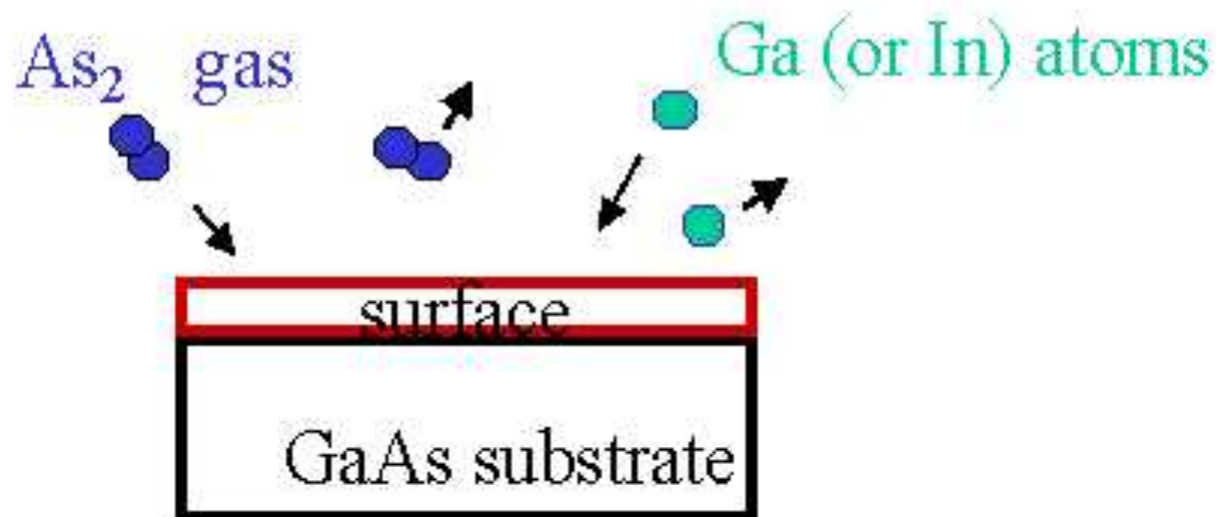
Surface diffusion, growth, and self-assembly of quantum dots at III-V semiconductor surfaces

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<http://www.fhi-berlin.mpg.de/th/th.html>

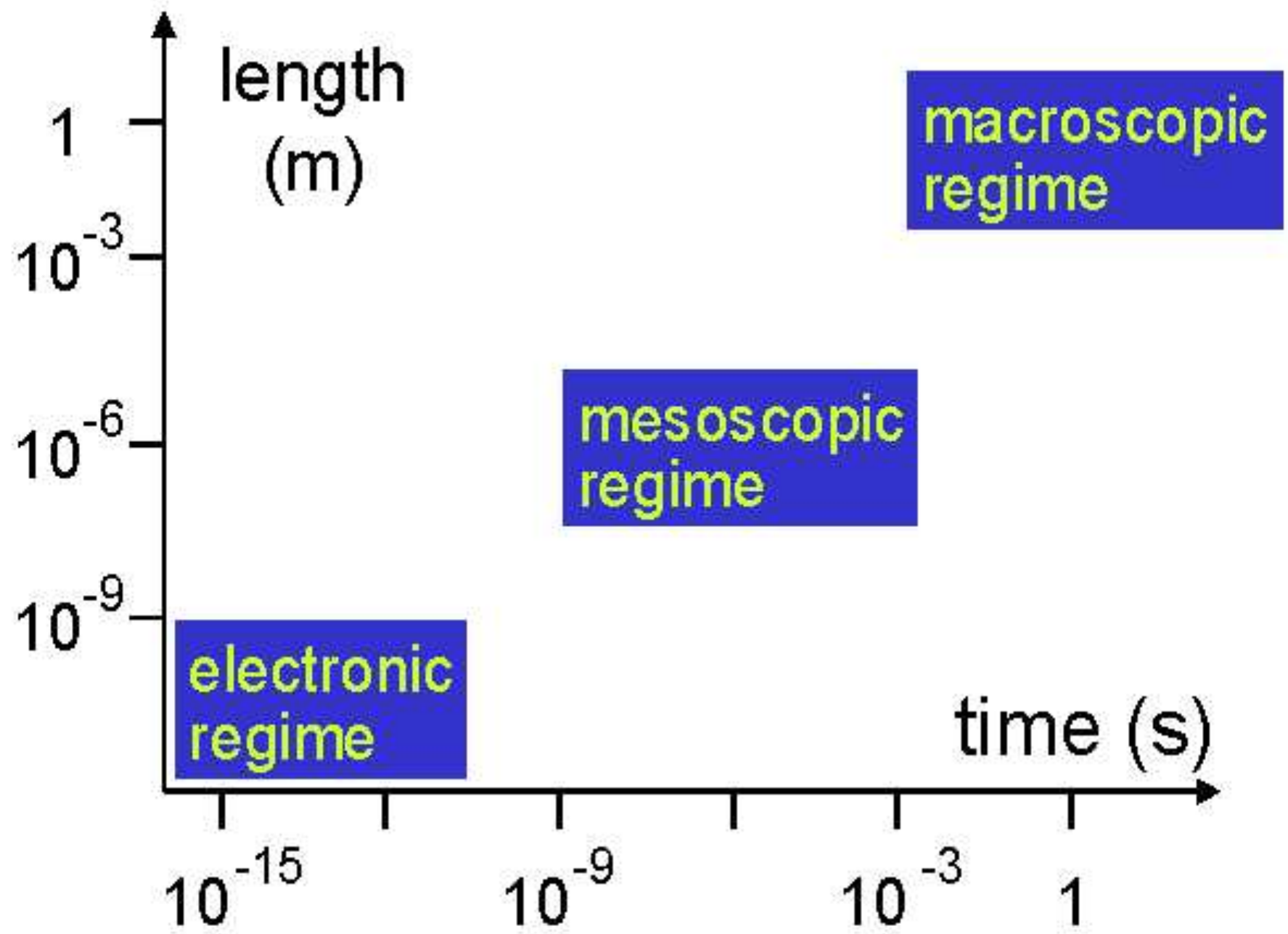
Microscopic processes controlling the growth of III-V semiconductors



- 1) deposition of Ga and As
- 2) adsorption of Ga
- 3) diffusion of Ga
- 4) desorption of Ga
- 5) adsorption of As₂ ?
- 6) dissociation of As₂ ?
- 7) diffusion of As
- 8) desorption of As
- 9) island nucleation
- 10) growth

key investigator: Peter Kratzer

- What are the microscopic processes to achieve stoichiometric growth?
- How do quantum dots form, and why do they assume a narrow size distribution (sometimes)



Density Functional Theory

The energy of the ground state of a many-electron system : $E_0(\{\mathbf{R}_I\}) = \text{Min}_{\Psi} \langle \Psi | H^e | \Psi \rangle$

Hohenberg and Kohn (1964): The functional

$$n(\mathbf{r}) = n[\Psi] = \langle \Psi | \sum_i \delta(\mathbf{r} - \mathbf{r}_i) | \Psi \rangle$$

can be inverted, *i.e.*,

$$\Psi(\mathbf{r}_1, \mathbf{r}_2, \dots, \mathbf{r}_N) = \Psi[n(\mathbf{r})] .$$

This implies:

$$E_0(\{\mathbf{R}_I\}) = \text{Min}_{n(\mathbf{r})} E_{\{\mathbf{R}\}}[n]$$

Kohn and Sham (1964):

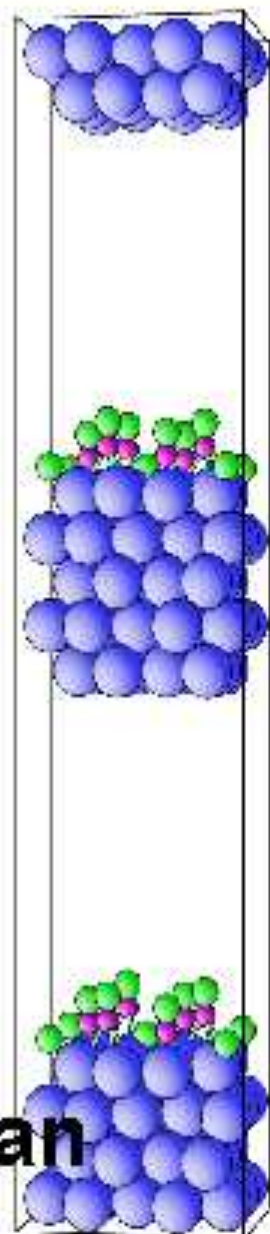
$$E_{\{\mathbf{R}_I\}}[n] = T_s[n] + \int d^3r v_{\{\mathbf{R}_I\}}^{\text{nuc}}(\mathbf{r})n(\mathbf{r}) \\ + \frac{1}{2} \int \int d^3r d^3r' \frac{n(\mathbf{r})n(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} + E^{\text{xc}}[n]$$

with **local-density approximation**
or **generalized gradient approximation**

Accuracy of geometries is better than 0.1 Å. Accuracy of calculated energies (relative) is better than 0.2 eV [for special cases better than 0.01 eV].

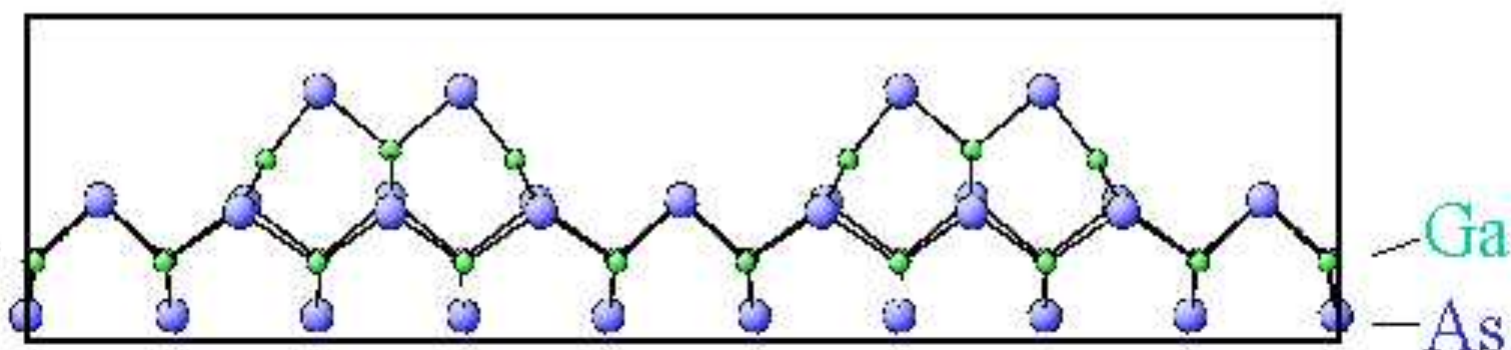
Methods

- I** {
- **Density functional theory**
 - *ab initio* pseudopotentials
(fhi96md --- <http://www.fhi-berlin.mpg.de/th/th.html>)
 - FP-LAPW
(the WIEN - code by *P. Blaha, K. Schwarz, et al.; M. Petersen et al., CPC XX (2000)*)
- II** {
- *ab initio* Molecular Dynamics
 - *ab initio* Quantum Dynamics
 - *ab initio* Lattice Gas Hamiltonian
 - *ab initio* kinetic Monte Carlo

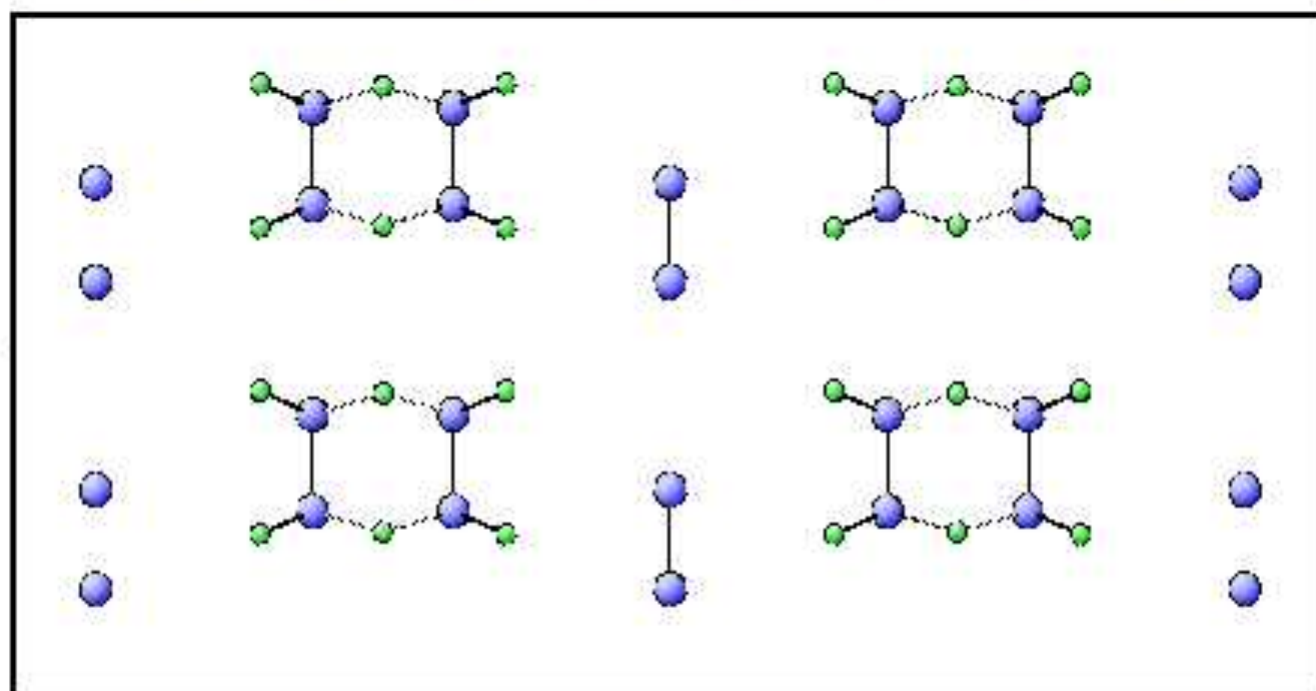


$\beta 2$ reconstruction of GaAs (001) (2x4) unit cell

side
view



top
view

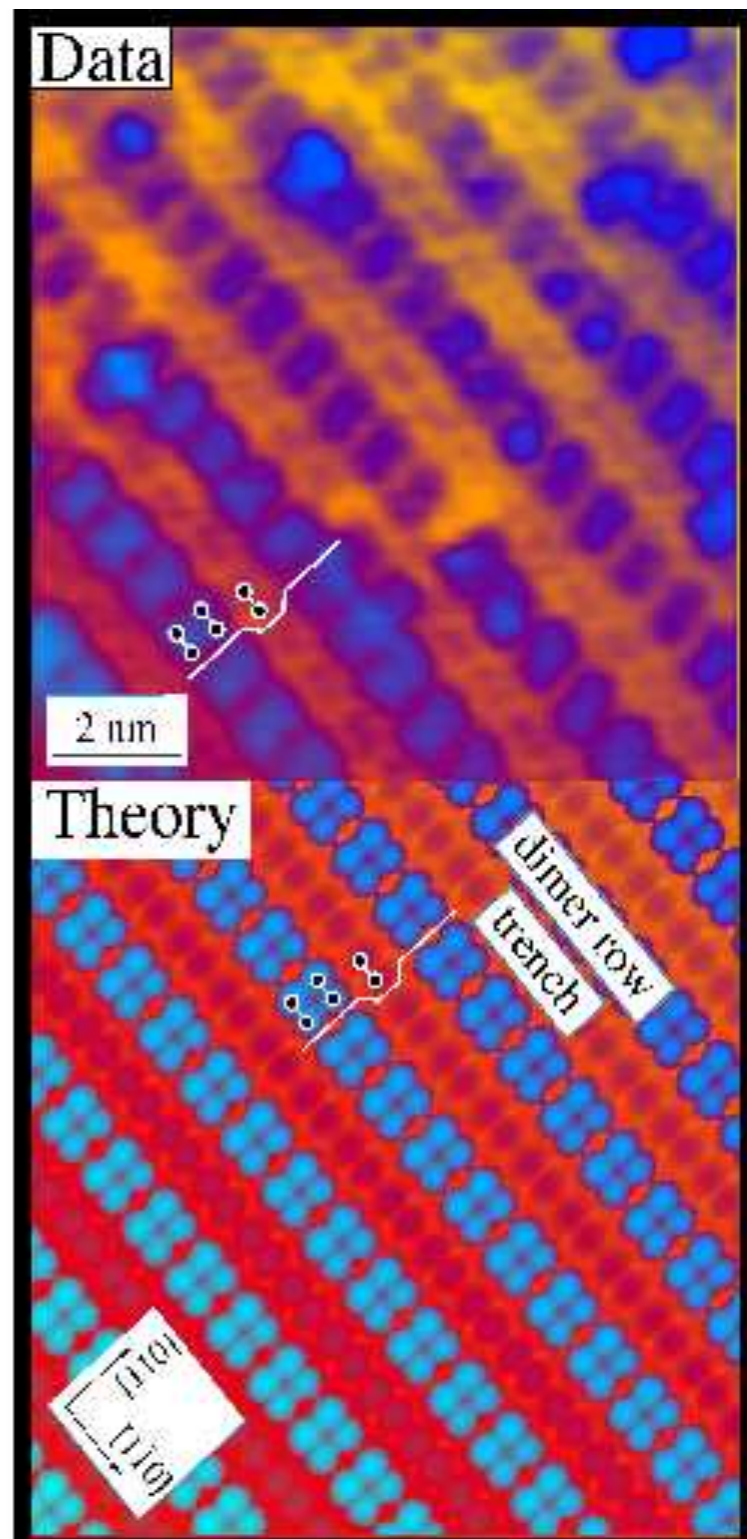


STM imaging of GaAs(001)

measured filled state image
at $V_{\text{tip}} = -2.1 \text{ eV}$ →

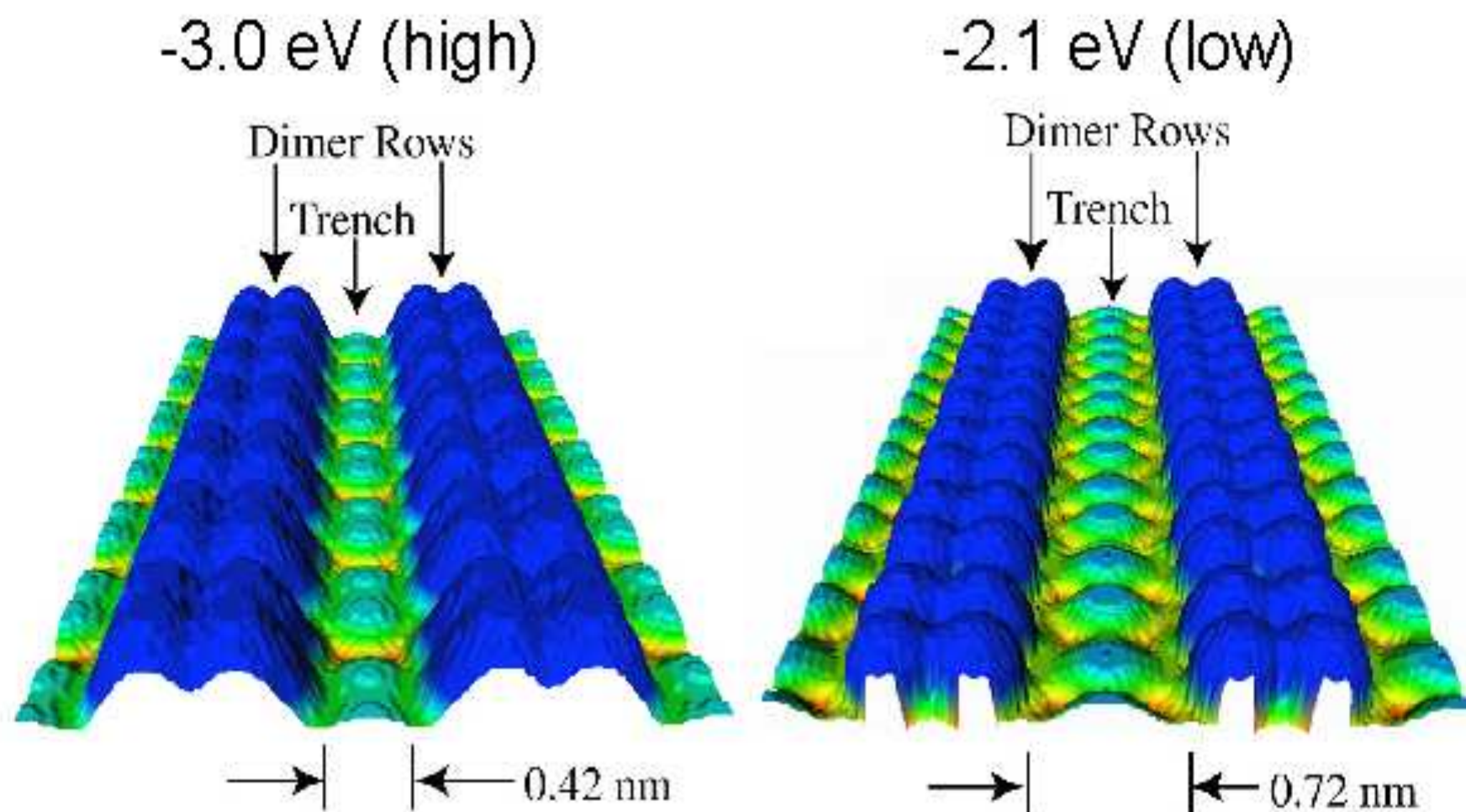
simulated image
local density of states
integrated to 0.3 eV
below the valence
band maximum →

*LaBella, Yang, Bullock,
Thibado, Kratzer & Scheffler,
PRL **83**, 2989 (1999).*



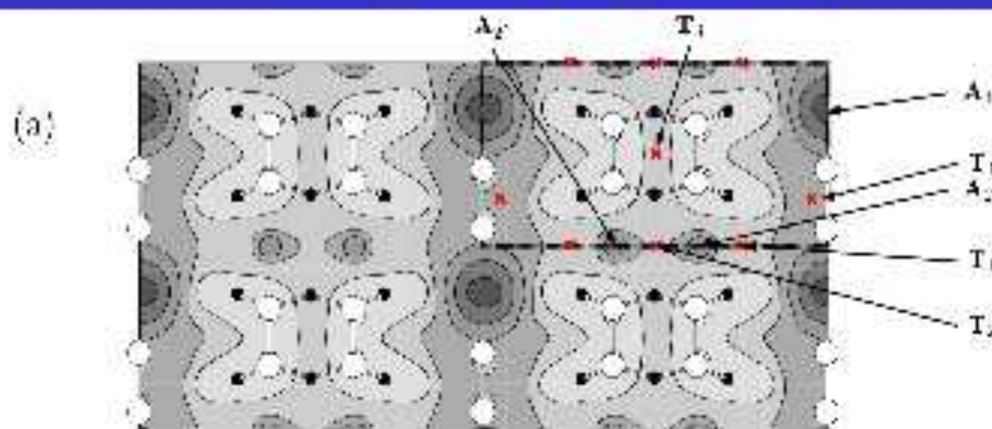
Voltage dependence of the STM current

STM simulation



*LaBella, Yang, Bullock,
Thibado, Kratzer & Scheffler,
PRL **83**, 2989 (1999).*

Total energy of a diffusing Ga atom at GaAs (001)



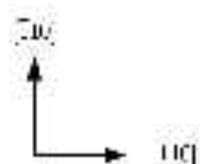
$$E_{A1} = -2.5 \text{ eV}$$

$$E^{TS} = -2.0 \text{ eV}$$

$$E_{A2} = -2.2 \text{ eV}$$

$$E^{TS} = -1.8 \text{ eV}$$

top
view



top view

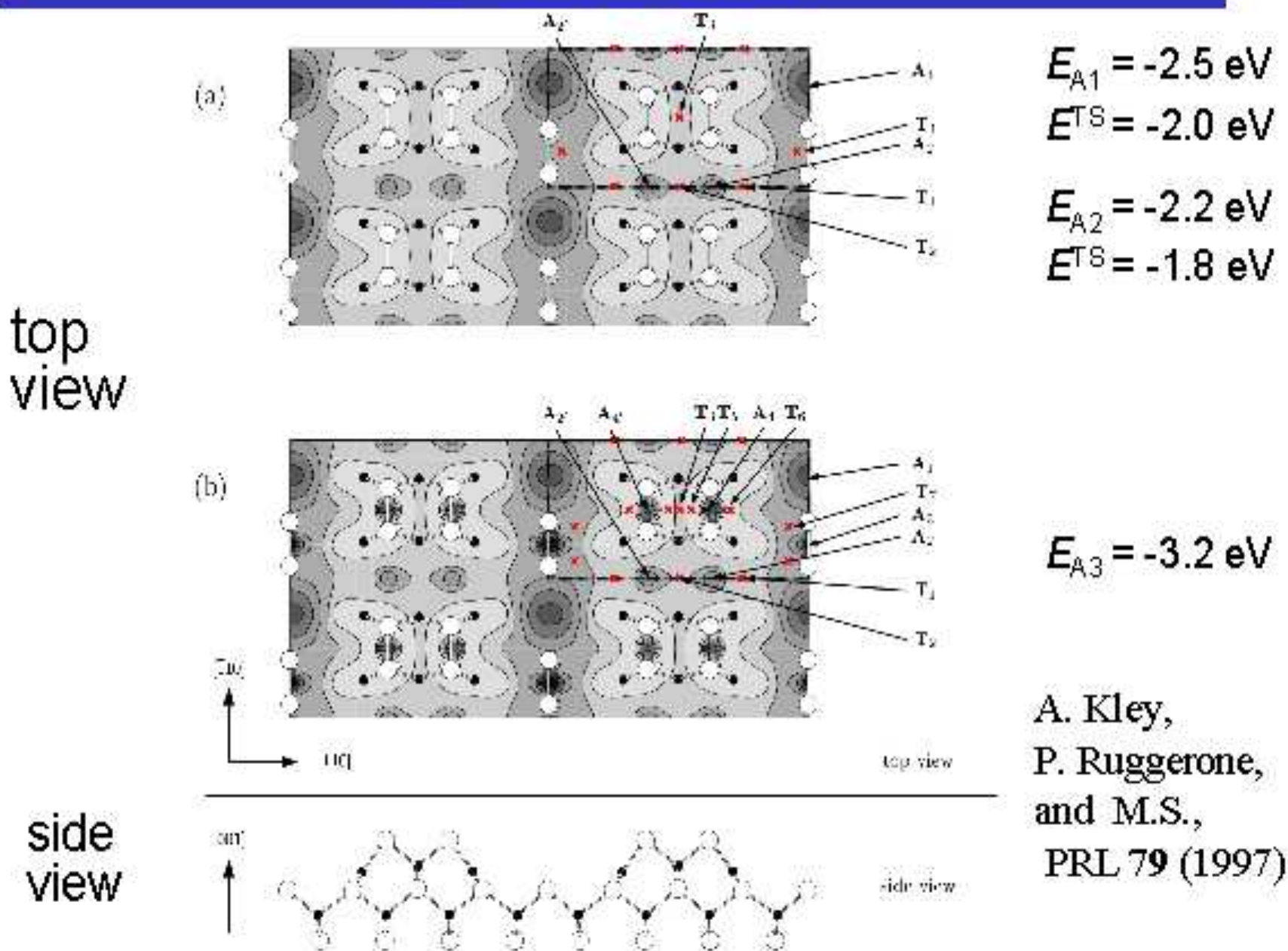
side
view



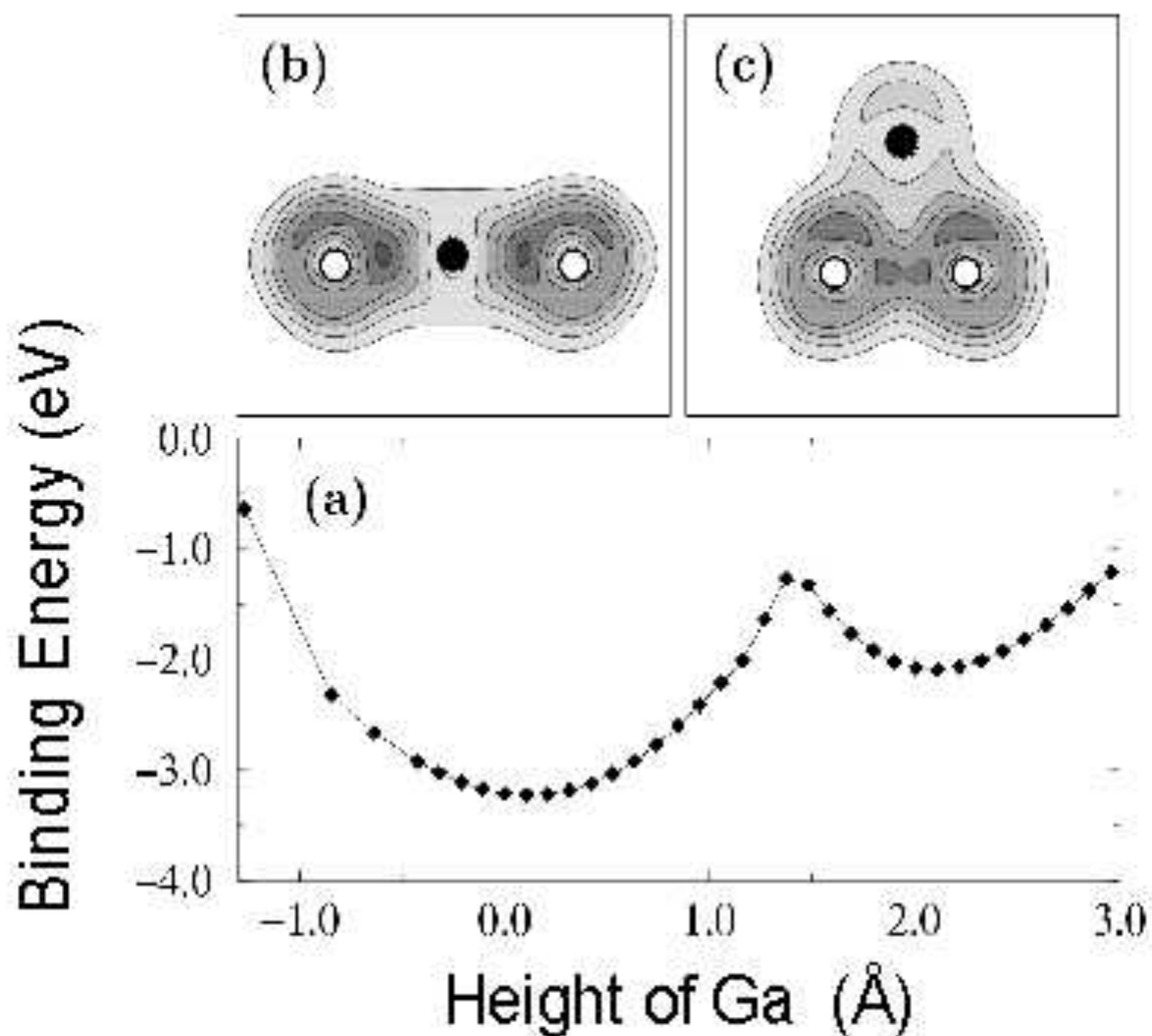
side view

A. Kley,
P. Ruggerone,
and M.S.,
PRL 79 (1997)

Total energy of a diffusing Ga atom at GaAs (001)



Unusually stable site for Ga adatom inside the trench-site As-dimer

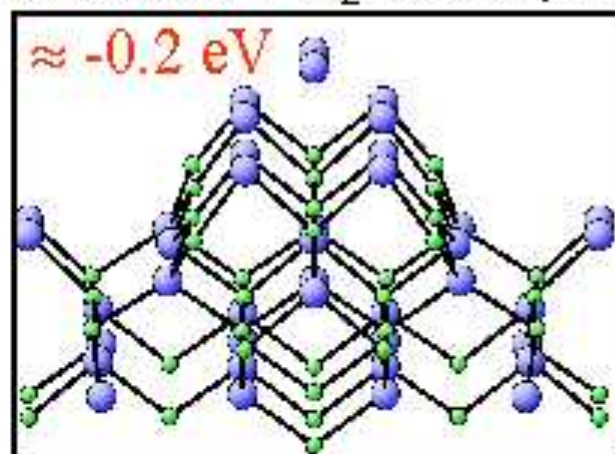


A. Kley, P. Ruggerone, M.S., PRL 79 (1997)

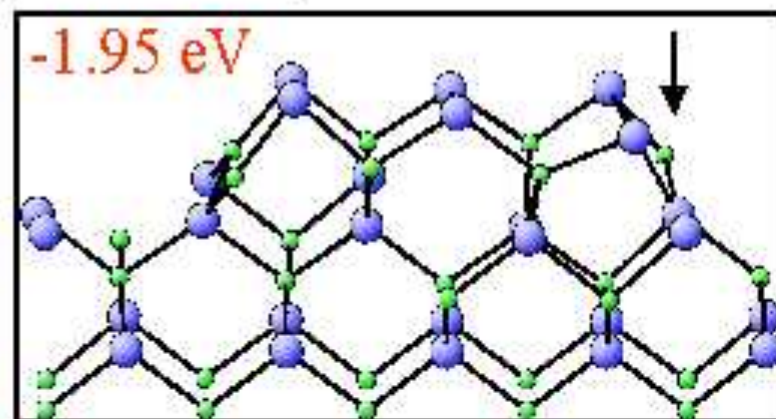
As₂ Adsorption

C. Morgan, P. Kratzer,
M.S., PRL **82** (1999)

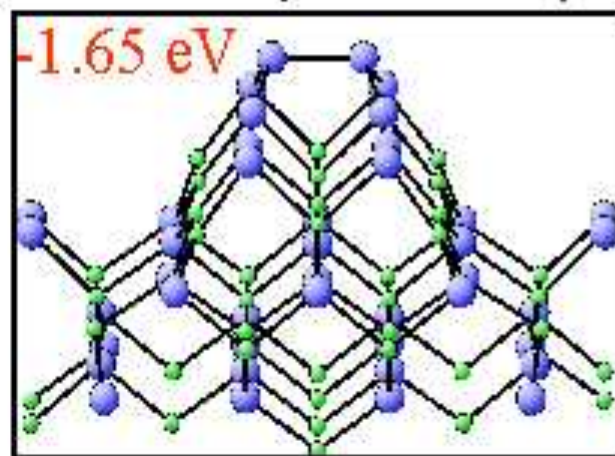
molecular As₂ adsorption



As₂ chemisorption at Ga adatom



As₂ chemisorption in top layer



As₂ does not need to break up to become incorporated. As₂ is readily available everywhere on the surface

Theory of the kinetics of growth

- 1) Analysis of all possibly relevant processes
- 2) Calculate the rates of all important processes

$$\Gamma^{(i)} = \Gamma_0^{(i)} \exp(\Delta E^{(i)} / kT)$$

- 3) Statistical approach to describe
 - deposition
 - diffusion
 - nucleation
 - growth



kinetic Monte Carlo method

movie

Adsorption, diffusion, island nucleation,
and growth of GaAs

DFT-GGA calculations

goto

<http://www.fhi-berlin.mpg.de/th/Highlights>
to see the movie

we display:

1/60 of the full simulation cell

total time = 1 sec

As₂ pressure = 1.33×10^{-3} Pa = 1.33×10^{-8} bar

Ga deposition rate = 0.1 ML/s

T=700 K (somewhat low; exp. uses about 850 K)

Summary

What makes GaAs(001) the preferred substrate?

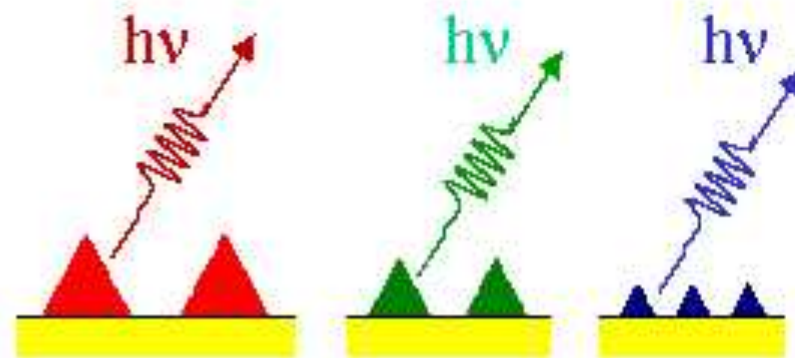
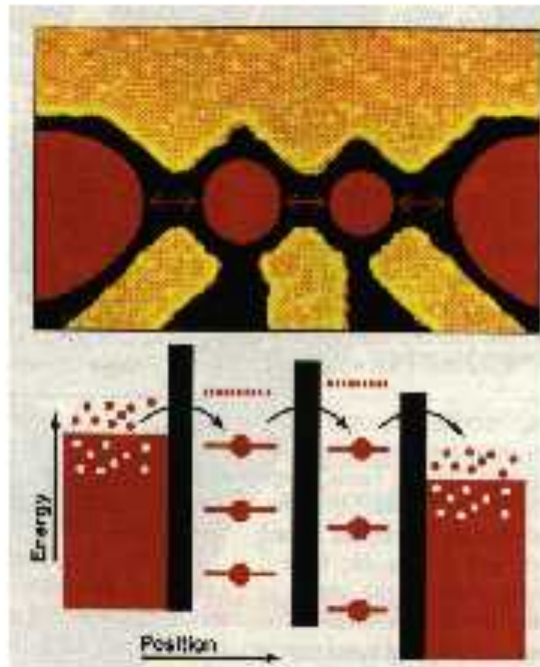
- principal structural element: As dimers
 - As_2 molecules are incorporated undissociatedly and without an activation barrier
 - several intermediate binding sites enhance the As_2 surface lifetime
- unusually stable two-fold coordinated cation adsorption state
 - cation evaporation is negligible even at high growth temperatures

Self-Assembly of Nano-Scale Structures at Semiconductor Surfaces

Motivation:

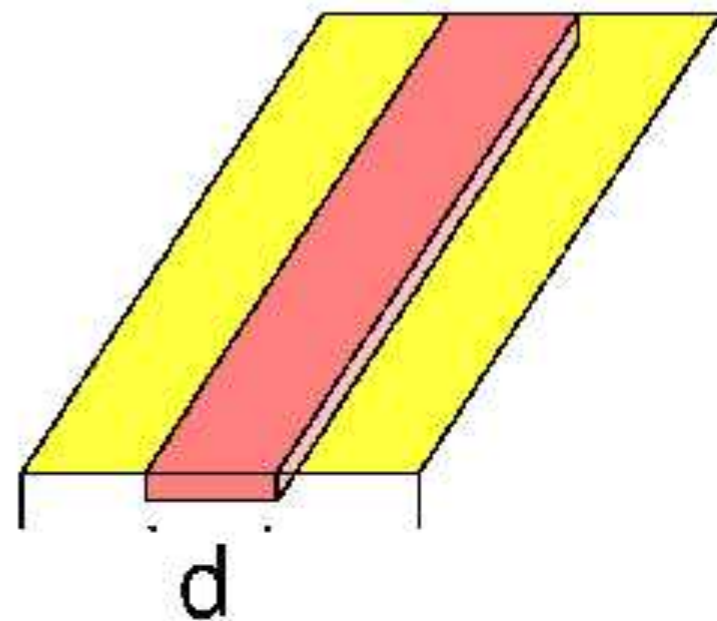
Single-electron transistor

LEDs and
laser diodes



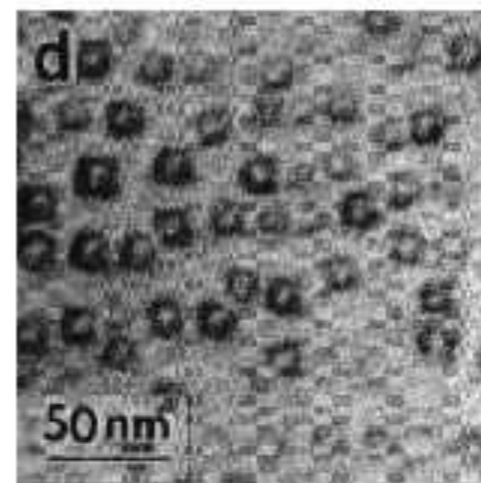
required dimensions: 50 - 200 Å

lithography ends
at $d > 300$ Å,
and has rough
edges



→ [100]

self-assembly of
InAs quantum-
dots on
GaAs (100)



[010]

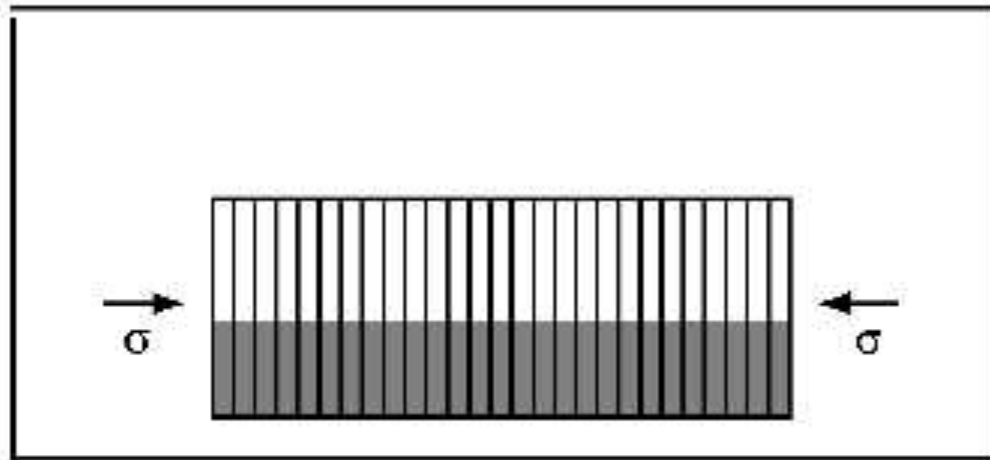
Ruvimov et al., PRB 1995

Thermal Equilibrium Shapes of InAs Quantum Dots on GaAs(100)

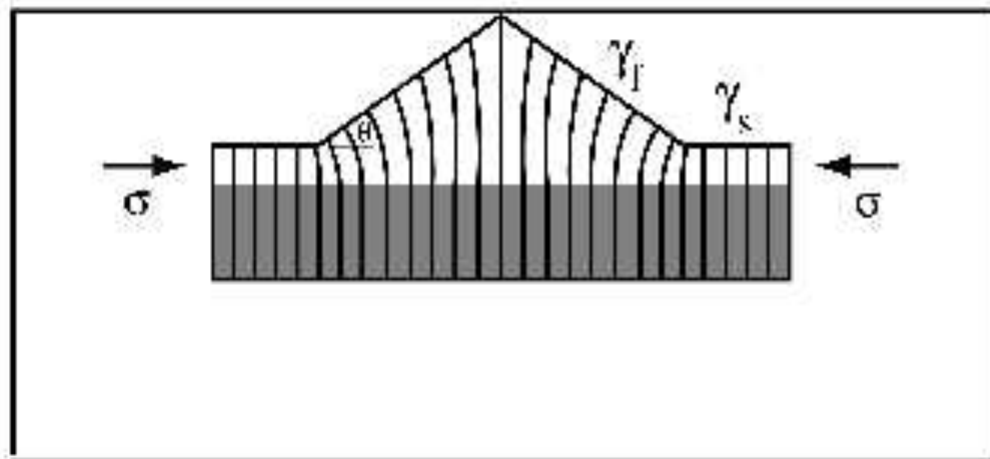
Stranski-Krastanov Growth Mode

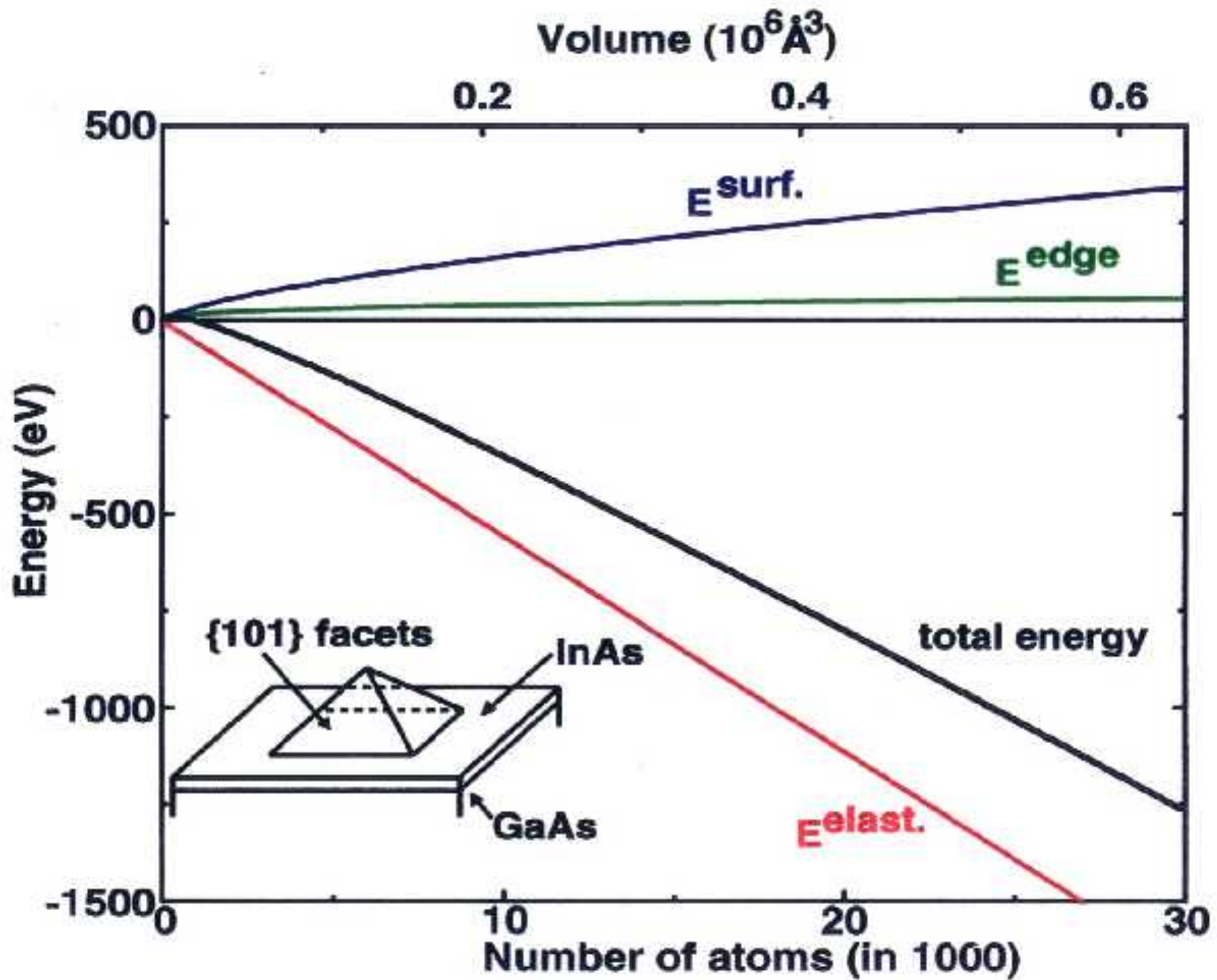
If the experimental quantum dot shape deviates from the equilibrium shape, equilibrium thermodynamics is not adequate to describe the island formation and size distribution

Stranski-Krastanov morphology as one way to reduce misfit strain energy

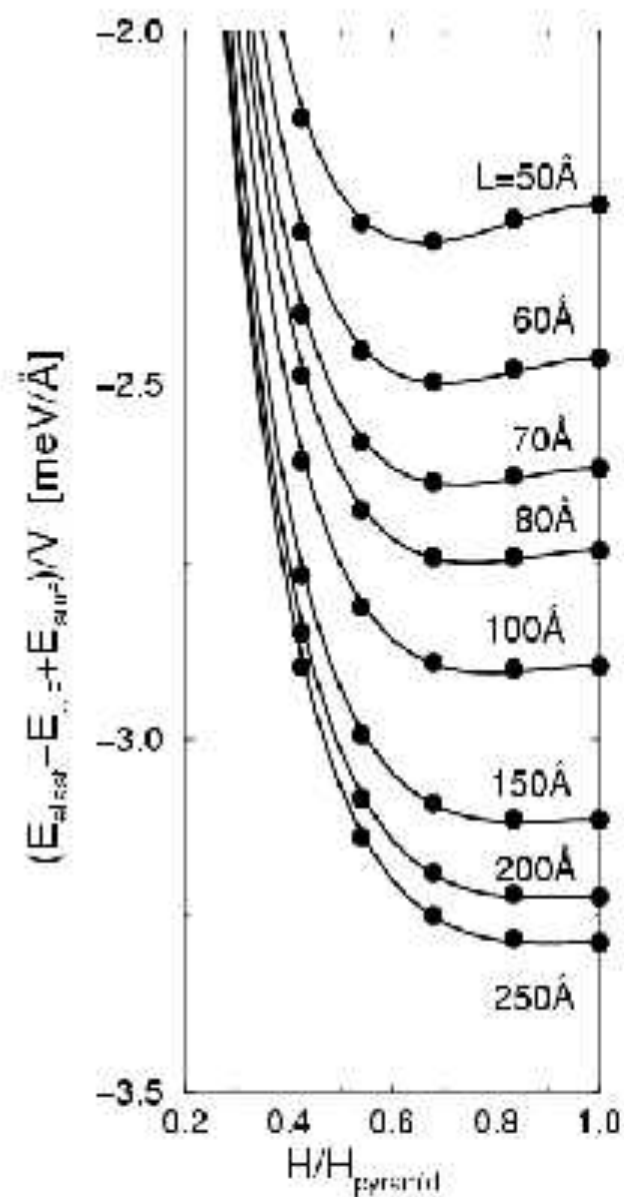
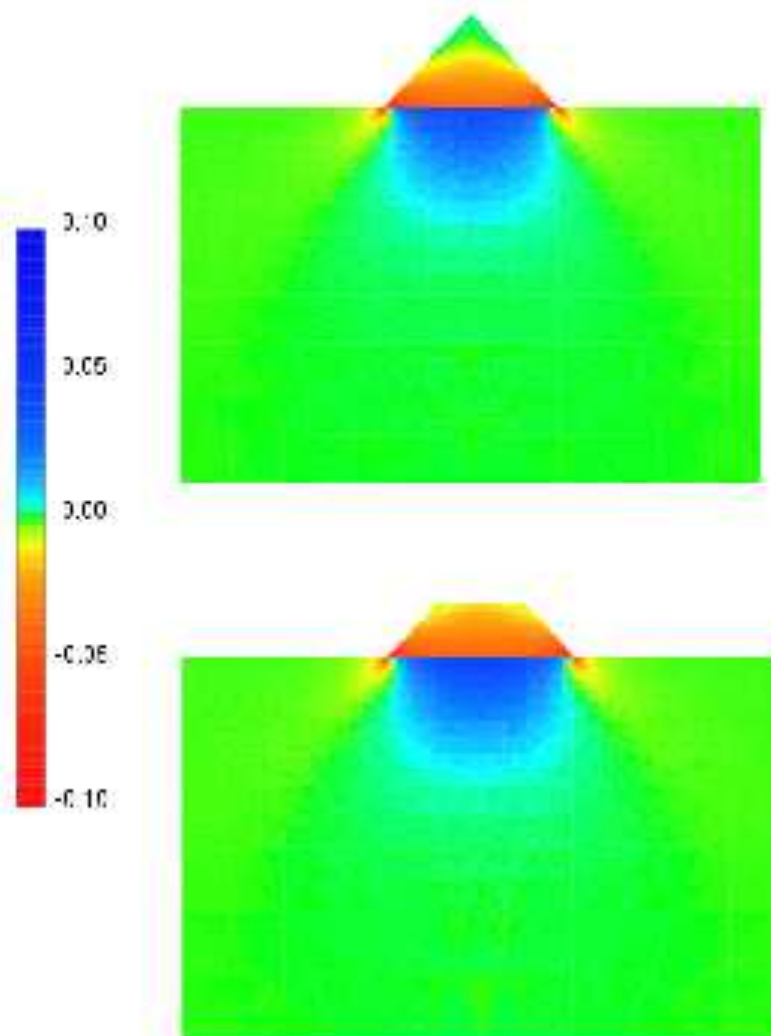


e.g.
InAs on
GaAs

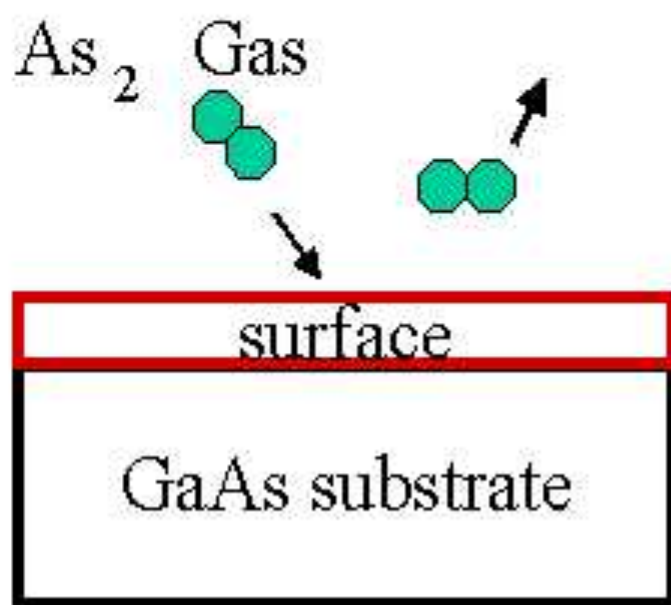




Stress tensor at strained InAs on GaAs



Stoichiometry and structure of the surface depend on the environment
(atomic chemical potentials)



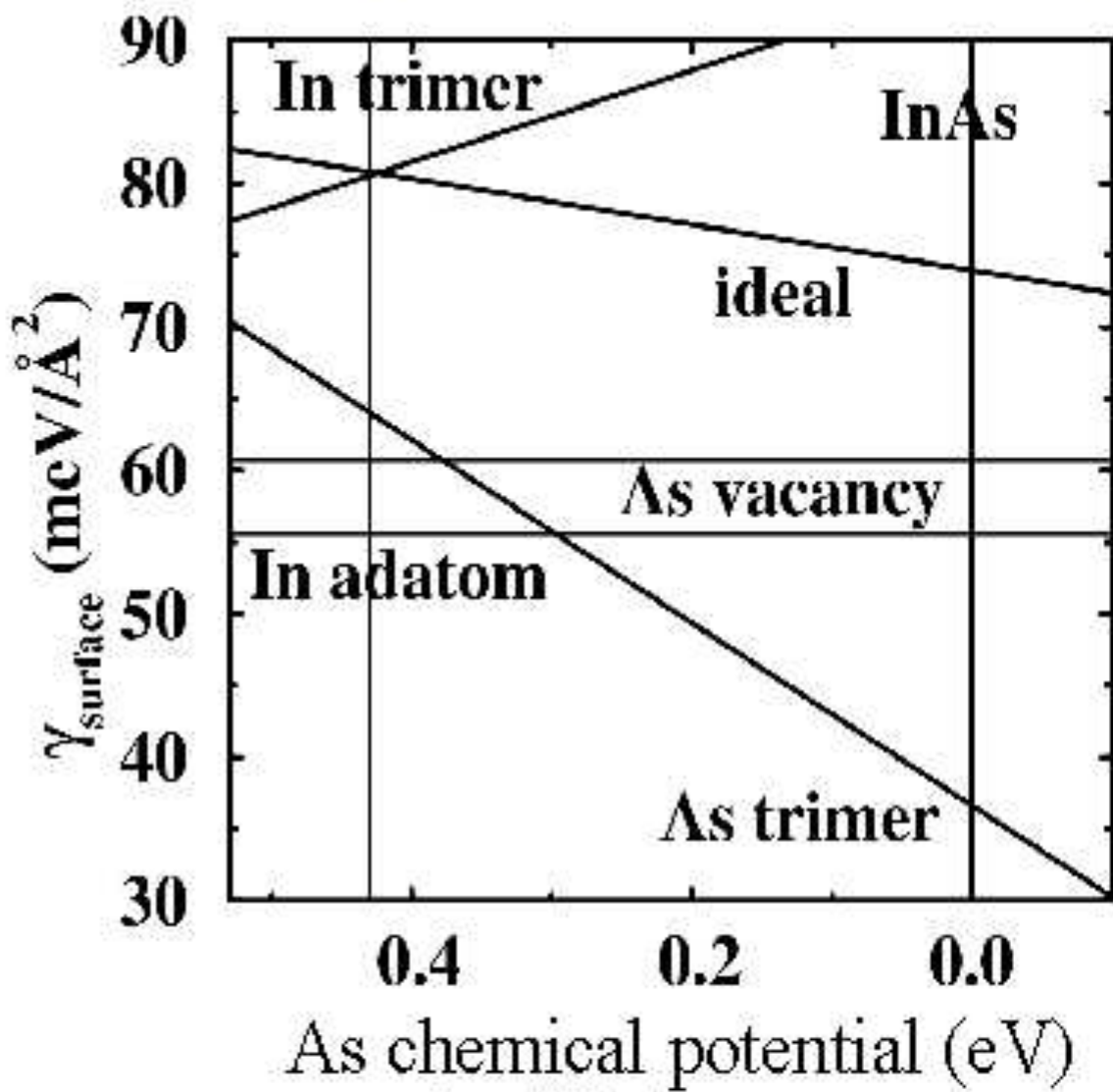
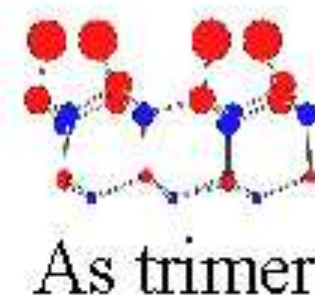
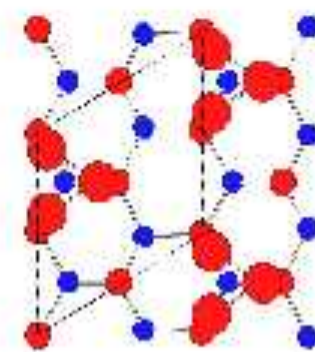
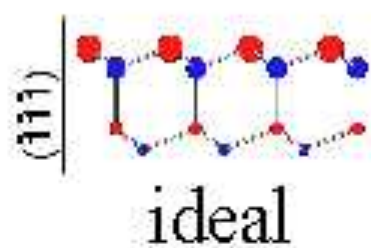
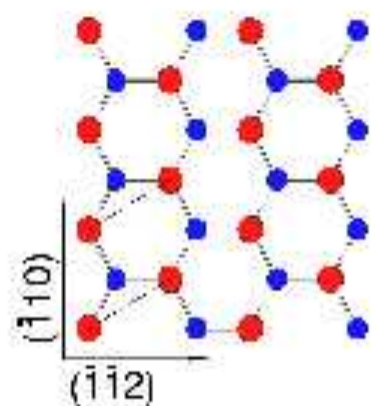
$$\mu_{\text{As}} < E^{\text{bulk}}(\text{As})$$

$$\mu_{\text{Ga}} < E^{\text{bulk}}(\text{Ga})$$

$$\mu_{\text{Ga}} + \mu_{\text{As}} = E^{\text{bulk}}(\text{GaAs})$$

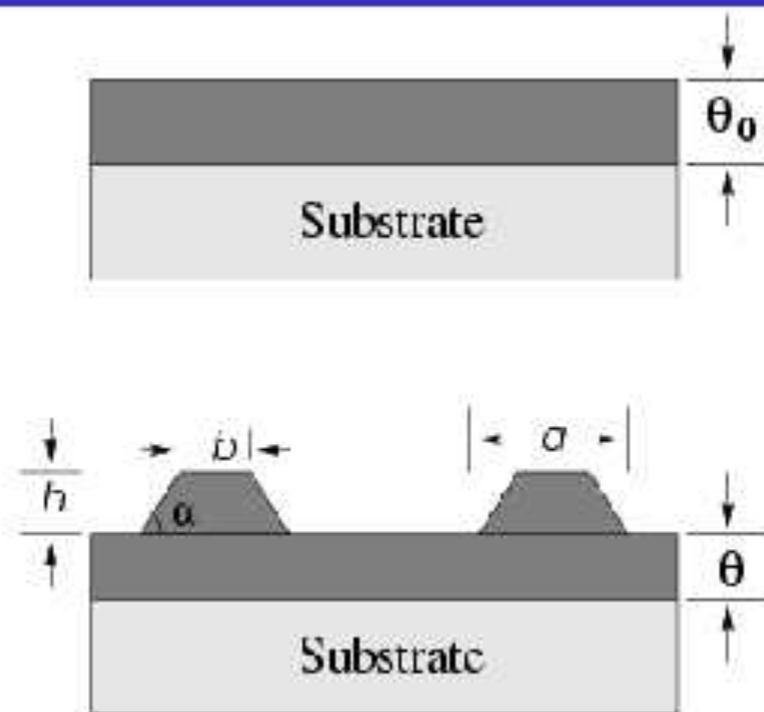
DFT Calculations of Surface Energies

Example: InAs (-1-1-1)



What determines the volume and shape of quantum dots? (how q-dots eat from the wetting layer)

e.g. InAs
on GaAs

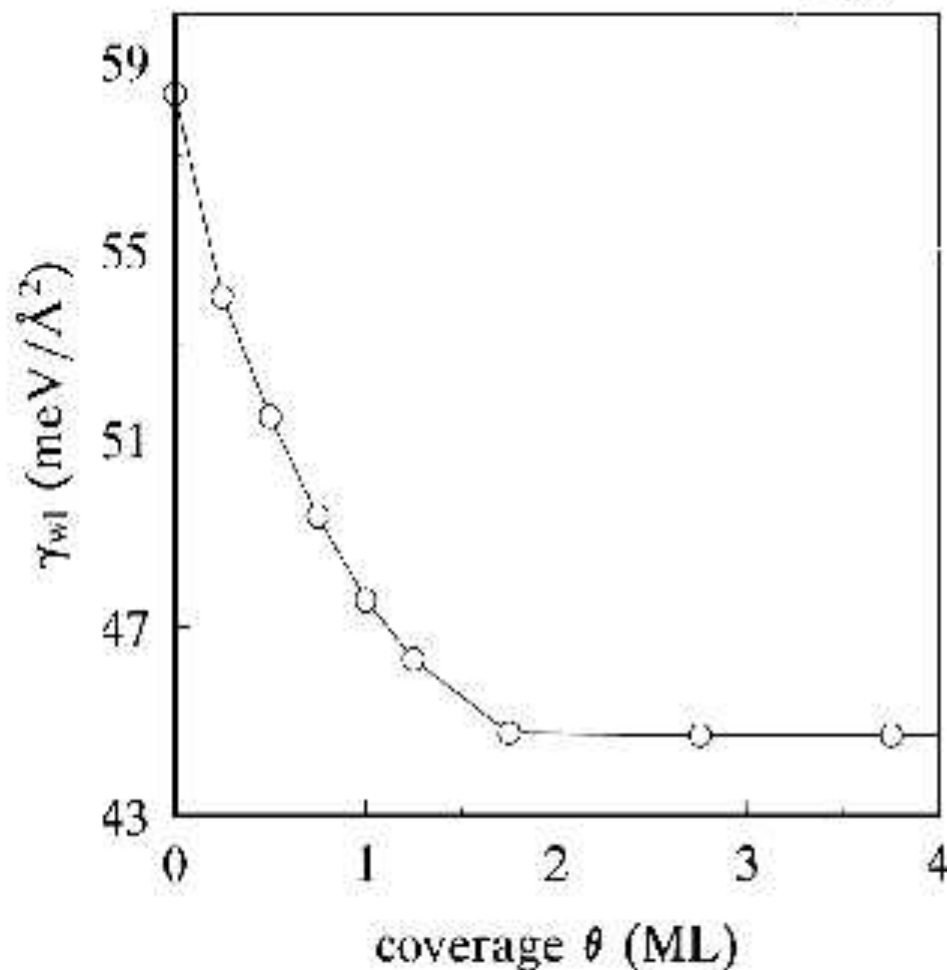


*L. Wang,
P. Kratzer,
and M.S.,
PRL 82 (1999)*

$$\begin{aligned}
 E/V &= E_{\text{is}}^{\text{cl}}/V - \epsilon_{\text{film}}^{\text{cl}} \\
 &+ [S\gamma_f + b^2\gamma_{\text{top}} - a^2\gamma_{\text{wl}}(\theta_0)]/V \\
 &+ (1/n - a^2) \times [\gamma_{\text{wl}}(\theta) - \gamma_{\text{wl}}(\theta_0)]/V
 \end{aligned}$$

InAs on GaAs (100)

surface plus interface energies



$$\gamma_{wl} A = E^{\text{tot}} - E^{\text{bulk}}(\text{GaAs}) N_{\text{Ga}}$$

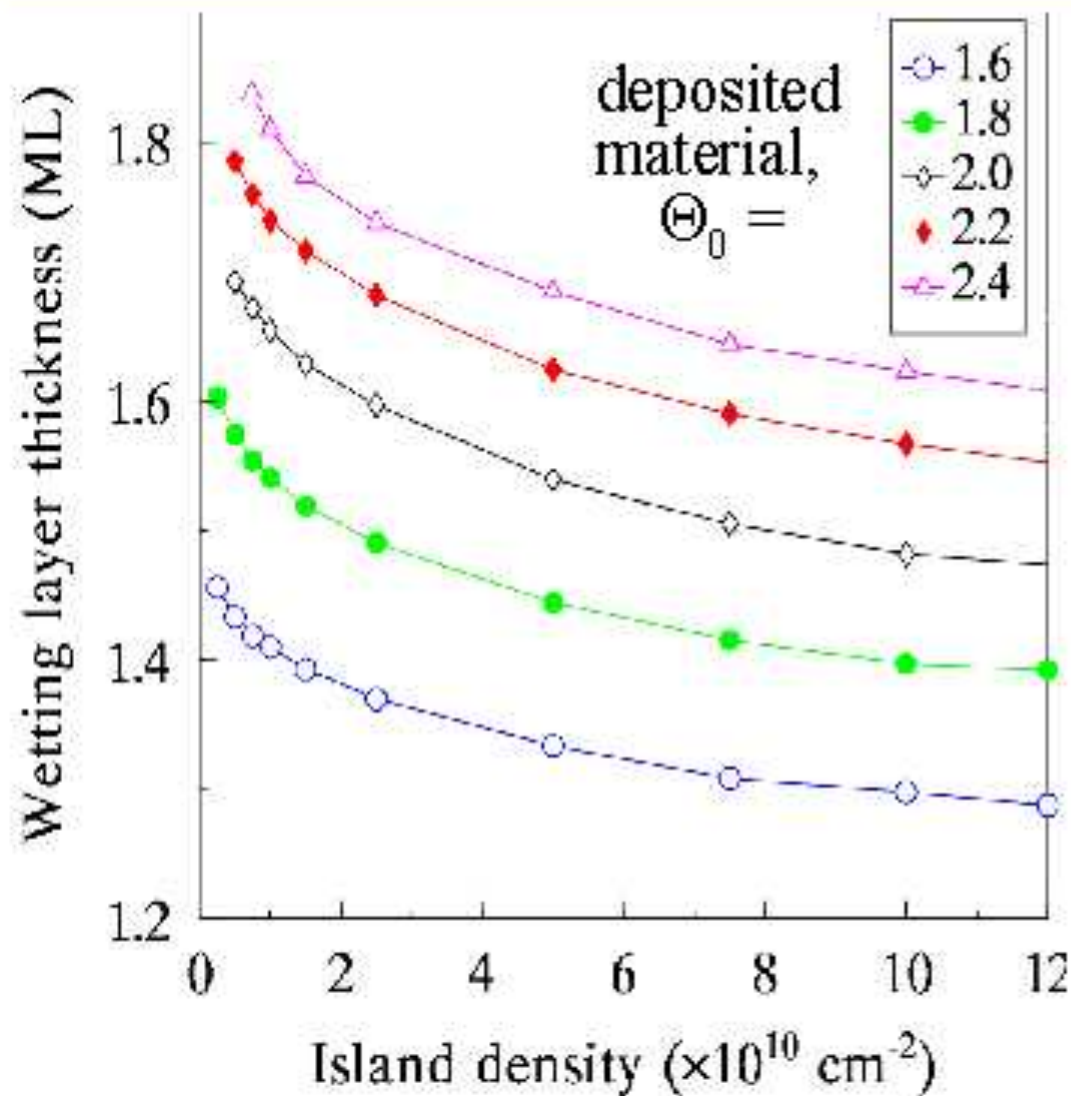
$$- E^{\text{strained-bulk}}(\text{InAs}) N_{\text{In}}$$

$$- \mu_{\text{As}} \Delta N_{\text{As}}$$

$$\mu_{\text{As}} = E^{\text{bulk}}(\text{As}) - 0.2 \text{ eV}$$

*L. Wang,
P. Kratzer,
and M.S.,
PRL 82 (1999)*

Thickness of the wetting layer depends on the island density

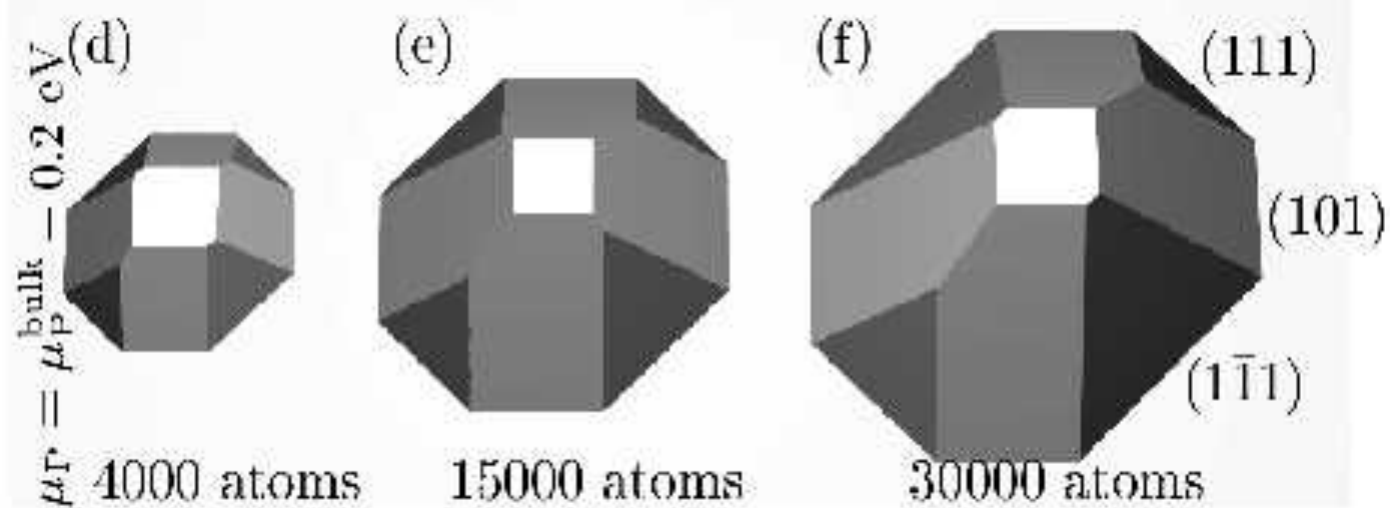
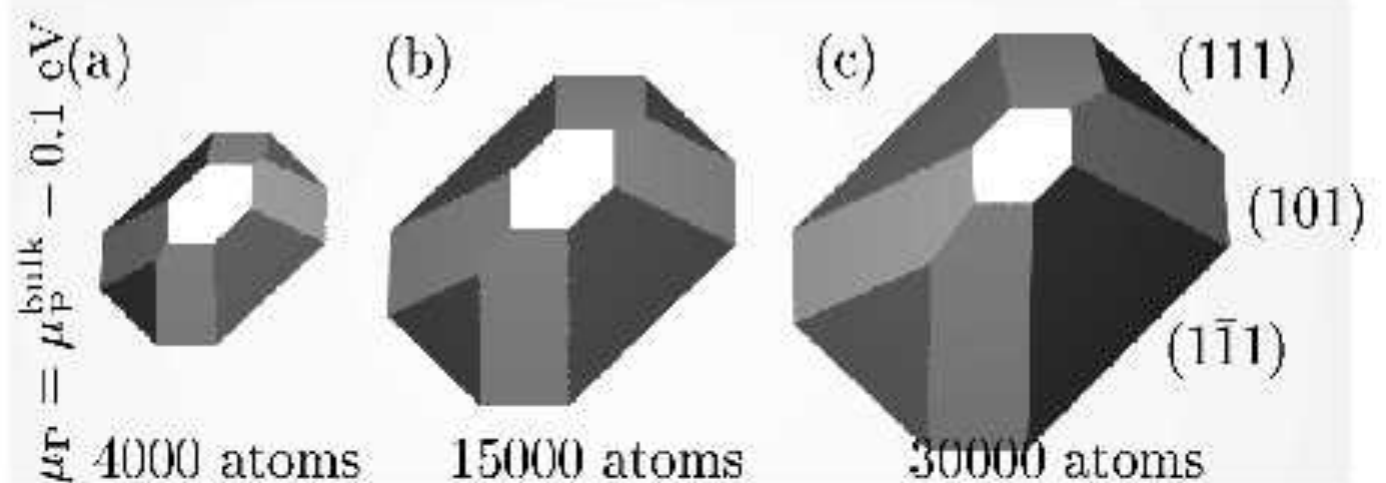


InAs on GaAs

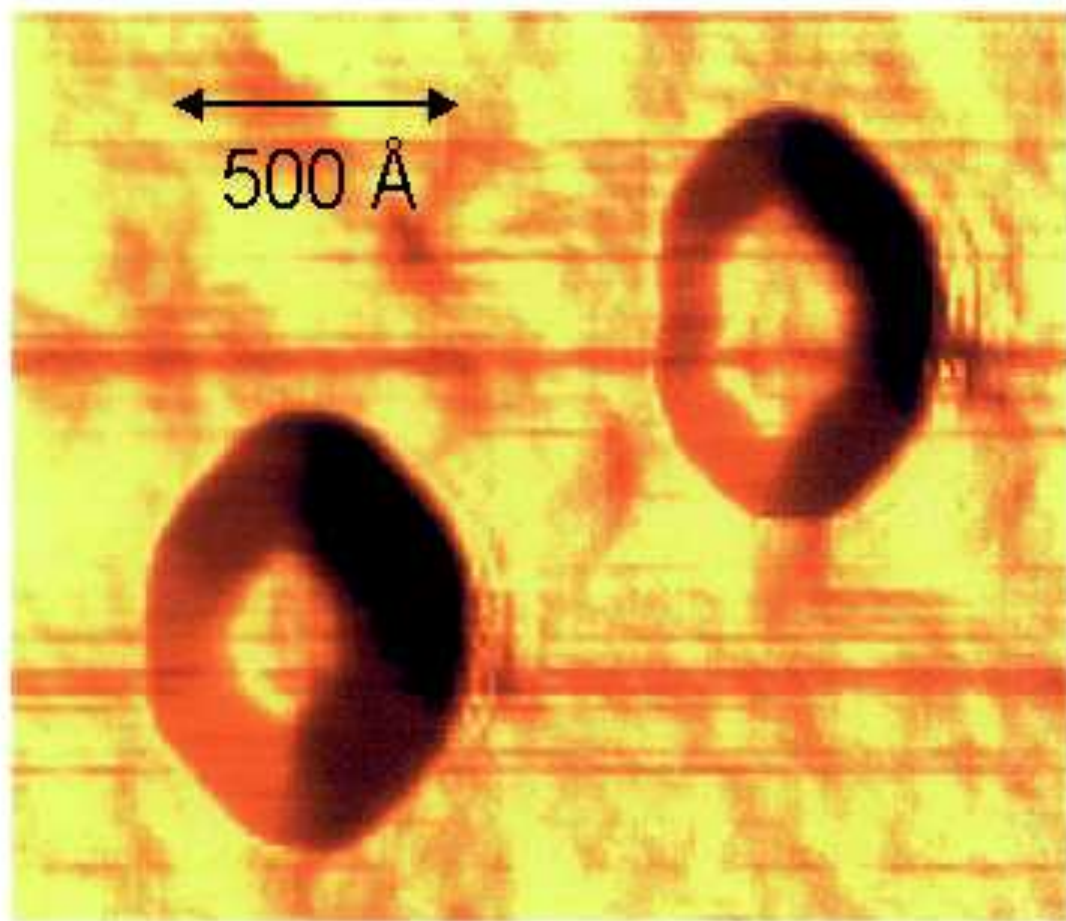
$$\mu_{\text{As}} = E^{\text{bulk}}(\text{As}) - 0.2 \text{ eV}$$

*L. Wang,
P. Kratzer,
and M.S.,
PRL 82 (1999)*

InP quantum dots on GaP(001)



Q. Liu, E. Pehlke, N. Moll, M.S., PRB 60, (1999)



Samuelson *et al.*
(1996)

MOVPE
grown InP
islands on GaInP

Electronic Structure Theory
(Density Functional Theory)

➔ **Potential Energy Surface**

Dynamics of the Nuclei
along this **PES**

Thermal
Equilibrium
Structures

Statistical Mechanics

Real World

